

Strongly coloured thiocyanate frameworks with perovskite-analogue structures

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October 24, 2018

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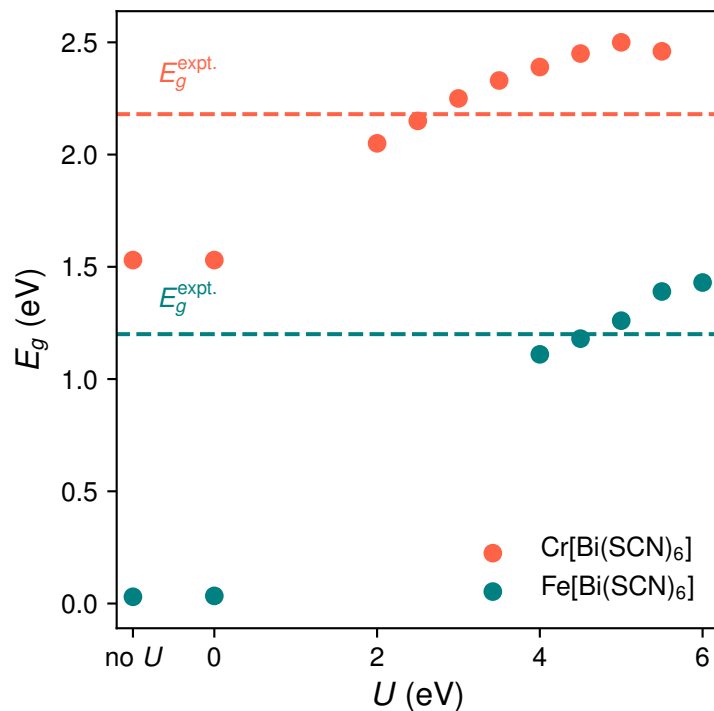
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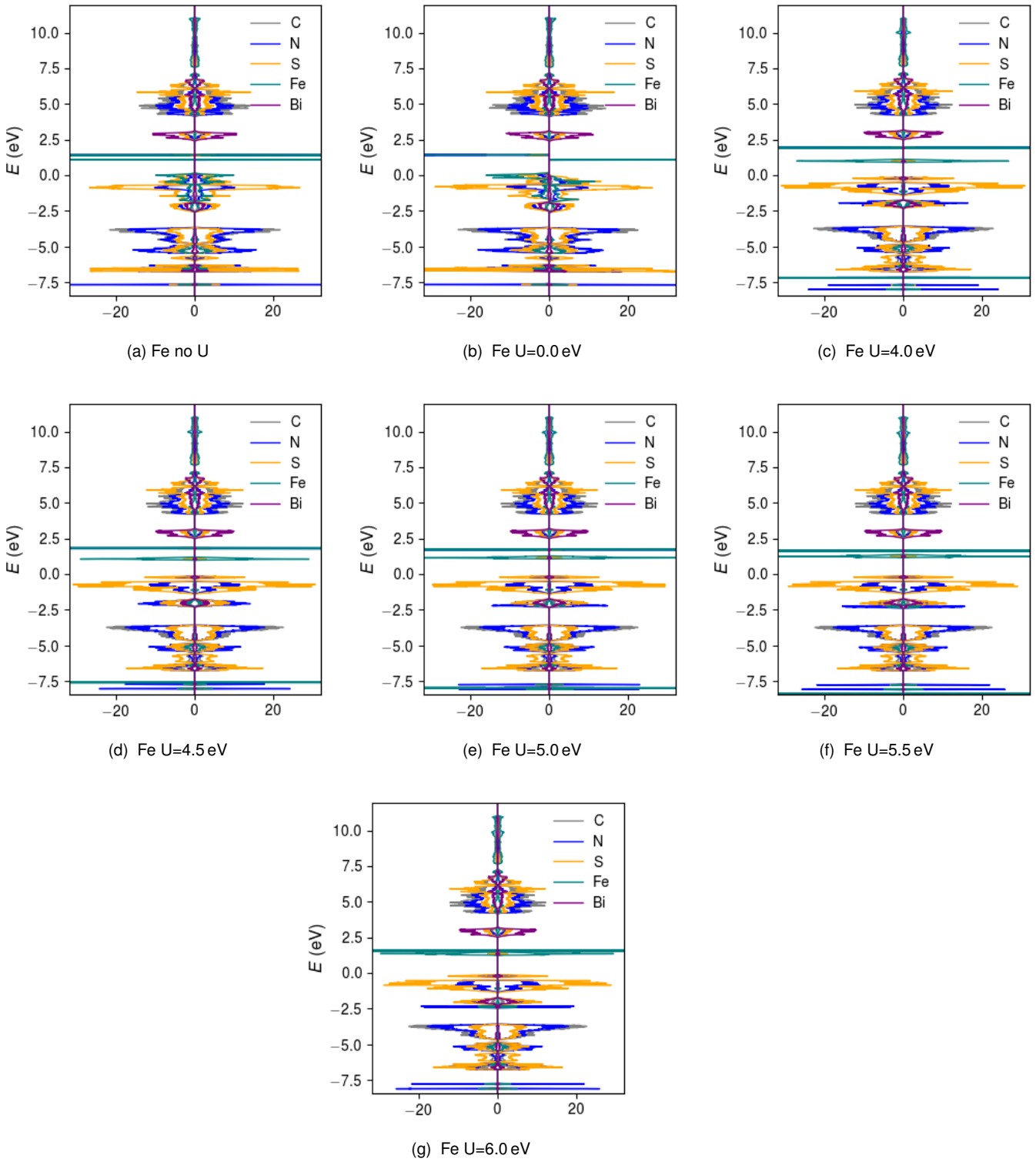
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1 DFT+ U details

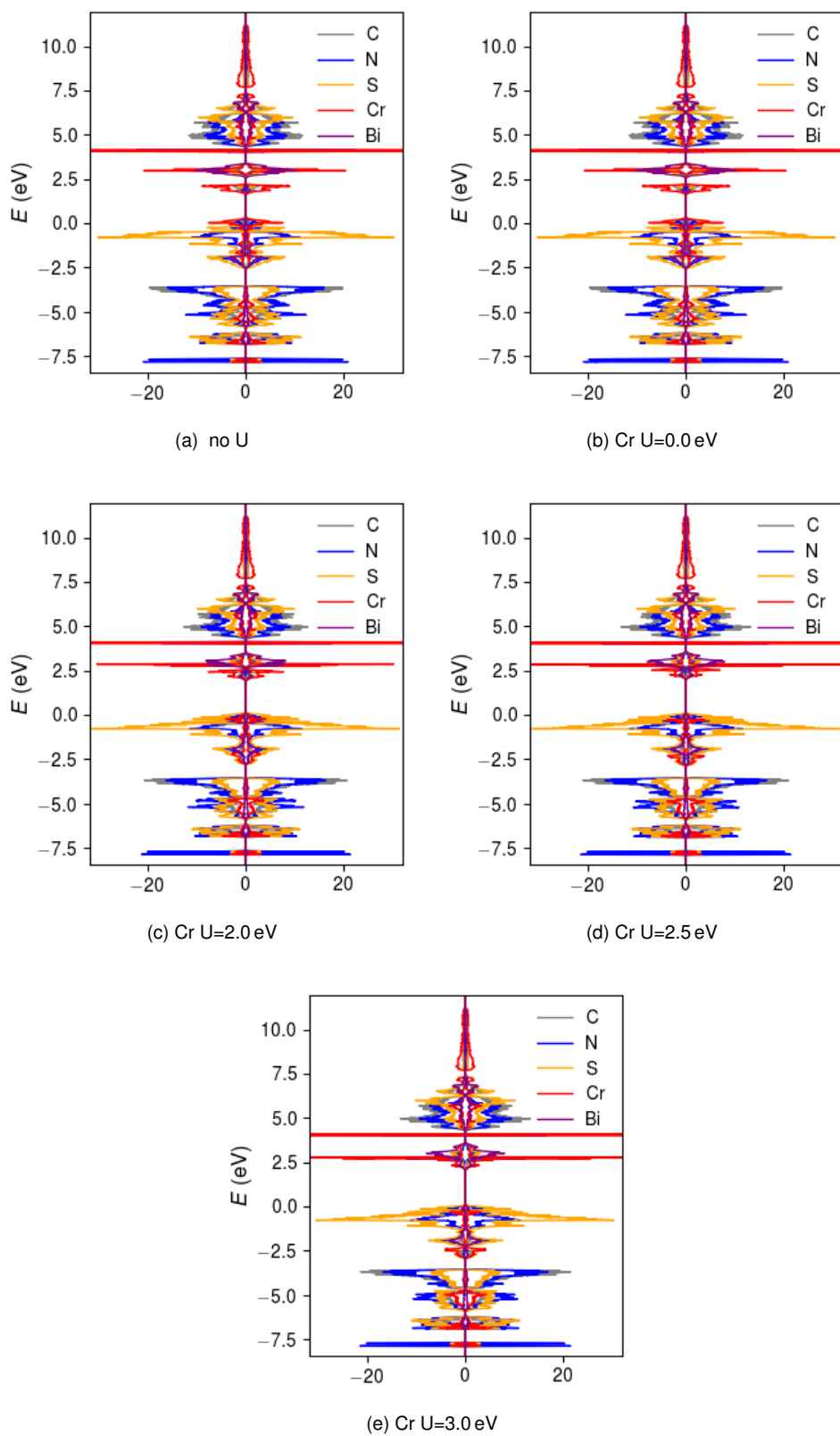
In order to account for the known self-interaction error found in DFT calculations, we carried out calculations including a Hubbard U -parameter for a range of values, around those previously reported in the literature.^{S1, S2} We found that although this did alter the energy of the metal d states, as expected, the band structure and density of states remained consistent over the full range explored. In the main text we present results for a value of U that produces a band gap consistent with that observed in the literature.



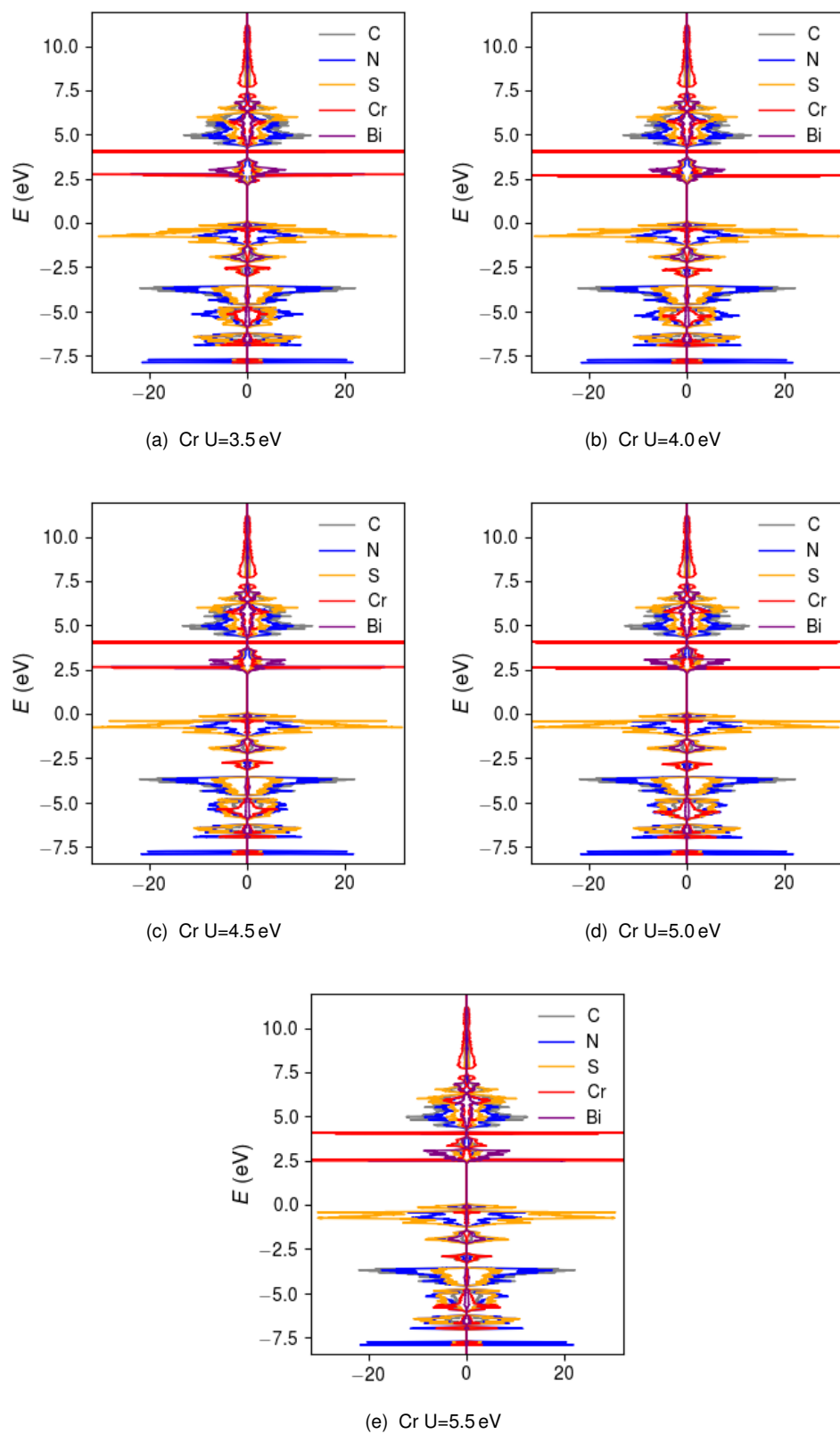
ESI Fig. 1: Variation in calculated optical (direct) band gap with the U parameter



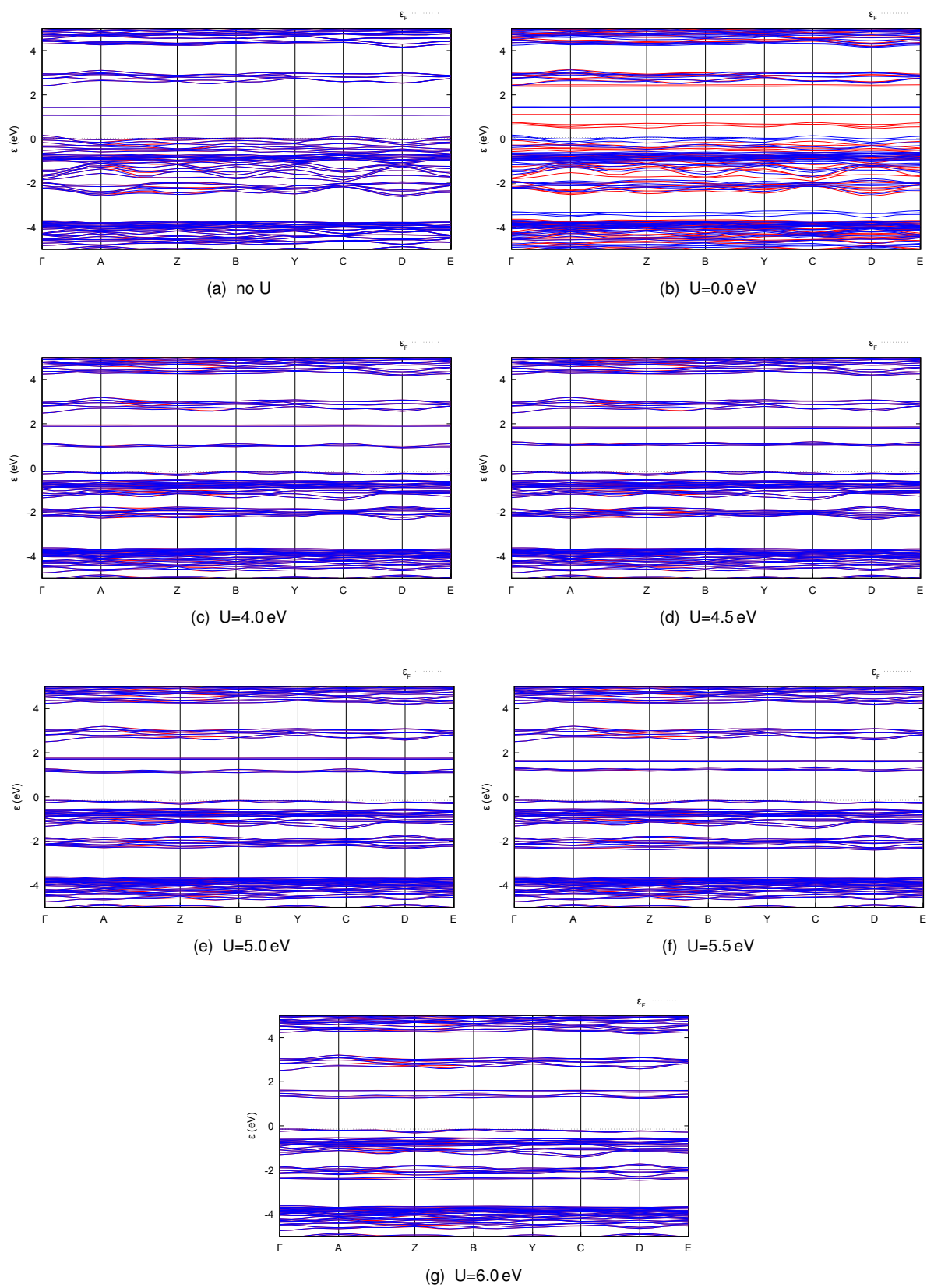
ESI Fig. 2: Variation of $\text{FeBi}(\text{SCN})_6$ DoS with U , spin polarisation indicated by positive or negative sign of density.



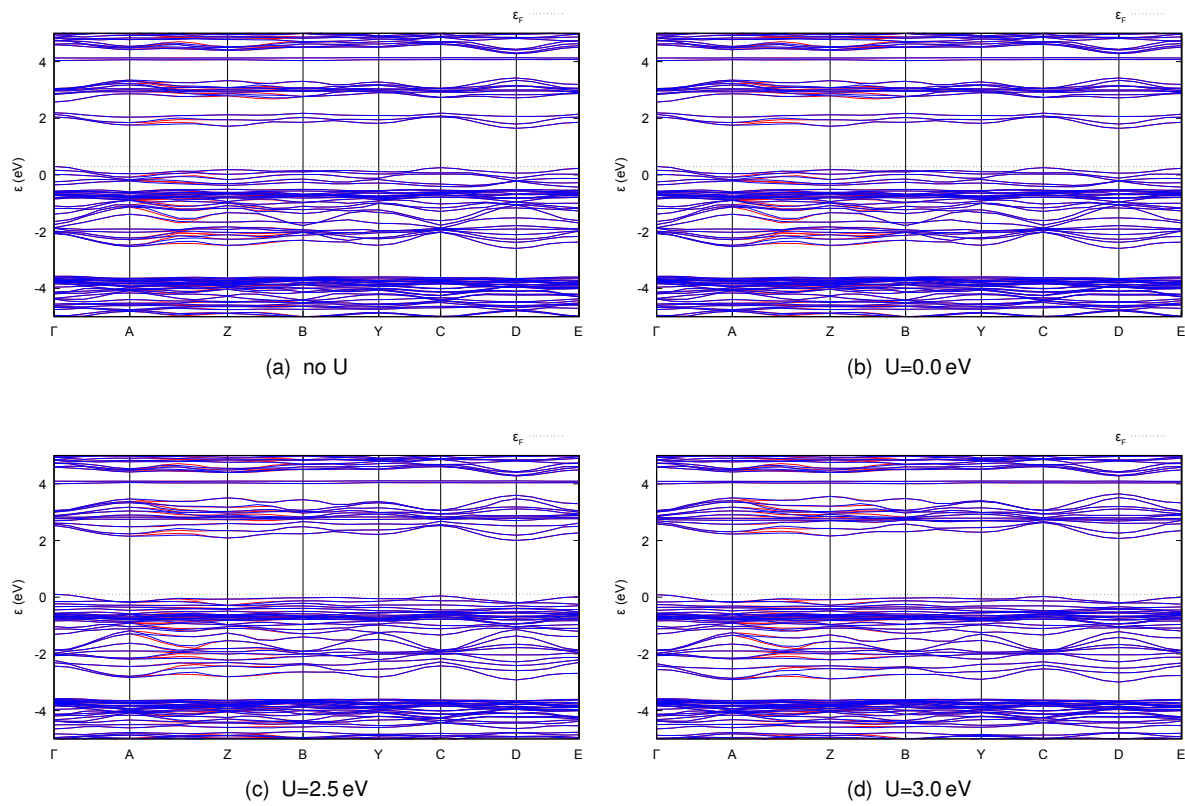
ESI Fig. 3: Variation of $\text{CrBi}(\text{SCN})_6$ DoS with U , spin polarisation indicated by positive or negative sign of density.



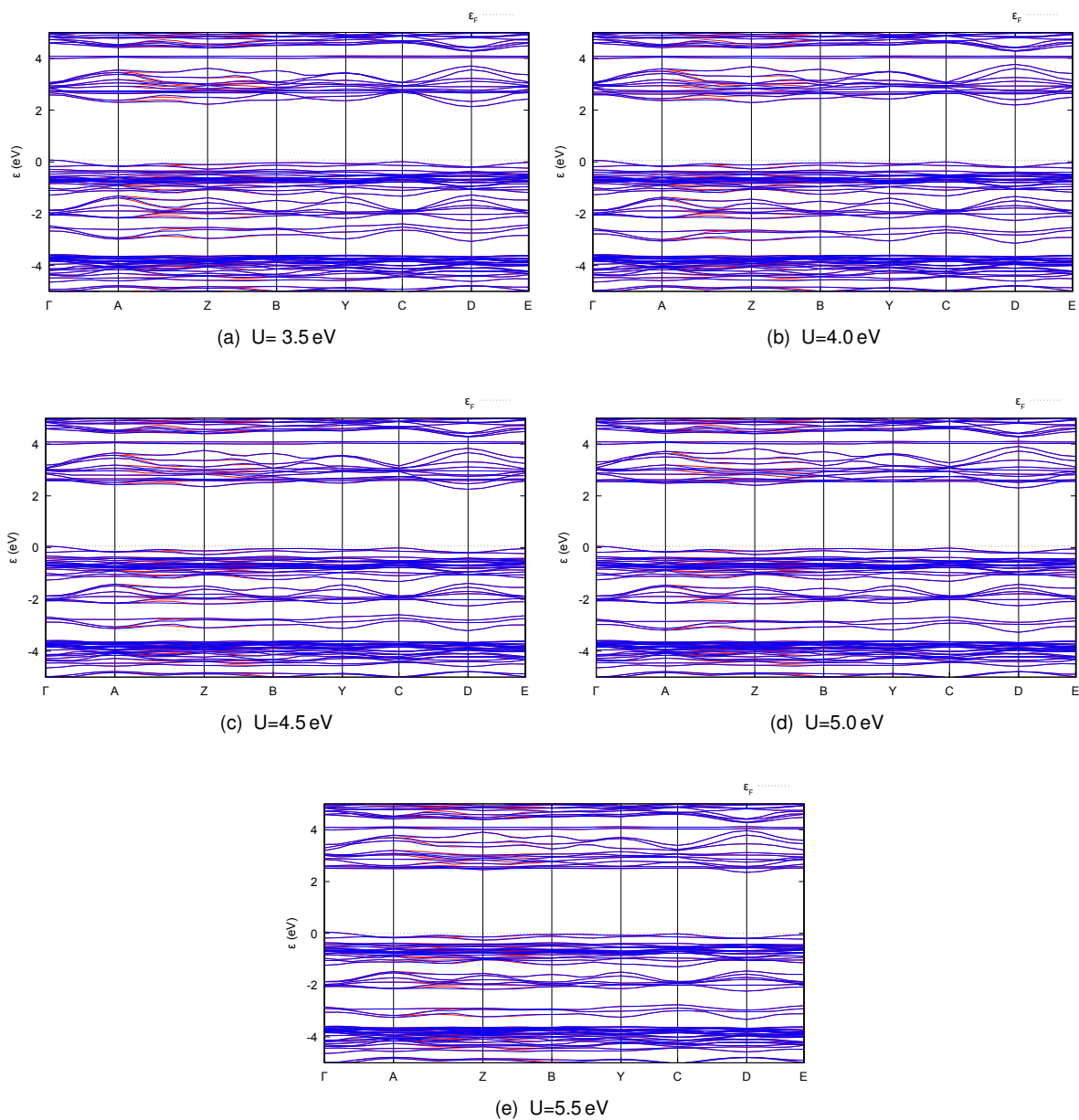
ESI Fig. 4: Variation of $\text{CrBi}(\text{SCN})_6$ DoS with U , spin polarisation indicated by positive or negative sign of density.



ESI Fig. 5: Variation of $\text{FeBi}(\text{SCN})_6$ band structure with U . Spin polarisation indicated by colour of line.



ESI Fig. 6: Variation of $\text{CrBi}(\text{SCN})_6$ band structure with U up to $U = 3.0$ eV. Spin polarisation indicated by colour of line.



ESI Fig. 7: Variation of $\text{CrBi}(\text{SCN})_6$ band structure with U greater than $U = 3.5\text{ eV}$. Spin polarisation indicated by colour of line.

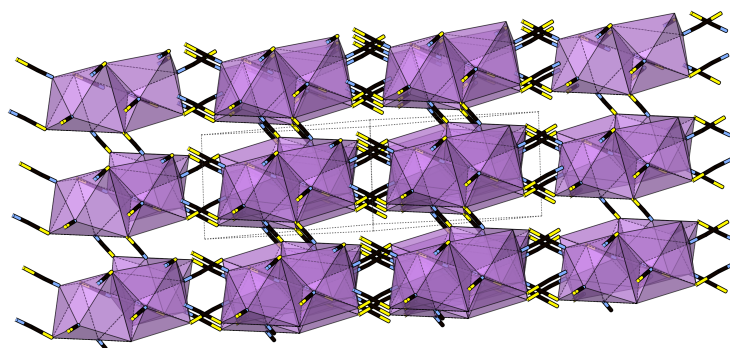
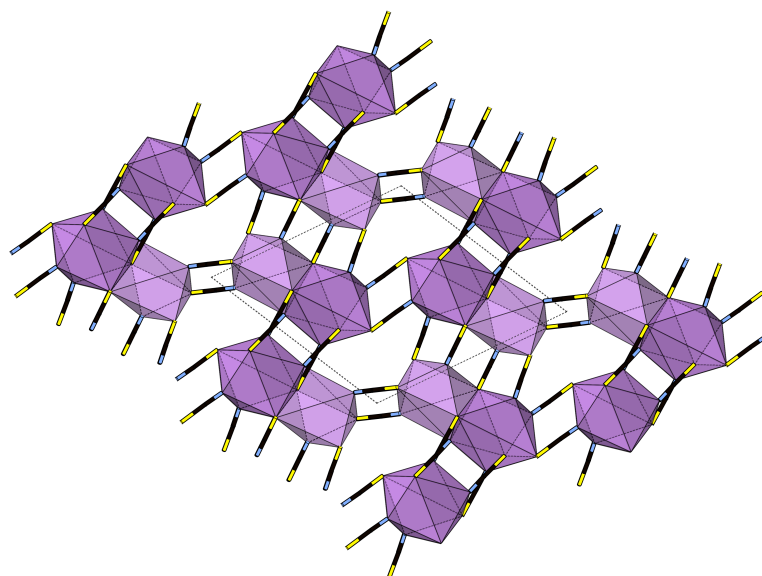
2 Bi(NCS)₃

During the synthesis of Fe[Bi(SCN)₆] we were able to obtain single crystals of Bi(SCN)₃ from the evaporation of H₃[Bi(SCN)₆] solutions in water. Here we briefly describe the structure of this new material. Bi(SCN)₃ crystallises in a low symmetry triclinic $P\bar{1}$ structure containing two distinct Bi³⁺ and six distinct NCS⁻ sites. The bonding around each Bi³⁺ is very irregular, Bi01 forming short bonds to two sulfur atoms (2.68 Å) and two nitrogen atoms (2.4 Å) and Bi02 forming short bonds to two sulfurs and one nitrogen. If only these short bonds are considered Bi01 is roughly square pyramidal forming a molecular dimer (NCS)₂Bi(NCS)₂Bi(NCS)₂ and Bi02 is a triangular pyramid forming an isolated Bi(SCN)₃ molecular. When longer interactions are included, these molecular species connect into a three dimensional framework structure formed from dimers of distorted Bi01N₄S₄ polyhedron face sharing with a distorted octahedral Bi02N₃S₃ which are linked through NCS to other dimers. This three dimensional network is illustrated in ESI Figs. 8.

ESI Table 1: Bi-X bond distances

Bi	X	r(Å)	Bi	X	r(Å)
Bi01	S003	2.679(3)	Bi02	S004	3.005(3)
Bi01	S005	2.686(3)	Bi02	S006	2.699(3)
Bi01	N00C	2.434(9)	Bi02	S007	3.059(3)
Bi01	N00F	2.462(9)	Bi02	S008	2.677(3)
Bi01	N00G	2.674(11)	Bi02	N00A	2.345(9)
			Bi02	N00D	2.506(9)

Estimated standard error in parantheses.



ESI Fig. 8: Crystal structure of $\text{Bi}(\text{SCN})_3$, Bi in purple, N in blue, C in black, and S in yellow. Bi shown in polyhedral representation.

3 Powder diffraction studies

3.1 Rietveld refinement of Cr[Bi(SCN)₆]

Rietveld refinement of Cr[Bi(SCN)₆] was carried out from the single crystal structure of Fe[Bi(SCN)₆], with Cr substituted for Fe. The Cr(NCS)₆ octahedron was constrained as a rigid body, with the bond angles retained but the lengths fixed to $d_{\text{CN}} = 1.15 \text{ \AA}$, $d_{\text{NS}} = 1.65 \text{ \AA}$, $d_{\text{CrN}} = 2.0 \text{ \AA}$ and uniform isotropic atomic displacement parameters for the entire octahedron. The background terms were refined using a Chebyshev polynomial, and lattice parameters, rotation of the rigid octahedron, and isotropic displacement parameters were allowed to refine freely, along with crystallographic size and strain (anisotropically modelled using fourth-order spherical harmonics). Refinement statistics: $R_{wp} = 10.998$, $R_{\text{Bragg}} = 3.50$, goodness of fit = 1.871.

3.2 Rietveld refinement of Cr[Bi(SCN)₆] · H₂O

Rietveld refinement of Cr[Bi(SCN)₆] · H₂O was carried out from the single crystal structure of Fe[Bi(SCN)₆], with Cr substituted for Fe. The Cr(NCS)₆ octahedron was constrained as a rigid body, with the bond angles retained but the lengths fixed to $d_{\text{CN}} = 1.15 \text{ \AA}$, $d_{\text{NS}} = 1.65 \text{ \AA}$, $d_{\text{CrN}} = 2.0 \text{ \AA}$ and uniform isotropic atomic displacement parameters for the entire octahedron. The background terms were refined using a Chebyshev polynomial, and lattice parameters, rotation of the rigid octahedron, and isotropic displacement parameters were allowed to refine freely, along with crystallographic size and strain (anisotropically modelled using fourth-order spherical harmonics). The guest oxygen was added at a position located from the void map produced in Mercury, its occupancy fixed the occupancy determined from TGA (i.e. fully occupied), and then its position and isotropic atomic displacement parameter allowed to refine freely. The hydrogen atoms were not included as they could not be located. Without the guest water included, the refinement statistics are appreciable worse: $R_{wp} = 3.706$ instead of $R_{wp} = 3.162$. Refinement statistics: $R_{wp} = 3.162$, $R_{\text{Bragg}} = 4.216$, goodness of fit = 7.542.

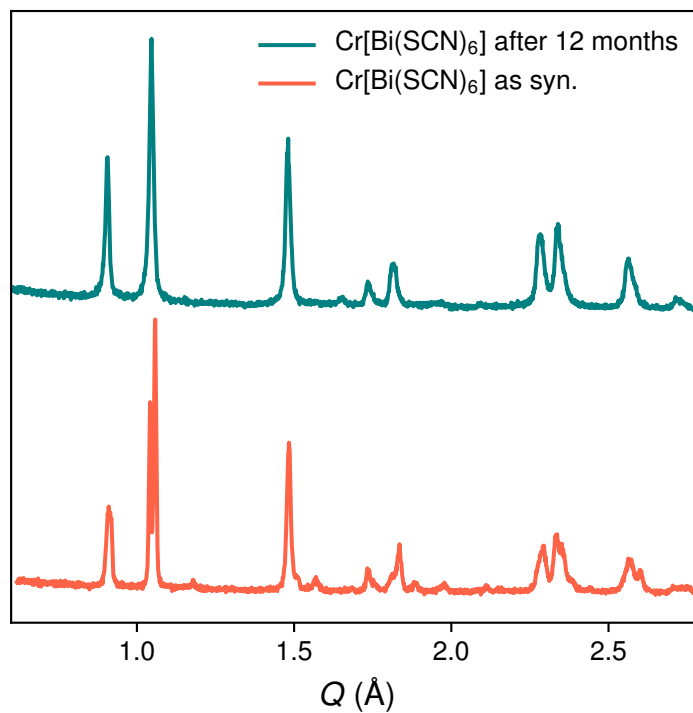
3.3 Sequential refinement of variable temperature data of Cr[Bi(SCN)₆]

Lattice parameters were extracted using Rietveld refinement of laboratory powder X-ray diffraction data. Due to the low resolution of the data sets and the acute pseudo symmetry, Pawley refinement proved unable to reliably distinguish the three crystallographic axes. We therefore resorted to Rietveld refinement, where the atomic positions were kept fixed of Fe[Bi(SCN)₆], to better deconvolute overlapping peaks. The background, strain, and lattice parameters were allowed to refine freely throughout the refinement.

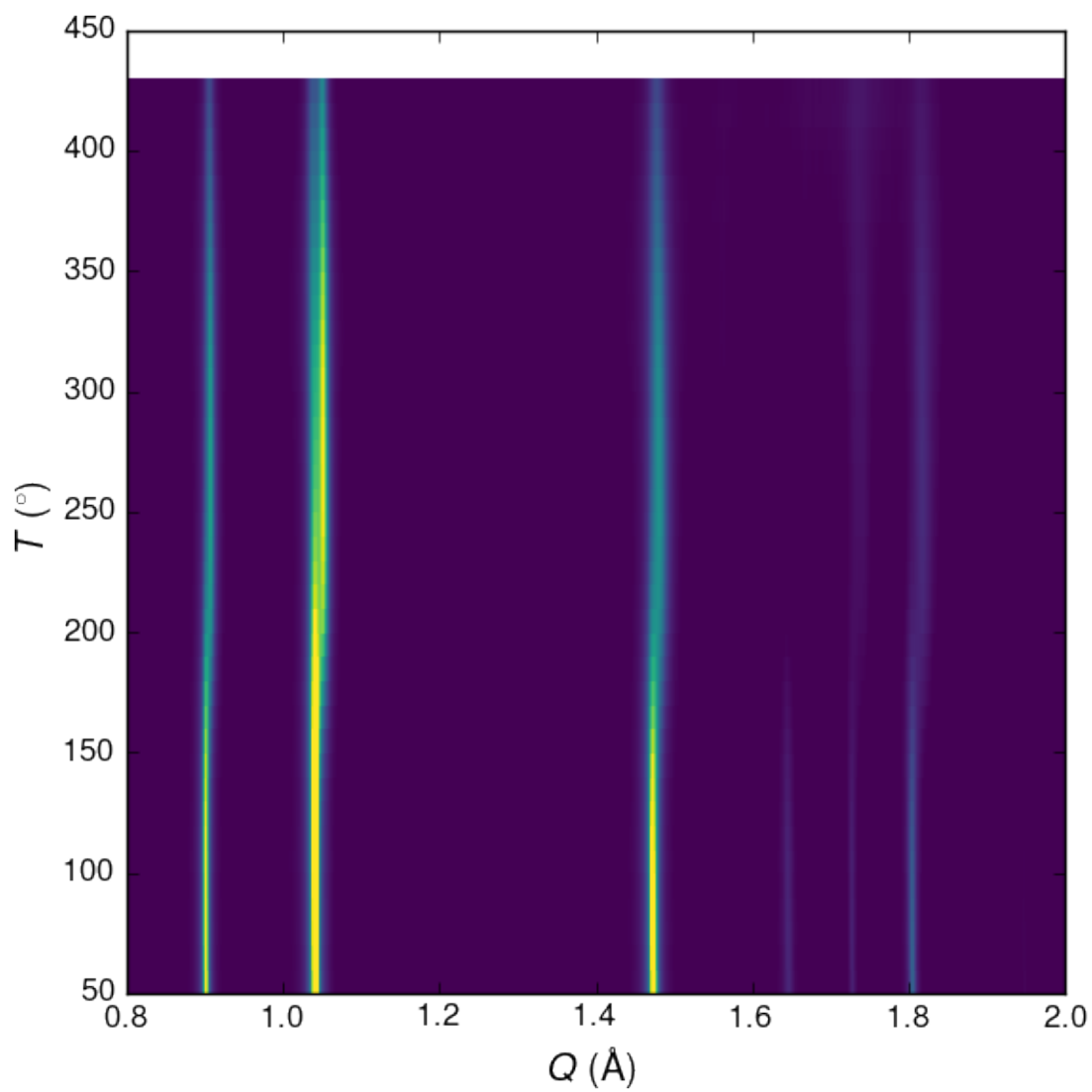
3.4 Sequential refinement of variable temperature data of Cr[Bi(SCN)₆] · H₂O

Lattice parameters were extracted using Pawley refinement of high resolution synchrotron powder X-ray diffraction data. Above 150 °C, the dehydration of the structure set in and the resulting combination of pseudosymmetry and phase mixtures prevented us from carrying out reliable quantitative fits of the data sets. The background, strain, and lattice parameters were allowed to refine freely throughout the refinement.

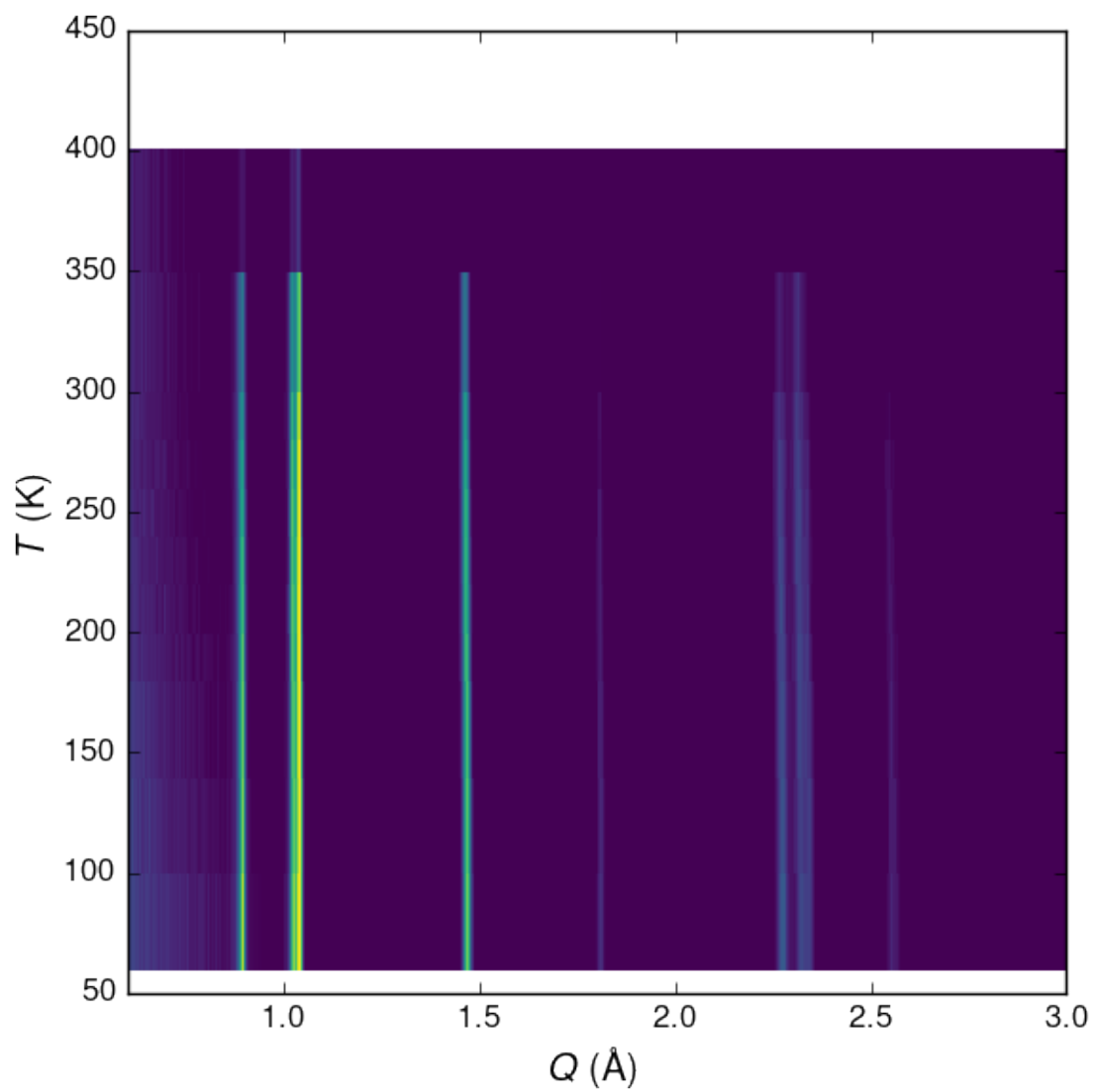
3.5 Powder diffraction studies of $\text{Cr}[\text{Bi}(\text{SCN})_6]$



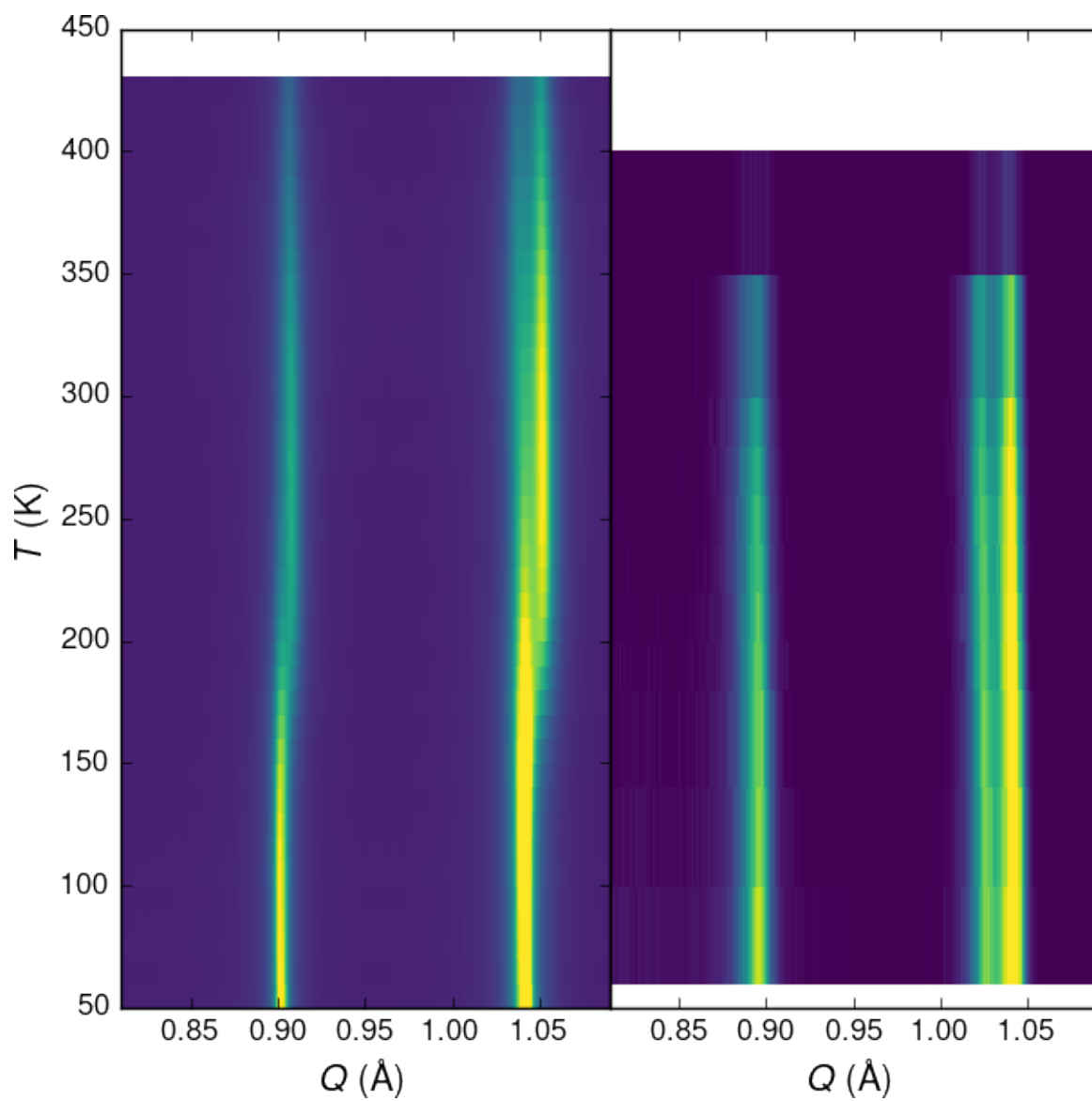
ESI Fig. 9: Hydration of anhydrous $\text{Cr}[\text{Bi}(\text{SCN})_6]$.



ESI Fig. 10: Heatmap representation of variable temperature powder X-ray diffraction measurements of hydrous $\text{Cr}[\text{Bi}(\text{SCN})_6] \cdot \text{H}_2\text{O}$.

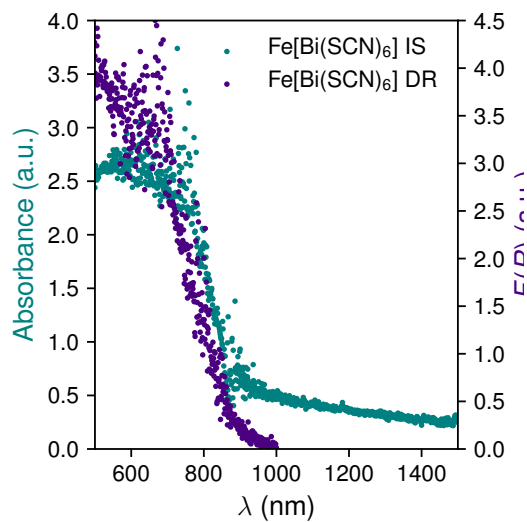


ESI Fig. 11: Heatmap representation of variable temperature powder X-ray diffraction measurements of anhydrous $\text{Cr}[\text{Bi}(\text{SCN})_6]$.

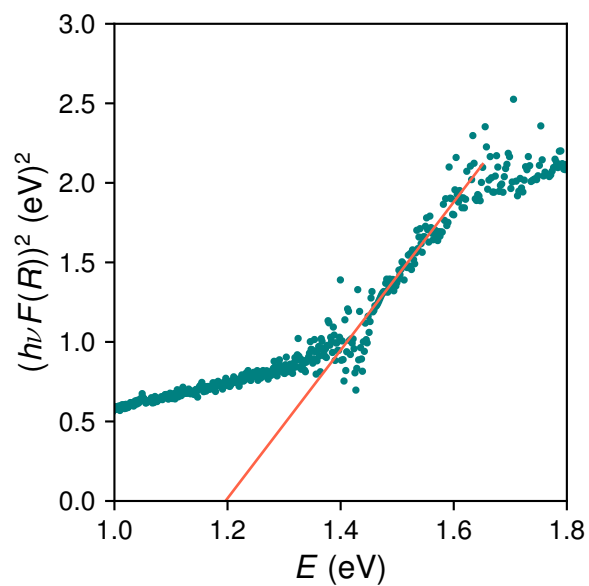


ESI Fig. 12: Side by side comparison of the heatmap representation of variable temperature powder X-ray diffraction measurements of anhydrous and hydrated $\text{Cr}[\text{Bi}(\text{SCN})_6]$ highlighting the low Q region.

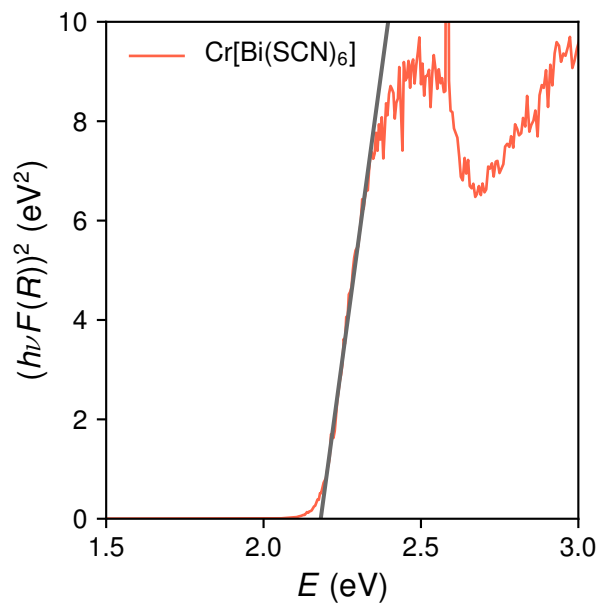
4 UV-Vis studies



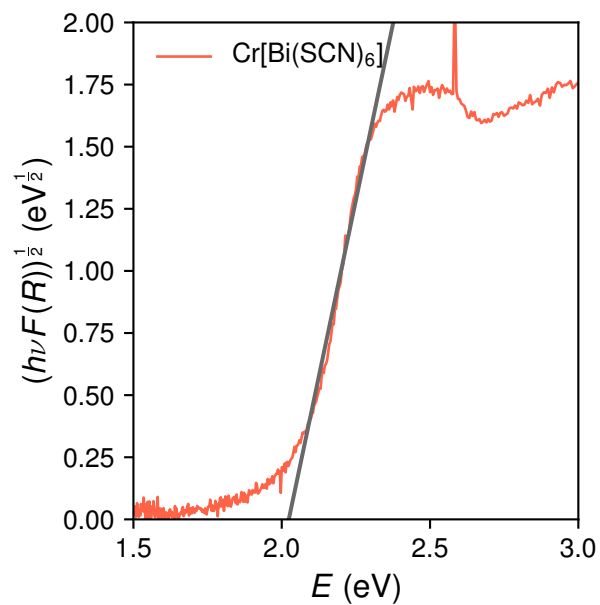
ESI Fig. 13: Comparison of powder diffuse reflectance (DR, purple, left) with integrating sphere absorption measurements carried out on a deposited film (IS, teal, right) for Fe[Bi(SCN)₆].



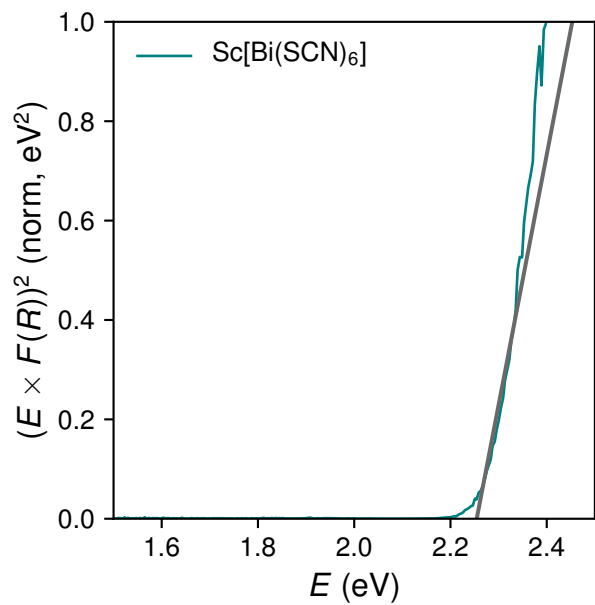
ESI Fig. 14: Calculation of the optical band gap for Fe[Bi(SCN)₆], using a Tauc plot assuming a direct band gap.



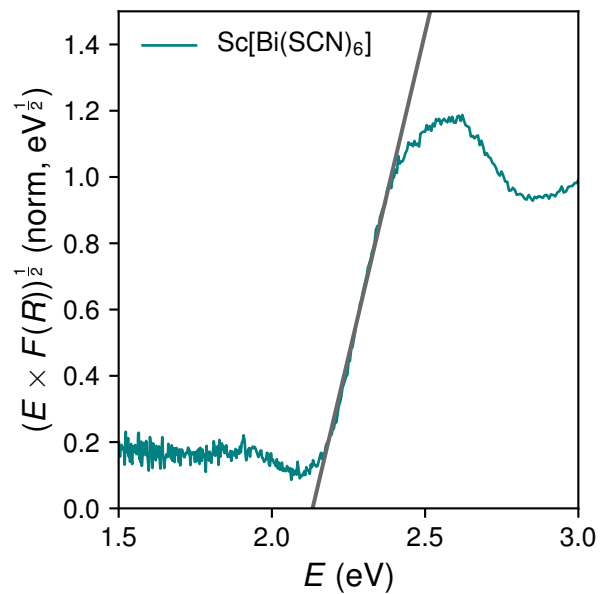
ESI Fig. 15: Calculation of the optical band gap for $\text{Cr}[\text{Bi}(\text{SCN})_6]$, using a Tauc plot assuming a direct band gap.



ESI Fig. 16: Calculation of the thermal band gap for $\text{Cr}[\text{Bi}(\text{SCN})_6]$, using a Tauc plot assuming an indirect band gap.

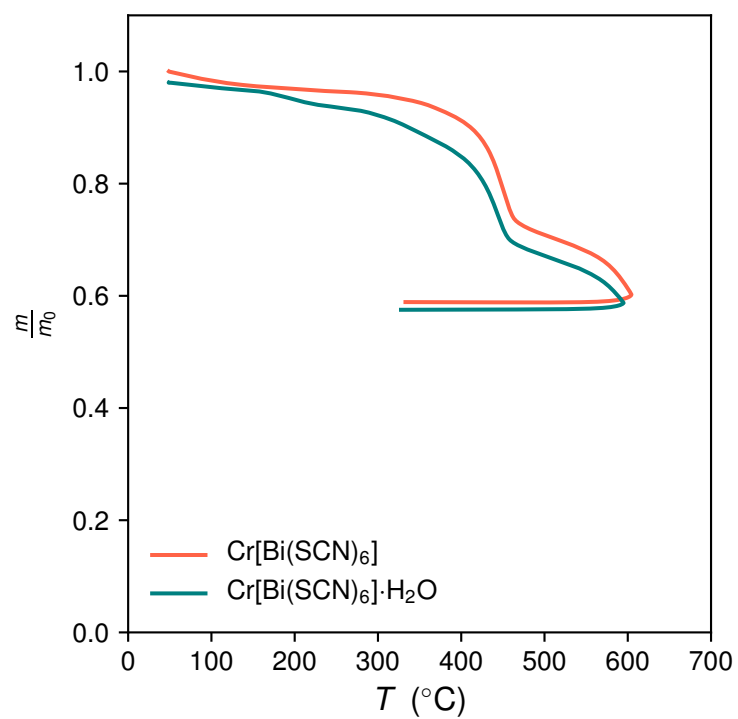


ESI Fig. 17: Calculation of the optical band gap for Sc[Bi(SCN)₆], using a Tauc plot assuming a direct band gap.

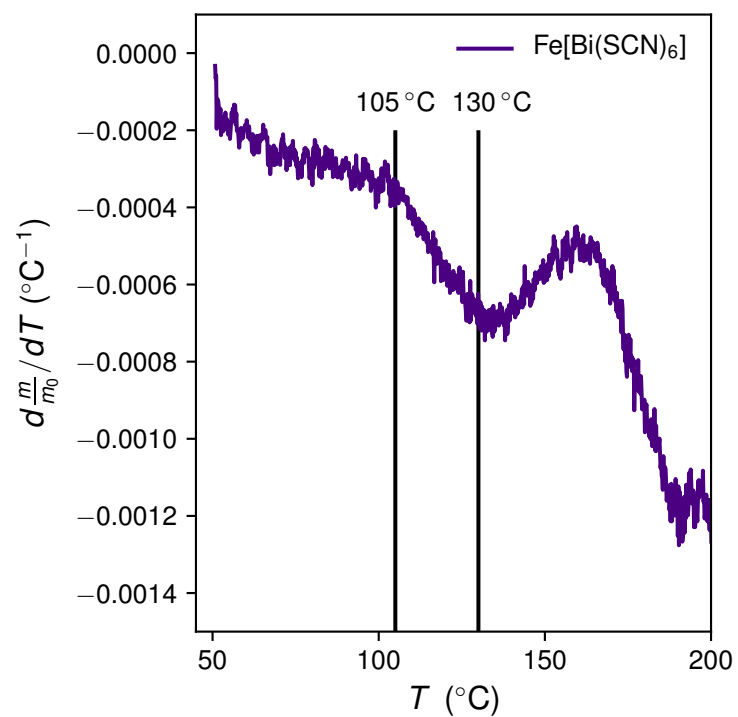
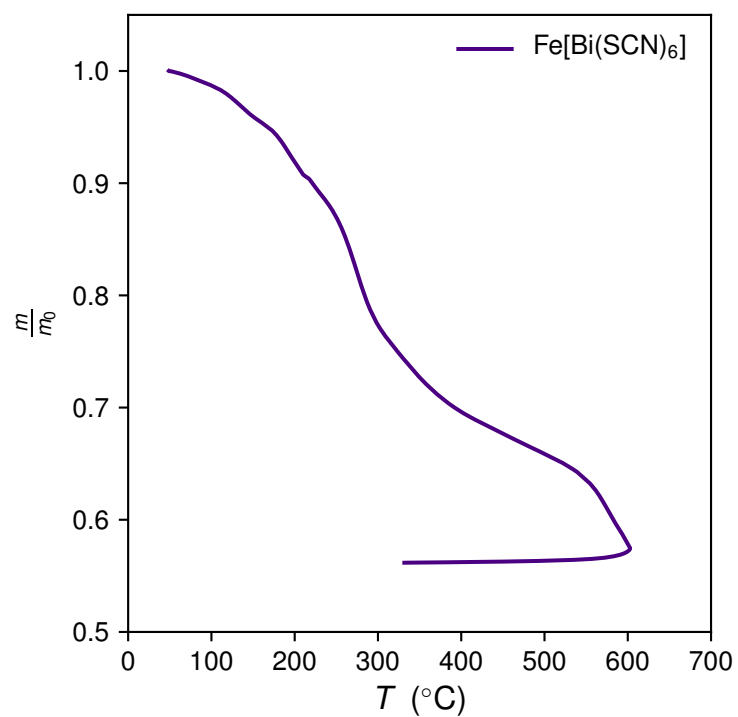


ESI Fig. 18: Calculation of the thermal band gap for Sc[Bi(SCN)₆], using a Tauc plot assuming an indirect band gap.

5 TGA



ESI Fig. 19: Full TGA mass loss curve for $\text{Cr}[\text{Bi}(\text{SCN})_6]$ and $\text{Cr}[\text{Bi}(\text{SCN})_6] \cdot \text{H}_2\text{O}$ measured under flowing N_2 .



ESI Fig. 20: Top: Full TGA mass loss curve for Fe[Bi(SCN)₆] measured under flowing N₂. Bottom: Temperature derivative of the TGA mass loss curve. The onset of the first mass loss (105 °) and the initial peak mass loss (130 °) are shown.

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- (S1) D. O. Scanlon, *et al.*, *Phys. Rev. B* **79**, 035101 (2009).
- (S2) M. Cococcioni, S. de Gironcoli, *Physical Review B* **71** (2005).