

## Supporting Information

# Sn-Doped Rutile TiO<sub>2</sub> Hollow Nanocrystals with Enhanced Lithium-Ion Batteries Performance

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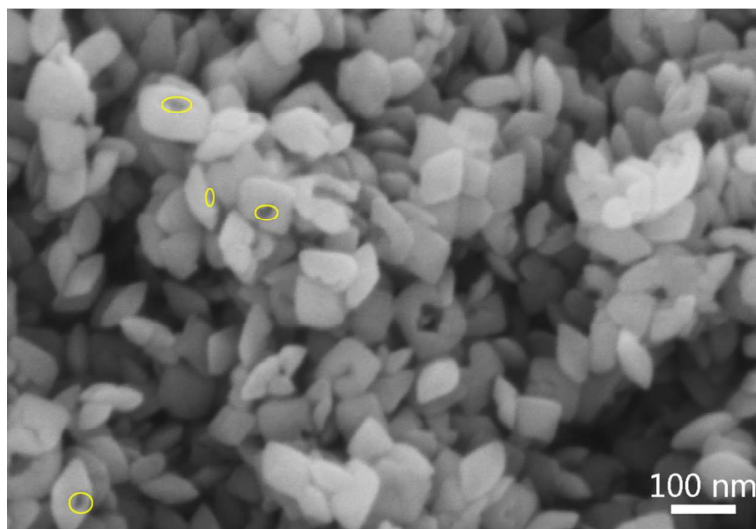
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### Supporting Information Contents

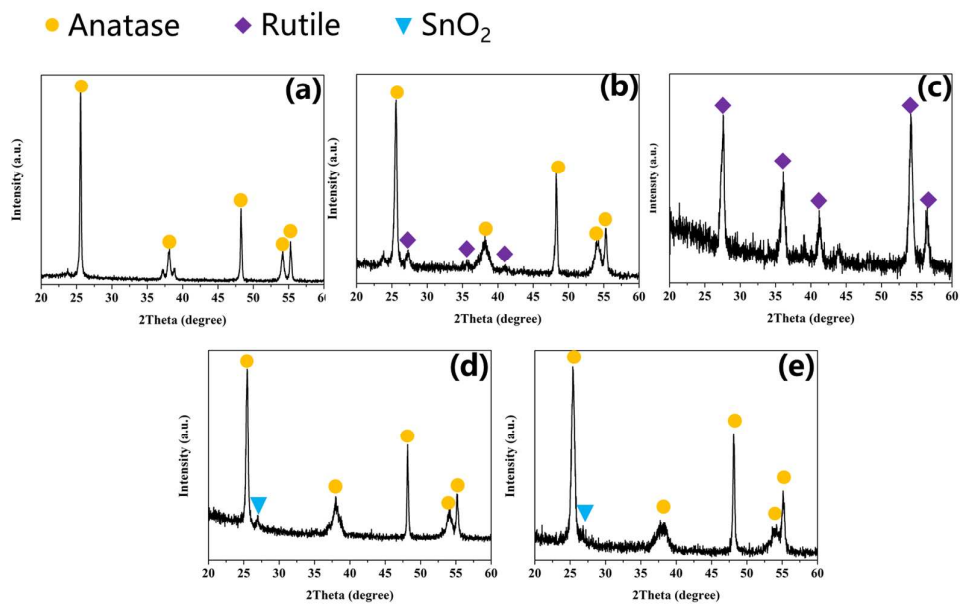
-Figures S1.-S14.

-Supporting Information Table S1. and Table S2.

-References

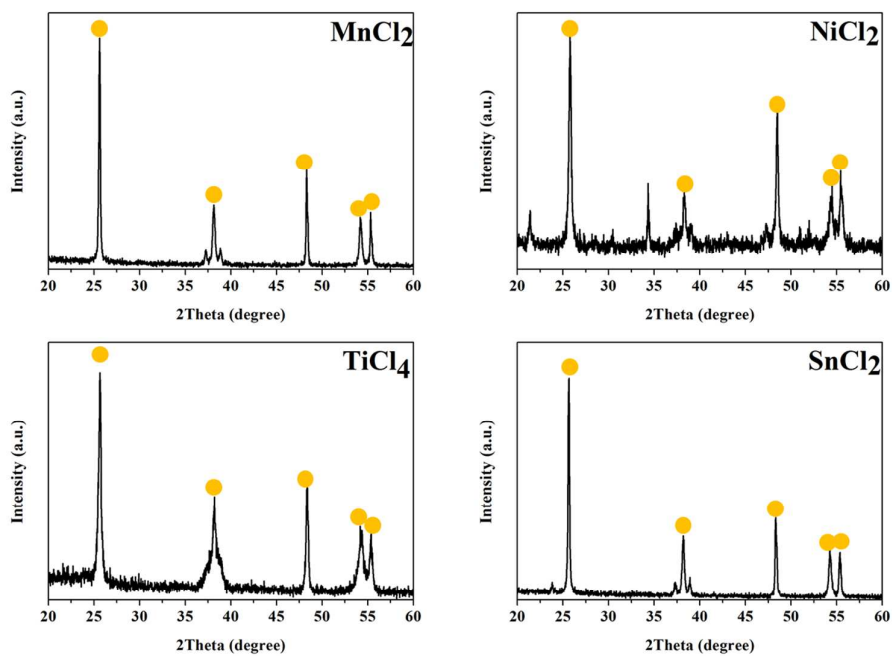


**Figure S1.** SEM image of the as-obtained Sn-doped rutile TiO<sub>2</sub> hollow nanocrystals.

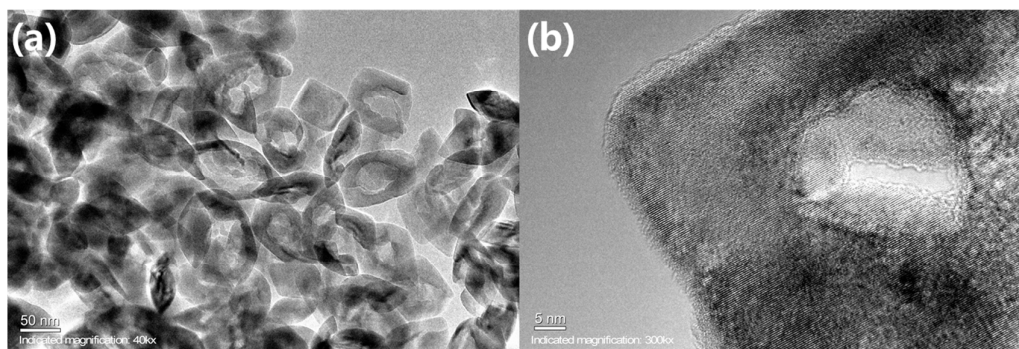


**Figure S2.** XRD patterns of the as-obtained TiO<sub>2</sub> synthesized with (a) 0 mmol, (b) 2 mmol, (c) 4 mmol, (d) 6 mmol and (e) 8 mmol SnCl<sub>4</sub>·5H<sub>2</sub>O.

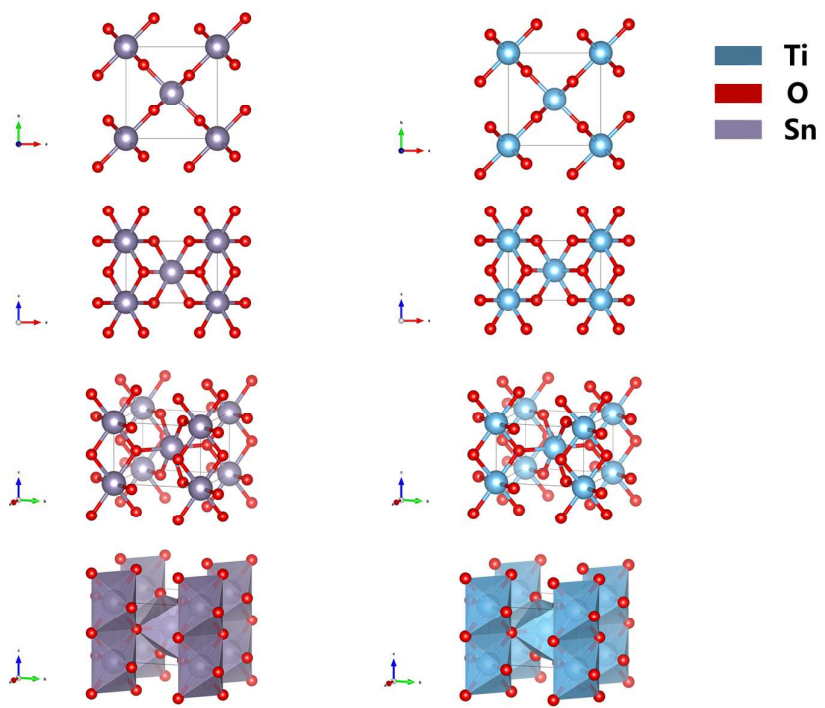
● Anatase TiO<sub>2</sub>



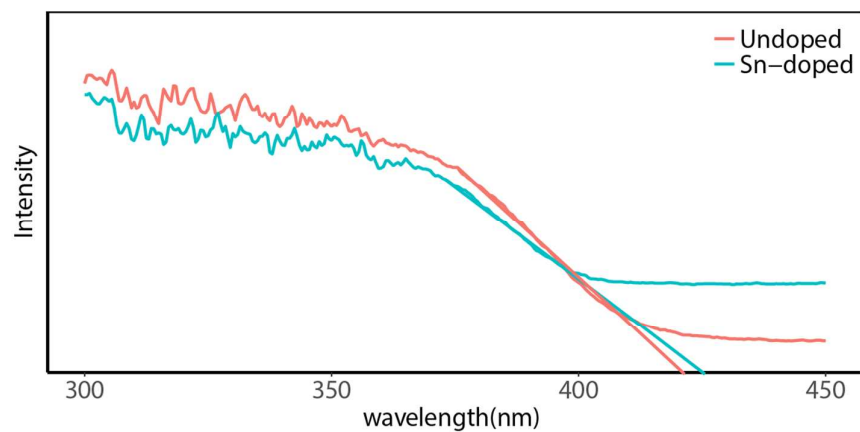
**Figure S1.** The XRD patterns of the products synthesized with other metal chlorides while keeping the molar amount of Cl<sup>-</sup> same with the SnCl<sub>4</sub> reaction system.



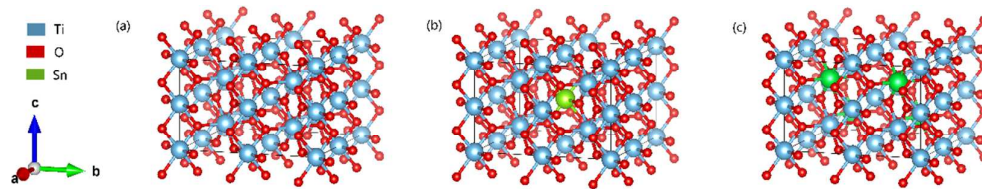
**Figure S4.** (a) TEM and (b) HRTEM images of the as-obtained Sn-doped rutile  $\text{TiO}_2$  hollow nanocrystals.



**Figure S5.** Schematic illustrations of SnO<sub>2</sub> and rutile TiO<sub>2</sub>.



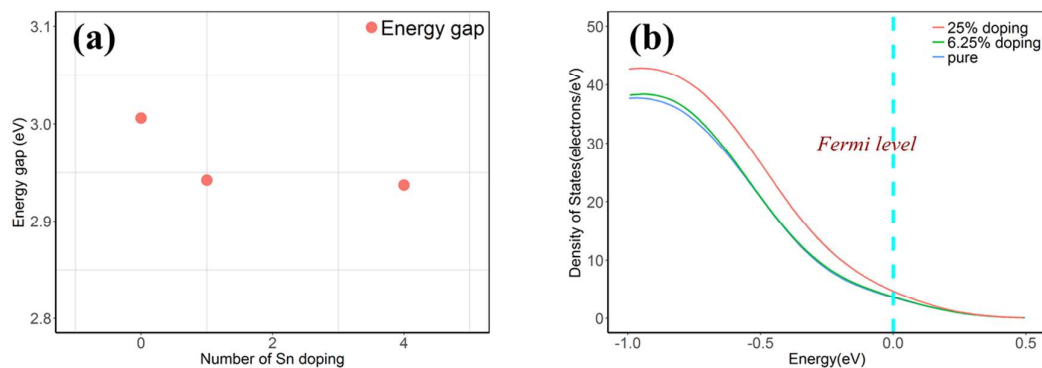
**Figure S6.** UV-vis spectra of the Sn-doped rutile TiO<sub>2</sub> hollow nanocrystals and the un-doped rutile TiO<sub>2</sub>. The calculated band gap of Sn-doped TiO<sub>2</sub> is ~2.9 eV, which is slightly lower than the pure rutile TiO<sub>2</sub>. DFT-based theoretical calculations shows that the doping of tin acts as the main reason for the decrease of the bandgap.



**Figure S7.** Schematic illustration of initial structures for the Sn-doped rutile TiO<sub>2</sub> with different doping amount. (a) pure rutile TiO<sub>2</sub>, (b) 6.25% Sn-doped rutile TiO<sub>2</sub>, (c) 25% Sn-doped rutile TiO<sub>2</sub>.

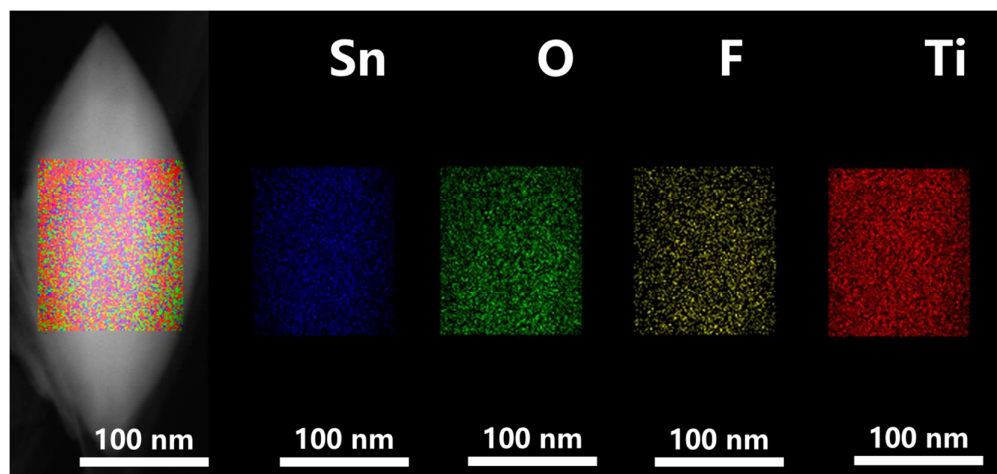
DFT-based (density functional theory) simulation was run using Dmol<sup>3</sup> package in the Materials Studio Software. A 2\*2\*2 supercell was created by replication of the elementary cell in the x, y, z directions. Generalized gradient approximation (GGA) functional proposed by Perdew–Burke–Ernzerhof<sup>1–3</sup>(PBE) was used. DSPP (DFT Semi-core Pseudopotential) was used to describe the electron and core interaction. A global cutoff of 5 Å was used which is the same setting for all the models.



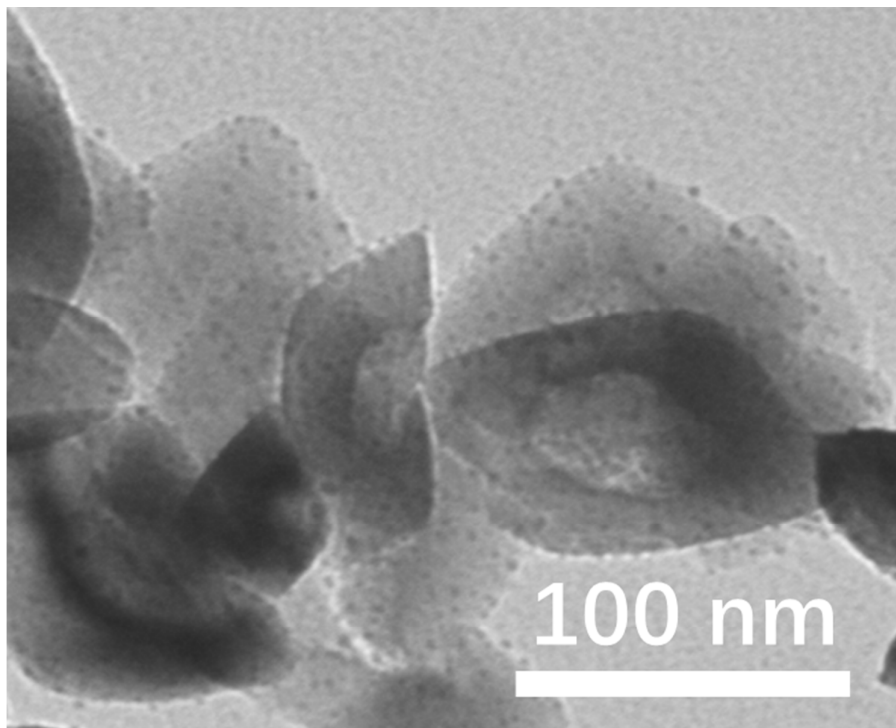


**Figure S8.** (a) Relationship between the energy gap and the number of Sn doping. (b) DOS spectra of the three different systems.

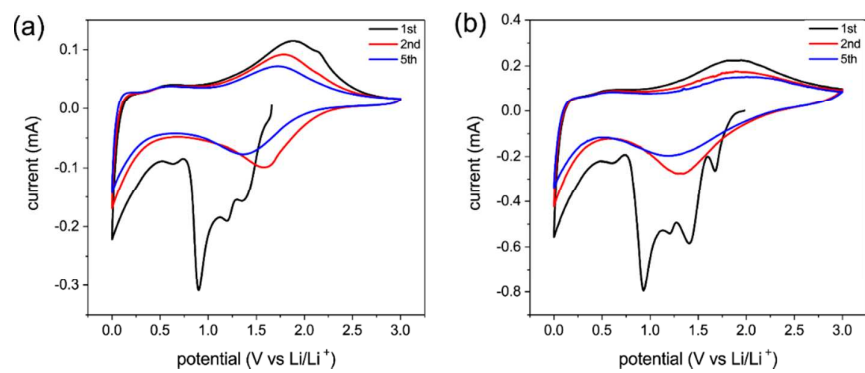
The calculated bandgaps were 3.006 eV for pure rutile  $\text{TiO}_2$ , 2.942 eV for 6.25% Sn-doped rutile  $\text{TiO}_2$ , and 2.937 eV for 25% Sn-doped rutile  $\text{TiO}_2$ . The results indicate that the bandgap narrows with increasing the amount of Sn in rutile  $\text{TiO}_2$ . Further analysis of the electron states distribution shows that the density of states near the Fermi level increases with the amount of Sn, which is beneficial for the photo-related catalysis or energy transformation process.



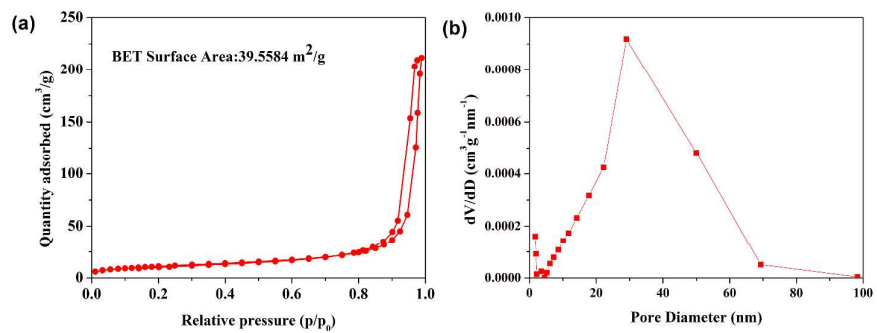
**Figure S9** STEM image of single Sn-doped TiOF<sub>2</sub> nanocrystal and EDS mappings of Sn, O, F and Ti elements.



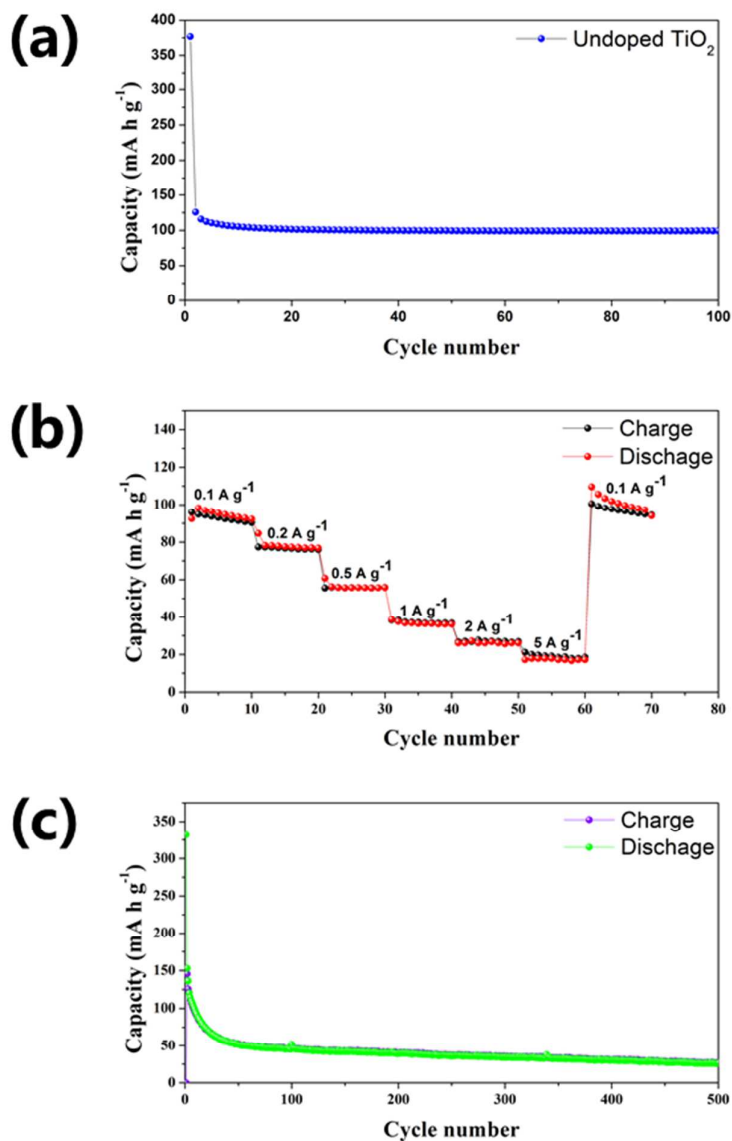
**Figure S10.** TEM image of the rutile Sn-doped  $\text{TiO}_2$  nanocrystals on the surfaces of solid Sn-doped  $\text{TiOF}_2$ .



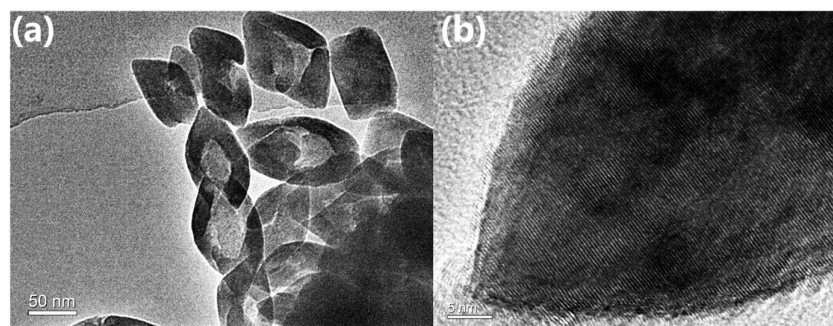
**Figure S11.** CV profiles of (a) un-annealed and (b) annealed Sn-doped rutile  $\text{TiO}_2$  hollow nanocrystals showing the 1st, 2nd and 5th cycles between 0.01 and 3.0 V at a scan rate of  $0.2 \text{ mV s}^{-1}$ .



**Figure S12** (a) The nitrogen adsorption/desorption isotherms and (b) the pore size distribution curves of the Sn-doped rutile TiO<sub>2</sub> nanocrystals. The BET specific surface area and BJH pore volume are 39.5584 m<sup>2</sup>/g and 0.32852 cm<sup>3</sup>/g .



**Figure S13.** (a) Cycle performance at 100 mA g<sup>-1</sup> in the 0.01-3 V range of the un-doped TiO<sub>2</sub>; (b) Rate capabilities of the un-doped TiO<sub>2</sub> at a current density ranging from 100-5000 mA h g<sup>-1</sup>; (c) The cycling performance of annealed Sn-doped rutile TiO<sub>2</sub> at the rate of 1000 mA h g<sup>-1</sup>.



**Figure S14** TEM (a) and HRTEM (b) images of the Sn-doped rutile TiO<sub>2</sub> after cycle performance test (1 A g<sup>-1</sup>, 100 cycles).

**Table S1.** Comparison of the lattice parameters of rutile SnO<sub>2</sub> and TiO<sub>2</sub>.

<b>Material</b>	<b>Rutile SnO<sub>2</sub></b>	<b>Rutile TiO<sub>2</sub></b>
a (Å)	4.73730	4.59400
b (Å)	4.73730	4.59400
c (Å)	3.18640	2.95900
$\alpha$	90	90
$\beta$	90	90
$\gamma$	90	90
Space Group	P42/mnm	P42/mnm



**Table S2.** Comparison of the electrochemical performances of the as-prepared Sn-doped rutile TiO<sub>2</sub> with the reported performances of rutile TiO<sub>2</sub> as anode materials in the LIBs.

Materials	Current rate	Reversible capacity (mA h g <sup>-1</sup> )	Current rate	Reversible capacity (mA h g <sup>-1</sup> )
TiO <sub>2</sub> submicroboxes <sup>4</sup>	30 C	68	1C	210
Dark TiO <sub>2</sub> nanorod <sup>5</sup>	50 C	93.6	1 C	166.7
TiO <sub>2</sub> mesocrystals <sup>6</sup>	5 C	100	1 C	171.3
Nanosized TiO <sub>2</sub> <sup>7</sup>	30 C	70	1 C	132
TiO <sub>2</sub> microspheres <sup>8</sup>	10 C	106	1 C	160.4
B-doped TiO <sub>2</sub> <sup>9</sup>	10 C	72	1 C	154
<b>Sn-doped TiO<sub>2</sub> structure (This work)</b>	<b>rutile hollow (This work)</b>	<b>5000 mA g<sup>-1</sup> (30C)</b>	<b>110</b>	<b>100 mA g<sup>-1</sup> (~0.6 C)</b>

## References

- (1) Perdew, J. P.; Wang, Y.; Pair-distribution function and its coupling-constant average for the spin-polarized electron gas. *Phys. Rev. B* **1992**, *46* (20), 947–954.
- (2) Perdew, J. P. Density-Functional Approximation for the Correlation Energy of the Inhomogeneous Electron Gas. *Phys. Rev. B* **1986**, *33* (12), 8822–8824.
- (3) Perdew, J. P.; Burke, K.; Ernzerhof, M. Generalized Gradient Approximation Made Simple. *Phys. Rev. Lett.* **1996**, *77* (18), 3865–3868.
- (4) Yu, X.-Y.; Wu, H. Bin; Yu, L.; Ma, F.-X.; Lou, X. W. D. Rutile TiO<sub>2</sub> Submicroboxes with Superior Lithium Storage Properties. *Angew. Chemie Int. Ed.* **2015**, *54* (13), 4001–4004.
- (5) Chen, J.; Song, W.; Hou, H.; Zhang, Y.; Jing, M.; Jia, X.; Ji, X. Ti<sup>3+</sup> Self-Doped Dark Rutile TiO<sub>2</sub> Ultrafine Nanorods with Durable High-Rate Capability for Lithium-Ion Batteries. *Adv. Funct. Mater.* **2015**, 6793–6801.
- (6) Hong, Z.; Wei, M.; Lan, T.; Jiang, L.; Cao, G. Additive-Free Synthesis of Unique TiO<sub>2</sub> Mesocrystals with Enhanced Lithium-Ion Intercalation Properties. *Energy Environ. Sci.* **2012**, *5* (1), 5408.
- (7) Hu, Y. S.; Kienle, L.; Guo, Y. G.; Maier, J. High Lithium Electroactivity of Nanometer-Sized Rutile TiO<sub>2</sub>. *Adv. Mater.* **2006**, *18* (11), 1421–1426.

- (8) Lan, T.; Liu, Y.; Dou, J.; Hong, Z.; Wei, M. Hierarchically Porous TiO<sub>2</sub> Microspheres as a High Performance Anode for Lithium-Ion Batteries. *J. Mater. Chem. A* **2014**, 2 (4), 1102–1106.
- (9) Tian, H.; Xin, F.; Tan, X.; Han, W. High Lithium Electroactivity of Boron-Doped Hierarchical Rutile Submicrosphere TiO<sub>2</sub>. *J. Mater. Chem. A* **2014**, 2 (27), 10599–10606.