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# Supplemental Information

# Quasi-Metal for Highly Sensitive and Stable

### Surface-Enhanced Raman Scattering

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#### **Supplemental Information**

#### **1. Transparent Methods**

### **Synthesis ofb-VO<sup>2</sup> nanosheets**

All chemicals used in the experiments are of analytical purity. In a typical synthesis, 0.1 g of vanadyl acetylacetonate (VAA) is dissolved in 30 mL of absolute ethanol to form a bright blue transparent solution. The resulting homogeneous solution is then transferred to a Teflon-lined high-pressure reactor, and a common glass sheet is placed above the liquid level as a growth platform for  $b$ -VO<sub>2</sub> nanosheets. The Teflon-lined stainless steel autoclave was sealed and heated for 20 h at 200  $^{\circ}$ C. After the reaction is completed, the glass sheet covered with black products was taken out. Finally, the black glass was washed with ethanol and distilled water for three times and dried at 50 °C in a vacuum drying oven.

#### **Synthesis of SiO<sub>2</sub>/VO<sub>2</sub>**

Open a bottle of analytically pure silicon tetrachloride (SiCl4) in air. Soon, a lot of "white-fog" formed at the mouth of the bottle. Place the glass sheet covered with  $VO<sub>2</sub>$ nanosheets in a white mist at the mouth of the bottle for 3 min. Then, the glass sheet is placed in a nitrogen-protected tubular furnace and heated to 400 °C for 1 hours. After natural cooling, take out for use.

#### **Synthesis of urchin-like**  $W_{18}O_{49}$  **nanostructures**

In a typical procedure, 3 g of  $WCl_6$  was dissolved in 100 mL of ethanol, and the obtained yellow solution was magnetically stirred for 20 min, and then transferred to a Teflon-lined stainless-steel autoclave and heated at 200 °C for 20 h with a heating rate of 4 °C/min. The autoclave was cool down naturally and a blue product was collected, washed, and dried in a vacuum drying oven at 50 °C for 4 h.

### **Characterization**

These samples were measured by a variety of characterization techniques. XRD patterns of the products were obtained on a Bruker D8 focus X-ray diffractometer by using CuKa radiation ( $\lambda = 1.54178$  Å). SEM images and EDS were obtained on a Hitachi S-4800. TEM and HRTEM characterizations were performed with a Tecnai G F30 operated at 300 kV. Ultraviolet–Vis (UV) absorption spectra were recorded with a Shimadzu UV3600. XPS were recorded on an ESCALab-250Xi of ThermoFisher Scientific. The Fourier transform infrared spectra were measured from THERMO Iz-10. The specific surface area was measured in a Micro Tristar II 3020. The work functions of b-VO<sub>2</sub> nanosheets was detected from Kelvin probe force microscope, Multimode-Picoforce-Veeco. ESR spectrum was obtained from ESP-300.

#### **Raman tests**

To study the SERS of these  $b$ -VO<sub>2</sub> nanosheets, a confocal micro Raman spectrometer (Renishaw-inVia Reflex) is used as the measuring instrument. In all SERS tests, unless specifically stated, the excitation wavelength is 532 nm, laser power is 0.5 mW and the specification of the objective is  $\times$  50 L. A series of standard solution (aqueous) of highly risk chemical with concentrations of  $10^{-7}$ - $10^{-10}$  M were used as the probe molecules. To improve the signal reproducibility and uniformity, before each test, the glass sheet (1 cm  $\times$  1 cm) covered with b-VO<sub>2</sub> nanosheets were immersed into a 20 mL of probe solution with specific concentration for 20 min, then taken out and dried

in air for 20 min. In all SERS tests, the laser beam is perpendicular to the top of the sample to be tested with a resultant beam spot diameter of  $5 \mu m$ . The calculation of EF are provided in Supplementary Methods.

### **Enhanced Factor Calculation**

To calculate the EF of the  $b$ -VO<sub>2</sub> nanosheets, the ratio of SERS to normal Raman spectra (NRS) of RH6G was determined by using the following calculating formula 1

$$
EF = (I_{SERS}/I_{NRS}) \times (C_{NRS}/C_{SERS})
$$
 (1)

where I<sub>SERS</sub> and I<sub>NRS</sub> refer to the peak intensities of the SERS and NRS, respectively. C<sub>NRS</sub> and C<sub>SERS</sub> refer to the concentrations of the probe molecules of the NRS and SERS, respectively. In the SERS measurements, two Raman scattering peaks,  $R_1$  at 612 cm<sup>-1</sup> and  $R_2$  at 773 cm<sup>-1</sup> were selected for the calculations of the EF. For comparison, the peak intensities of the R6G  $(1 \times 10^{-2} \text{ M})$ , aqueous solution) directly placed on bare glass slide were detected as NRS data. For the NRS data, the integration time is 4000 s, while for the SERS data, the integration time is 10 s.

#### **Calculations details**

All the density functional theory (DFT) calculations were carried out using the Vienna Ab initio Simulation Package (VASP)<sup>1-3</sup>. The Perdew-Burke-Ernzerhof (PBE) was used to describe the exchange-correlation 4 . The optimization process was performed using a conjugate gradient algorithm with a force tolerance of 0.02 eV/Å, with kinetic energy cutoff set as 500 eV. A Gamma k-point sampling of  $3 \times 7 \times 5$  was used to optimize the VO<sup>2</sup> cell parameters. The optimized lattice parameters were 11.99, 3.74, 6.40 Å, and 90.0, 107.0, 90.0°, respectively.

To calculate the VO<sub>2</sub> (112) surface, a  $2 \times 2 \times 1$  supercell was built by cut the original cell along (112) facet, with a 20 Å additional vacuum layer in z-direction. Thus, the lattice parameter of the layer slab model was 24.82 Å  $\times$  19.68 Å  $\times$  26.97 Å.<br>A 2D-shaped graphene layer was calculated to adsorb RH6G molecule as well. containing 200 carbon atoms, with the supercell parameters equal to 24.47 Å  $\times$  24.67 Å  $\times$  20 Å and 90, 90, 120 $^{\circ}$ , respectively. The structure of RH6G molecule was optimized in the same cell as well. For these simulations, the plane wave cutoff energy set as 500 eV and the k-point mesh set as  $1 \times 1 \times 1$ .

1. Kresse, G., and Hafner, J. (1994). Ab initio molecular-dynamics simulation of the liquid-metal–amorphous-semiconductor transition in germanium. Phys. Rev. B *49*, 14251-14269 .

2. Kresse, G., and Furthmüller, J. (1996). Efficiency of ab-initio total energy calculations for metals and semiconductors using a plane-wave basis set. J. Comp. Mater. Sci. *6*, 15-50.

3. Kresse, G., and Furthmüller, (1996). J. Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set. Phys. Rev. B *54*, 11169-11186.

4. Perdew, J. P., Burke, K., and Ernzerhof, M. (1996). Generalized gradient approximation made simple. Phys. Rev. Lett. *77*, 3865-3868 .



# **2. Supporting Figures**

**Figure S1.** SEM image (a) and XRD pattern (b) of the V<sub>2</sub>O<sub>5</sub> nanobelts obtained when

ethanol is replaced by deionized water. Related to Figure 1.



**Figure S2.** N<sub>2</sub> adsorption/desorption isotherms of the as-synthesized b-VO<sub>2</sub> nanosheets, related to Figure 1.



Figure S3. FTIR spectrum of the as-synthesized b-VO<sub>2</sub> nanosheets, related to Figure

1.



**Figure S4.** Raman spectrum of the b-VO<sub>2</sub> nanosheets, related to Figure 1.



**Figure S5.** TEM image of the  $V_2O_5$  nanosheets prepared by heating the b-VO<sub>2</sub> nanosheets at 400 °C, related to Figure 2.



**Figure S6.** XRD pattern (a) and SEM image (b) of the prepared urchin-like  $W_{18}O_{49}$ nanostructures. Related to Figure 2.



**Figure S7.** The comparison of oxidation resistance between b-VO<sub>2</sub> and W<sub>18</sub>O<sub>49</sub>. The change of color and absorbance before and after heating from two kinds of samples clearly indicated that the oxidation resistance of  $W_{18}O_{49}$  is much lower than that of b-VO2. Related to Figure 2.



Figure S8. The differential thermal analysis (DTA) curve of the b-VO<sub>2</sub> nanosheets. Below 350  $\degree$ C, the b-VO<sub>2</sub> nanosheets are stable. When the temperature continues to rise, they were gradually oxidized to  $V_2O_5$ . Related to Figure 2.



**Figure S9.** A series of XRD patterns demonstrated that no detectable change in the  $XRD$  patterns of  $b-VO<sub>2</sub>$  nanosheets after the heating, irradiating, and long-term storage. Related to Figure 2.



**Figure S10.** A series of XPS spectra demonstrated that no detectable change in the valence state of the  $b$ -VO<sub>2</sub> nanosheets after the heating, irradiation, and long-term storage. The XPS and XRD (Figure S6) results demonstrate that the stability of these b-VO<sup>2</sup> nanosheets is extraordinary high. Related to Figure 2.



**Figure S11.** The standard Raman spectrum of R6G reference material, related to Figure 3.



Figure S12. The bare b-VO<sub>2</sub> nanosheet arrays without R6G solution only showed the typical Raman scattering peaks of themself, related to Figure 3.



**Figure 13.** SERS spectra of the dye molecules of MO (a) and RhB (b). Related to Figure 3.



**Figure S14.** SERS spectra of 10<sup>-10</sup> M R6G on b-VO<sub>2</sub> nanosheets and graphene, respectively, related to Figure 6.



Figure S15. The measured surface potential difference profiles. Considering that the work function of Au reference is 4.8 eV, the work function of  $b$ -VO<sub>2</sub> nanosheets is estimated to be 4.55 eV. The b-VO<sub>2</sub> thickness dependence on its work function can be neglected. Related to Figure 6.



Figure S16. When these b-VO<sub>2</sub> nanosheets were heated in air for a period of time, their colors have changed dramatically. Related to Figure 6.



**Figure S17.** The XPS spectrum of the sample after heating at 400 °C for 1h in air, which suggests that the  $VO<sub>2</sub>$  has been transformed into  $V<sub>2</sub>O<sub>5</sub>$ . Related to Figure 6.



**Figure S18.** The UV-Vis absorption of the sample after heating at 400 °C for 1h in air, related to Figure 6.



Figure 19. The UV-Vis absorption of the VO<sub>2</sub> samples reduced by NaBH<sub>4</sub> aqueous solution, related to Figure 6.



Figure S20. The obtained SERS signals from the reduced VO<sub>2</sub> nanosheet substrates, related to Figure 6.



**Figure** 21. (a) Schematic illustrating the synthesis of the  $SiO_2/VO_2$ . SiCl<sub>4</sub> is easily hydrolyzed into  $H_2SiO_3$  in air. These formed  $H_2SiO_3$  species were coated on  $VO_2$ nanosheets and formed H<sub>2</sub>SiO<sub>3</sub>/VO<sub>2</sub>. The VO<sub>2</sub> nanosheets coated with amorphous  $SiO<sub>2</sub>$  layers were obtained by heating under N<sub>2</sub> protection. (b) The HRTEM image of the obtained  $SiO_2/VO_2$ , revealing the thickness of  $SiO_2$  layer is about 2 nm. Related to Figure 6.



Figure S22. (a) The obtained SERS signals from VO<sub>2</sub> nanosheet substrate and SiO2/VO<sup>2</sup> substrate, respectively. Related to Figure 6.

Substrate	Probe	Excited	Author	$\rm EF$	Detectable	Stability
	molecule	wavelength			Limit	
		(nm)			(M)	
TiO <sub>2</sub>	<b>MB</b>	532	D. Qi et al.,	$2\times10^4$	$10^{-5}$	stable
			2014			
	4-Mpy	514.5		10 <sup>4</sup>	$10^{-3}$	Liable to
CdTe			Y. F. Wang			oxidation
			et al., 2007			and
						corrosion
ZnO	D <sub>266</sub>	488	H. Wen et	50	$10^{-5}$	Liable to
			al., 1996			corrosion
CdS	4-Mpy	514.5		10 <sup>2</sup>	$10^{-3}$	Liable to
			Y. F. Wang			oxidation
			et al., 2008			and
						corrosion
$\alpha$ -Fe <sub>2</sub> O <sub>3</sub>	4-Mpy	514.5	X. Q. Fu et	$2.7 \times 10^{4}$	$10^{-3}$	Liable to
			al., 2009			corrosion
	4-MBA	488				Liable to
Cu <sub>2</sub> O			L. Jiang et	10 <sup>5</sup>	$10^{-3}$	oxidation
			al., 2013			and
						corrosion
CuO	4-Mpy	514.5	Y. Wang et	$10^2$	$10^{-1}$	Liable to
			al., 2007			corrosion
$W_{18}O_{49}$	R <sub>6</sub> G	532.8	S. Cong et	$3.4 \times 10^{5}$	$10^{-7}$	Liable to
			al., 2015			oxidation
Cu <sub>2</sub> O	R6G	532			$10^{-9}$	Liable to
			L. Guo et	$8\times10^5$		oxidation
			al., 2017			and
						corrosion
ZnO	4-MBA	633	Wang, X. T.	$6.6 \times 10^{5}$		Liable to
			et al., 2017			corrosion
MoS <sub>2</sub>	R6G	532.8	Zheng, Z.	$1.6 \times 10^{5}$	$10^{-7}$	Liable to
			H. et al.,			oxidation
			2017			
						Liable to

**Table S1: Some of the previously reported EFs for Non-Noble Metal Enhanced Raman** Substrate materials, related to Figure 3

<b>MOF</b>	R6G	532.8	Sun, H.Z.	10 <sup>6</sup>	$10^{-8}$	oxidation
			et al., 2019			and
						corrosion
Organic						Liable to
Semiconductor	DFH-4T	532	Yilmaz M. et al., 2017	$3.4 \times 10^3$	$10^{-5}$	oxidation
						and
						corrosion
Nb <sub>2</sub> O <sub>5</sub>	MB	532	Shan, Y. F.	$7.1 \times 10^6$	$10^{-9}$	stable
			et al., 2017			
WTe <sub>2</sub> /WTe <sub>2</sub>	R6G	532	Tao, L. et	$6.2 \times 10^{9}$	$10^{-15}$	Liable to
			al., 2018			oxidation
						and
						corrosion
MoO <sub>2</sub>	R6G	532	Zhang, Q. et	$3.75 \times 10^{6}$	$10^{-7}$	stable
			al., 2017			
$b-VO2$	R <sub>6</sub> G	532	Tian, Z. et	$6.7 \times 10^{7}$	$10^{-10}$	
			al (this			stable
			work)			

Qi, D., Lu, L., Wang, L., and Zhang, J. (2014). Improved SERS sensitivity on plasmon-free TiO<sup>2</sup> photonic microarray by enhancing light-matter coupling. J. Am. Chem. Soc*. 136*, 9886–9889.

Wang, Y. F., Suna, Z. H., Wanga, Y. X., Hua, H. L., Zhaoa, B., Xua, W. Q., and Lombardi, G. R. (2007). Surface-enhanced Raman scattering on mercaptopyridine-capped CdS microclusters. Spectrochim. Acta, Part A. *66*, 1199–1203.

Wen, H., He, T. J., Xu, C. Y., Zuo, J., and Liu, F. C. (1996). Surface enhancement of Raman and absorption spectra from cyanine dye D266 adsorbed on ZnO colloids. Molecular Physics. *88*, 281–290.

Wang, Y. F., Zhang, J. H., Jia, H. Y., Li, M. J., Zeng, J. B., Yang, B., Zhao, B., and Xu, W. Q. (2008). Mercaptopyridine surface-functionalized CdTe quantum dots with enhanced Raman scattering properties. J. Phys. Chem. C. *112*, 996–1000.

Fu, X. Q., Bei, F. L., Wang, X., Yang, X. J., and Lu, L. D. (2009). Surface-enhanced Raman scattering of 4-mercaptopyridine on sub-monolayers of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanocrystals (sphere, spindle, cube). J. Raman Spectrosc*. 40*, 1290–1295.

Jiang, L.,You, T. T., Yin, P. J., Shang, Y., Zhang, D. F., Guo, L., and Yang, S. (2013). Surface-enhanced Raman scattering spectra of adsorbates on Cu<sub>2</sub>O nanospheres: charge-transfer and electromagnetic enhancement. Nanoscale. *5*, 2784–2789.

Wang, Y., Hu, H. L., Jing, S. Y., Wang, Y. X., Sun, Z. H., Zhao, B., Zhao, C., and Lombardi, J. R. (2007). Enhanced Raman scattering as a probe for 4-mercaptopyridinesurface-modified copper oxide nanocrystals. Anal. Sci. *23*, 787–791.

Cong, S., Yuan1, Y. Y., Chen, Z. G., Hou, J. Y., Yang, M., Su, Y. L., Zhang, Y. Y., Li, L., Li, Q. W., Geng, F. X., and Zhao, Z. G. (2015). Noble metal-comparable SERS enhancement from semiconducting metal oxides by making oxygen vacancies. Nat. Commun. *6*, 7800.

Lin, J., Shang, Y., Li, X. X., Yu, J., Wang, X. T., and Guo L. (2017). Ultrasensitive SERS detection by defect engineering on single  $Cu<sub>2</sub>O$  superstructure particle. Adv. Mater. *29*, 1604797.

Guo, L., Wang, X. T., Shi, W. X., Jin, Z., Huang, W. F., Lin, J., Ma, G. S., and Li, S. Z. (2017). Remarkable SERS activity observed from amorphous ZnO nanocages. Angew. Chem. Int. Ed*. 56*, 9851–9855.

Zheng, Z. H., Cong, S., Gong, W. B., Xuan, J. N., Li, G. H., Lu, W. B., Geng F. X.,

and Zhao Z. G. (2017). Semiconductor SERS enhancement enabled by oxygen incorporation. Nat. Commun*. 8*, 1993.

Sun, H. Z., Cong, S., Zheng, Z. H., Wang, Z., Chen, Z. G., and Zhao, Z. G. (2019). Metal–organic frameworks as surface enhanced Raman scattering substrates with high tailorability. J. Am. Chem. Soc*. 141*, 870–878.

Yilmaz, M., Babur, E., Ozdemir, M., Gieseking, R. L., Dede, Y., Tamer, U., Schatz, G. C., Facchetti, A., Usta, H., and Demirel, G. (2017). Nanostructured organic semiconductor films for molecular detection with surface-enhanced Raman spectroscopy. Nat. Mater. *16*, 918–924.

Shan, Y. F., Zheng, Z. H., Liu, J. J., Yang, Y., Li, Z. Y., Huang, Z. R., and Jiang D. L. (2017). Niobium pentoxide: a promising surface-enhanced Raman scattering active semiconductor substrate. npj Comput. Mater. *3*, 11.

Li, T., Chen, K., Chen, Z. F., Cong, C. X., Qiu, C. Y., Chen, J. J., Wang, X. M., Chen, H. J., Yu, T., Xie, W. J., et al. (2018). 1T' transition metal telluride atomic Layers for plasmon-free SERS at femtomolar levels. J. Am. Chem. Soc. *140*, 8696–8704.

Zhang, Q. Q., Li, X. S., Ma, Q., Zhang, Q., Bai, H., Yi, W. C., Liu, J. Y., Han, J., and Xi, G. C. (2017). A metallic molybdenum dioxide with high stability for surface enhanced Raman spectroscopy. Nat. Commun*. 8*, 14903.