

Supplementary Information:

Advantages of eutectic alloys for creating catalysts in the realm of nanotechnology-enabled metallurgy

Jianbo Tang¹, Rahman Daiyan¹, Mohammad B. Ghasemian¹, Shuhada A. Idrus-Saidi¹, Ali Zavabeti^{2,3}, Torben Daeneke², Jiong Yang¹, Pramod Koshy⁴, Soshan Cheong⁵, Richard D. Tilley^{5,6,7}, Richard B. Kaner^{8,9}, Rose Amal¹ & Kouros Kalantar-Zadeh^{1*}

¹ School of Chemical Engineering, University of New South Wales (UNSW), Sydney, NSW 2052, Australia

² School of Engineering, RMIT University, Melbourne, VIC 3001, Australia

³ College of Material Science and Technology, Nanjing University of Aeronautics and Astronautics, 29 Jiangjun Ave, 211100 Nanjing, China

⁴ School of Materials Science and Engineering, UNSW, Sydney, NSW 2052, Australia

⁵ Mark Wainwright Analytical Centre, UNSW, Sydney, NSW 2052, Australia

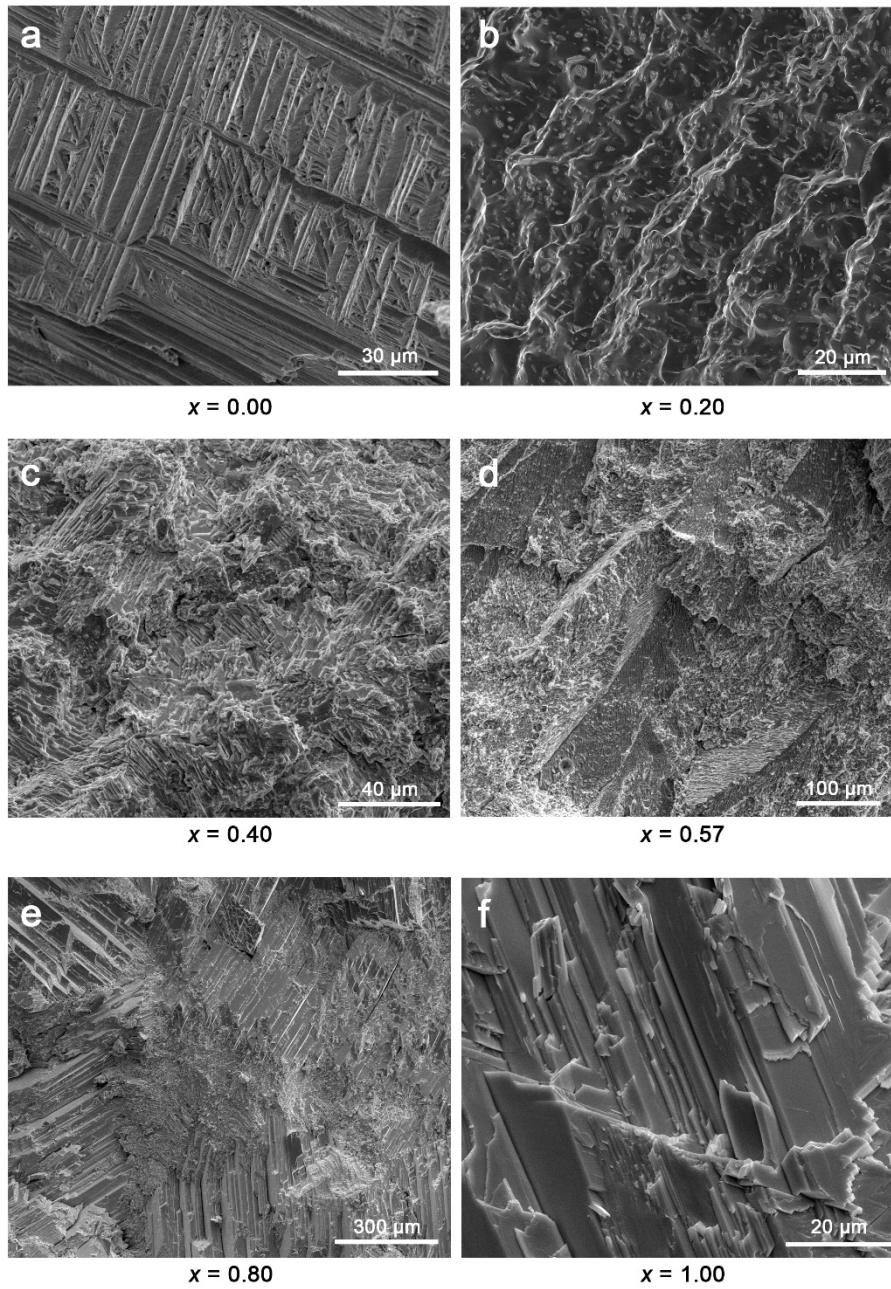
⁶ School of Chemistry, UNSW, Sydney, NSW 2052, Australia

⁷ Australian Centre for NanoMedicine, UNSW, Sydney, NSW 2052, Australia

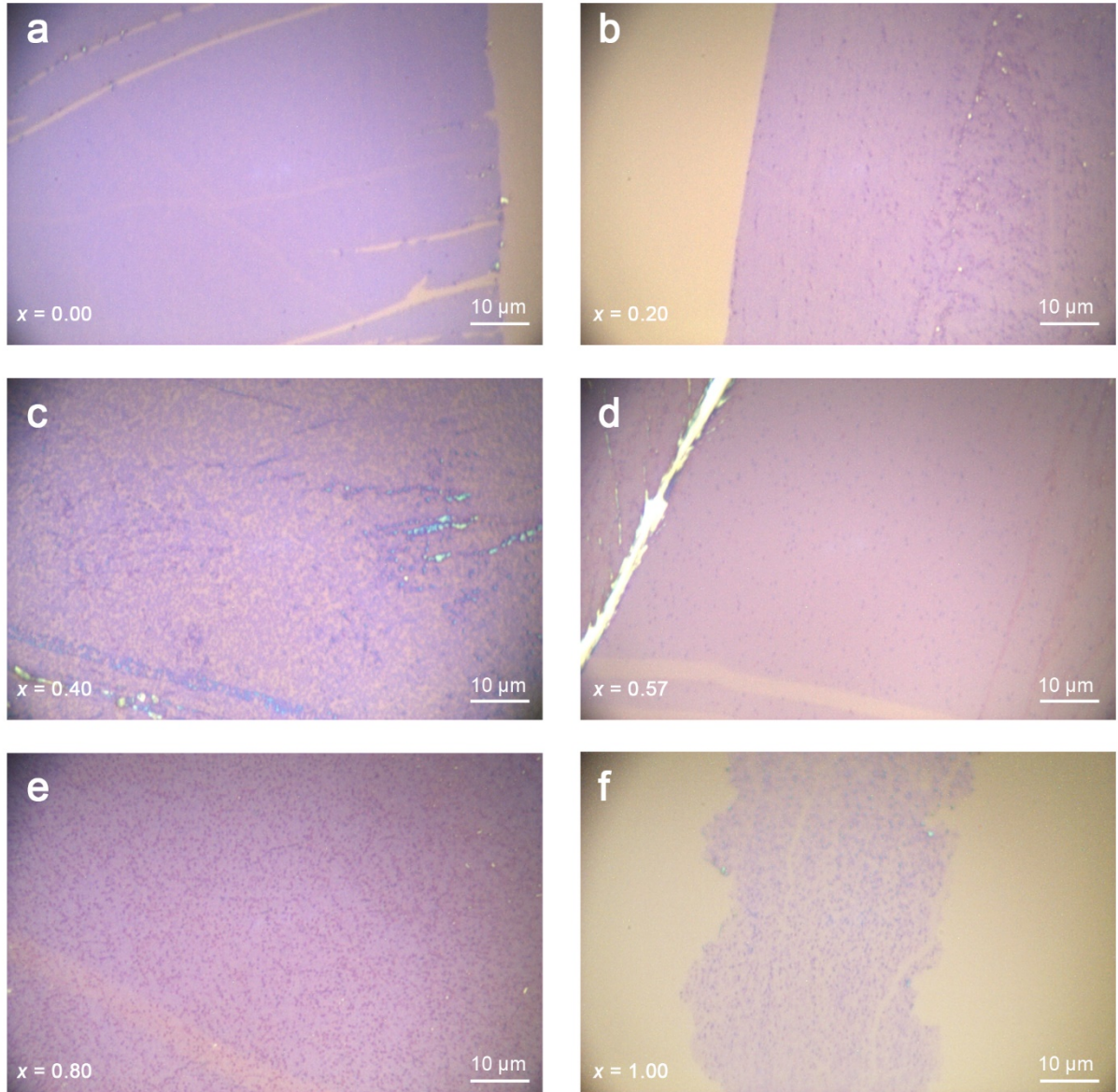
⁸ Department of Chemistry and Biochemistry, University of California, Los Angeles (UCLA), Los Angeles, CA 90095, USA

⁹ Department of Materials Science and Engineering, UCLA, Los Angeles, CA 90095, USA

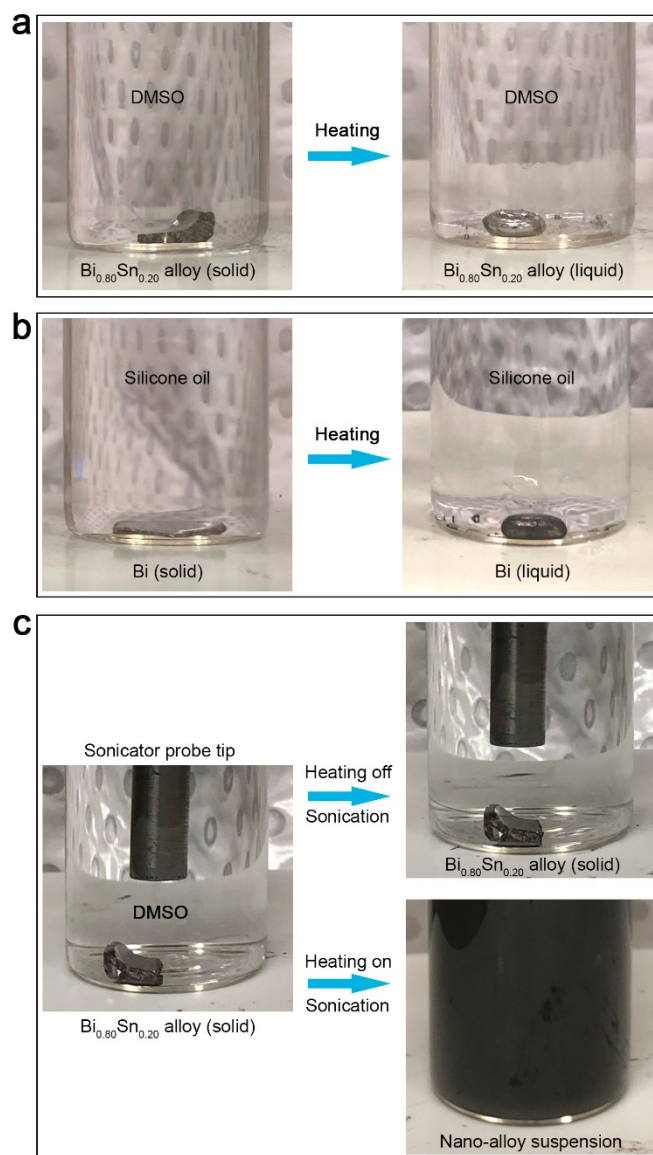
*Correspondence to K.K.-Z. (email: k.kalantar-zadeh@unsw.edu.au)



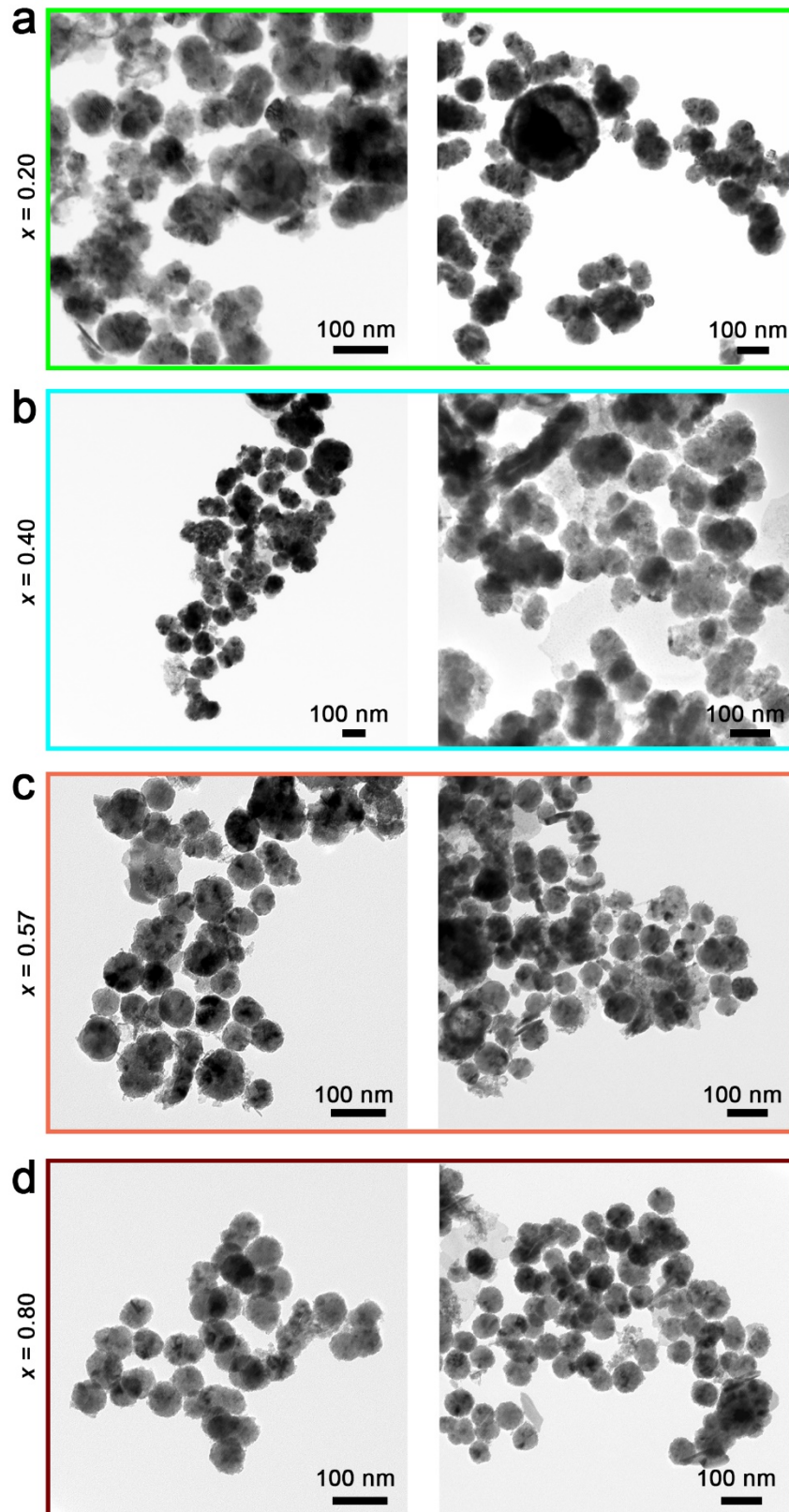
Supplementary Figure 1. Additional SEM images of the microstructures of the $\text{Bi}_x\text{Sn}_{1-x}$ bulk alloy samples and the Sn, Bi metals.



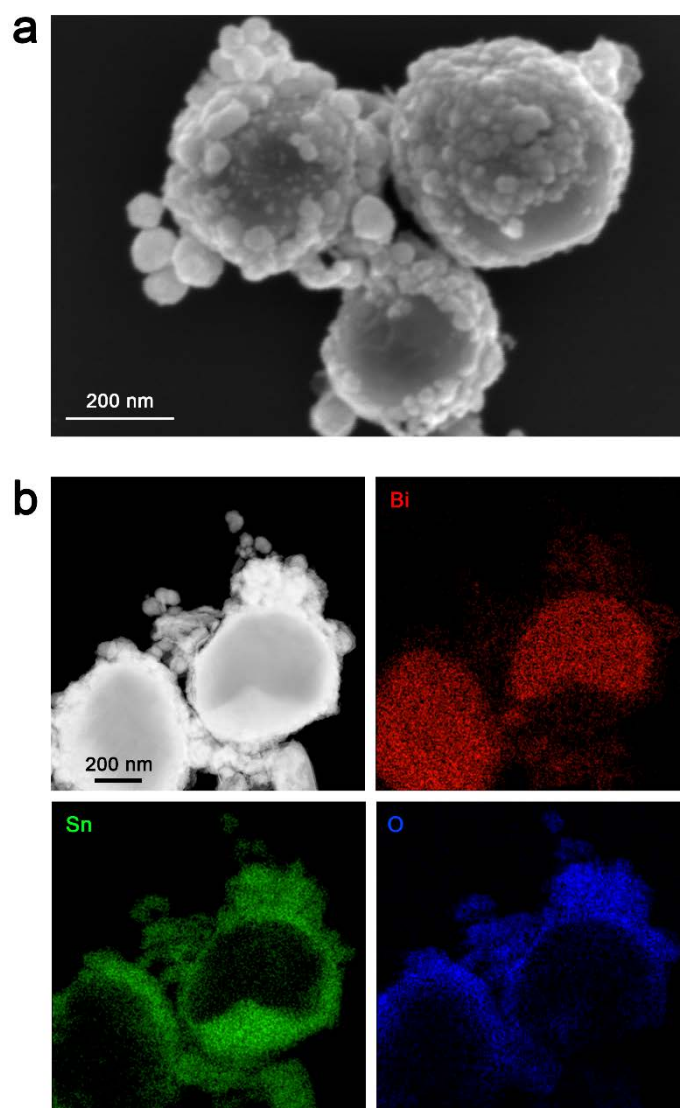
Supplementary Figure 2. Optical images of surface oxide layers for Raman tests. The oxide layers are touch-printed from liquid $\text{Bi}_x\text{Sn}_{1-x}$ bulk alloys in ambient air using silicon substrates.



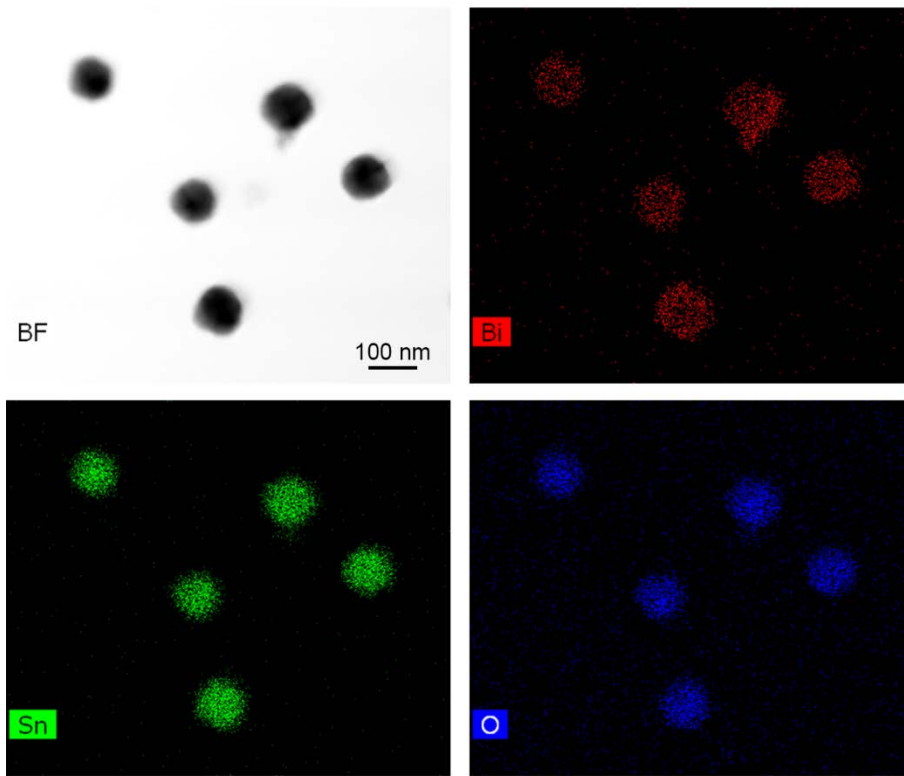
Supplementary Figure 3. Demonstration of the liquefied state of the samples and the necessity of heating. **a** The $\text{Bi}_{0.80}\text{Sn}_{0.20}$ sample in DMSO solvent. **b** The Bi metal control sample in silicone oil. **c** Comparison of sonicating with and without heating to show that heating is necessary to fragment the liquid alloy into droplets. The hotplate is set to 400°C and its surface temperature is measured to be $397.4 \pm 0.4^\circ\text{C}$ by an infrared thermometer.



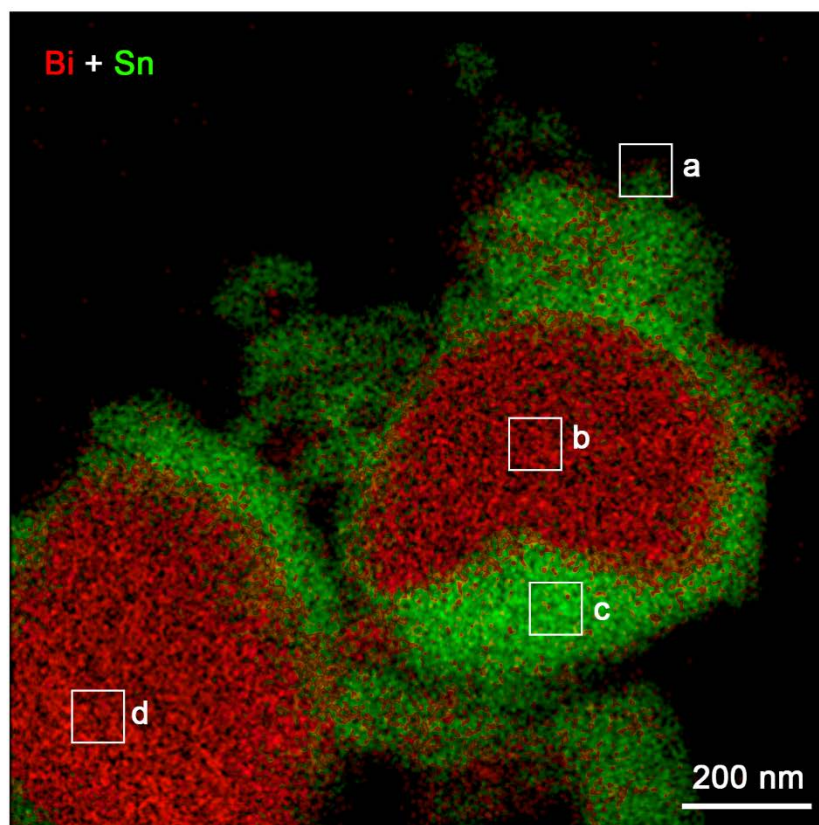
Supplementary Figure 4. Additional TEM images of the $\text{Bi}_x\text{Sn}_{1-x}$ nano-alloy samples prepared in this study. Some large particles typically > 200 nm are occasionally seen.



Supplementary Figure 5. **a** SEM images of the large particles that are usually partially coated by small particles to form core-shell structures. **b** TEM image and STEM-EDX mapping showing the element distribution in the Bi-rich cores and the shells composed of small particles.

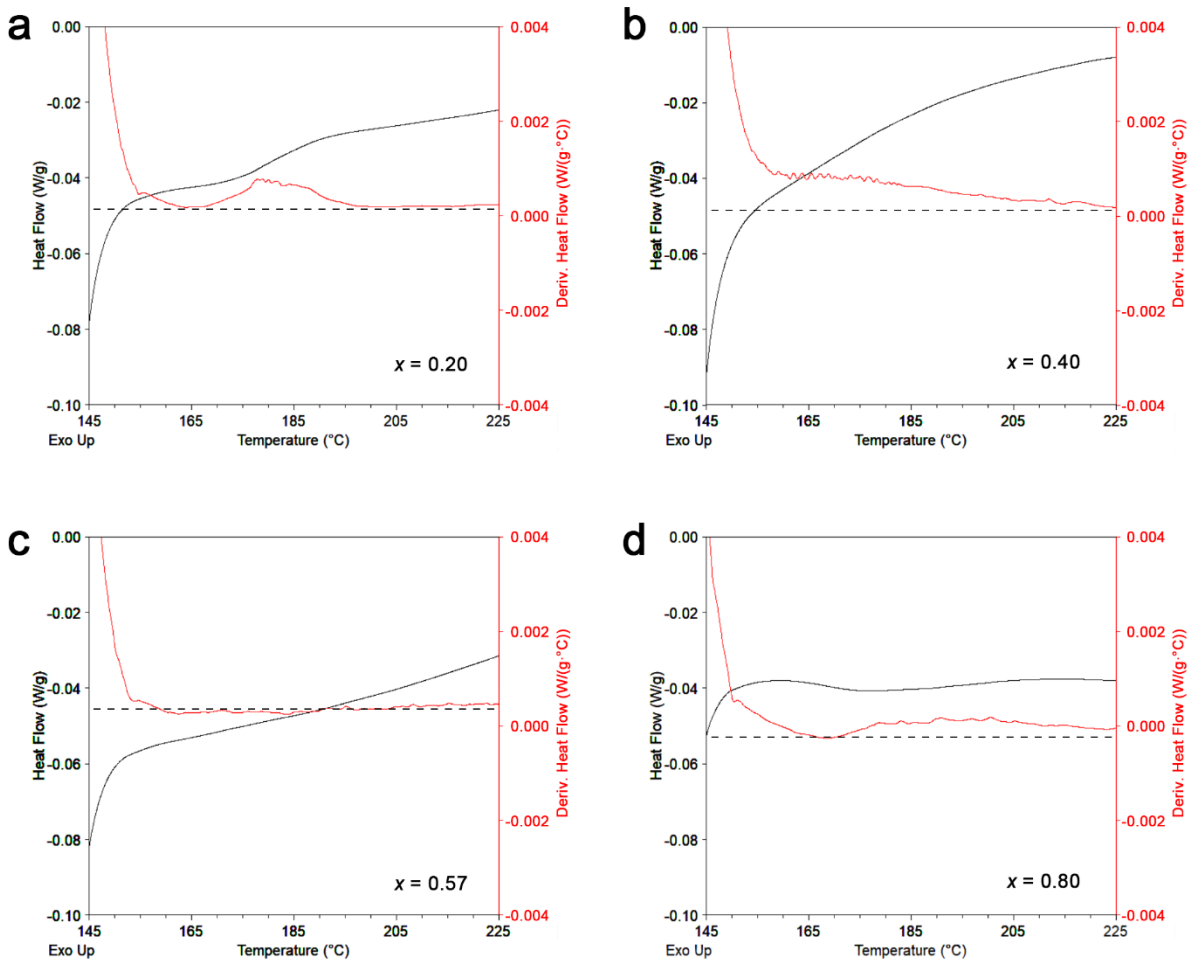


Supplementary Figure 6. Additional TEM-EDX mapping results of small particles of the eutectic nano-alloy sample.

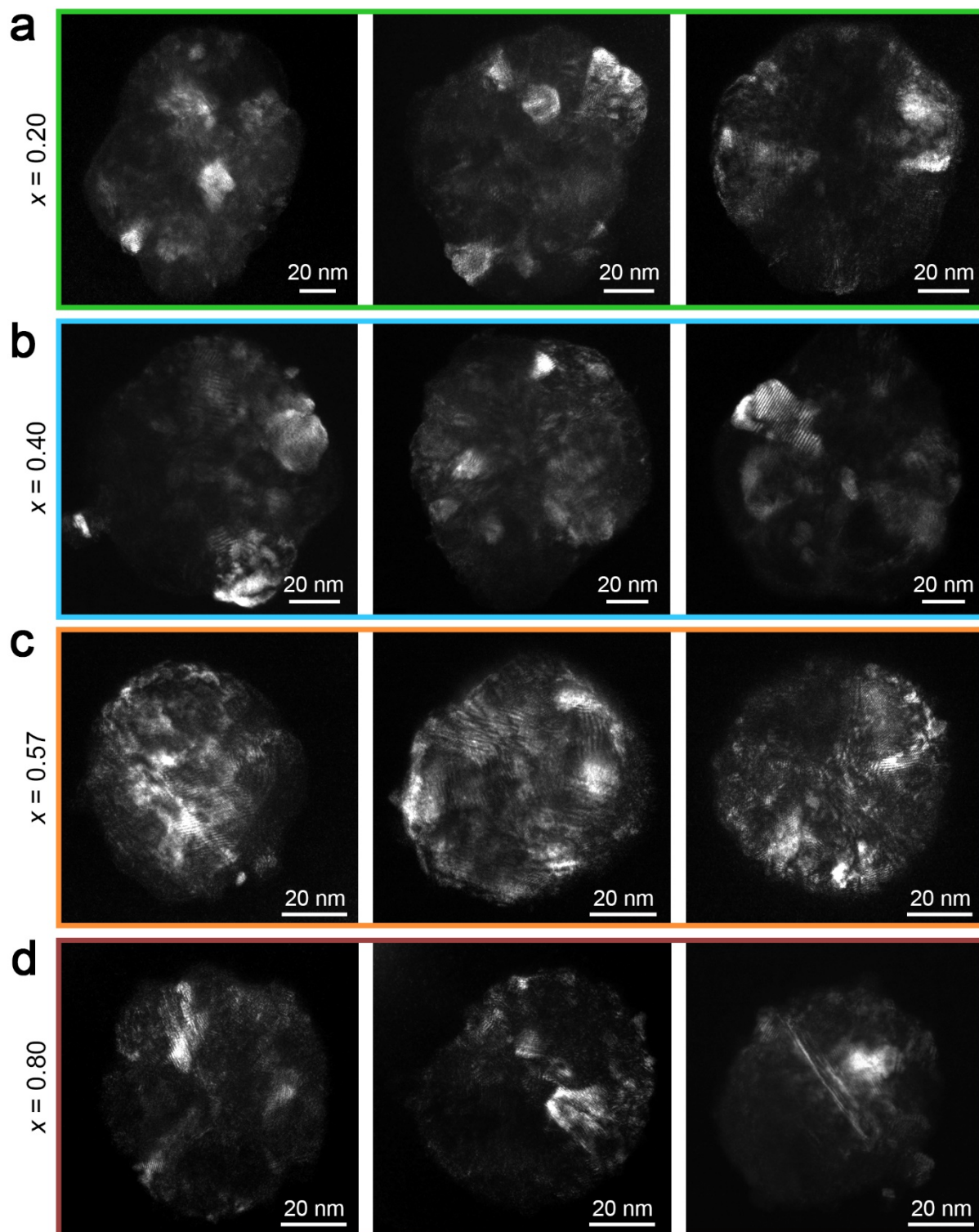


Element	Atom %			
	a	b	c	d
Sn	91	30	95	10
Bi	9	70	5	90

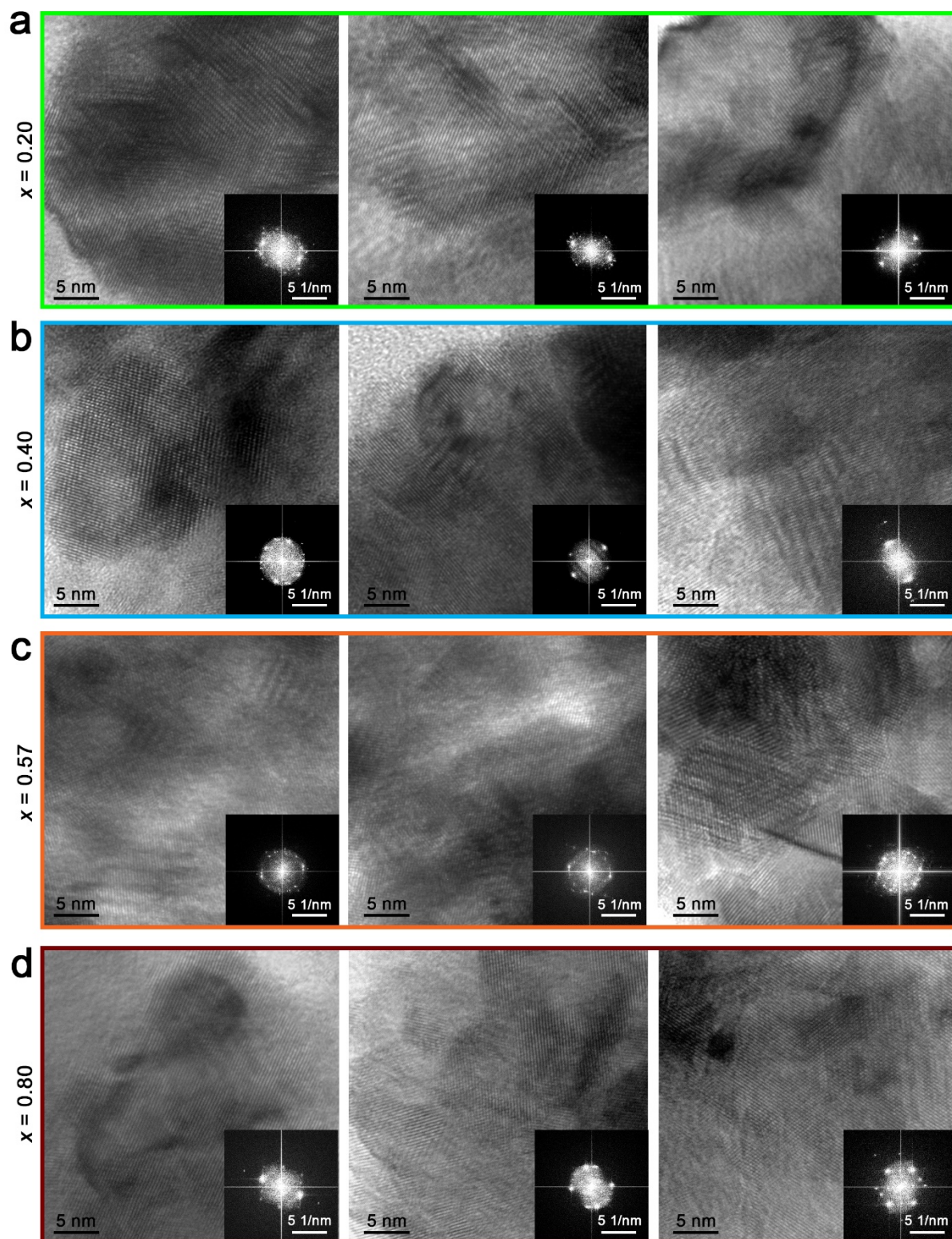
Supplementary Figure 7. Relative Bi-Sn atomic ratio in the small particle (a), Bi-rich cores of the large particles (b, d), and the small-particle-containing shell (c). The values are obtained from EDX spectra.



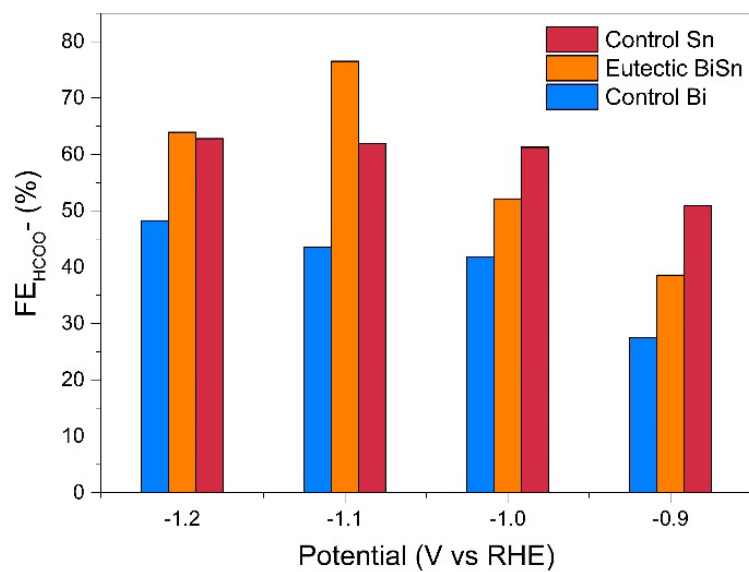
Supplementary Figure 8. Plots of heat flow (black curve) and its temperature derivative (red curve) for the different Bi-Sn nano-alloy samples shown in the insets of Fig. 3k in the main text. The temperature derivative of heat flow of all non-eutectic samples shows temperature-dependent variations while that of the eutectic sample $x=0.57$ remains constant. The results mean that the phase transition takes place beyond the main peak region for the non-eutectic samples and it is therefore confirmed that the small particles in the $x=0.57$ nano-alloy sample, which contribute to the majority of the sample surface area, still remains eutectic after ultrasonication.



Supplementary Figure 9. Additional dark-field TEM images of different $\text{Bi}_x\text{Sn}_{1-x}$ nano-alloy samples with Bi ratio x indicated.



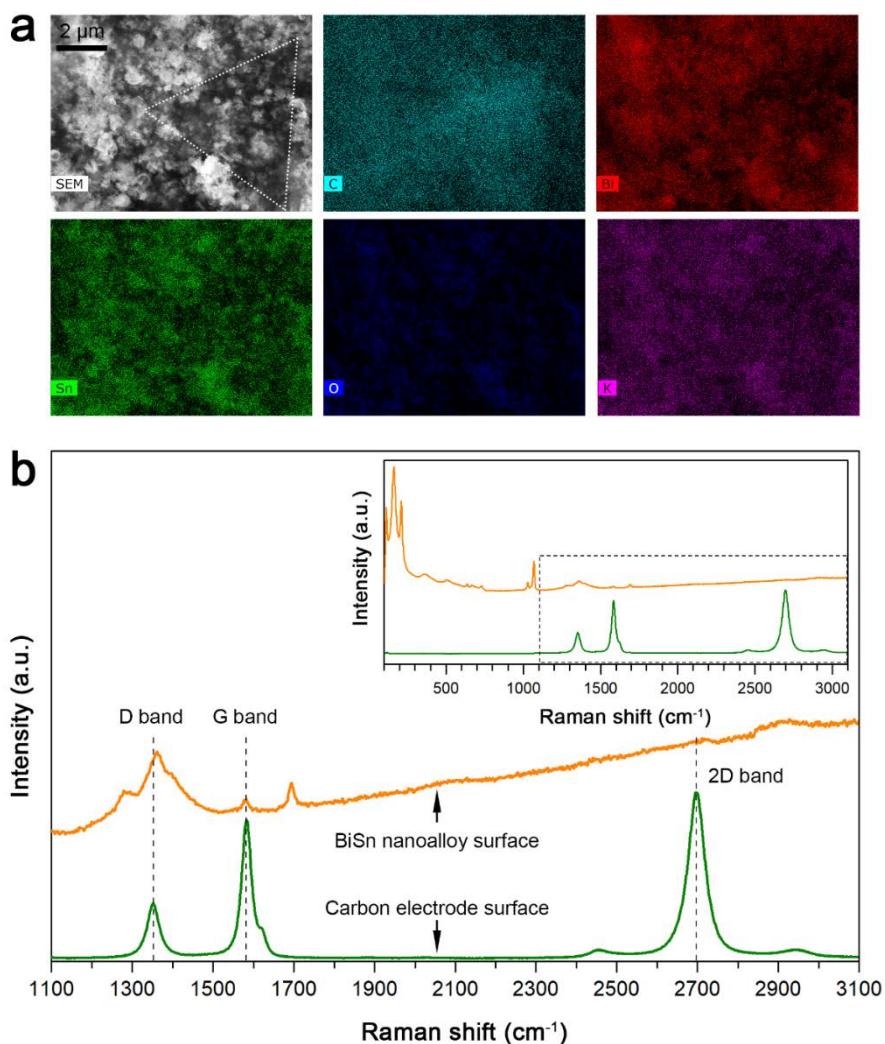
Supplementary Figure 10. Additional BF-HR-TEM images with the insets showing the FFTs of the Bi-Sn nano-alloys of different compositions as indicated. The eutectic nanoparticles typically show smaller grains than the non-eutectics and their omnidirectionally oriented grains render the FFT patterns circular.



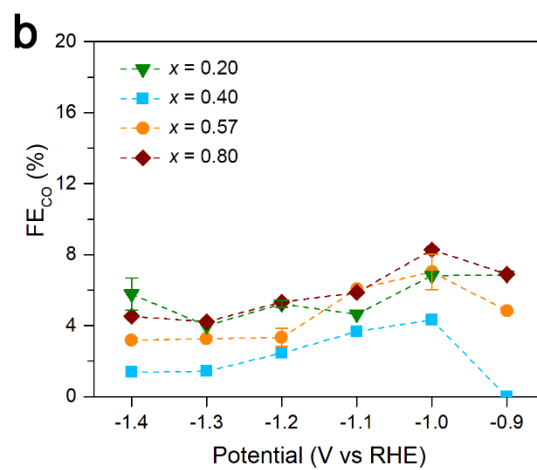
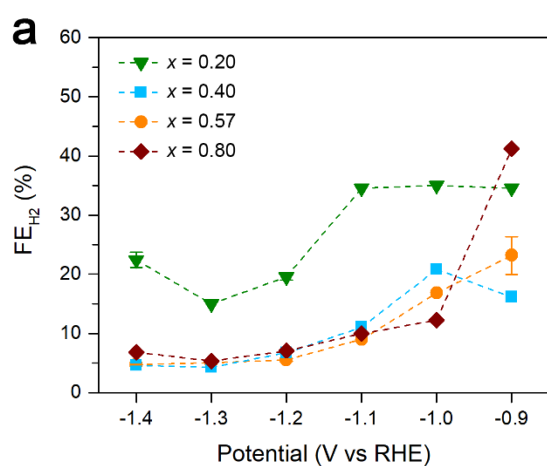
Supplementary Figure 11. Dependence of Faradaic efficiency for HCOO^- on applied potential for eutectic Bi-Sn nano-alloy sample and control Sn and Bi samples.

Supplementary Table 1. CO₂RR catalytic performances of various Sn- and Bi-based catalysts.

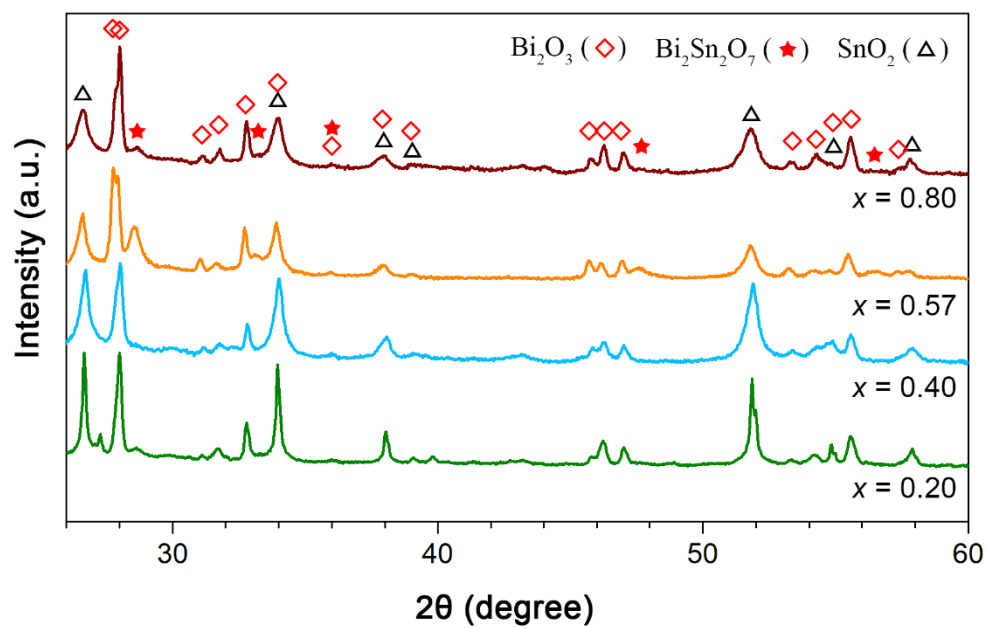
Catalyst	Electrolyte	Operating potential (V vs RHE)	Current density (mA/cm ²)	Faradaic efficiency for HCOO ⁻ (%)	Reference
Eutectic Bi-Sn	0.1 M KHCO₃	-1.1	10.7	78	This work
m-SnO ₂	0.1 M KHCO ₃	-1.15	10.8	75.2	1
SnO ₂ /Graphene	0.1M NaHCO ₃	-1.16	10.2	93.6	2
1D SnO ₂ wire in tube	0.1 M KHCO ₃	-0.99	7	63	3
Heat-treated Sn dendrite	0.1 M KHCO ₃	-1.36	17.1	71.6	4
SnO ₂ nanosheets on carbon cloth	0.5 M NaHCO ₃	-0.99	45	87	5
Bi-Sn	0.5 M KHCO ₃ with 600 rpm stirring	-1.14	-61	94	6
Sn/SnO _x /Ti	0.5 M NaHCO ₃	-0.7	1.8	55	6
Annealed Sn NPs	0.1M KHCO ₃	-1.2	4	51.5	7
Bi nanosheets	0.5 M NaHCO ₃	-0.8	-5	94	8
Sn quantum sheets/Graphene	0.1 M NaHCO ₃	-1.15	21.5	89	9
SnO ₂ nanowires (plasma treated)	0.1 M KHCO ₃	-0.8	6	81	10



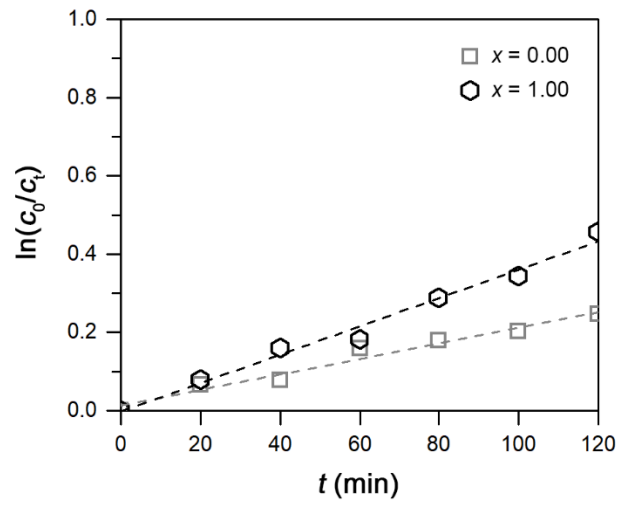
Supplementary Figure 12. Evidence for the formation of carbonaceous species during the CO₂RR reaction. **a** SEM-EDX mapping of the nano-alloy (eutectic) surface after CO₂RR reaction. The outlined triangle region, which is darker than the Bi-Sn nano-alloys under SEM, shows higher C density than other regions. This enrichment of C is supposed to be a result of direct reduction of CO₂ to solid C during the CO₂RR reaction. **b** Raman spectra collected from the Bi-Sn nano-alloy surface (yellow curve) and from the surface of the carbon electrode used to support the Bi-Sn nano-alloy (green curve). The inset shows the extended Raman spectra of **b**. When compared with the Raman spectra of the carbon electrode, the D band of the carbon material on the Bi-Sn nano-alloy sample is slightly shifted. In addition, it does not have a 2D band at near 2700 cm⁻¹. Such differences confirm that the carbon Raman signal should come from carbon species produced during the CO₂RR reaction, not from the carbon electrode.



Supplementary Figure 13. **a** Dependence of the Faradic efficiency of H₂ and **b** that of CO on the applied potential for different Bi-Sn nano-alloy samples.



Supplementary Figure 14. XRD patterns of the annealed Bi-Sn nano-alloy samples with their starting Bi ratio indicated.



Supplementary Figure 15. Plots of $\ln(c_0/c_t)$ vs t for the control Sn ($x = 0.00$) and Bi ($x = 1.00$) samples.

Supplementary Table 2. Eutectic composition and melting point of some binary alloy systems¹¹.

Alloy system A-B	Eutectic composition A _{wt%} B _{wt%}	Melting point of A (°C)	Melting point of B (°C)	Eutectic melting point of A-B (°C)
Bi-Cd	Bi ₆₀ Cd ₄₀	271.4	321.1	146
Bi-In	Bi _{43.3} In _{56.7}	271.4	156.6	72.7
Bi-Li	Bi ₂₃ Li ₇₇	271.4	180.6	175.0
Bi-Pb	Bi _{55.2} Pb _{44.8}	271.4	327.5	125.5
Bi-Pd	Bi ₉₇ Pd ₃	271.4	1555	256
Bi-Pt	Bi _{99.2} Pt _{0.8}	271.4	1769	259
Bi-Sm	Bi ₉₉ Sm ₁	271.4	1074	252
Bi-Sn	Bi ₅₇ Sn ₄₃	271.4	232.0	139
Bi-Te	Bi _{98.3} Te _{1.7}	271.4	449.6	266
Bi-Yb	Bi ₉₅ Yb ₅	271.4	819	250
Bi-Zn	Bi _{97.3} Zn _{2.7}	271.4	419.6	254.5
Cd-In	Cd _{25.3} In _{74.4}	321.1	156.6	126
Cd-Pb	Cd _{17.5} Pb _{82.5}	321.1	327.5	246
Cd-Sb	Cd ₉₂ Sb ₈	321.1	630.8	290
Cd-Sn	Cd _{32.25} Sn _{67.75}	321.1	232.0	176
Cd-Tl	Cd ₁₇ Tl ₈₃	321.1	304	203.5
Cd-Zn	Cd _{82.6} Zn _{17.4}	321.1	419.6	266
Cu-In	Cu _{0.9} In _{99.1}	1064.9	156.6	153
Cu-Tin	Cu _{0.7} Sn _{99.3}	1064.9	232.0	227
Dy-Sn	Dy _{1.2} Sn _{98.8}	1412	232.0	215
Ga-Ag	Ga _{94.5} Ag _{5.5}	29.8	951.9	25.0
Ga-In	Ga _{78.6} In _{21.4}	29.8	156.6	15.3
Ga-Mn	Ga ₉₉ Mn ₁	29.8	1246	29.8
Ga-Sn	Ga _{86.5} Sn _{13.5}	29.8	232.0	20.5
Ga-Yb	Ga ₉₉ Yb ₁	29.8	819	27
Ga-Zn	Ga _{96.4} Zn _{3.4}	29.8	419.6	24.7
Li-Pd	Li ₅₁ Pd ₄₉	180.6	1555	145
Li-Sn	Li ₉₈ Sn ₂	180.6	232.0	179
Li-Sr	Li ₃₇ Sr ₆₃	180.6	769	134
Li-Tl	Li ₇₇ Tl ₂₃	180.6	304	175
Li-Zn	Li _{69.3} Zn _{30.7}	180.6	419.6	161
Mg-Sn	Mg _{2.1} Sn _{97.9}	650	232.0	203.5
Mg-Tl	Mg ₃ Tl ₉₇	650	304	202
Pb-Pd	Pb _{95.5} Pd _{4.5}	327.5	1555	260
Pb-Pt	Pb ₉₅ Pt ₅	327.5	1769	290
Pb-Sb	Pb _{88.9} Sb _{11.1}	327.5	530.8	251.7
Pb-Sn	Pb _{38.1} Sn _{61.9}	327.5	232.0	183
Pd-Sn	Pd ₉₉ Sn ₁	1555	232	230
Pd-Tl	Pd ₉₉ Tl ₁	1555	302	293
Pt-Sn	Pt _{0.8} Sn _{99.2}	1769	232.0	226
Sb-Tl	Sb _{19.7} Tl _{80.3}	630.8	304	195
Se-Tl	Se _{47.5} Tl _{52.5}	221	304	199
Sn-Sr	Sn ₉₉ Sr ₁	232.0	759	230
Sn-Tl	Sn ₅₇ Tl ₄₃	232.0	304	166
Sn-Zn	Sn _{91.2} Zn _{8.8}	232.0	419.6	198.5
Te-Tl	Te ₄₂ Tl ₅₈	449.6	304	224
Tl-Zn	Tl ₉₇ Zn ₃	304	419.6	292

Supplementary Table 3. Properties of some proposed liquids for ultrasonication¹².

Liquid	Melting point	Boiling point	Solubility in water	Flammability
Water	0	100	-	Non-flammable
Dimethyl sulfoxide (DMSO)	19	189	Miscible	Flammable
Glycerol	17.8	290	Miscible	Flammable
Paraffin wax	37	> 370	Immiscible	Flammable
Silicone oil	< 0 (pour point)	> 300	Immiscible	Non-flammable
AlBr ₃	97.5	257	Soluble	Non-flammable
AlCl ₃ -KCl (33:67 mole %)	90		Soluble	Non-flammable

Supplementary References

- 1 Daiyan, R., Lu, X., Saputera, W. H., Ng, Y. H. & Amal, R. Highly selective reduction of CO₂ to formate at low overpotentials achieved by a mesoporous tin oxide electrocatalyst. *ACS Sustain. Chem. Eng.* **6**, 1670-1679 (2018).
- 2 Zhang, S., Kang, P. & Meyer, T. J. Nanostructured tin catalysts for selective electrochemical reduction of carbon dioxide to formate. *J. Am. Chem. Soc.* **136**, 1734-1737 (2014).
- 3 Fan, L., Xia, Z., Xu, M., Lu, Y. & Li, Z. 1D SnO₂ with wire-in-tube architectures for highly selective electrochemical reduction of CO₂ to C1 products. *Adv. Funct. Mater.* **28**, 1706289 (2018).
- 4 Won, D. H. *et al.* Rational design of a hierarchical tin dendrite electrode for efficient electrochemical reduction of CO₂. *ChemSusChem* **8**, 3092-3098 (2015).
- 5 Li, F., Chen, L., Knowles, G. P., MacFarlane, D. R. & Zhang, J. Hierarchical mesoporous SnO₂ nanosheets on carbon cloth: A robust and flexible electrocatalyst for CO₂ reduction with high efficiency and selectivity. *Angew. Chem. Int. Ed.* **56**, 505-509 (2017).
- 6 Chen, Y. & Kanan, M. W. Tin oxide dependence of the CO₂ reduction efficiency on tin electrodes and enhanced activity for tin/tin oxide thin-film catalysts. *J. Am. Chem. Soc.* **134**, 1986-1989 (2012).
- 7 Wu, J., Risalvato, F. G., Ma, S. & Zhou, X.-D. Electrochemical reduction of carbon dioxide III. The role of oxide layer thickness on the performance of Sn electrode in a full electrochemical cell. *J. Mater. Chem. A* **2**, 1647-1651 (2014).
- 8 Han, N. *et al.* Ultrathin bismuth nanosheets from in situ topotactic transformation for selective electrocatalytic CO₂ reduction to formate. *Nat. Commun.* **9**, 1320 (2018).
- 9 Lei, F. *et al.* Metallic tin quantum sheets confined in graphene toward high-efficiency carbon dioxide electroreduction. *Nat. Commun.* **7**, 12697 (2016).
- 10 Kumar, B. *et al.* Reduced SnO₂ porous nanowires with a high density of grain boundaries as catalysts for efficient electrochemical CO₂-into-HCOOH conversion. *Angew. Chem. Int. Ed.* **56**, 3645-3649 (2017).
- 11 Baker, H. & Okamoto, H. *ASM Handbook. Vol. 3. Alloy Phase Diagrams* (ASM International, 1992).
- 12 Lide, D. R. *CRC Handbook of Chemistry and Physics Vol. 85*(CRC Press, 2004).