Supplementary Information

Spatial quantification of dynamic inter and intra particle crystallographic heterogeneities within lithium ion electrodes

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Supplementary Figure 22: A diffraction peak of interest from Particle 1 as derived from the five XRD-CT datasets (blue: dataset 1, red: dataset 5).

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Supplementary Figure 27: Region of interest from XRD-CT dataset 3 which exhibits the abnormal behavior for the diffraction peak at *ca.* Q = 3.05 Å-1 .

Supplementary Figure 28: Summed diffraction patterns from the two regions-of-interest from XRD-CT dataset 3. Blue: ROI presented in Supplementary Figure 26, Red: region corresponding to the inverse image presented in Supplementary Figure 26.

Supplementary Figure 29: Summed diffraction patterns from the two regions-of-interest from XRD-CT dataset 3 focusing on the (004) reflection. Blue: ROI presented in Supplementary Figure 26, Red: region corresponding to the inverse image presented in Supplementary Figure 26.

Supplementary Figure 30: Illustration of the cubic spinel LiMn₂O₄ structure using the Mercury software.

Supplementary Figure 31: Region of interest from XRD-CT dataset 5 which exhibits the abnormal behavior for the diffraction peak at *ca.* Q = 3.05 Å-1 .

Supplementary Figure 32: Fit result using 23 pseudo-Voigt peaks for a diffraction pattern extracted from a region from XRD-CT dataset 5 which does not contain the new diffraction peak (region corresponding to the inverse image presented in Supplementary Figure 31). Black: Observed pattern, Red: calculated peaks, Green: difference plot.

Supplementary Figure 33: Fit result using 23 pseudo-Voigt peaks for a diffraction pattern extracted from a region from XRD-CT dataset 5 which contains the new diffraction peak (ROI shown in Supplementary Figure 31).

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Supplementary Figure 35: The difference in Q position for certain peaks in the two diffraction patterns of interest.

Supplementary Figure 36: The ratio of the FWHM of certain peaks in the two diffraction patterns of interest.

Supplementary Figure 37: Histograms of LiMn₂O₄ lattice parameter for the five XRD-CT datasets and the main phases present in these datasets.

Supplementary Figure 38: Williamson-Hall plot using the peak shape parameter values from the peaks obtained after fitting the summed diffraction patterns from particle 2 for the five XRD-CT datasets. Theta values in degrees (°).

Supplementary Figure 39: Maps obtained from the Williamson-Hall plot spatially-resolved analysis using the five XRD-CT datasets. First row: Maps corresponding to the slopes of the fitted FWHM*cos(theta) vs sin(theta) lines. Second row: Maps corresponding to the intercepts of the fitted FWHM*cos(theta) vs sin(theta) lines. Third row: Maps corresponding to the norm of the residuals of the fitted FWHM*cos(theta) vs sin(theta) lines. Fourth row: Rwp maps corresponding to the error of the fitted Gaussian peaks during the spatially-resolved batch fitting.

Supplementary Figure 40: Williamson-Hall plot using the peak shape parameter values from the peaks obtained after fitting the summed diffraction patterns from the three segmented regions of interest from the cycled cell in the main manuscript. Theta values in degrees (°).

Supplementary Table 1. Corresponding error values for quantification of mass fractions for Particle 1 in the main manuscript.

Supplementary Table 2. Corresponding error values for quantification of mass fractions for Particle 2 in the main manuscript.

Supplementary Table 3. Corresponding error values for quantification of mass fractions for the 3 segmentations of the degraded electrode sample in the main manuscript.

Supplementary Note 1. Lattice parameter error

Operando cell

The quality of the fits obtained from the Rietveld analysis of the XRD-CT data was high. As an example, the fit from the Rietveld analysis of the summed diffraction pattern from XRD-CT dataset 1 is presented in Supplementary Figure 1 (Rwp = 5.731 %).

The error maps for the $Li_xMn_2O_4$ lattice parameter values, as calculated from the Topas v5 software, presented in Figure 3 and 4 in the main manuscript are shown in Supplementary Figure 2. The error in most regions is below 0.002 Å, indicating that the confidence in the lattice parameter values is very high.

For Rietveld quantification of the mass fractions present in the single particles (Particle 1 and Particle 2) and the degraded electrode shown in Figures 6 and 7 in the main manuscript, the corresponding error values are shown in Supplementary Tables 1 and 2. The process of Rietveld refinement is discussed in a later section.

Cycled cell

The quality of the fit from the Rietveld analysis of the summed diffraction pattern from Segmentation 1 in Figure 7 of the main manuscript was high (Rwp = 7.93 %). The result is shown in Supplementary Figure 3.

There were also some minor peaks that could not be assigned to any known phase after performing phase identification using the ICSD and ICCD PDF databases. Peak indexing was also attempted with the Topas software but did not yield any promising candidate unit cells. These minor peaks appear at the following Q positions: 1.5, 2.31, 2.36, 2.62, 3.32, 3.55, 3.75, 3.9 and 4.2 \AA ⁻¹.

A table listing the errors for the mass fractions (Wt. %) and lattice parameters in Å (LPA) of the different phases for the 3 segmentations of the degraded cycled sample shown in Figure 7 of the main manuscript is shown in Supplementary Table 3.

Supplementary Note 2. XRF spectra of the LMO graphite electrodes

The capacity fade of the degraded cell is presented in Figure 7 of the main manuscript. Using X-ray fluorescence (XRF), significant amounts of elemental Mn were found on the degraded graphite electrode, indicating dissolution, migration and deposition on the graphite, as seen in Supplementary Figure 4. The system used for XRF was an Attomap Micro (Sigray, CA, USA).

Supplementary Note 3. Voltage profiles from the operando cell with Li vs. LMO

The voltage profile of the operando cell during the initial charge step outside the beam is shown in Supplementary Figure 5. The voltage profile of the operando cell from during the experiment is shown in Supplementary Figure 6a where the XRD-CT data was gathered during the open-circuit periods. A comparison of this voltage profile (without the open-circuit periods) from the micro-cell with Li vs. LMO, and the voltage profile from a standard coin cell with graphite vs. LMO is presented in Supplementary Figure 6b. The voltage profiles are similar in shape, but the voltage of the operando cell is higher, which is attributed to the counter electrode being Li rather than graphite.

Supplementary Note 4. Reconstruction of region-of-interest (ROI) XRD-CT data

The data processing strategy used for the high-resolution region-of-interest (ROI) XRD-CT data was the same as in our previous work¹. This sequence is shown in Supplementary Figure 7.

- 1. First, a coarse XRD-CT scan was performed. The translation step size for this coarse scan was 20 μm. This dataset can be reconstructed independently of the ROI scan to investigate the sample at a different length scale.
- 2. A ROI XRD-CT scan was acquired after the coarse scan. Here, the translation step size was 1 μ m and the total area covered was $351 \mu m \times 351 \mu m$.
- 3. The sinogram data volume from the coarse and the ROI scans were combined to yield a new data volume. To achieve this and at the same time suppress the formation of any artefacts in the reconstructed data, the coarse XRD-CT sinogram data were expanded by a factor of 20 times (linear interpolation). This expansion factor is equal to the ratio of the translation step size used for the coarse scan over the translation step size used the ROI scan.
- 4. Each data point of each sinogram (i.e. coarse and high resolution) is associated to a beam monitor value (diode). The sinograms were normalized with respect to the recorded beam intensity values leading to two sinograms with matching intensities. In case the beam intensity cannot be monitored, a scale factor has to be applied to the expanded coarse sinogram data so that the intensity between the two datasets is matched (i.e. to take into account the intensity mismatch between the two datasets).
- 5. The new combined sinogram data volume is reconstructed and to minimize the required time, only a region of diameter equal to the size of the ROI scan is reconstructed (351 pixels in this experiment).

Supplementary Note 5. Segmentation of the LMO particles in the XRD-CT data

The signal-to-noise ratio was not high enough to perform automatic segmentation to separate the particles/agglomerates from the background. In order to overcome this problem, a global $Li_xMn_2O_4$ image was calculated from the sum of the Li_xMn₂O₄ phase distribution maps (i.e. five images in total) as derived from the Rietveld analysis of the respective XRD-CT datasets. This image was then filtered using a mean filter and then a manual segmentation was performed. The result of the segmentation is shown in Supplementary Figure 8. A binary mask was created based on the results of the segmentation process. The mask is shown in Supplementary Figure 9. The particles were then labelled showing that a total number of ca. 290 particles are present in the XRD-CT ROI images. The results are presented in Supplementary Figure 10.

Supplementary Note 6. Identification of the secondary cubic LMO phase

This mask was then applied to the Li_xMn₂O₄ lattice parameter maps obtained from the Rietveld analysis of the in situ LMO XRD-CT data which led to clearer images. This is shown in Supplementary Figure 11 where one can clearly observe that different particles exhibit different values for $Li_xMn_2O_4$ lattice

parameter. This image was then imported into the Ilastik software² and image segmentation was performed using supervised machine learning. In total five regions-of-interest (ROIs) displaying different lattice parameter values were selected. The results are shown in Supplementary Figure 12. The summed diffraction patterns from these five regions are presented in Supplementary Figure 13. The same diffraction peaks are present in the summed diffraction patterns exported from all five regions but there is a shift to lower scattering angles (i.e. increase in unit cell size). This increase in unit cell size corresponds to different amounts of Li being incorporated in the unit cell. However, as shown in Supplementary Figure 14 there is shoulder peak to the right of all diffraction peaks corresponding to region 5. The following structures (ICSD codes in brackets) were used to model the diffraction pattern from region 5 but none of them yielded satisfactory results:

Li₂Mn₂O₄ (201557), Li₂MnO₂ (37327), Li₂MnO₃ (239796), LiMn₂O₄ (50415 – cubic spinel and 87774 – tetragonal), LiMn₃O₄ (201556), LiMnO₂ (81053 – orthorhombic and 173138 - rhombohedral), Li₂O (54368), Li_2O_2 (24143), Mn₂O₃ (290640 – orthorhombic, 236255 – triclinic, 236254 – rhombohedral, 9091 – cubic, 9090 – orthorhombic), Mn_2O_7 (60821 – monoclinic), Mn_3O_4 (68174 – tetragonal, 40110 – orthorhombic, 30005 – orthorhombic), Mn_5O_8 (16956 – monoclinic), MnO (657304 and 9864 – cubic, 262928 – hexagonal), MnO₂ (78331 – orthorhombic, 20229 – tetragonal, 248069 – orthorhombic, 78331 – orthorhombic, 393 – tetragonal, 54114 – orthorhombic and 76430 – hexagonal).

It was realized that the shoulder peaks correspond to a separate phase rather than a lower symmetry LMO phase than the cubic spinel Li_xMn₂O₄. As the shoulder peaks appear on the right-hand side of all $Li_xMn₂O₄$ diffraction peaks, this was new phase was assigned to another cubic $Li_xMn₂O₄$ phase. It should be noted that peak indexing of this minor phase was also attempted using Topas software but it was not possible to yield any useful results (peak indexing after assuming that all the peaks belong to the same phase did not give any promising candidate unit cells). Our observation is in agreement with the work of Bianchini et al.^{3, 4} who reported the presence of metastable cubic Li_{0.5}Mn₂O₄ (P2₁3 space group). The addition of the cubic Li_{0.5}Mn₂O₄ unit cell to the Rietveld model led to superior fits. The quality of the fit of the summed diffraction pattern from XRD-CT dataset 2 is shown in Supplementary Figure 15 (Rwp = 4.563 %). In Supplementary Figure 16 it is clearly demonstrated why the addition of the Li_{0.5}Mn₂O₄ phase is necessary and that it helps model the right-hand shoulder asymmetry very well.

Supplementary Note 7. Behavior of two distinct particles during lithiation

To investigate the heterogeneity in chemistry between distinct particles in the LMO electrode, two masks were created (Supplementary Figures 17 and 18), focusing on the two particles of interest discussed in the manuscript. The diffraction pattern from Particle 1, which contains the extra peaks, can be modelled well by using two cubic $Li_xMn_2O_4$ unit cells to model the data. Phase 1 shows lattice parameter equal to 8.13 Å while phase 2 shows lattice parameter equal to 8.047 Å. The results are shown in Supplementary Figure 19 (Rwp = 5.26%). The quality of the fit from the Rietveld analysis of the particle 2 is also high (Rwp = 4.62 %). The result is shown in Supplementary Figure 20. The summed diffraction patterns from Particle 1 from the five XRD-CT datasets are shown in Supplementary Figure 21. No new peaks appear during the lithiation experiment. Interestingly though, the shoulder peaks seem to disappear after XRD-CT dataset 1. It should be noted that inspection of the raw 2D diffraction data did not shown any indication of apparent strain. The presence of the shoulder peak can be more easily seen in Supplementary Figure 22 where a narrow Q range is used.

The summed diffraction patterns from Particle 2 from the five XRD-CT datasets are shown in Supplementary Figure 23. As shown in Supplementary Figure 24, Particle 2 behaves "normally" for the duration of the experiment. No new peaks or shoulder peaks appear and the peaks shift towards lower scattering angles showing that the particle is continuously being lithiated and the corresponding unit cell expanding during the experiment. A similar data processing strategy was used for XRD-CT dataset 2. Specifically, first the Li_xMn₂O₄ lattice parameter map was used to segment different regions-of-interest as shown in Supplementary Figure 25. The summed diffraction patterns from these four regions are shown in Supplementary Figure 26. Region 1 (blue peak) corresponds to the particles exhibiting the lowest lattice parameter value (between 8.1 – 8.15 Å). There is a right-hand shoulder present in these data which is in agreement with the observations from Particle 1 from XRD-CT dataset 1.

Supplementary Note 8. Identification of the cubic rock-salt LiMnO² phase

XRD-CT datasets 3, 4 and 5 did not reveal the presence of any other peaks/phases. However, careful inspection of the data showed that there are regions in the sample where the intensity of the Li_xMn₂O₄ diffraction peak at *ca.* Q = 3.05 Å⁻¹ shows abnormal behavior. The region of interest from XRD-CT dataset 3 which exhibits the abnormal behavior for the diffraction peak is shown in Supplementary Figure 27.

It is important to note that this peak corresponds to the (004) reflection. As shown in Supplementary Figures 28 and 29, the intensity of the rest of the diffraction peaks remain the same. It is also clearly shown that there is more than one peak present in this region. This implies that the cubic symmetry is broken and indeed it is not possible to obtain a good fit using a single cubic spinel $Lim₂O₄$ unit cell.

Interestingly, as shown in Supplementary Figure 20 (created with the Mercury software), there are no Li atoms present in the unit cell at the (004) plane. However, one can imagine that the place of interstitial atoms and the expansion along the z axis can lead to a unit cell with lower symmetry. It has previously been suggested that the cubic spinel-type $Li_xMn_2O_4$ changes to first to a cubic rock-salt and then to a tetragonal phase during lithiation^{5, 6}.

To investigate this further, the two summed diffraction patterns from two regions of interest from XRD-CT dataset 5 were investigated in more detail. The region of interest from XRD-CT dataset 5 which exhibits the abnormal behavior for the diffraction peak is shown in Supplementary Figure 31. The properties of the peaks in the two summed diffraction patterns were examined using the Fityk software⁷. A pseudo-Voigt peak shape function was used to model the peaks present in two diffraction patterns exported from two different regions from XRD-CT dataset 5. Specifically the following 23 peak positions were used (in Q $(A-1)$:

1.35, 2.56, 2.67, 3.07, 3.35, 3.98, 4.34, 4.53, 5.02, 5.08, 5.30, 5.46, 5.88, 6.12, 6.63, 6.67, 6.84, 6.96, 7.29, 7.48, 7.60, 7.92 and 8.18

As expected the fits are excellent (i.e. since there is no chemistry to constraint the relative peak intensities) and the results are presented in Supplementary Figures 32 and 33. A Williamson-Hall plot was constructed using the results of the fits (i.e. position of the peaks and their full width at half maximum (FWHM))obtained from the Fityk software (Supplementary Figure 34). In Supplementary Figure 35, the shift of the peaks is presented. Specifically, the difference in Q position for each peak in the two diffraction patterns is calculated. In general, the values are negative implying that the peaks from the lower symmetry peak are shifted towards lower scattering angles. As expected, the difference is most significant

at the (004) reflection. Similarly, the ratio of the FWHM of each peak was also calculated. It can be clearly seen that the values for the lower symmetry phase are more than 1, indicating that all peaks have become broader. As expected, the peak corresponding to the (004) reflection is significantly broader.

The analysis led us to the conclusion that the observed left-hand side shoulder of the peak corresponding to the (004) reflection of the cubic spinel $Li_xMn_2O_4$ is attributed to the presence of the cubic rock-salt LiMnO₂ (ICSD: 194998) phase (Li_{0.5}Mn_{0.5}O). This phase generates its main diffraction peak at this Q region. The addition of this structure to the Rietveld model led to an improved fit and it was possible to account for the left-hand side peak asymmetry at ca. Q = 3.05 Å⁻¹.

Supplementary Note 9. Strain analysis for the LiMn2O⁴ phase

The same analysis was performed for the summed diffraction patterns from particle 2 from the five XRD-CT datasets. Particle 2 was chosen for the strain analysis as it is representative of the average electrode particles (in contrast to particle 1). As previously, the peak fitting was performed using the Fityk software using pseudo-Voigt functions to fit the LiMn₂O₄ peaks and considering the instrumental broadening of the peaks using the peak profile calculated using the $CeO₂$ pattern. The results are presented in Supplementary Figure 38 and it can be clearly seen that the slope of the fitted lines increases from XRD-CT dataset 1 to 5 while the offset remains stable. This result clearly demonstrates that in this case it is possible to decouple the crystallite size-strain effects. The broadening of the LiMn₂O₄ peaks is attributed to the increase of strain which could be caused by a variety of reasons including phase segregation (the formation of the LiMnO₂ rock-salt phase is observed from XRD-CT dataset 2) and the lithiation process itself. The values for the slopes for the five datasets were 0.39, 0.53, 0.66, 0.74 and 0.77 respectively. It should be noted that, as shown in Supplementary Figure 38, the fit is worse for XRD-CT datasets 4 and 5 as the points deviate more from the linear model implying the presence of other features in the LiMn₂O₄ peaks (this coincident with Li occupancy exceeding 1).

It was also attempted to perform this analysis in a spatially-resolved manner. As the signal-to-noise ratio is worse for the spatially-resolved diffraction patterns present in the reconstructed XRD-CT data, only the first seven LiMn₂O₄ diffraction peaks (corresponding to hkl reflections: (111), (311), (222), (004), (331), (511), (044) and 442) were used for the analysis. We developed in-house python code for batch multipeak fitting using the scipy.optimise package using gaussian peak shapes (http://pd.chem.ucl.ac.uk/pdnn/peaks/gauss.htm). Initially, the peak fitting was performed using the CeO² pattern and the FWHM of the peaks was calculated. The 2theta broadening of the peaks was modelled using the Caglioti formula and refining the U, V and W parameters. The batch peak fitting was performed using the Gaussian peak shape and the instrumental broadening (calculated using the Caglioti formula for the LiMn₂O₄ peak positions) was subtracted from the obtained FWHM values (http://pd.chem.ucl.ac.uk/pdnn/peaks/broad.htm) before performing the first-degree polynomial fitting for the Williamson-Hall plots (FWHM*cos(theta) vs sin(theta)). The results obtained from this spatiallyresolved analysis are presented in Supplementary Figure 39. It can be seen that the results are in agreement with the ones obtained from the representative particle (Supplementary Figure 38) as it is shown that the slope (corresponding to strain) increases from XRD-CT dataset 1 to 5 with a small decrease in intercept (corresponding to crystallite size). However, it should be pointed out that one should be careful when interpreting these results as although the Rwp values from the peak fitting are low, the error in the slopes and intercepts is more significant (norm of the residuals) and the number of peaks used can have a strong impact on the obtained values. For these reasons, we maintained our analysis of the cycled cell using the summed diffraction patterns from the three segmented regions rather than performing the spatially-resolved analysis.

The Williamson-Hall plot analysis was also performed for the cycled cell and specifically for the summed diffraction patterns exported from the three segmented regions presented in Figure 7 in the main manuscript. The obtained results are in full agreement with the results obtained from the *in situ* experiment (Supplementary Figure 40). Specifically, it is seen that the offset of the three fitted lines is the same within the experimental errors while the slope is increasing from Segmentation 3 to Segmentation 1. As discussed in the main manuscript, Segmentation 3 corresponds to the region of the sample where there is mainly the LiMn₂O₄ cubic spinel present while Segmentation 1 contains the various undesired phases such as the Li₂MnO₃ and the LiMnO₂ cubic rock-salt. The obtained values for the three slopes are 0.82, 0.66 and 0.50.

Supplementary References

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