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Supplementary Materials for

Robust, high-performance n-type organic semiconductors

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Section S1. Details of syntheses and characterizations

Reagents and Starting Materials

1,5-Dinitroanthraquinone, methyl cyanoacetate, potassium t-butoxide (KOtBu), 1,3dimethyl-2-imidazolidinone (DMI), sulfuric acid (H₂SO₄), 4-dimethylaminopyridine (DMAP), Nphenylbis(trifluoromethanesulfonimide) formic (Tf₂NPh), acid. tetrakis(triphenylphophine)palladium (0) (Pd(PPh₃)₄), N-bromosuccinimide (NBS), 2,4,6trichlorophenyl formate, palladium acetate (Pd(OAc)₂), 4,5-bis(diphenylphosphino)-9,9dimethylxanthene (Xantphos), p-toluenesulfonic acid monohydrate (p-TsOH · H₂O), propionic 2-phenylethylamine. octylamine, 4-heptylamine, perylene-3,4,9,10-tetracarboxylic acid. dianhydride were purchased from either Tokyo Chemical Industry Co., Ltd, Sigma-Aldrich Inc., KANTO chemical Co., Ltd., or Fujifilm-Wako Pure Chemical Industries, Ltd. Nuclear magnetic resonance (NMR) spectra were recorded with chloroform-D (CDCl₃) (D, 99.8%) containing 0.05% (v/v) tetramethylsilane (TMS) or 1,1,2,2-tetrachloroethane- d_2 (CDCl₂CDCl₂) (D, 99.5%) purchased from Cambridge Isotope Laboratries, Inc.

Methods

All the reactions were carried out under argon, and air- or moisture-sensitive liquids and solutions were transferred via a syringe or a Teflon cannula. Analytical thin-layer chromatography (TLC) was performed on glass plates coated with a 0.25 mm-thick layer of 230-400 mesh silica gel containing a fluorescent indicator (Merck silica gel 60 F254). TLC plates were visualized by exposure to an ultraviolet lamp (254 or 365 nm) and by immersion in 10% phosphomolybdic acid in ethanol followed by heating on a hot plate. Flash column chromatography was performed on Kanto silica gel 60 and open column chromatography was performed on Wakogel C-200 (75-150 μm). Melting points (mp) were collected on a Mettler Toledo MP70 Melting Point System. ¹H and ¹³C NMR spectra were measured with a JEOL ECS400 spectrometer (400 MHz for ¹H and 100 MHz for ¹³C) in CDCl₃ at room temperature or CDCl₂CDCl₂ at 100 °C. Chemical shifts are reported in δ ppm. ¹H NMR and ¹³C NMR spectra are referenced to residual protons (δ 7.26 ppm for chloroform, 5.93 ppm for 1,1,2,2-tetrachloroethane (TCE)) and carbon-13 in the deuterated solvent (δ 77.16 ppm for chloroform, 74.00 ppm for TCE) as an internal standard, respectively. The data are presented in the following format: chemical shift, multiplicity (s = singlet, d = doublet, t = triplet, q = quartet, m = multiplet), coupling constant in Hertz (Hz), signal area integration in natural numbers. Recycling preparative gel permeation chromatography (GPC) was performed with a Japan Analytical Industry (JAI) LC-918 equipped with polystyrene gel columns (JAIGEL 1H and 2H) using chloroform as an eluent. Time-of-flight high-resolution mass (TOF-MS) spectrometry was performed with a Thermo Fisher LTQ-Orbitrap XL with the atmospheric pressure chemical ionization (APCI) method at Comprehensive Analysis Center, The Institute of Scientific and Industrial Research, Osaka University. Elemental analysis was carried out on a J-Science Lab JM10 CHN analyzer at Comprehensive Analysis Center, the Institute of Scientific and Industrial Research, Osaka University or on an Elementar vario MICRO cube at The Microanalytical Laboratory, Department of Chemistry, Graduate School of Science, The University of Tokyo.



Fig. S1. ¹H NMR spectrum of 1 (CDCl₂CDCl₂, 100 °C)



Fig. S2. ¹³C NMR spectrum of compound 1 (CDCl₂CDCl₂, 100 °C)



Fig. S3. ¹H NMR spectrum of compound 2 (CDCl₂CDCl₂, 100 °C)



Fig. S4. ¹³C NMR spectrum of compound 2 (CDCl₂CDCl₂, 100 °C)



Fig. S5. ¹H NMR spectrum of compound 3 (CDCl₂CDCl₂, 100 °C)



Fig. S6. ¹³C NMR spectrum of compound 3 (CDCl₂CDCl₂, 100 °C)



Fig. S7. ¹H NMR spectrum of compound 4 (CDCl₂CDCl₂, 100 °C)



Fig. S8. ¹³C NMR spectrum of compound 4 (CDCl₂CDCl₂, 100 °C)



Fig. S9. ¹H NMR spectrum of compound 5 (CDCl₂CDCl₂, 100 °C)



Fig. S10. ¹³C NMR spectrum of compound 5 (CDCl₂CDCl₂, 100 °C)



Fig. S11. ¹H NMR spectrum of compound 6 (CDCl₂CDCl₂, 100 °C)



Fig. S12. ¹³C NMR spectrum of compound 6 (CDCl₂CDCl₂, 100 °C)



Fig. S13. ¹H NMR spectrum of BQQ-TCDA (CDCl₂CDCl₂, 100 °C)



Fig. S14. ¹H NMR spectrum of PhC₂–BQQDI (CDCl₂CDCl₂, 100 °C)



Fig. S15. ¹H NMR spectrum of C₈–BQQDI (CDCl₂CDCl₂, 100 °C)

¹H, 400MHz, CDCl₃, 30 °C



Fig. S16. ¹H NMR spectrum of 4-Hep–BQQDI (CDCl₃, 30 °C)

¹³C, 100MHz, CDCl₃, 30 °C



Fig. S17. ¹³C NMR spectrum of 4-Hep–BQQDI (CDCl₃, 30 °C)

<u>General Synthesis of PDI Derivatives ($R = PhC_2$, C_8 and 4-Hep)</u>

R–**PDI** derivatives were synthesized in the same manner as **BQQDI** derivatives. A mixture of perylene-3,4,9,10-tetracarboxylic dianhydride and respective amine (2.4 equiv.) in quinoline (0.025 M) was heated at 160 °C for 3–22 h. After cooling to room temperature, an excess amount of MeOH was added to the reaction mixture to obtain the crude precipitates which were collected by filtration to afford the desired compounds. Further purifications were performed by sublimation or column chromatography.

N,N-Diphenethyl-3,4,9,10-perylenetetracarboxylic acid diimide (PhC2–PDI)



PhC₂–PDI

Reaction period was 16 h. Crude yield 98%. Purification was carried out by sublimation. Purification yield: 78% (dark-brown solid). m.p.: >350 °C. ¹H NMR (400 MHz, CDCl₂CDCl₂ at 100 °C): δ 8.68 (d, *J* = 8.0 Hz, 4H), 8.62 (d, *J* = 8.4 Hz, 4H), 7.36–7.27 (m, 8H), 7.20 (td, *J* = 7.2 Hz, *J* = 4.0 Hz, 2H), 4.44 (td, *J* = 7.6 Hz, *J* = 4.8 Hz, 4H), 3.07 (t, *J* = 7.8 Hz, 4H). Anal. Calcd for C₄₀H₂₆N₂O₄: C, 80.25; H, 4.38; N, 4.68, found: C, 80.48; H, 4.51; N, 4.74.

N,N-Dioctyl-3,4,9,10-perylenetetracarboxylic acid diimide (C₈–PDI)



C₈–PDI

Reaction period was 3 h. Crude yield 96%. Purification was carried out by sublimation. Purification yield: 86% (brown solid). ¹H NMR (400 MHz, CDCl₃ at 40 °C): δ 8.70 (d, *J* = 8.0 Hz, 4H), 8.63 (d, *J* = 8.0 Hz, 4H), 4.21 (t, *J* = 7.6 Hz, 4H), 1.77 (m, 4H), 1.48–1.26 (m, 20H), 0.88 (t, *J* = 7.0 Hz, 6H). Anal. Calcd for C₄₀H₄₂N₂O₄: C, 78.15; H, 6.89; N, 4.56, found: C, 78.17; H, 6.96; N, 4.63. Plate-shaped red single crystals of **C**₈–**PDI** were grown by recrystallization from nitrobenzene under ambient atmosphere. The single crystal obtained was measured at 298 K. A total of 6423 reflections were measured at the maximum 2 θ angle of 75.786°, of which 3359 were independent reflections ($R_{int} = 0.0190$). The analysis was carried out with space group of $P\overline{1}$ without disorder expediently. The crystal data are as follows: $C_{40}H_{42}N_2O_4$; FW = 614.76, Triclinic, $P\overline{1}$, a = 4.7647(2) Å, b = 8.5099(3) Å, c = 20.2613(6) Å, $\alpha = 85.332(2)^\circ$, $\beta = 89.683(2)^\circ$, $\gamma = 82.285(3)^\circ$, V = 811.39(5) Å³, Z = 1, $D_{calcd} = 1.258$ g cm⁻³, F(000) = 328.0, $\mu = 0.639$ (mm⁻¹), the refinement converged to R_1 [$I > 2\sigma(I)$] = 0.0514, w R_2 (all data) = 0.1594, GOF = 1.089 (CCDC-1938482).

N,*N*-Di(4-heptyl)-3,4,9,10-perylenetetracarboxylic acid diimide (4-Hep-PDI)(61)



4-Hep–PDI

Reaction period was 22 h. Purification was carried out by silica-gel column chromatography (CHCl₃–acetone (99:1)). Yield: 61% (red solid). ¹H NMR (400 MHz, CDCl₃ at 40 °C): δ 8.69–8.61 (m, 4H), 5.28–5.20 (m, 2H), 2.32–2.22 (m, 4H), 1.88–1.78 (m, 4H), 1.43–1.23 (m, 8H), 0.93 (t, *J* = 7.4 Hz, 6H). Anal. Calcd for C₃₈H₃₈N₂O₄: C, 77.79; H, 6.53; N, 4.77, found: C, 77.70; H, 6.54; N, 4.73.

A mixture of 2,9-Bis(2,2,3,3,4,4,4-heptafluorobutyl)-1,2,3,8,9,10-hexahydro-1,3,8,10-tetraoxoanthra[2,1,9-*def*:6,5,10-*d'e'f'*]diisoquinoline-5,12-dicarbonitrile (PDI–FCN₂[5,12-]) and 2,9-Bis(2,2,3,3,4,4,4-heptafluorobutyl)-1,2,3,8,9,10-hexahydro-1,3,8,10-tetraoxoanthra[2,1,9-*def*:6,5,10-*d'e'f'*]diisoquinoline-5,13-dicarbonitrile (PDI–FCN₂[5,13-])



PDI–**FCN**₂ was synthesized according to the literature(*30*). Since **PDI**–**FCN**₂ was composed of two CN–substituted isomers at 5,12- and 5,13-positions due to the synthetic issue, their ratio was determined by ¹H NMR to be [5,12] : [5,13] \approx 0.83 : 0.17. Dark-brown solid after sublimation. ¹H NMR (400 MHz, CDCl₃ at r.t.) ([] indicates 5,12- or 5,13- isomer): δ 9.76 (d, *J*

= 7.6 Hz, [5,13], 2H), 9.74 (d, J = 8.0 Hz, [5,12], 2H), 9.10 (s, [5,13], 2H), 9.05 (s, [5,12], 2H), 9.00 (d, J = 8.4 Hz, [5,12], 2H), 8.95 (d, J = 8.4 Hz, [5,13], 2H), 5.05 (t, J = 15.2 Hz, 4H). Anal. Calcd for C₃₄H₁₀F₁₄N₄O₄: C, 50.76; H, 1.25; N, 6.96, found: C, 50.88; H, 1.41; N, 7.17. Plateshaped brown single crystals of **PDI**–**FCN**₂ were grown by physical vapor transport technique under ambient atmosphere. The single crystal obtained was measured at 298 K. A total of 7133 reflections were measured at the maximum 2θ angle of 68.250°, of which 2609 were independent reflections ($R_{int} = 0.0983$). The analysis was carried out with space group of $P\overline{1}$ without disorder expediently. The crystal data are as follows: C₃₄H₁₀F₁₄N₄O₄; FW = 804.46, Triclinic, $P\overline{1}$, a =5.2871(19) Å, b = 7.5621(19) Å, c = 19.404(5) Å, $a = 90.49(2)^\circ$, $\beta = 94.95(3)^\circ$, $\gamma = 105.14(3)^\circ$, V =745.7(4) Å³, Z = 1, $D_{calcd} = 1.791$ g cm⁻³, F(000) = 400.0, $\mu = 1.602$ (mm⁻¹), the refinement converged to R_1 [$I > 2\sigma(I)$] = 0.1616, w R_2 (all data) = 0.4586, GOF = 1.222 (CCDC-1938639).

Section S2. Results of DFT calculations

Table S1.	Intermolecular interaction energy (kcal/mol) of BQQDIs and PDIs based on the
crystal str	cture with substitution of side chains with hydrogen.

	C ₈ –BQQDI	PhC ₂ –BQQDI	C ₈ –PDI	PDI-FCN ₂
M06-2X/6-31++G**	-7.97	-7.36	-5.40	-3.37
MP2/6-31++G**	-8.62	-8.22	-5.78	-4.19

Table S2. Intermolecular interaction energy (kcal/mol) of BQQDIs and PDIs based on the crystal structure.

	C ₈ –BQQDI	PhC ₂ –BQQDI	C ₈ –PDI
M06-2X/6-31++G**	-7.79	-8.48	-5.16

Section S3. Details of instrumental characterizations

Compounds	$E_{0/-1}$ (V)	$E_{-1/-2}$ (V)	$E_{\text{LUMO}}(V)$
4-Hep-BQQDI	-0.69	-0.97	-4.11
4-Hep-PDI	-1.02	-1.26	-3.78 (^{lit.} -3.84(61))
PDI-FCN ₂	-0.53	-0.86	-4.27

Table S3. Summary of cyclic voltammetry.

* $E_{0/-1}$ and $E_{-1/-2}$ attributes to the half-wave reduction potentials corresponding to the first- and second-redox reactions, respectively (vs. Fc/Fc⁺). E_{LUMO} (vs. vacuum level) is estimated from $E_{0/-1}$: $E_{LUMO} = -4.8 - E_{0/-1}$.

Table S4. Summary of TG-DTA.

Compounds	T_{onset} (°C)	<i>T</i> ^{95%} (°C)
PhC ₂ -BQQDI	354	421
C ₈ –BQQDI	303	377
PhC ₂ –PDI	370	429
C ₈ –PDI	306	402
PDI-FCN ₂	279	342

[†] T_{onset} : the onset temperature of weight loss; T_{95} : the temperature at which the weight fraction is 95%.



Fig. S18. TG–DTA profiles. Measurements were carried out under a nitrogen stream at the heating rate of 1 K min⁻¹. (A) PhC₂–BQQDI, (B) C₈–BQQDI, (C) PhC₂–PDI, (D) C₈–PDI and (E) PDI–FCN₂. Red and blue lines represent the residual weight fraction (left axis) and the heat flow (right axis), respectively. Insets: magnifications of the first inflection point of the weight fraction.

Photophysical properties

Intense optical absorptions in the UV-vis region are characteristic features of BQQDI and **PDI** derivatives. Fig. S2A displays a photograph of benzonitrile solutions $(4 \times 10^{-5} \text{ M})$ of 4-Hep-BQQDI and 4-Hep-PDI. While 4-Hep-PDI results in an orange solution, 4-Hep-BQQDI affords a yellow solution. The difference in the color can be explained by the UVvis absorption spectra shown in Fig. S2B. 4-Hep-BQQDI showed absorption peaks at 520, 485, 455 and 430 nm. On the other hand, 4-Hep–PDI showed absorption peaks at 529, 492, 461 and 433 (shoulder) nm. Thus, **4-Hep–BQQDI** exhibits quite similar photophysical properties to those of **4-Hep–PDI**, except the red-shift features which is principally due to a reduced molecular symmetry in **BOODI** structure compared with that of **PDI**. A slightly smaller absorbance in 4-Hep–BOODI is also attributable to the same origin. In addition, we compared optical absorption properties of PhC2-BQQDI in a benzonitrile solution and in a vacuumdeposited solid film (Figs. S2C-D). Although the solubility was much poorer than that of 4-Hep-BOODI, PhC₂-BOODI in a solution state exhibited an UV-vis absorption spectrum identical to **BQQDI** core. In the solid film, the absorption bands were broadened, and a new band appeared at red-shifted region, which was attributable to the aggregation in the solid state. As well, optical absorption spectra were measured and used for investigating materials stabilities of PhC2-BOODI, C8-BOODI and PhC2-PDI under ambient atmosphere (Figs. S19C and S19E-G). Interestingly, PhC2-BQQDI and C8-BQQDI films showed remarkably different UVvis absorption spectra, while they were composed of the same π -electron system. This might be due to different aggregated structures as likely seen in their single crystal structure determinations. UV-vis absorption spectrum of the vacuum-deposited PhC2-PDI film is likely a mixture of the amorphous and crystalline phases(62). With time spent, the absorbance at 607 nm became increased while that at 492 nm decreased, which indicated that the amorphous phase partly turned into crystalline phase.



Fig. S19. Photophysical characterization. (A) Photograph of 4-Hep–BQQDI (left) and 4-Hep–PDI (right) solutions in benzonitrile (4×10^{-5} M). (B) UV-vis absorption spectra of 4-Hep–BQQDI (blue) and 4-Hep–PDI (magenta) in 4×10^{-6} M benzonitrile solutions. (C) Photograph of vacuum-deposited thin films (100 nm-thick) on quartz substrates under room light. Left: PhC₂–BQQDI, Center: C₈–BQQDI, Right: PhC₂–PDI. (D) Comparison of UV-vis absorption spectra of PhC₂–BQQDI in a benzonitrile solution (10^{-6} M; green) and in a vacuum-deposited thin film state (100 nm thick; pink). (E-G), Time-dependent UV-vis absorption spectra of vacuum-deposited 100 nm-thick films of PhC₂–BQQDI, C₈–BQQDI and PhC₂–PDI, respectively.

Section S4. Characterization of vacuum-deposited PC-TFTs

Device evaluations of vacuum-deposited PC-TFTs

Analyses of the vacuum-deposited PC-TFTs in an argon-filled glovebox and in an air are summarized in Table S5 and Fig. S20.

Table S5. Summary of vacuum-deposited PC-TFT performances in an argon-filled glovebox and in an air.

		in glovebox		in air		
Compounds	$\begin{array}{c} \text{Max. } \mu_{\text{e}} \\ [\text{cm}^2 \text{V}^{-1} \\ \text{s}^{-1}] \end{array}$	$\begin{array}{c} \operatorname{Avg.} \mu_{\rm e} \\ [\operatorname{cm}^2 \operatorname{V}^{-1} \\ \operatorname{s}^{-1}] \end{array}$	$V_{ m th}$ [V]	$\begin{array}{c} {\rm Max.} \ \mu_{\rm e} \\ [{\rm cm}^2 \\ {\rm V}^{-1} \ {\rm s}^{-1}] \end{array}$	$\begin{array}{c} \operatorname{Avg.} \mu_{\mathrm{e}} \\ [\operatorname{cm}^2 \mathrm{V}^{-1} \\ \mathrm{s}^{-1}] \end{array}$	$V_{ m th}$ [V]
PhC ₂ –BQQDI	0.55	0.49(2)	-0.6(6)	0.49	0.43(3)	1.7(4)
C ₈ –BQQDI	0.29	0.27(2)	4.2(1)	0.34	0.33(1)	5.3(3)
PhC ₂ –PDI	0.01	0.098(1)	5.3(2)	2.5×10^{-2}	0.021(1)	22.2(7)
C ₈ –PDI	0.44	0.41(4)	15.5(6)	0.50	0.28(15)	25(12)
PDI-FCN ₂	0.38	0.33(7)	-4.0(5)	0.31	0.27(3)	-5.0(7)

[†]Data are extracted from 2–8 devices. Devices were thermally annealed at 100 °C for 2 h in a glovebox before measurement. $L = 200 \ \mu\text{m}$, $W = 1000 \ \mu\text{m}$. $V_D = 50 \ \text{V}$.



Fig. S20. Properties and durability of vacuum-deposited PC-TFTs in ambient air. TFT structures: Sub = G = n⁺⁺-Si; Ins = SiO₂ (200 nm) + DTS-SAM, $C_i = 17.3 \text{ nF cm}^{-2}$; S/D = Au (60 nm). Transfer characteristics of (A) C₈–BQQDI and (B) C₈–PDI with $V_D = 50$ V, under Ar (red) and air (blue). Black dashed lines are the fits to $|I_D|^{1/2}$ – V_G plots to estimate μ_e and V_{th} . (C) PC-TFT μ_e values as functions of V_g . The μ_e values determined from the plateaus in the plots for PhC2–BQQDI, C8–BQQDI, PhC2–PDI, and PDI–FCN₂ confirm the validity of the mobility assessment. In contrast, the peak-shaped plot for C8–PDI indicates that μ_e was overestimated(63). (D-F) Output curves for PhC2–BQQDI, C8–BQQDI, and PDI–FCN₂, respectively, in air in the V_G range of 0–50 V with 10 V steps. (G-I) Changes in the transfer properties of TFTs incorporating PhC2–BQQDI, C8–BQQDI, and PDI–FCN₂ during storage as determined under ambient conditions for approximately one month. While a gradual negative shift in V_{th} is observed for PhC2–BQQDI and C8–BQQDI, a significant positive change in V_{th}

accompanied by an increase in the off current is evident for **PDI–FCN2**. (**J**,**K**) μ_{e} (normalized relative to the initial value, μ_{0}) and V_{th} shifts (ΔV_{th}) as functions of time, respectively.

Surface morphology of vacuum-deposited PC films

Surface morphologies of the PC thin films were studied by AFM. Comparing PhC₂–BQQDI and C₈–BQQDI, fiber-like grains can be seen for the former, while plate-like grains are predominant for the latter. The morphology of PhC₂–PDI is similar to that of PhC₂–BQQDI, while the grain size seems smaller. C8–PDI formed relatively larger plate-like grains, suitable for charge-carrier transport. PDI–FCN₂ formed plate-like grains as seen in C₈–BQQDI, while the grains are small.



Fig. S21. AFM images of vacuum-deposited thin films. 40 nm-thick films were deposited on DTS-SAM modified SiO₂. (A) PhC₂–BQQDI, (B) C₈–BQQDI, (C) PhC₂–PDI, (D) C₈–PDI and (E) PDI–FCN₂.

Thin-film X-ray diffraction on vacuum-deposited PC thin films

Thin-film out-of-plane X-ray diffraction profiles are shown in Fig. S22. PhC₂–BQQDI showed diffraction peaks at 2θ of 4.00° (d = 22.0 Å), 4.82° (18.3 Å) and 15.92 (5.6 Å), 19.81 (4.5 Å) and 24.9° (3.6 Å), suggesting that PC film of PhC₂–BQQDI is composed of mixture, one of which is consistent with the single crystal structural data. For C₈–BQQDI, the diffractions were observed at 2θ of 3.31° (26.7 Å), 10.02° (8.8 Å), 13.28° (6.6 Å), 16.58 (5.3 Å) and 23.32° (3.8 Å), being consistent with the single crystal structural data. PhC₂–PDI showed the diffractions at $2\theta = 5.75^{\circ}$ (15.3 Å), being consistent with the literature(64). For C₈–PDI, the observed diffractions were at 2θ of 4.34° (20.3 Å), 8.69° (10.2 Å), 13.00° (6.8 Å), 17.32° (5.1 Å), 21.75° (4.1 Å) and 26.13° (3.4 Å), indicating a high degree of layer-by-layer structural order and being consistent with the literature(65). As for PDI–FCN₂, the diffraction peaks were observed at 8.81° (10.0 Å) and 13.32° (6.6 Å), where the latter was consistent with the literature while the former had not been visible.



Fig. S22. Out-of-plane X-ray diffraction profiles of vacuum-deposited thin films. (A) **PhC2–BQQDI**, (B) **Cs–BQQDI**, (C) **PhC2–PDI**, (D) **Cs–PDI** and (E) **PDI–FCN2** vacuum-deposited 40 nm-thick PC films on DTS-SAM modified SiO₂ at 140 °C.



Section S5. Characterization of solution-deposited SC-TFTs

Fig. S23. SC-TFT characterizations under ambient conditions. TFT structures: Sub = G = n^{++} -Si, Ins = SiO₂ (200 nm) + AL-X601 (25–60 nm), $C_i = 12.5-15$ nF cm⁻²; S/D = Au (40 nm). Output curves, transfer characteristics, and effects of V_G on μ_e for (A-C) PhC₂–BQQDI, (D-F) C₈–BQQDI, and (G-I) PDI–FCN₂, respectively, having L/W (μ m/ μ m) ratios of 190/136 (1.4), 91/75 (1.2) and 94/660 (0.14). Transfer characteristics determined at V_D values of 75, 75, and 50 V, respectively. The black dashed lines in the transfer characteristic plots are fits to $|I_D|^{1/2}-V_G$ used to estimate μ_e and V_{th} .



Fig. S24. Determination of the direction of the crystallographic axis. (A) PhC₂–BQQDI, (B) C₈–BQQDI, and (C) PDI–FCN₂. For each row, the left panel schematically illustrates the experimental setup (with "IP" indicating the imaging plate), the central panel shows a polarized optical microscopy image of the SC film (with the crystallographic axis indicated), and the right panel shows the X-ray oscillation image acquired along the out-of-plane direction (*i.e.*, the X-ray was almost parallel to the substrate) at room temperature. The oscillation images were obtained in the ω -scan mode with $\varphi = 0^{\circ}$ and $\chi = 0^{\circ}$ for PhC₂–BQQDI, $\chi = 20^{\circ}$ for C₈–BQQDI, and $\chi = 25^{\circ}$ for PDI–FCN₂. From these analyses, the crystallographic axes along the channel of each TFT, which also correspond to the direction of thin-film crystal growth, were determined to be approximately [010], [1-10], and [110] for the PhC₂–BQQDI, C₈–BQQDI, and PDI–FCN₂, respectively.

<u>PhC₂-BQQDI with β -PTS-SAM/SiO₂ dielectric layer: sample-to-sample variation</u>

Although the AL-X601 polymer insulator surface has exhibited well wettability and electron mobility up to 3.0 cm² V⁻¹ s⁻¹ for **PhC₂–BQQDI** SC-TFT, sample-to-sample mobility variation likely due to the quality of formed AL-X601 layer has been observed. On the other hand, **PhC₂-BQQDI** SC-TFTs can also be fabricated using silicon wafer substrates with β -PTS-SAM modified SiO₂ layer, where the variation of surface condition should be suppressed compared with polymer insulator. Therefore, the device anisotropy and sample-to-sample variation of the **PhC₂–BQQDI** SC-TFT were evaluated using the β –PTS-SAM functionalized SiO₂ dielectric surface. Representative output and transfer ($V_D = 50$ V) characteristics are shown in Fig. S25A and 25B, respectively, with $\mu_e = 2.23 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$. 15 SC-TFTs studied herein revealed high carrier mobilities of $\mu_e = 2.2\pm0.3 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ (the maximum $\mu_e = 2.57 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$) and $V_{\text{th}} = 4.3 \pm 1.6$ V. The intrinsic nature of small sample-to-sample variation and high μ_{e} is verified in the superimposed V_G-dependent μ_e curves (Fig. S25C). In Fig. S25B, the fit to $|I_D|^{1/2}$ (black broken line) and the slope of an electrically equivalent ideal TFT (magenta solid line) are shown, estimating the reliability factor (r)(41) of 87% in this device. The r for all 15 TFTs are summarized in Fig. S25D, where the r is ranging from 81% to 91%. Hence, the effective mobility (μ_{eff}), which is estimated by $\mu_e \times r$, has a distribution between 1.64 and 2.16 cm² V⁻¹ s⁻¹ as shown in Fig. S25E.



Fig. S25. SC-TFT performances of PhC₂–BQQDI on a β –PTS-SAM modified silicon wafer. TFT structure: **G** = n⁺⁺-Si, **Ins** = SiO₂ (200 nm) + β –PTS-SAM; C_i = 17.3 nF cm⁻², **OSC** =

PhC₂–BQQDI, **S/D** = Au (40 nm). Representative (**A**) output and (**B**) transfer ($V_D = 50$ V) curves under ambient condition. Black broken line is the fit to $|I_D|^{1/2} - V_G$ plot. Magenta solid line represents the slope of an electrically equivalent ideal TFT(41), indicating the reliability factor (*r*) of 87%. (**C**) μ_e plotted as a function of V_G of 15 SC-TFTs. (**D-E**) Statics of *r* and the effective mobility (μ_{eff} ; $\mu_{eff} = \mu_e \times r$) corresponding to the 15 SC-TFTs.

Section S6. Gated Hall effect measurement on PhC₂–BQQDI

The SC-TFT device of **PhC₂–BQQDI** used for gated Hall effect measurements was fabricated in the same manner as the SC-TFT units described above, except for the use of a metal shadow mask having a Hall bar geometry. An n⁺⁺-doped silicon wafer with a 200 nm-thick SiO₂ layer and coated with AL-X601 (~31 nm) was used as the substrate. The laser etching allowed the formation of a well-defined Hall bar architecture with L/W = 235/48 and a longitudinal interprobe (V₁–V₂ and V₃–V₄ shown in Fig. S26A) distance (*D*) of 115 µm. The gated Hall effect measurements were performed using a helium gas-exchanged cryostat with a superconducting magnet at 300 K. While the magnetic field (*B*) was swept between 12 and –12 T, transistor operation was conducted using a Keithley 2634B source meter with $V_D = 1.5$ V. Longitudinal (V_{long}) and transverse (V_{trans}) electromotive forces were monitored with Agilent 34410A 6¹/₂ digit multimeters.

Sheet conductivity (σ_{sh}) at each V_G was calculated by $I_D D / V_{long} W$, shown in Fig. S26B. As a result, the four-terminal field-effect mobility (μ_{4T}) was estimated to be 4.1 cm² V⁻¹ s⁻¹ with $V_{th} = 13.0$ V.

Hall coefficient (R_{Hall}) was estimated from V_{trans} by using the following equation:

$$V_{\text{trans}} - V_{B=0} = R_{\text{Hall}} I_{\text{D}} B,$$

where $V_{B=0}$ stands for the V_{trans} when B = 0 T. $(R_{\text{Hall}})^{-1}$ was plotted as a function of V_{G} (Fig. S26C), in which the negative sign of R_{Hall} was consistent with the *n*-type TFT operation. In addition, the Hall carrier density (n_{Hall}) was estimated from the following relationship:

$$n_{\text{Hall}} = \frac{1}{qR_{\text{Hall}}},$$

where *q* is the elementary charge. $(R_{\text{Hall}})^{-1}$ dependent on V_{G} was thus transformed into V_{G} dependent n_{Hall} , as shown in Fig. S26D. Simultaneously, the sheet carrier density accumulated by field effect (n_{FE}) , which was calculated from C_i $(n_{\text{FE}} = C_i|V_{\text{G}} - V_{\text{th}}|/q)$, was also plotted as a function of $|V_{\text{G}} - V_{\text{th}}|$ (blue solid line). Here, the slopes of calculated n_{FE} and the linear fit to n_{Hall} are directly compared due to the ideal relationship $n = (qR_{\text{Hall}})^{-1}$, confirming a well-consistent feature with the so-called coherence factor determined by $n_{\text{FE}}/n_{\text{Hall}}$ of 1.0. The coherence factor of 1 means the entire field-accumulated charge carriers are coherently mobile, which indicates an ideal band-like electron transport at the semiconductor–dielectric interface. The Hall mobility (μ_{Hall}) estimated by $R_{\text{Hall}} \cdot \sigma_{\text{sh}}$ was 4.0 cm² V⁻¹ s⁻¹, being consistent with the $\mu_{4\text{T}}$ and the coherence factor of 1.



Fig. S26. Gated Hall effect measurement on PhC₂–**BQQDI SC-TFT.** TFT structure: **Sub** = **G** = n⁺⁺-Si, **Ins** = SiO₂ (200 nm) + AL-X601 (31 nm), C_i = 12.8 nF cm⁻²; **S/D** = Au (40 nm). (**A**) Optical microscope image of the channel region. (**B**) *V*_G-dependent σ_{sh} estimated by the 4terminal geometry. Black dashed line is the linear fit, affording the μ_{4T} of 4.13 cm² V⁻¹ s⁻¹. (**C**) $(R_{Hall})^{-1}$ as a function of $|V_G - V_{th}|$ at 300 K (V_{th} = 13.0 V). Black dashed line is the linear fit. (**D**) Sheet carrier densities as a function of $|V_G - V_{th}|$. Black broken line is the linear fit to n_{Hall} . Blue solid line represents the theoretical sheet carrier density accumulated by the field effect (n_{FE}), calculated as $n_{FE} = C_i(V_G - V_{th})/q$.

Section S7. Thermal stress tests on PhC2-BQQDI SC-TFTs on different substrates



Fig. S27. Thermal stress test data for PhC₂–BQQDI SC-TFTs using different substrates. (A, C) and (B, D) provide the transfer curves obtained at various T_{anneal} values and the extracted μ_e/μ_{100} and $\Delta V_{th}/V_{th,100}$ values, respectively, with different substrates. TFT structures for A-B: Sub = G = n⁺⁺-Si, Ins = SiO₂ (200 nm) + AL-X601 (45 nm), $C_i = 13.0$ nF cm⁻²; S/D = Au (40 nm). TFT structures for C-D: Sub = G = n⁺⁺-Si, Ins = SiO₂ (100 nm) + β –PTS-SAM, $C_i = 34.5$ nF cm⁻²; S/D = Au (40 nm).

Section S8. Details of MD simulation and estimated dynamic fluctuations

	PhC ₂ -BQQDI	C ₈ –BQQDI	PDI-FCN ₂	PhC ₂ -PDI
Num. of molecules	672	560	576	672
Temperature (K)	296	300	298	298
Crystal syst.	monoclinic	triclinic	triclinic	monoclinic
Space group	$P2_{1}/n$	$P\overline{1}$	$P\overline{1}$	$P2_{1}/c$
a (nm)	7.705	4.710	5.287	4.737
b (nm)	5.022	6.671	7.562	32.450
<i>c</i> (nm)	35.810	26.091	19.404	9.507
α (°)	90.00	87.69	90.49	90.00
β (°)	92.47	89.71	94.95	100.27
γ (°)	90.00	74.79	105.14	90.00

Table S6. Number of molecules, temperature, and MD cell size and shape.



Fig. S28. Colour-coded *B*-factor distributions (unit: $Å^2 s^{-1}$) obtained from the trajectories during the last 20 ns of 100 ns MD runs. (A-B) PhC₂–BQQDI (at 296 and 373 K, respectively). and (C) PDI–FCN₂ (at 298 K). The initial structures for both were constructed from single-crystal structures obtained experimentally.



Fig. S29. Superimposed snapshots of the representative molecules after 100 ns and 100.1 ns of MD simulations. (A) Top and side views for PhC₂–BQQDI are shown as sticks colored by pale pink for 100 ns and magenta for 100.1 ns. (B) Top and side views for PDI–FCN₂ are shown as cyan sticks for 100 ns and as azure sticks for 100.1 ns.

Dynamical fluctuation of transfer integrals

By using the atomic coordinates at the 100 ns acquired by the MD simulation, transfer integrals t_{1-3} corresponding to Figs. 3E-F in the main text were calculated over 141–209 dimers.



Fig. S30. Simulated dynamical fluctuation of transfer integral. Transfer integrals t_{1-3} , corresponding to Figs. 3E-F, of PhC₂–BQQDI and PDI–FCN₂ at room temperature based on the atomic coordinates acquired by single crystal structure-based MD simulation.

Section S9. Evaluation of CMOS device

We evaluated a single CMOS inverter performance, confirming the full rail-to-rail swing (Fig. S31C), the voltage gain over 100 (Fig. S31C), and the standby power on the order of nano-watt (Fig. S31D).



Fig. S31. CMOS inverters. (**A**) Polarized optical microscopy image of the photolithographically fabricated CMOS inverters. (**B**) Schematic circuit diagram of a single CMOS inverter. (**C**) Voltage transfer curves (corresponding to those shown in Fig. 6G) and corresponding voltage gains. (**D**) Through current (I_{through}) during the inverter operation.

Captions for Movie S1 to S4:

Movie S1. MD simulation of PhC₂–BQQDI at 296 K viewed from the side of the brickwork assembly during 80–100 ns run. Representative seven molecules are displayed, and the phenethyl moieties are omitted for clarity.

Movie S2. MD simulation of PhC₂–BQQDI at 296 K viewed from the front of the brickwork assembly during 80–100 ns run. Representative seven molecules are displayed, and the phenethyl moieties are omitted for clarity.

Movie S3. MD simulation of PDI–FCN₂ at 298 K viewed from the side of the brickwork assembly during 80–100 ns run. Representative seven molecules are displayed, and the 1*H*,1*H*-heptafluorobutyl moieties are omitted for clarity.

Movie S4. MD simulation of PDI–FCN₂ at 298 K viewed from the front of the brickwork assembly during 80–100 ns run. Representative seven molecules are displayed, and the 1*H*,1*H*-heptafluorobutyl moieties are omitted for clarity.

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