

Supplementary Materials for

Ultrastrong lightweight compositionally complex steels via dual-nanoprecipitation

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Table S1
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Legends for movie S1

Other Supplementary Material for this manuscript includes the following:

(available at advances.sciencemag.org/cgi/content/full/6/46/eaba9543/DC1)

Movie S1

Table S1. Chemical composition of the model CCS according to wet-chemical analysis.

	Fe	Mn	Al	Ni	C
wt. %	55.23	29.6	8.75	5.22	1.20
at. %	48.34	26.38	15.89	4.49	4.90

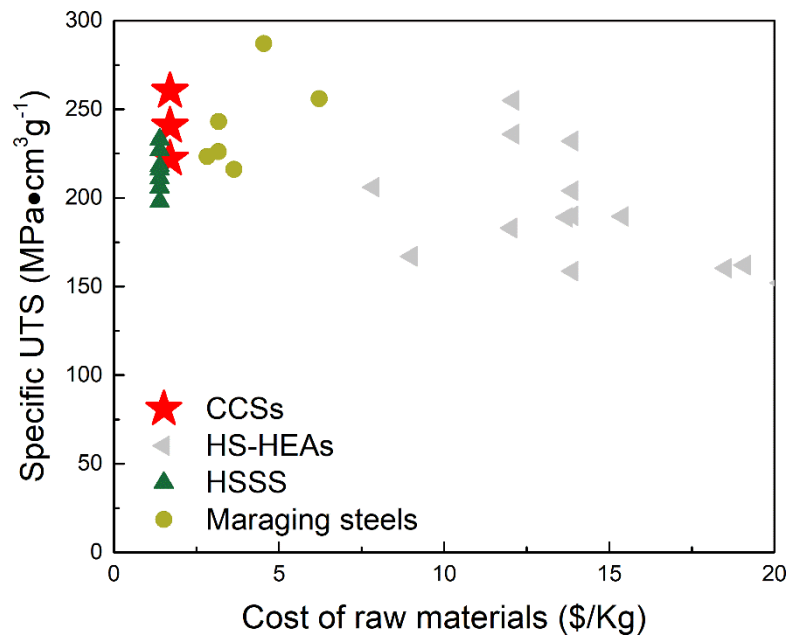


Fig. S1. Cost of raw materials versus specific UTS. Our CCSs, high-strength (HS) HEAs (21, 24-33), HSSs (6, 7), and maraging steels (22, 23) are shown for comparison.

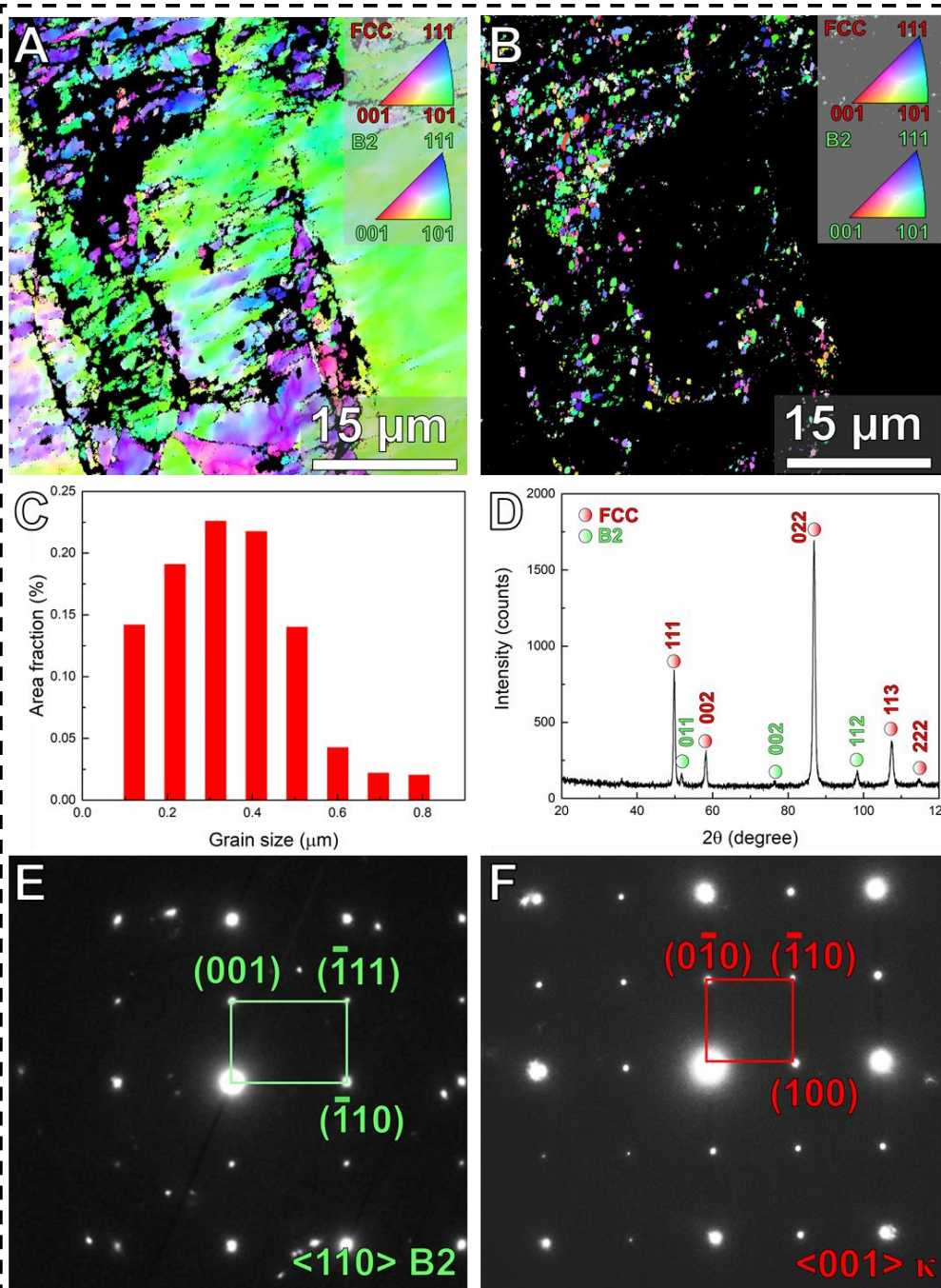
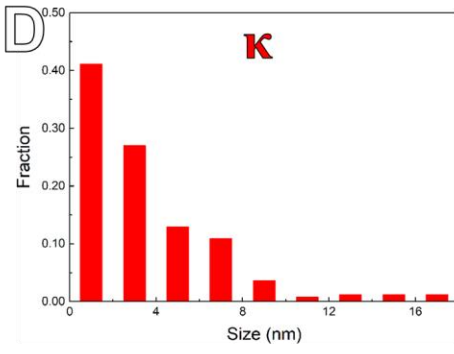
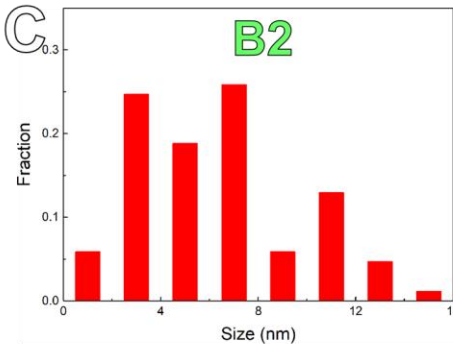
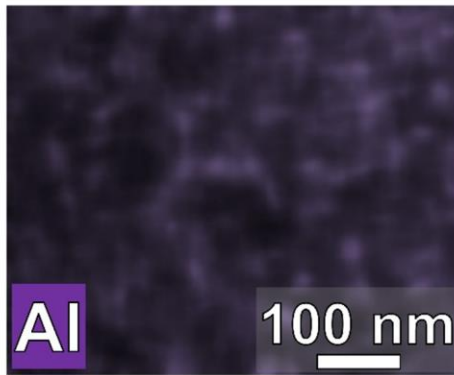
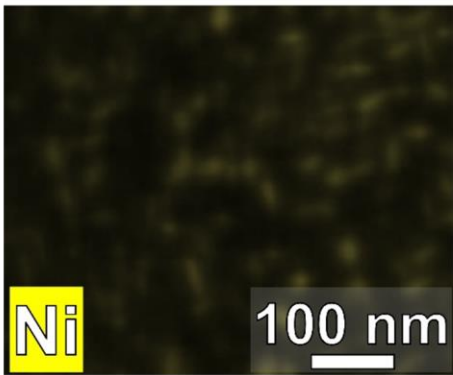
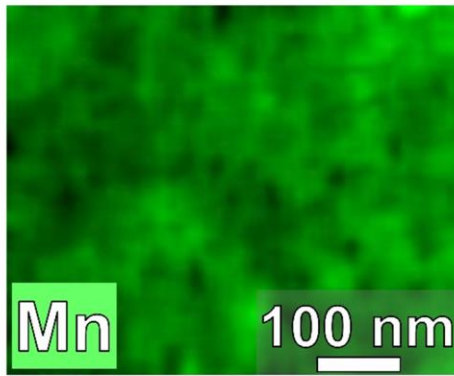
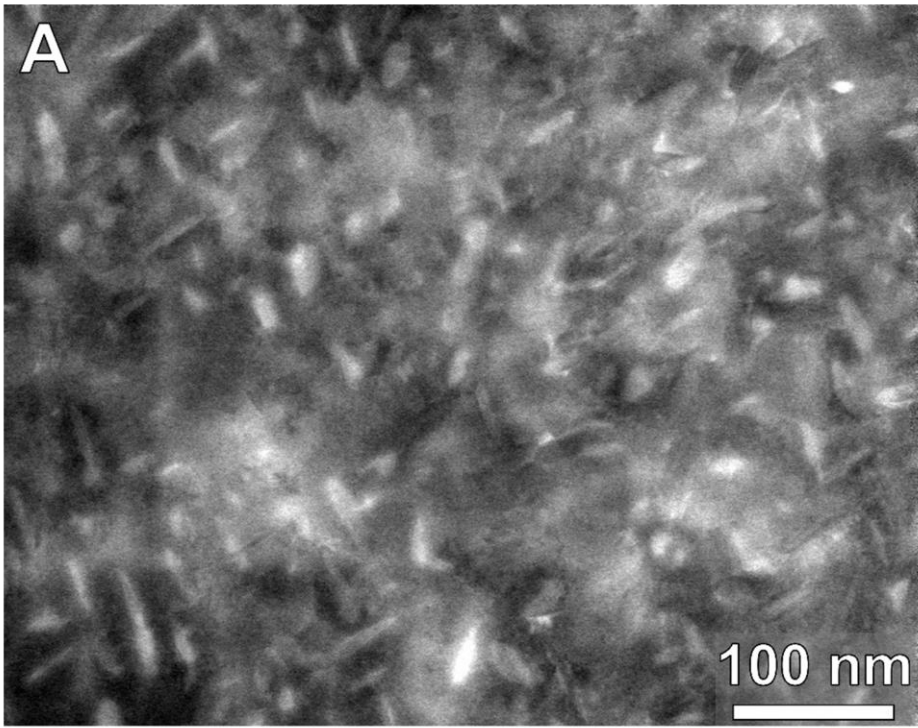


Fig. S2. Microstructures of the model CCS annealed at 800 °C. IPF maps showing the (A) non-recrystallized area and (B) recrystallized area. (C) The grain size distribution in the recrystallized area. The area fraction of the recrystallized area is 8 %, and the corresponding average grain size is 0.35 μm in this area. (D) XRD pattern. The peaks of FCC structure contain the information from both austenite matrix and κ-carbides, since the XRD analysis cannot differentiate between them. SAED patterns of B2 (inset in Figure 1D) and κ-carbide (inset in Figure 1G) are indexed in (E) and (F), respectively. Due to the very small size of B2 particle, austenite matrix was also included when performing the SAED, causing the appearance of satellite spots in (E).



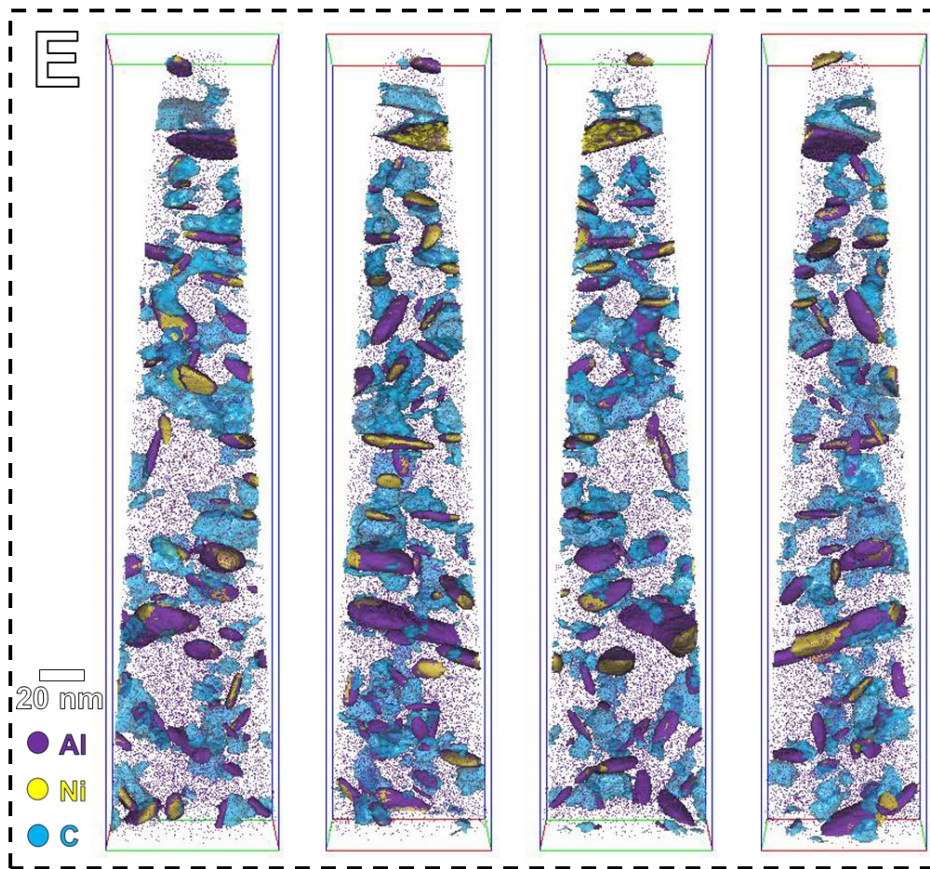


Fig. S3. Characterization of dual-nanoprecipitation in the model CCS annealed at 800 °C. (A) HAADF-STEM image. (B) EDS maps. The size distributions of very fine (C) B2 particles and (D) κ -carbides. The average sizes of B2 and κ -carbide are ~6 nm and ~4 nm, respectively, which have been determined by APT analysis. (E) Reconstructed maps from atom probe tomography probing of a typical tip shown in Fig. 2E, viewed from four different directions (highlighted by iso-composition surfaces of 30 at. % Ni, 30 at. % Al, and 10 at. % C), which clearly reveal the co-precipitation of B2 and κ -carbide with adjacent positions, referred to here as backpack topology.

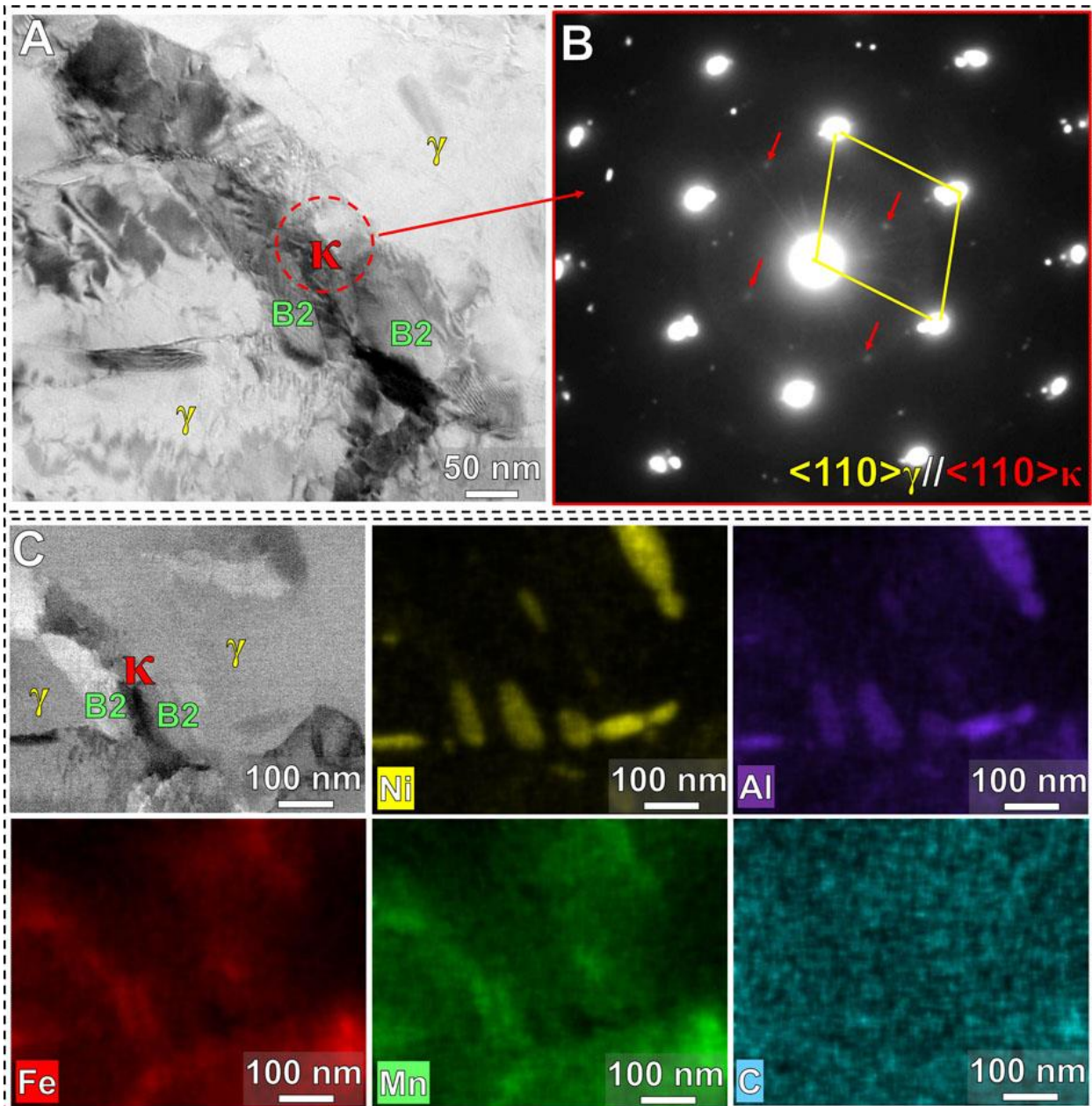


Fig. S4. Dual-nanoprecipitation in the model CCS annealed at 900 °C. (A) Bright-field TEM image and (B) SAED pattern. The formation of κ -carbide is revealed by the SAED pattern in (B), and the existence of B2 phase is identified by the EDS maps in (C). The co-precipitation of κ -carbide and B2 phase still exists even when annealed at higher temperatures such as 900°C.

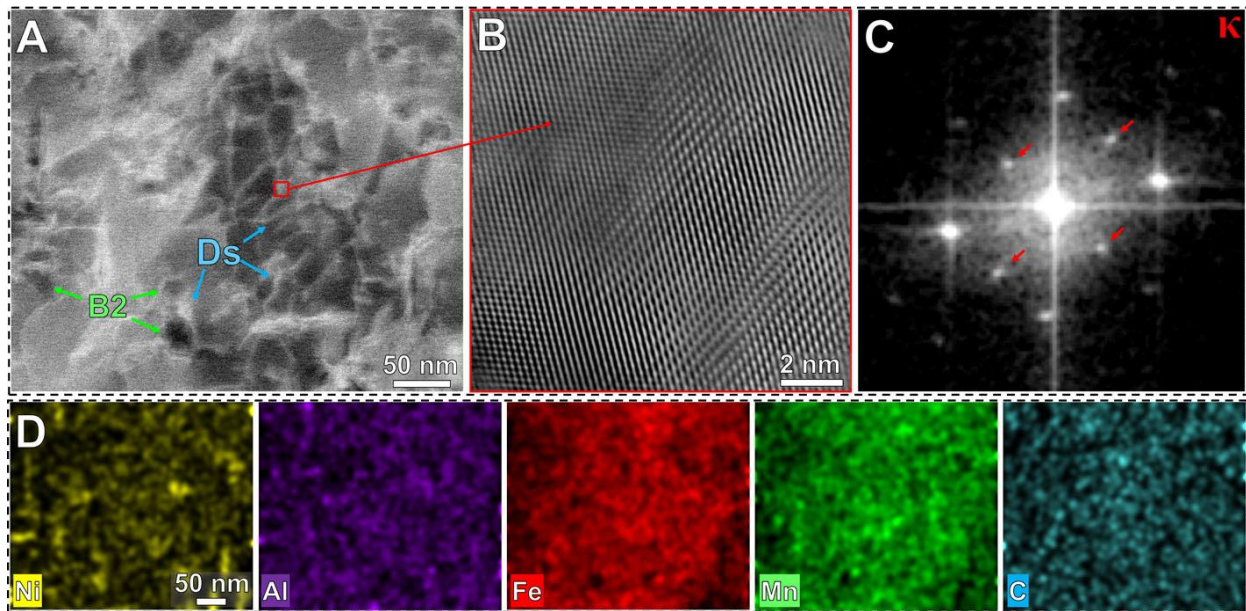


Fig. S5. Deformation microstructures of the model CCS annealed at 800 °C at 7 % strain. (A) LAADF-STEM image showing the dislocations (Ds, in bright contrast). **(B)** HR HAADF-STEM image of the marked region in (A). **(C)** The corresponding FFT pattern for (B) showing the existence of κ -carbides. **(D)** EDS maps for (A) showing the existence of B2 particles. Much more dislocations were seen at a higher strain. Again, B2 particles pinned the dislocation movements, while κ -carbides were cut through by dislocations.

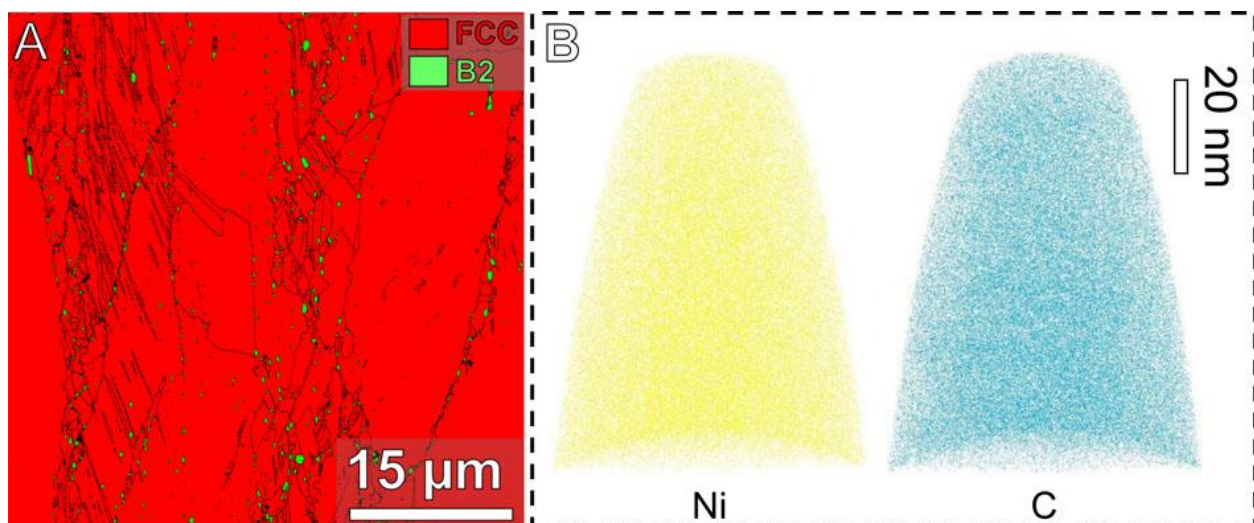


Fig. S6. Microstructure and nanoscale elemental distribution of the hot-rolled CCS. (A) The EBSD phase map shows that the B2 particles were formed during hot-rolling. The black lines in (A) represent the high angle grain boundaries. (B) Reconstructed atomic maps of Ni and C based on APT analysis show that the very fine nano-precipitation did not occur during the hot-rolling process.

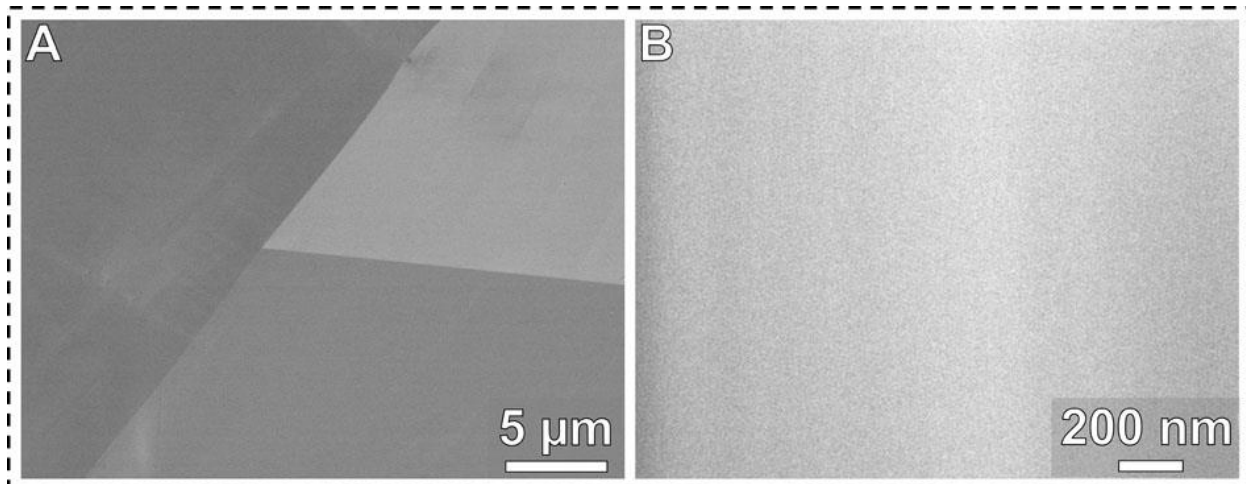


Fig. S7. Microstructures of the model CCS homogenized at 1200 °C for 1 h. (A) BSE image and (B) HAADF-STEM image show that nearly all precipitates dissolve into the austenite matrix after homogenization.

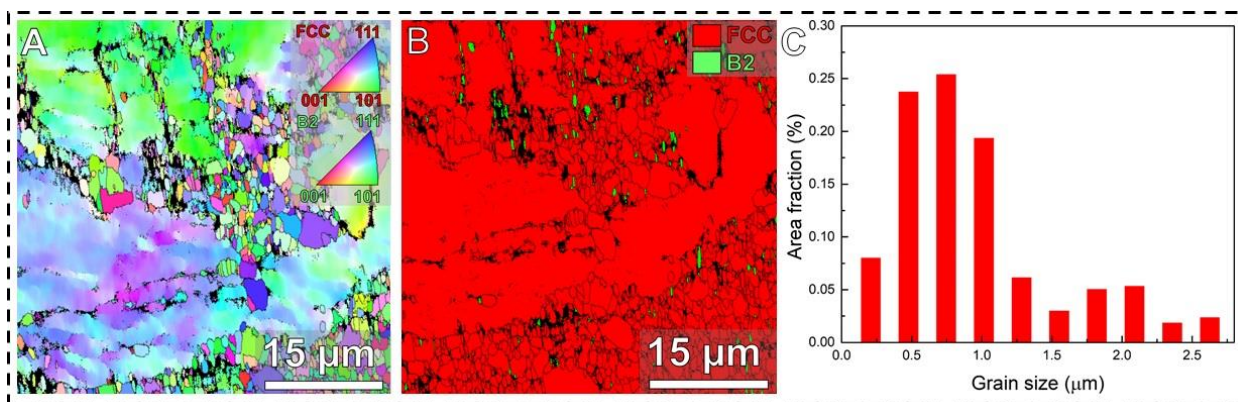


Fig. S8. EBSD maps of the model CCS annealed at 850 °C. (A) IPF map and (B) phase map, and (C) the grain size distribution in the recrystallized area. The black lines in (B) represent the high angle grain boundaries. The area fraction of the recrystallized area is 28 %, and the corresponding average grain size is 1.0 μm in this area.

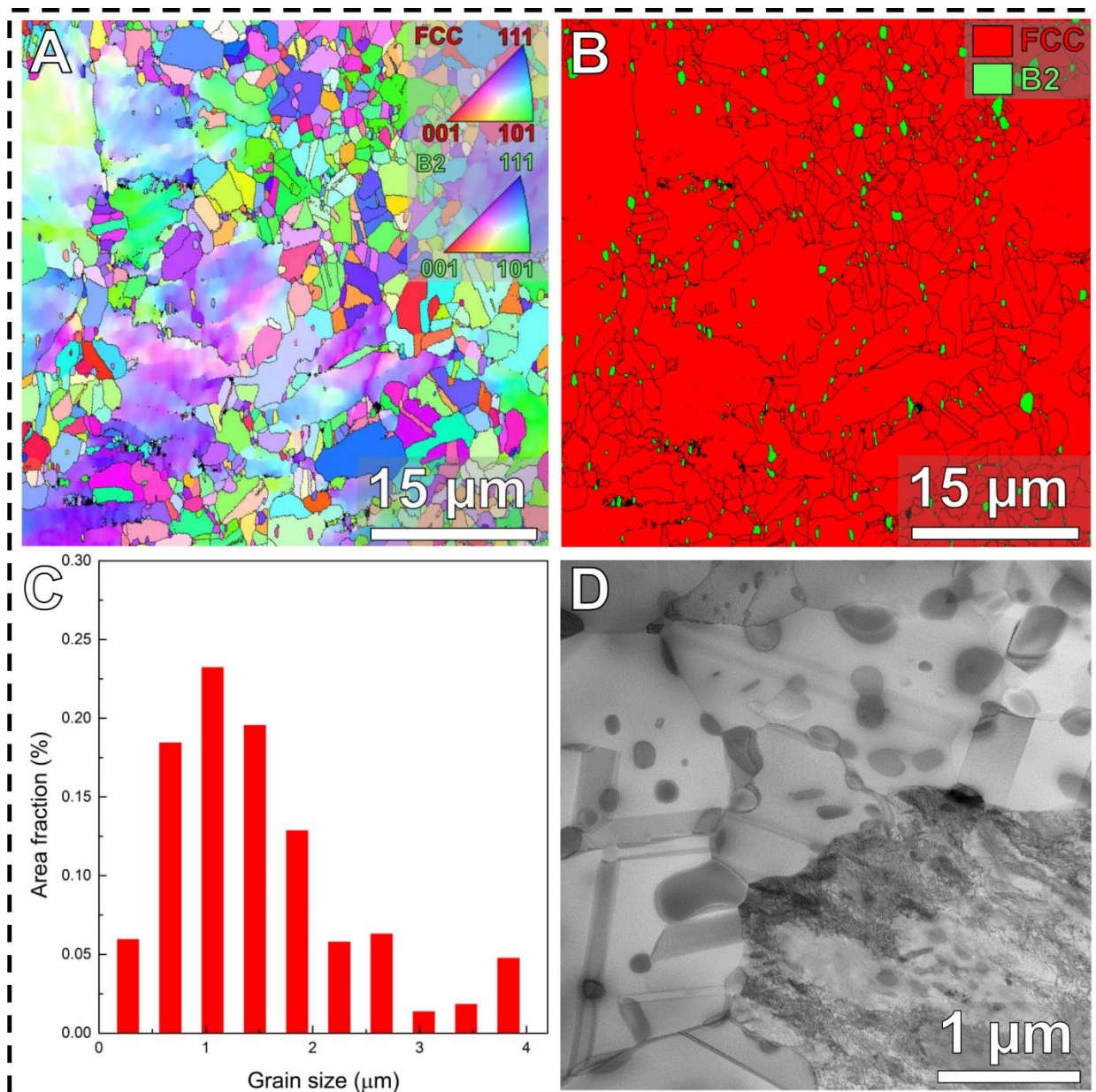


Fig. S9. EBSD maps and STEM image of the model CCS annealed at 900 °C. (A) IPF map, (B) phase map, (C) the grain size distribution in the recrystallized area, and (D) bright-field STEM image. The black lines in (A) and (B) represent the high angle grain boundaries. The area fraction of the recrystallized area is 57 %, and the corresponding average grain size is 1.5 μm in this area.

Movie S1. 3D exhibition of APT dataset in Fig. 2E showing the dual-nanoprecipitation of κ -carbides and B2 particles.