#### **Appendix A. Supplementary Materials**

#### **1. Isotopic measurements**

 The aliquots for Os isotopic analysis were added into chilled Pyrex® Carius tubes, along with 5 ml of concentrated HNO<sup>3</sup> and 2.3 ml of concentrated HCl. The Carius tubes were sealed and then heated for at least 24 h at 220 °C (Shirey and Walker, 1995). Osmium was extracted from the solution using CCl<sup>4</sup> solvent-extraction methods (Cohen and Waters, 1996), and then further purified using microdistillation from a dichromate solution (Birck et al., 1997). This procedure resulted in ~70 % yield for Os. Between 70-300 ng of purified Os were loaded onto outgassed Pt 787 filaments in HBr, activated with Ba(OH)<sub>2</sub>, and analyzed as OsO<sub>3</sub><sup>-</sup> by a *Thermo-Fisher Triton*  thermal ionization mass spectrometer (TIMS) at the University of Maryland (UMd) (Walker, 2012). Osmium isotopic data were corrected for instrumental and natural mass fractionation by normalizing Os/ $188$ Os to 3.08271 (Allègre and Luck, 1980). Isobaric interferences from OsO<sub>3</sub><sup>-</sup> species with <sup>17</sup>O or <sup>18</sup>O were corrected using the O isotopic composition reported by Nier (1950). Molybdenum aliquots were prepared following the methods described in Worsham et al. (2016a). In brief, aliquots were dried, dissolved in 1 M HF, and loaded onto an anion column of AG 1-X8 (200-400 mesh) resin. Molybdenum was eluted using 6 M HNO3-3 M HF, dried, and dissolved in 6 M HCl. The Mo solution was then added to a smaller anion column with ~0.3 mL AG 1-X8 (200-400 mesh) resin and eluted with 1 M HCl. This elution was repeated a second time. This procedure resulted in ~50 % yield for Mo. Purified Mo aliquots were treated with concentrated HCl and HNO<sup>3</sup> multiple times in order to destroy organics. The samples were then dissolved in 6 M HCl, about 500-1000 ng Mo were loaded onto outgassed Re filaments, and 800 activated with  $\sim 2 \mu L$  of a 5  $\mu$ g/ $\mu$ L La(NO<sub>3</sub>)<sub>3</sub> solution. A double filament assembly was used and 801 the same amount of  $La(NO<sub>3</sub>)<sub>3</sub>$  was added to the ionization Re filament. Molybdenum was measured

802 as MoO<sub>3</sub> by the *Thermo-Fisher Triton Plus* TIMS at UMd. The <sup>100</sup>Mo<sup>18</sup>O<sup>16</sup>O<sub>2</sub> species was measured using a  $10^{13} \Omega$  resistor amplifier for O isotope interference corrections. Molybdenum 804 isotopic data were corrected for instrumental and natural mass fractionation by normalizing  $805 \frac{98}{100}$ %Mo to 1.453171 (Lu and Masuda, 1994).

806 Aliquots for Ru isotopic analysis were prepared following the procedure discussed by 807 Bermingham et al. (2016). In brief, aliquots of the primary dissolution were dried and dissolved in 808 0.15 M HCl and then processed through a cation column of AG50WX8 (200-400 mesh) resin. 809 Ruthenium was immediately eluted using 0.15 M HCl and dried to about 5 µL for microdistillation, 810 in which  $H_2Cr_2O_7$  was used as the oxidant and 6 M HCl was used as the trap solution and reductant. 811 This procedure resulted in ~50 % yield for Ru. About 1000 ng of purified Ru were loaded onto 812 outgassed single Pt filaments in HBr, activated with Ba(OH)<sub>2</sub>, and analyzed as RuO<sub>3</sub> by a *Thermo-*813 *Fisher Triton Plus* TIMS. Ruthenium isotopic data were corrected for instrumental and natural 814 mass fractionation by normalizing  $^{99}Ru/101Ru$  to 0.745075 (Chen et al., 2010). The O isotopic 815 composition reported by Nier (1950) was used to determine and correct for in-run  $^{17}O$  and  $^{18}O$ 816 molecular interferences.

 Tungsten aliquots from the primary dissolution were exposed to air for 1-2 weeks in order to 818 oxidize Fe in the solution from  $+2$  to  $+3$ . Ferrous Fe was then removed using a diisopropyl ether extraction procedure (Dodson et al., 1936). The resulting solution was dried, dissolved in 1 M HCl-0.1 M HF, and eluted through a cation column of AG50-X8 (200-400 mesh) resin with 1 M 821 HCl-0.1 M HF. The solution was then dried, dissolved in 0.5 M HCl-0.5 M HF, loaded onto a column of AG1-X8 (100-200 mesh) resin, and eluted with 6 M HCl-1 M HF. This elution was repeated twice using progressively less resin and acid (Touboul and Walker, 2012). This procedure 824 resulted in  $~60$  % yield for W. Organics accrued during the column chemistry were removed by

 drying the sample multiple times in concentrated HCl and HNO3. About 300-1000 ng of purified W were then dissolved in 1 M HCl- 0.01 M HF, loaded onto outgassed single Re filaments, activated with 1  $\mu$ L of a 5  $\mu$ g/ $\mu$ L La-5  $\mu$ g/ $\mu$ L Gd solution, and measured as WO<sub>3</sub> using a *Thermo*- *Fisher Triton* TIMS following the method of Archer et al. (2017). This method allowed 829 independent measurement of both  $182 \text{W}/184 \text{W}$  and  $183 \text{W}/184 \text{W}$ . Potential mass interference from 830 ReO<sub>3</sub> species were also monitored and corrected for. Tungsten isotopic data were corrected for 831 instrumental and natural mass fractionation by normalizing  $186W/184W$  to 0.92767 (Volkening et al., 1991).

 Some prior studies of terrestrial and cosmochemical materials have observed deviations in  $183$ W/ $184$ W ratios for reference standards and samples that were processed through certain column chemistry procedures and analyzed by multi-collector inductively-coupled plasma mass spectrometry (e.g., Kruijer et al., 2013; Cook and Schӧnbächler, 2016; Kruijer and Kleine, 2018). Such deviations have been interpreted to be the result of nuclear field shift effects, requiring 838 corrections to be made to account for the observed deviations (Cook and Schönbächler, 2016). However, no such effects have been observed for reference standards, terrestrial samples, or meteorite samples using the column chemistry procedures described here, coupled with thermal ionization mass spectrometry (Archer et al., 2017; Mundl et al., 2017; Mundl et al., 2018; Archer et al., 2019). Consequently, no corrections to the measured  $183 \text{W}/^{184}\text{W}$  ratios from this study have been made.

Values of  $\mu^{182}W$  for the SBT were corrected for nucleosynthetic variations using the method reported by Kruijer et al. (2014a). In this correction, variation in the *r*-process component for irons 846 is subtracted from the  $\mu^{182}W$  value using  $\mu^{183}W$  as an *r*-process component proxy. This correction is based on the <sup>182/184</sup>W *vs.* <sup>183/184</sup>W isotopic composition of fine- and coarse-grained CAI, which give a <sup>182/184</sup>W *vs.* <sup>183/184</sup>W (186/184 normalized) correlation of +1.41  $\pm$  0.06. The  $\Delta T_{CAI}$  ages were then calculated using the CAI initial value  $\mu^{182}W$  of -349  $\pm$  7 (Kruijer et al., 2014b), the present-850 day chondritic  $\mu^{182}$ W value of -190  $\pm$  10 (Kleine et al., 2004), and the decay constant for <sup>182</sup>Hf of 851 0.078  $\pm$  0.002 Myr<sup>-1</sup> (Vockenhuber et al., 2004). The  $\Delta T_{\text{CAI}}$  ages for Babb's Mill, South Byron, and ILD 83500 were calculated from the  $\mu^{182}W$  and  $\mu^{183}W$  values determined by normalizing to  $186W/184W$ . The uncertainties for the  $\Delta T_{CAI}$  ages include the 2SD determined from the analysis of standards run during an analytical campaign for  $\mu^{182}W$  and  $\mu^{183}W$  compositions, and the 855 uncertainty associated with the  $\mu^{183}W$  *r*-process correction only (Kruijer et al., 2014a). Tungsten 856 isotopic data were not obtained for Milton because there was insufficient material to make this 857 measurement.

858

### 859 **2. Fractional crystallization modeling**

860 In this study, Eq. S1 is used to calculate the concentration of an element throughout the 861 evolution of a liquid melt, where  $F_n$  is the fraction of liquid (n = 100 is pure liquid),  $C_{Ln}$  is the 862 concentration of an element in the liquid phase at  $F_n$ , and  $D_n$  is the partition coefficient of an 863 element. The concentration of an element is calculated at each 0.1 % fraction of liquid with respect 864 to the concentration of the previous liquid fraction. For S, a constant partition coefficient of 0.001 865 is used (Walker et al., 2008). The partition coefficients for P and C are determined at each  $F_n$  by 866 considering the concentration of S and P or S and C, respectively, in the liquid at  $F_{n+1}$ . Equation 867 S2 is used to account for the effects of S on P and C partitioning behavior (Chabot and Jones, 868 2003).

869 Values of D<sub>o</sub> for P and C are taken from Chabot et al. (2017) and Worsham et al. (2016b), 870 respectively. The  $\beta_{SPC}$  variable is determined using Eq. S3 (Jones and Malvin, 1990), in which  $\beta_S$  871 and  $\beta_P$  are taken from Chabot et al. (2017) and  $\beta_C$  is taken from Worsham et al. (2016b). For P, the 872 effects of C are not considered and for C, the effects of P are not considered. The  $\beta_{SPC}$  and "Fe 873 domains" (Eq. S4; Chabot et al., 2017) variables are calculated at each  $F_n$ , where  $X_i$  is the mole 874 fraction of an element.

875  
Eq. S1: 
$$
C_{Ln-1} = \frac{C_{Ln}}{(F_{n-1} + 1 - F_{n-1}D_{n-1})}
$$

876  
Eq. S2: 
$$
\frac{1}{D} = \frac{[Fe \text{ domains}]^{Bsrc}}{D_o}
$$

$$
877 \tEq. S3: \beta_{SPC} = \left[\frac{2X_S}{2X_S + 4X_P + 4X_C}\right] \beta_S + \left[\frac{4X_P}{2X_S + 4X_P + 4X_C}\right] \beta_P + \left[\frac{4X_C}{2X_S + 4X_P + 4X_C}\right] \beta_C
$$

878  
Eq. S4: Fe domains = 
$$
\frac{1 - 2X_s - 4X_P - 4X_C}{1 - X_s - 3X_P - 3X_C}
$$

$$
g_{79} \qquad \qquad Eq. S5: C_{Sn} = C_{Ln}D_n
$$

880 After determining the concentration of S, P, and C at each  $F_n$ , the D values for the HSE are 881 calculated at each  $F_n$  using Eq. S2-S4 by collectively considering the changes in S, P, and C content 882 in the liquid (Jones and Malvin, 1990; Chabot and Jones, 2003; Worsham et al., 2016b; Chabot et 883 al., 2017). The concentration of HSE in the liquid at each  $F_n$  is then determined using Eq. S1 and 884 the solid composition  $(C_{Sn})$  at each  $F_n$  is determined using Eq. S5.

885

### 886 **3. Thermal accretion model**

 This study uses a thermal accretion model (Eq. S6) that describes a solid sphere, which gains heat through  $26$ Al decay and loses heat by conduction (Carslaw and Jaeger, 1959). The model solves for the temperature at a certain depth within the sphere at any time (t) after accretion and we assume that differentiation occurred instantaneously when the temperature halfway to the

891 center of the body from the surface reached 1600 K (Kruijer et al., 2017). This assumption is made 892 since ~50 % of silicate melt fractions may be necessary for core formation (Taylor, 1992). In order 893 to be consistent with past studies, we assume a radius of  $R = 40$  km for the parent body and the 894 temperature is evaluated at a depth of  $r = 20$  km (Kruijer et al., 2017). The background nebular 895 temperature is represented as  $T_0$ , thermal conductivity is represented as K, thermal diffusivity is 896 represented as k, and a <sup>26</sup>Al decay constant of  $\lambda = 9.83 \times 10^{-7}$  yr<sup>-1</sup> is used with a Solar System initial  $26$ Al/<sup>27</sup>Al ratio of 5.23 x 10<sup>-5</sup> (Norris et al., 1983; Jacobsen et al., 2008). All values used in this 898 model are listed in Table S1.

899 Eq. S6: T = T<sub>0</sub> + 
$$
\frac{kA}{K\lambda}e^{-\lambda t}\left(\frac{R\sin\left(r*\left(\frac{\lambda}{k}\right)^{\frac{1}{2}}\right)}{\sin\left(R*\left(\frac{\lambda}{k}\right)^{\frac{1}{2}}\right)} - 1\right) + \frac{2R^3A}{r\pi^3K}\sum_{n=1}^{\infty} \frac{(-1)^n}{n(n^2 - \frac{\lambda R^2}{k\pi^2})}\sin\left(\frac{n\pi r}{R}\right)e^{-kn^2\pi^2t/R^2}
$$

900

901 The background temperature is assumed to be 250 K, consistent with models by Hevey and 902 Sanders (2006), Kruijer et al. (2014b), and Kruijer et al. (2017), which is the average temperature 903 estimated for a circumsolar disk at 2.5 AU and 1 Myr after T-Tauri star formation (Woolum and 904 Cassen, 1999). The power per unit volume A (W m<sup>-3</sup>) =  $H^*\rho$ , where H is the power per unit mass 905 (W kg<sup>-1</sup>) and  $\rho$  is the density of the planetesimal (kg m<sup>-3</sup>). The power per unit mass, H, is equal to 906 (total Al atoms/kg)\*( $^{26}$ Al/ $^{27}$ Al)\*(2.0 x 10<sup>-26</sup> W/atom), where ( $^{26}$ Al/ $^{27}$ Al) is a function of time, the 907 parent body concentration of Al is based off of the range observed in carbonaceous chondrites 908 (0.86-1.68 wt. %; Lodders and Fegley, 1998), and the decay power of <sup>26</sup>Al is calculated from a 909 decay energy of <sup>26</sup>Al of 6.4 x 10<sup>-13</sup> J/atom with its half-life of 9.83 x 10<sup>-7</sup> yr<sup>-1</sup> (Hevey and Sanders, 910 2006). We note that this calculation of heat production differs from that reported by Kruijer et al. 911 (2017). Kruijer et al. (2017) report using a range of Al concentration from 8.65-16.8 wt. %, which 912 is an order of magnitude greater than used here. In addition, Kruijer et al. (2017) report calculating

913 heat production as A = Al concentration\*( $^{26}$ Al/ $^{27}$ Al) and it is calculated in this study as A = (total 914 Al atoms/kg)\*( $^{26}$ Al/<sup>27</sup>Al)\*(2.0 x 10<sup>-26</sup> W/atom)\*(ρ), where ( $^{26}$ Al/<sup>27</sup>Al) is a function of time in both 915 models. Despite these differences between the model inputs, the results of both models differ by 916 less than 0.2 Myr. For this study, 0.86 wt. % Al relates to  $A = 12.3*(^{26}Al)^{27}Al$ , 1.20 wt. % Al 917 relates to A = 17.1\*( $^{26}$ Al/<sup>27</sup>Al) and 1.68 wt. % Al relates to A = 23.9\*( $^{26}$ Al/<sup>27</sup>Al). Therefore, the 918 results in this study for 0.86 wt. % Al and 1.20 wt. % Al are similar to the results by Kruijer et al. 919 (2017) for 12 wt. % Al and 16.8 wt. % Al, respectively.

### 920 **Supplementary Tables**

### 921 **Table S1**

Sample	$n^a$	$\mu^{189}$ Os	2SD	$\mu^{190}\overline{\text{Os}}$	2SD
Milton	1	$-4$	8	25	18
Babb's Mill	2	$\overline{1}$	6	-4	7
Babb's Mill (rep) 1		8	7	$-1$	4
South Byron	3	8	7	$-4$	7
ILD 83500	1	3	6	$-3$	7

922 Osmium isotopic composition data for Milton and the SBT.

 $\frac{a}{\text{m}}$  is the number of analyses. The reported  $\mu^{189}$ Os and  $\mu^{190}$ Os values reflect the average values obtained for each meteorite piece. Uncertainties reflect the largest (of n analyses) 2SD of the standards run during an analytical campaign.

## 927 **Table S2**





929  $aH = (Al atoms/kg)*(26Al)^{27}Al)*(E*\lambda_{sec})$ , where  $(26Al)^{27}Al)$  is a function of time and  $\lambda$  is in units of s<sup>-1</sup>.



**Fig. S1.** Compilation of CRE-corrected  $\mu^{97}$ Mo *vs*.  $\mu^{100}$ Ru data for iron meteorites from Bermingham et al. (2018). Blue squares represent meteorites classified as CC type and red diamonds represent meteorites classified as NC type. Data from this study for the SBT and Milton are also plotted. The black line represents the regression of the Mo-Ru cosmic correlation from Bermingham et al. (2018). The inset shows the Mo and Ru isotopic composition of the SBT and Milton without other meteorites.



Fig. S2. Compilation of  $\mu^{183}W$  *vs.*  $\mu^{182}W$  data for iron meteorites from Kruijer et al. (2017). Blue 940 squares represent meteorites classified as CC type and red diamonds represent meteorites classified 941 as NC type. The average SBT W isotopic composition is also plotted (green triangle). The average 942  $\mu^{183}$ W value for the SBT (+9  $\pm$  5, 2SE, n = 4) is resolved from the W standard value obtained 943 during the analytical campaigns for this work  $(0 \pm 2, 2SE, n = 16)$ .



 **Fig. S3**. CI-normalized HSE patterns calculated from a constant HSE initial concentration from Table 4 and varying S and P parental melt compositions. Assuming Milton metal crystallized in 947 the first ~10% of solid, the Re/Os of Milton metal resembles a high S system with 17 wt. % S, 1.5 948 wt. % P, and <0.05 wt. % C. However, the Pt/Os of Milton metal resembles a lower S system with 8 wt. % S, 1 wt. % P, and <0.05 wt. % C.



 **Fig. S4**. Fractional crystallization models for Re (ppb) *vs*. Re/Os and Pt (ppb) *vs.* Pt/Os for two pairs of parental melt compositions. Black lines represent mixing between solid and liquid, at 0 % solid-100 % liquid, 20 % solid-80 % liquid, and 40 % solid-60 % liquid. (*Upper left and right*) Fractional crystallization model, which reproduce the HSE patterns observed for the SBT, calculated for the initial parameters listed in Table 4. (*Bottom left and right*) Fractional crystallization model with 7 wt. % S and 1 wt. % P, which can reproduce the observed HSE patterns for Milton metal. This model requires a starting liquid composition of Re = 1,350 ppb, Os  $963 = 11,000$  ppb, and Pt = 17,500 ppb. Neither model can reproduce both Re/Os and Pt/Os observed in Milton metal and the SBT through solid metal-liquid metal mixing.



 **Fig. S5**. CI-normalized HSE parental melt compositions, with varying S and P contents, which reproduce the Milton HSE pattern as an early (<10 %) fractional crystallizing metal. The parental melt composition of the SBT is shown for comparison.



 **Fig. S6**. Thermal model results for a 40 km radius body when evaluating the temperature midway 971 to the center of the body  $(r = 20 \text{ km})$ . The model is based on an assumption of instantaneous differentiation when the temperature reaches 1600 K. Thermal model results are shown for Al 973 concentrations of 0.86, 1.20, and 1.68 wt. %. The differentiation age of the SBT (2.1  $\pm$  0.8 Myr 974 after CAI formation; blue solid line  $\pm$  blue dashed lines) fits an accretion age range of 1.1  $\pm$  0.5 Myr after CAI formation.

# **Supplementary References**











