

Supplementary Information: Dynamic fingerprint of fractionalized excitations in single-crystalline $\text{Cu}_3\text{Zn}(\text{OH})_6\text{FBr}$

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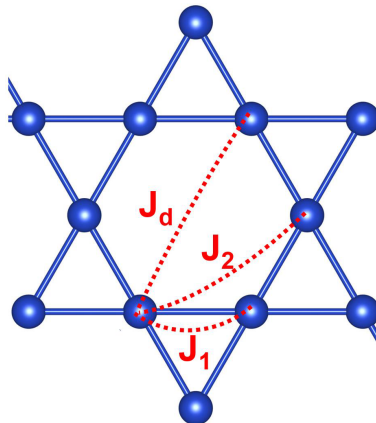
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Supplementary Note 1. ESTIMATION OF EXCHANGE PARAMETERS IN THE KAGOME COMPOUNDS



Supplementary Figure 1. Exchange interactions J_1 , J_2 , and J_d in the Kagome lattice.

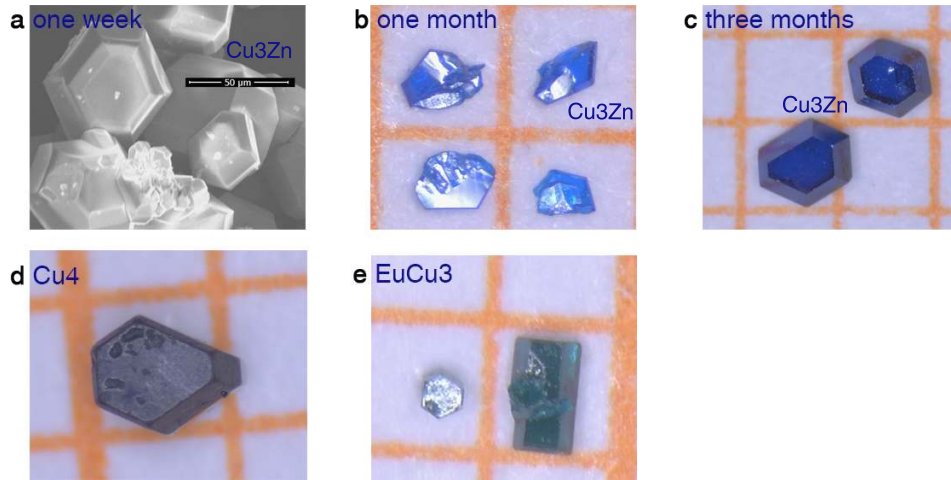
We implement the density functional theory (DFT) [1] to estimate the exchange parameters in the kagome compounds. We performed first-principles calculations with the Perdew–Burke–Ernzerhoff revised for solids (PBEsol) functional in generalized gradient approximation (GGA) [2, 3] as implemented in the Vienna Ab Initio Simulation Package (VASP).[4–6] An energy cutoff of 620 eV was used. We used $6 \times 6 \times 4$ Monkhorst-Pack grids [7] for all calculations. All results were obtained with Cu $3d$ valence electrons pseudopotential within GGA+U ($U_{3d} = 6$ eV) scheme.[8]

We fix the lattice constants and relax the atomic positions with a coplanar magnetic structure with negative vector spin chirality in the presence of spin-orbit couplings in our calculations. Exchange interactions can be determined from total energies of various different spin configurations. To determine J_1 , J_2 , and J_d (see Supplementary Fig. 1), we used the ferromagnetic state, antiferromagnetic state ($\mathbf{q} = 0$), cuboc2 state, and cuboc1 state for a $2 \times 2 \times 1$ supercell as discussed in Fig. 4 of Ref. [9]. Inter-layer couplings J_c are determined by comparing the energies for different stacking patterns of spin configurations. Note for the above interaction terms, we turn off the spin-orbit coupling in our simulations. For the Dzyaloshinski-Moriya (DM) interaction, we turn on spin-orbit couplings and compare the energies of $\mathbf{q} = 0$ antiferromagnetic states with positive and negative vector spin chiralities. The results are listed in Supplementary Table. 1. The nearest neighbor interactions in Cu₃Zn and EuCu₃ are larger than the experimental values.

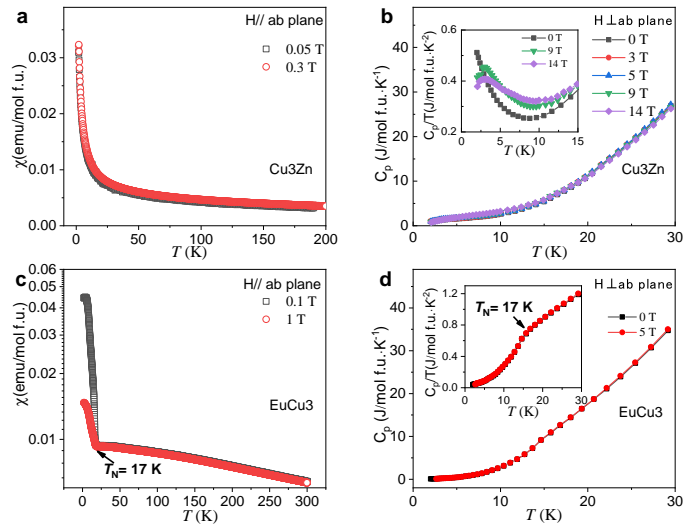
Supplementary Table 1. Theoretical results of exchange interaction (in meV) and Cu-O-Cu bonding angle for various materials with the kagome structure. SG denotes the space group. J_c denotes the inter-layer coupling. The references for the lattice constants are also listed.

Formula	SG	J_1 (meV)	J_2 (meV)	J_d (meV)	J_c (meV)	DM (meV)	DM/ J_1	\angle Cu-O-Cu ($^\circ$)	Reference
Cu ₃ Zn(OH) ₆ FBr	$P6_3/mmc$	24.13	-0.01	-0.65	1.442	1.12	0.05	117.47	[10]
YCu ₃ (OH) ₆ Cl ₃	$P\bar{3}m1$	10.21	0.12	-0.09	0.040	3.45	0.3	118.60	[11]
EuCu ₃ (OH) ₆ Cl ₃	$P\bar{3}m1$	13.02	0.16	-0.08	0.003	3.83	0.3	120.33	[12]
SmCu ₃ (OH) ₆ Cl ₃	$P\bar{3}m1$	13.55	0.14	-0.08	-0.004	5.91	0.4	120.36	[13]

Supplementary Note 2. CRYSTAL PHOTOGRAPH AND THERMODYNAMIC CHARACTERIZATION

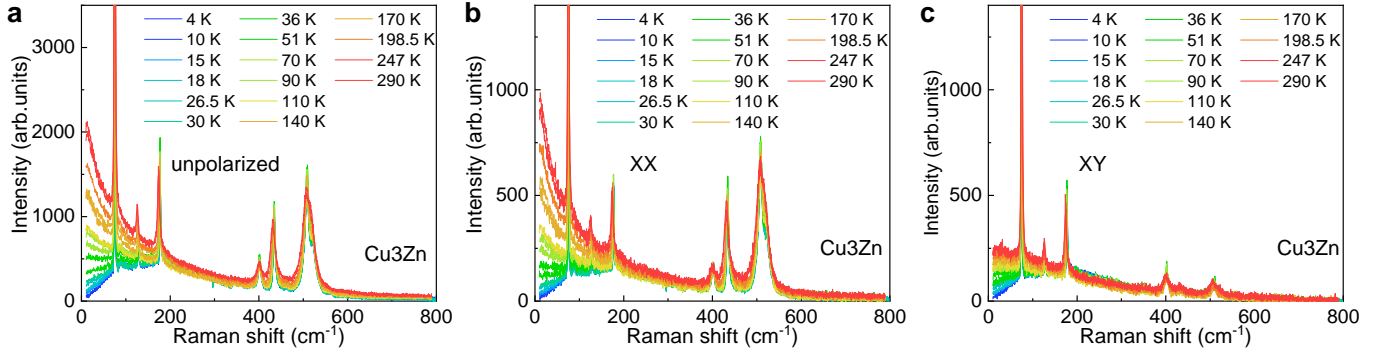


Supplementary Figure 2. Photographs for single crystals of Cu_3Zn , Cu_4 and EuCu_3 . Crystal sizes and morphologies of $\text{Cu}_3\text{Zn}(\text{OH})_6\text{FBr}$ (Cu_3Zn) for different growth periods: (a) one week, (b) one month, and (c) three months. (d) $\text{Cu}_4(\text{OH})_6\text{FBr}$ (Cu_4); (e) $\text{EuCu}_3(\text{OH})_6\text{Cl}_3$ (EuCu_3). The yellow grid in (b), (c), (d) and (e) is $1 \times 1 \text{ mm}^2$.



Supplementary Figure 3. Thermodynamic properties of single crystals for Cu_3Zn and EuCu_3 . (a) Temperature dependent magnetic susceptibilities ($\chi = M/H$) at 0.05 T and 0.3 T fields. (b) The temperature dependent specific heat C_p at different magnetic fields in Cu_3Zn . The thermodynamic properties of single crystals for Cu_3Zn agree well with previous results on the powder samples.[10, 14] (c) The magnetic susceptibilities show the ordering temperature $T_N = 17 \text{ K}$ in EuCu_3 . The Eu^{3+} with ground state of 7F_0 contributes to the Van Vleck paramagnetism (d) The temperature dependent heat capacities C_p in EuCu_3 .

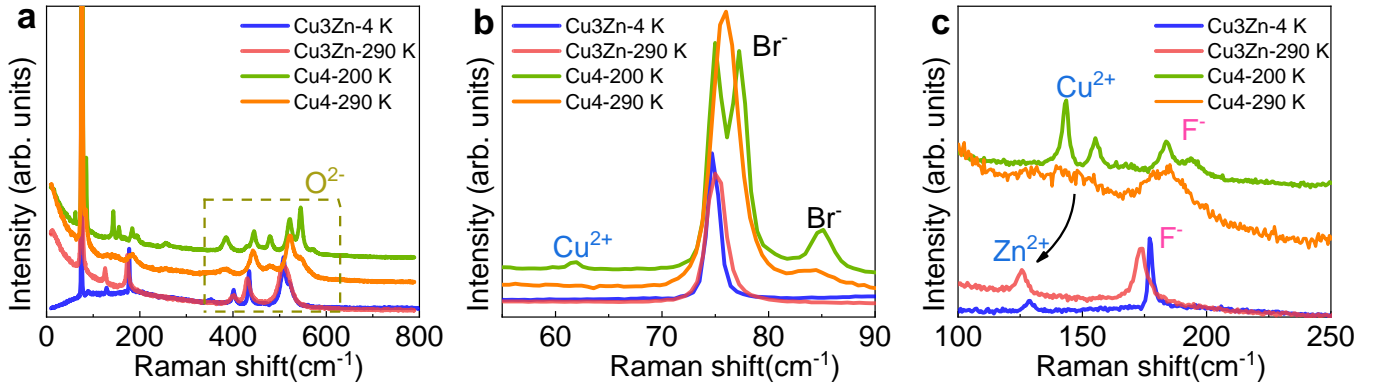
Supplementary Note 3. TEMPERATURE EVOLUTION OF THE RAMAN SPECTRA AND PHONON MODE ASSIGNMENT IN Cu3Zn



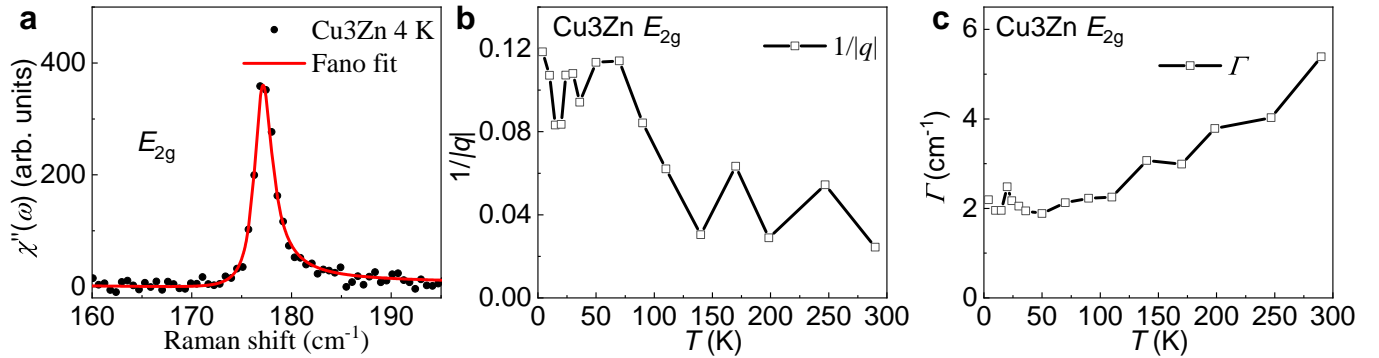
Supplementary Figure 4. Raman spectra in Cu3Zn at different temperatures. (a) Unpolarized Raman spectra in Cu3Zn. (b) Raman spectra in the XX configuration contain the A_{1g} and E_{2g} channel. (c) Raman spectra in the XY configuration contain the E_{2g} channel.

Supplementary Table 2. Phonon mode assignment for Cu3Zn. Cu3Zn crystallizes the space group $P6_3/mmc$ (No. 194) and has Raman-active A_{1g} , E_{1g} , and E_{2g} modes according to the point group representation of D_{6h} ($6/mmm$). E_{1g} is not visible when the light polarization lies in the kagome ab plane, and we have Raman-active phonon modes $\Gamma_{\text{Raman}} = 4A_{1g} + 9E_{2g}$.

Frequency (Exp.) (cm^{-1})	Modes (Exp.)	Frequency (Cal.) (cm^{-1})	Modes (Cal.)	Associated vibrating irons
74.6	E_{2g}	71.2	E_{2g}	Br^-
126.4	E_{2g}	124.11	E_{2g}	Zn^{2+}
172.2	E_{2g}	184.18	E_{2g}	F^-
355.5	E_{2g}	345.37	E_{2g}	O^{2-}
401.5	E_{2g}	396.21	E_{2g}	O^{2-}
430.8	A_{1g}	426.06	A_{1g}	O^{2-}
488.6	E_{2g} , visible in 532 nm	493.47	E_{2g}	O^{2-}
521.1	A_{1g}	508.84	A_{1g}	O^{2-}
920.3	E_{2g}	920.33	E_{2g}	H^+
1016.7	A_{1g}	1082.26	A_{1g}	H^+
1028.2	E_{2g} , weak	1020.11	E_{2g}	H^+
3352.3	?	3333.19	E_{2g}	H^+
3467.5	A_{1g}	3512.14	A_{1g}	H^+

Supplementary Note 4. RAMAN SPECTRA EVOLUTION FROM Cu₄ TO Cu₃Zn AND FANO EFFECT IN Cu₃Zn


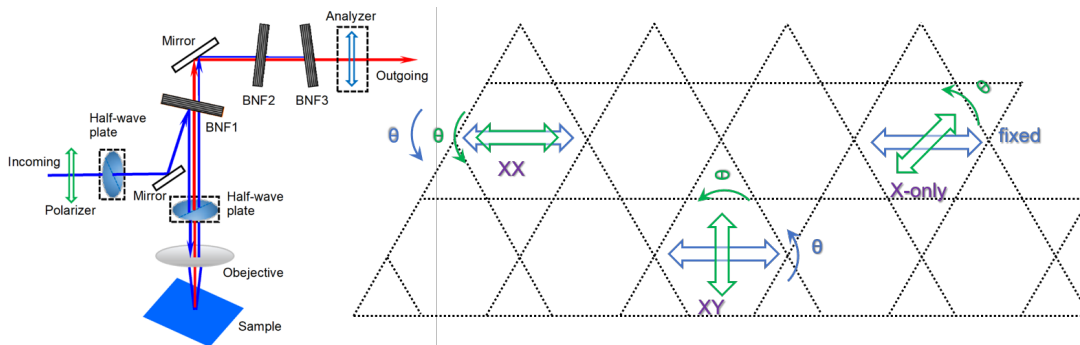
Supplementary Figure 5. Raman spectral evolution from Cu₄ to Cu₃Zn. (a) Unpolarized Raman spectra for Cu₄ and Cu₃Zn at selected temperatures. Comparison for phonon modes between 40 cm⁻¹ and 90 cm⁻¹ in (b), and between 100 cm⁻¹ and 250 cm⁻¹ in (c) for Cu₄ and Cu₃Zn. The Cu₄ spectra in (a), (b) and (c) have been offset vertically for clarity. The phonon evolution from Cu₄ to Cu₃Zn displays the difference by substituting the interlayer Cu²⁺ site of Cu₄ with Zn²⁺ in Cu₃Zn. The parent Barlowite Cu₄ transforms to orthorhombic *Pnma* below $T \approx 265$ K, characterized by changes in the relative occupancies of the interlayer Cu²⁺ site. Between 300 cm⁻¹ and 600 cm⁻¹, there are several phonon peaks associated with O²⁻ vibrations in Cu₄ and Cu₃Zn. Cu₃Zn displays the in-plane relative vibration of Br⁻ (E_{2g} mode) at 75 cm⁻¹, and has no Raman-active mode related to the kagome Cu²⁺ vibrations since Cu²⁺ is the inversion center. The Br⁻ phonon mode splits into two peaks in Cu₄ due to the superlattice folding in the orthorhombic *Pnma* phase at low temperature. An additional Br⁻ peak at 85 cm⁻¹ appears in Cu₄, related to the Br vibrations along the *c*-axis. The kagome layers in Cu₄ are distorted at low temperature, signaled by a new phonon mode for the kagome Cu²⁺ vibration at 62 cm⁻¹. Cu₃Zn displays sharp E_{2g} modes at 125 cm⁻¹ and 173 cm⁻¹ corresponding to in-plane relative movements for Zn²⁺ and F⁻, respectively. The corresponding modes (interlayer Cu²⁺ and F⁻ vibrations) in Cu₄ are broad at 290 K due to the randomly distributed interlayer Cu²⁺ and split into two peaks at 200 K.



Supplementary Figure 6. Fano lineshape of the E_{2g} F⁻ phonon peak at 173 cm⁻¹ in Cu₃Zn. (a) Fano lineshape for the E_{2g} in-plane phonon mode related to F atomic movement. Temperature dependent Fano asymmetric parameter $1/|q|$ in (b) and the width Γ in (c). The asymmetric Fano lineshape provides an additional probe of the magnetic degree of freedom.

Supplementary Note 5. LIGHT POLARIZATION CONFIGURATIONS IN ANGLE-RESOLVED POLARIZED RAMAN SCATTERING

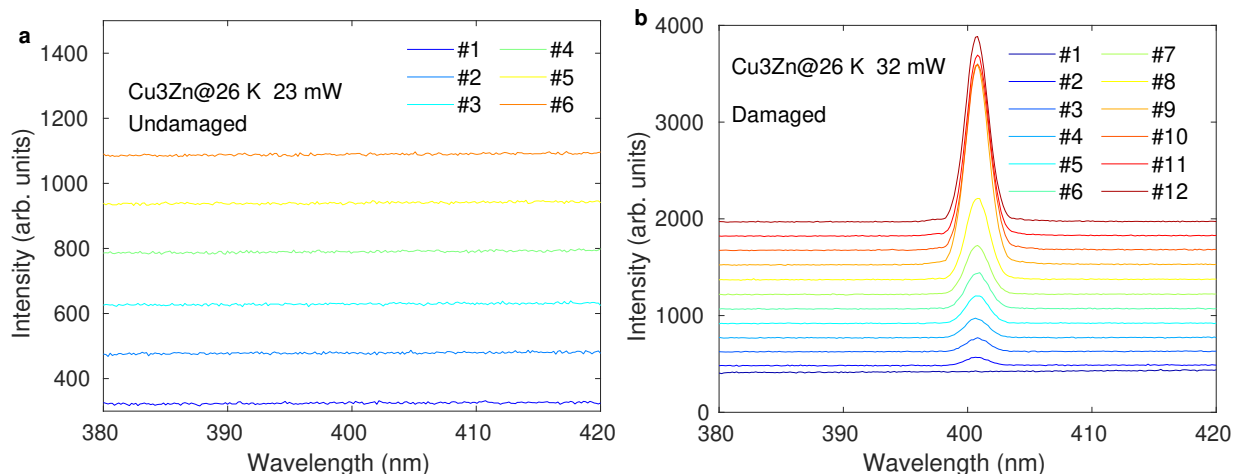
The polarized Raman measurements with light polarized in the ab kagome plane of samples were performed in parallel (XX), perpendicular (XY), and X-only polarization configurations. Two typical polarization configurations were utilized to measure the angle-resolved polarized Raman (ARPR) spectra: i) a half-wave plate was put after the polarizer in the incident path to vary the angles between the polarization of incident laser and the analyzer with the fixed vertical polarization, which can be denoted as the X-only configuration; ii) a half-wave plate is allocated in the common path of the incident and scattered light to simultaneously vary their polarization directions, while the polarizations of incident laser and analyzer were parallel or perpendicular to each other. By rotating the fast axis of the half-wave plate with an angle of $\theta/2$, the polarization of incident and/or scattered light is rotated by θ .



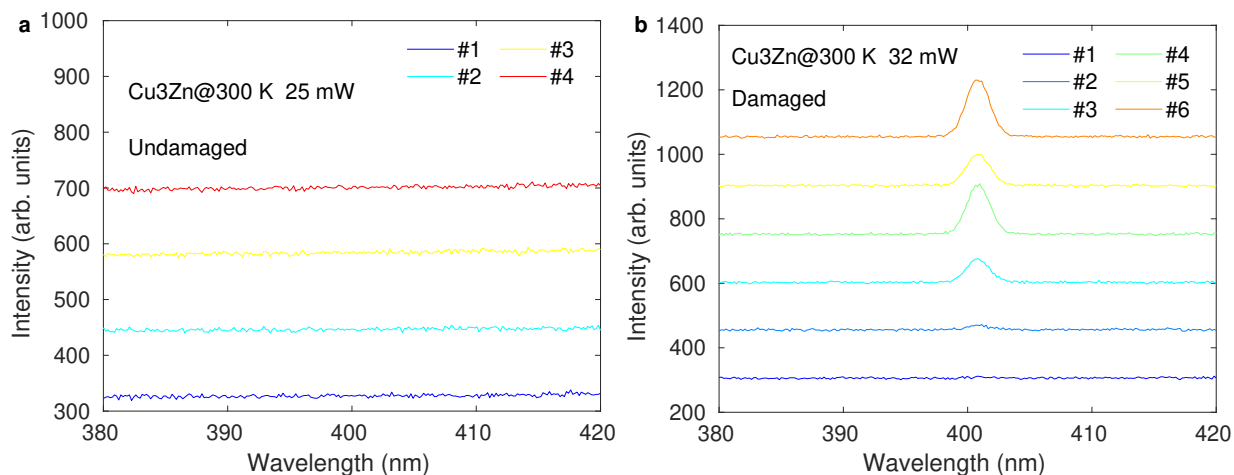
Supplementary Figure 7. Three polarization configurations in the ARPR response. In the XX (XY) configuration, the incoming and outgoing light polarizations are parallel (perpendicular) and we rotate both of them simultaneously. In the X-only configuration, the outgoing light polarization is fixed and we rotate the incoming light polarization only.

Supplementary Note 6. SECOND-HARMONIC-GENERATION (SHG) RESULTS OF Cu₃Zn

SHG measurements were performed using a homemade confocal microscope in a back-scattering geometry. A fundamental wave centered at 800 nm was used as excitation source, which was generated from a Ti-sapphire oscillator (Chameleon Ultra II) with an 80 MHz repetition frequency and a 150 fs pulse width. After passing through a 50 \times objective, the pump beam was focused on the sample with a diameter of 2 μ m. The scattering SHG signals at 400 nm were collected by the same objective and led to the entrance slit of a spectrometer equipped with a thermoelectrically cooled CCD. Two shortpass filters were employed to cut the fundamental wave.

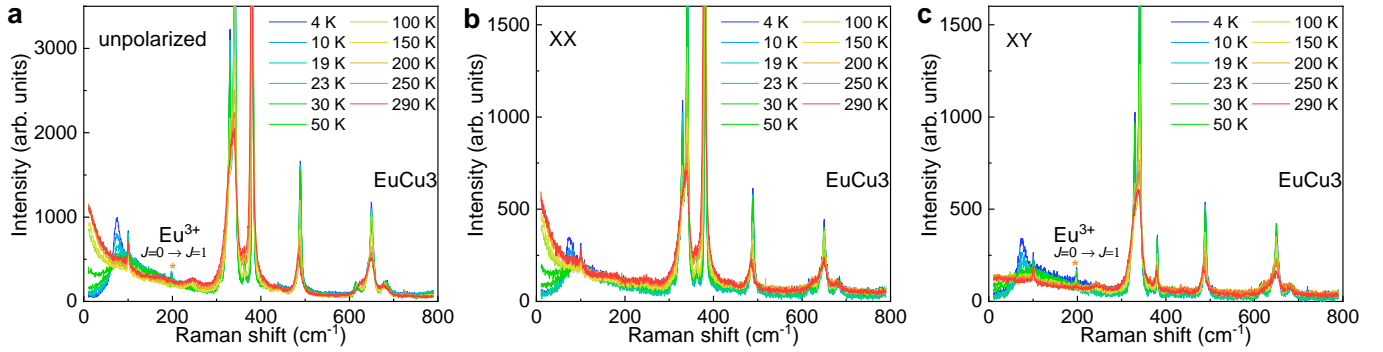


Supplementary Figure 8. SHG in Cu₃Zn at 26 K with different laser powers. (a) SHG measurements in the same spot of sample taken every 5 seconds (from #1 to #6). At 23 mW, SHG signals in Cu₃Zn sample are absent, implying that inversion symmetry remains preserved. (b) A series of SHG measurements under the excitation power of 32 mW in the same point of the sample taken every 5 seconds (from #1 to #12). A remarkable SHG signal at 400 nm is detectable after a 10-second exposure, which dramatically enhances as the time increases. Due to the damage or degradation of Cu₃Zn under high power excitation, the inversion symmetry breaking induces a strong SHG signals in sample. By comparison, we conclude that undamaged Cu₃Zn single crystal presents spatial inversion symmetry at low temperature. The lines have been offset vertically for clarity.

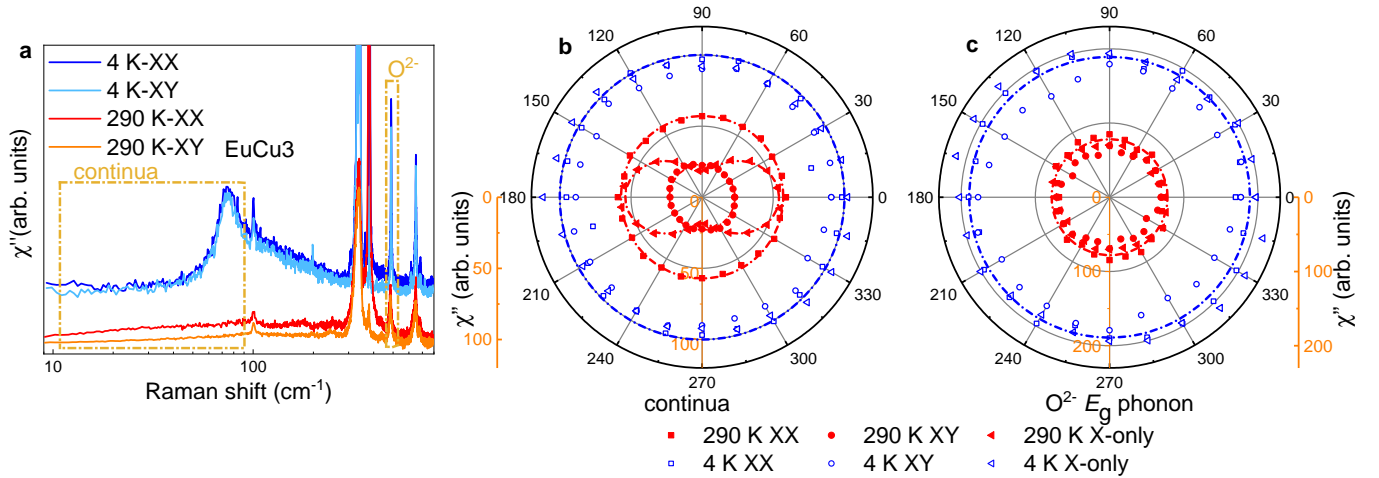


Supplementary Figure 9. SHG in Cu₃Zn at 300 K with different laser powers. (a) and (b) represent the successive SHG measurements in the same point of sample taken every 5 seconds with excitation powers at 25 mW and 32 mW, respectively. There are no SHG signals at the excitation power of 25 mW, whereas strong SHG signals appear at the excitation power above 32 mW after a 10-second exposure. By comparison, damage or degradation in crystal structure under high power excitation induces a detectable SHG signal, implying that inversion symmetry presents in undamaged Cu₃Zn at room temperature. The lines have been offset vertically for clarity.

Supplementary Note 7. RAMAN RESPONSES IN EuCu3



Supplementary Figure 10. Raman spectra of EuCu3 at different temperatures. (a) Unpolarized Raman spectra. (b) Raman spectra in the XX configuration in EuCu3 contain the A_g and E_g channel. (c) Raman spectra in the XY configuration contain the E_g and A_{2g} channel. For Eu^{3+} , we observe the A_{2g} excitation of the $4f^6$ configuration with the transition from ${}^7F_{J=0}$ to ${}^7F_{J=1}$.



Supplementary Figure 11. Rotation symmetry of Raman dynamics for lattice vibrations and magnetic excitations in EuCu3. We monitor the selected magnetic continuum at low frequency and the $\text{O}^{2-} E_g$ mode in (a). (b) ARPR dependence of the integrated Raman continuum from 9 to 80 cm^{-1} . The continua at 290 K follow the $\cos^2(\theta)$ function for the A_{1g} channel, while in other cases, the continua remain constant. (c) ARPR dependence of the $\text{O}^{2-} E_g$ phonon (487 cm^{-1}) scattering intensity. Its Raman intensity is independent of θ .

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