## **Electronic Supplementary Information for:**

## Trends in C-O and N-O bond scission on rutile oxides described by oxygen vacancy formation energies

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Rutile	DFT	EXP
TiO <sub>2</sub>	2.97,4.67	2.96,4.59
$VO_2$	2.84,4.59	2.85,4.56
CrO <sub>2</sub>	2.93,4.45	2.92,4.54
MnO <sub>2</sub>	2.89,4.45	2.88,4.40
FeO <sub>2</sub>	2.80,4.44	
CoO <sub>2</sub>	2.88,4.31	
NiO <sub>2</sub>	2.93,4.35	

**Table S1.** Comparison between DFT-calculated and experimental (EXP) lattice constants (in Å) of 3*d* rutile oxides.

Rutile	No dispersion	Dispersion
TiO <sub>2</sub>	-0.25	-0.51
$VO_2$	-0.20	-0.42
CrO <sub>2</sub>	-0.17	-0.38
$MnO_2$	-0.06	-0.23
FeO <sub>2</sub>	0.00	-0.16
$CoO_2$	-0.06	-0.23
$NiO_2$	-0.04	-0.20

**Table S2.** Adsorption energies ( $\Delta E_{Ads}$  in eV) of CO<sub>2</sub> with or without dispersion corrections<sup>1</sup> on the 3*d* rutile (110) surfaces.

Rutile	CC	$D_2 \rightarrow CO^+$	0	N <sub>2</sub>	$_2O \rightarrow N_2 + 0$	С
	$\Delta E_{Act}$	$\Delta E_{Diss}$	d	$\Delta E_{Act}$	$\Delta E_{Diss}$	d
TiO <sub>2</sub>	0.62	-0.51	1.28	0.13	-3.80	1.27
$VO_2$	1.29	0.29	1.59	0.43	-3.01	1.35
CrO <sub>2</sub>	1.65	1.22	1.73	0.57	-2.03	1.44
$MnO_2$	1.85	1.49	2.05	0.67	-1.46	1.36
FeO <sub>2</sub>	2.24	1.81	1.91	0.83	-1.27	1.42
CoO <sub>2</sub>	2.36	2.22	2.12	0.89	-0.53	1.49
NiO <sub>2</sub>	3.23	3.19	2.42	1.25	0.23	1.55

**Table S3.** Calculated activation energies ( $\Delta E_{Act}$  in eV), reaction energies ( $\Delta E_{Diss}$  in eV) and C(N)-O distance (*d* in Å) at the TSs for C-O and N-O bond scission on 3*d* rutile (110) surfaces.

Rutile	CH <sub>3</sub> C	H→CH <sub>3</sub>	+OH	NH <sub>2</sub> C	$H \rightarrow NH_2$	+OH
	$\Delta E_{Act}$	$\Delta E_{Diss}$	d	$\Delta E_{Act}$	$\Delta E_{Diss}$	d
TiO <sub>2</sub>	1.03	-0.44	2.05	0.18	-2.67	1.46
$VO_2$	1.25	0.55	2.55	0.17	-1.65	1.45
CrO <sub>2</sub>	1.56	0.76	2.85	0.49	-1.52	1.85
$MnO_2$	1.49	0.54	2.81	0.50	-1.48	1.89
FeO <sub>2</sub>	1.83	0.95	2.75	0.59	-0.59	1.99
$CoO_2$	2.00	1.03	2.75	0.65	-0.78	2.55
NiO <sub>2</sub>	2.21	1.40	2.55	0.94	-0.03	2.65

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Rutile	$\Delta E_{CO2}$	$\Delta E_{CO}$	$\Delta E_O$	$\Delta E_{N2O}$	$\Delta E_{N2}$
TiO <sub>2</sub>	-0.25	-0.13	-3.95	-0.20	-0.04
VO <sub>2</sub>	-0.20	-0.07	-3.17	-0.14	-0.03
CrO <sub>2</sub>	-0.17	-0.21	-2.06	-0.12	-0.03
$MnO_2$	-0.06	-0.01	-1.52	-0.04	-0.03
FeO <sub>2</sub>	0.00	-0.16	-1.29	0.02	-0.02
CoO <sub>2</sub>	-0.06	-0.37	-0.61	-0.03	-0.03
NiO <sub>2</sub>	-0.04	-0.02	0.15	-0.03	-0.01

**Table S4.** Adsorption energies ( $\Delta E_{Ads}$  in eV) of the intermediates on the 3*d* rutile (110) surfaces.

Rutile	$\Delta E_{CH3OH}$	$\Delta E_{CH3}$	$\Delta E_{OH}$	$\Delta E_{NH2OH}$	$\Delta E_{NH2}$
TiO <sub>2</sub>	-1.00	-1.04	-4.66	-0.86	-2.19
$VO_2$	-0.90	-0.51	-4.06	-0.73	-1.58
CrO <sub>2</sub>	-0.91	-0.67	-3.67	-0.74	-1.54
$MnO_2$	-0.76	-0.82	-3.61	-0.58	-1.18
FeO <sub>2</sub>	-0.68	-0.88	-3.21	-0.44	-1.11
CoO <sub>2</sub>	-0.83	-0.98	-3.10	-0.64	-1.17
NiO <sub>2</sub>	-0.65	-1.03	-2.68	-0.52	-0.94

Rutile	$\Delta E_{Vac}$
TiO <sub>2</sub>	3.95
$VO_2$	3.17
CrO <sub>2</sub>	2.06
$MnO_2$	1.52
FeO <sub>2</sub>	1.29
CoO <sub>2</sub>	0.61
NiO <sub>2</sub>	-0.15

**Table S5.** Vacancy formation energies ( $\Delta E_{Vac}$  in eV) for the 3*d* rutile (110) surfaces.



**Figure S1.** (a) Effect of +U on the adsorption energy of oxygen ( $\Delta E_{Ads}(O)$ ) using  $\frac{1}{2}O_2$  as a reference. (b) Effect of dispersion on the adsorption energy of CO<sub>2</sub>,  $\Delta E_{Ads}(CO_2)$ , at the bridge oxygen vacancy on rutile-type (110) surfaces.  $N_M$  is the number of outer electrons of 3*d* metals in the rutile oxides, which are calculated based on the periodic table and the oxidation state of the oxide.<sup>2</sup> For instance, Mn cations in MnO<sub>2</sub> have 3 outer electrons, as Mn's electronic distribution ends at 3s<sup>2</sup>4d<sup>5</sup> and its oxidation state is +4. The U values of 5.5, 3.32, 2.1, 3, 4.9, 5.37 and 6.37 are used for Ti to Ni, taken from refs 3-7.



**Figure S2.** (a) BEP Relations for CO<sub>2</sub> and N<sub>2</sub>O scission at the (110) surface of rutile oxides. Data are provided for fully converged PBE calculations and single-point PBE+U calculations. (b) Correlation between the activation energies for CO<sub>2</sub> and N<sub>2</sub>O scission (single-point PBE+U calculations) with the formation energies of oxygen vacancies (fully relaxed PBE+U calculations). The mean absolute errors (MAEs, in eV) are provided in each case.

**Note:** In line with the trends for oxygen adsorption energies in Fig. S1(a), the slopes are steeper upon including U in the calculations. The larger MAEs with respect to PBE are presumably due to the use of single-point calculations to assess the reaction energies and barriers, instead of fully relaxed data.



**Figure S3.** Optimized configurations for (a)  $CO_2$ , (b) CO, (c) O, (d)  $N_2O$ , (e)  $N_2$ , (f)  $CH_3OH$ , (g)  $CH_3$ , (h) OH, (i)  $NH_2OH$ , and (j)  $NH_2$  adsorption on 3*d* rutile (110) surfaces. The blue, vermilion, grey, ochre and white balls represent 3d metals (Ti-Ni), O, C, N and H atoms, respectively. The bridge O vacancy is denoted by the rectangle on the reduced surfaces.

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