

Supplementary Information:

Coexistence of structural and magnetic phases in van der Waals magnet CrI_3

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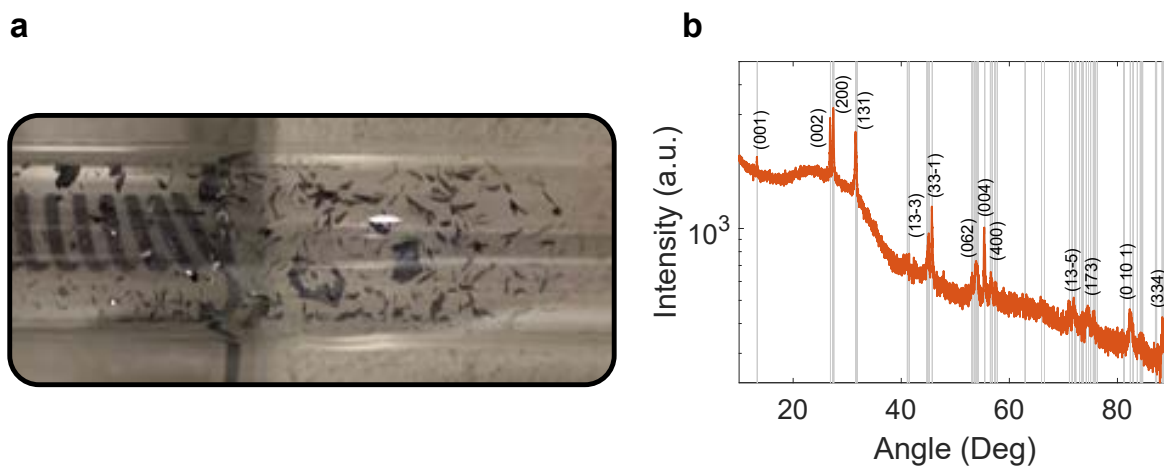
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25 **1 CrI₃ bulk crystal growth**

26 Chromium triiodide crystals were grown using the chemical vapour transport technique. Chromium
27 powder (99.5% Sigma-Aldrich) and anhydrous iodine beads (99.999% Sigma-Aldrich) were mixed
28 in a 1:3 ratio in an argon atmosphere inside a glovebox. 972 mg of the mixture were loaded into
29 a silica ampoule with a length, inner diameter and outer diameter of 500 mm, 15 mm and 16 mm
30 respectively (Supplementary Figure 1a). The ampoule was extracted from the glovebox with a ball
31 valve covering the open end to prevent air exposure and then it was immediately was evacuated
32 using a turbo-molecular pump down to 6×10^{-6} . Once the pressure stabilised, the closed end was
33 dipped in liquid nitrogen to prevent the sublimation of the iodide beads. The ampoule was then
34 flame sealed and introduced into a three-zone furnace. An initial inverted gradient step was used
35 to minimize nucleation sites in the growth zone. The gradient was then inverted so that the source

36 zone would remain at 650 °C, the middle growth zone at 550 °C and the third zone at 600 °C for
37 3 days. Crystal growth takes place in the centre and source zones. The single crystals and the
38 microcrystalline powder were both extracted from the ampoule and stored in an argon atmosphere
39 inside the glovebox to prevent oxidation and hydration. A sample of the bigger single crystals was
40 ground into a powder using a razor blade. X-Rays diffraction analysis was performed on samples
41 of the powder and single crystalline CrI₃ by loading the material into a capillary and flame-sealing
42 it inside the glove box. The powder pattern of both samples were consistent with the monoclinic
43 AlCl₃-type structure (C2/m) reported for CrI₃ (Supplementary Figure 1b).



Supplementary Figure 1: **CrI₃ samples.** **a**, Picture of the ampoule used for sample fabrication. **b**, X-Ray diffraction pattern of the CrI₃ microcrystalline powder studied in the μ -SR experiment. The most prominent peaks of the orange trace (experiment) have been labelled (Miller indices) and marked (grey lines) with peaks positions of the monoclinic (room temperature) crystalline phase of CrI₃.

44 2 μ -SR experiment and analysis

45 The μ -SR method is based on the observation of the time evolution of the spin polarization $\vec{P}(t)$
46 of the muon ensemble. In μ -SR experiments an intense beam ($p_\mu = 29$ MeV/c) of 100 % spin-
47 polarized muons is stopped in the sample. Currently available instruments allow essentially a back-
48 ground free μ -SR measurement at ambient conditions ¹. The positively charged muons thermalize
49 in the sample at interstitial lattice sites, where they act as magnetic microprobes. In a magnetic
50 material the muons spin precess in the local field B_μ at the muon site with the Larmor frequency
51 $\nu_\mu = \gamma_\mu/(2\pi)B_\mu$ (muon gyromagnetic ratio $\gamma_\mu/(2\pi) = 135.5$ MHz T⁻¹). The muons μ^+ implanted
52 into the sample will decay after a mean life time of $\tau_\mu = 2.2$ μ s, emitting a fast positron e^+ prefer-
53 entially along their spin direction. Various detectors placed around the sample track the incoming
54 μ^+ and the outgoing e^+ . When the muon detector records the arrival of a μ in the specimen, the
55 electronic clock starts. The clock is stopped when the decay positron e^+ is registered in one of the
56 e^+ detectors, and the measured time interval is stored in a histogramming memory. In this way a
57 positron-count versus time histogram is formed. A muon decay event requires that within a certain
58 time interval after a μ^+ has stopped in the sample a e^+ is detected. This time interval extends
59 usually over several muon lifetimes (*e.g.* 10 μ s).

60

61 The GPS (π M3 beamline) μ -SR instrument at the Paul Scherrer Institute, Switzerland, was
62 used to study the CrI₃ sample. The specimen was mounted in a He gas-flow cryostat and the tem-
63 perature was varied between 5 and 200 K.

65 **Analysis of Weak TF- μ SR data** The weak TF asymmetry spectra were analyzed ^{2,3} using the
 66 function:

$$A_S(t) = A_p \exp(-\lambda t) \cos(\omega t + \phi). \quad (1)$$

67 where t is the time after muon implantation, $A(t)$ is the time-dependent asymmetry, A_p is the ampli-
 68 tude of the oscillating component (related to the paramagnetic volume fraction), λ is an exponential
 69 damping rate due to paramagnetic spin fluctuations and/or nuclear dipolar moments, $\omega = 2\pi\nu_\mu$ is
 70 the Larmor precession frequency set by the strength of the transverse magnetic field, and ϕ is a
 71 phase offset. The zero for $A(t)$ was allowed to vary for each temperature to deal with the asymmetry
 72 baseline shift occurring for magnetically ordered samples. From these refinements, the magneti-
 73 cally ordered volume fraction at each temperature T was obtained by $V_M = 1 - A_p(T)/A_p(T_{max})$,
 74 where $A_p(T_{max})$ is the amplitude in the paramagnetic phase at high temperature.

76 **Analysis of ZF μ -SR data** In the paramagnetic state, the μ -SR spectra are described by the com-
 77 bination of Lorentzian and Gaussian Kubo-Toyabe depolarization function:

$$A_{\text{ZF,PM}}(t) = (1 - V_M)A_0 \left(\frac{1}{3} + \frac{2}{3} (1 - (\sigma t)^2 - \Lambda t) e^{-\frac{1}{2}(\sigma t)^2 - \Lambda t} \right) \quad (2)$$

78 Here, the depolarisation rates σ and Λ are due to the nuclear dipole moments and randomly
 79 oriented diluted local electronic moments, respectively. A_0 is the initial asymmetry. $V_{PM} = 1 - V_M$
 80 is the paramagnetic fraction, which acquires maximum value of 1 above 60 K.

81 In the ordered state, the response of the sample consists of three magnetic components, de-
 82 pending on temperature:

$$A_{\text{ZF,M}}(t) = V_M A_0 \omega_1 \left(f_{T1} \cos(\gamma_\mu B_{\mu 1} t + \phi) e^{-\lambda_{T1} t} + f_{L1} e^{-\lambda_{L1} t} \right) + \dots \quad (3)$$

$$V_M A_0 \omega_2 \left(f_{T2} \cos(\gamma_\mu B_{\mu 2} t + \phi) e^{-\lambda_{T2} t} + f_{L2} e^{-\lambda_{L2} t} \right) + \dots \quad (4)$$

$$V_M A_0 \omega_3 \left(f_{T3} e^{-\lambda_{T3} t} + f_{L3} e^{-\lambda_{L3} t} \right) \quad (5)$$

83 Here, ω_1 , ω_2 , and ω_3 are the relative fractions of the magnetic components. The first and
 84 second terms describe the high and low frequency magnetic components, respectively. The com-
 85 ponents are characterised by an oscillating “transverse” component and a slowly relaxing “lon-
 86 gitudinal” component. The longitudinal component arises due to muons experiencing local field

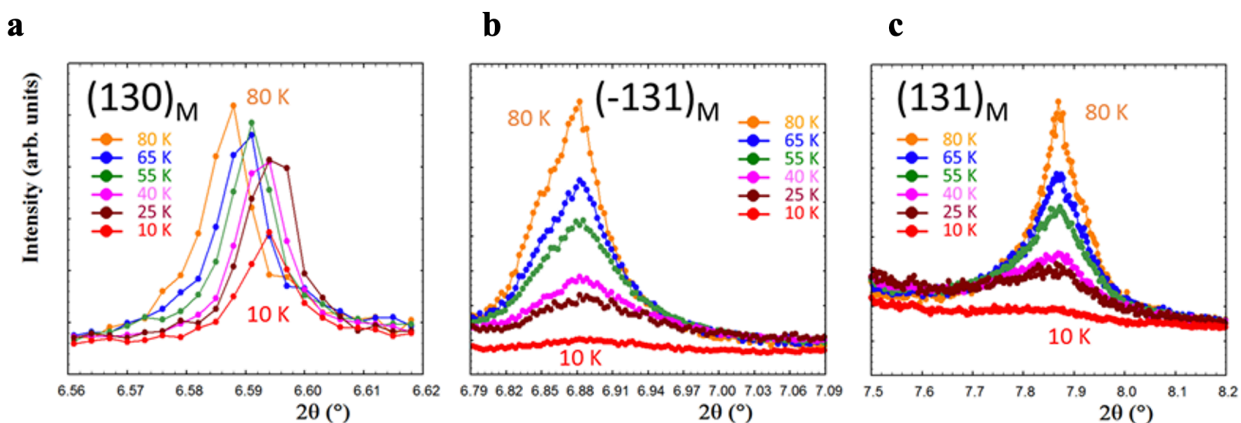
87 components which are parallel to the initial muon spin polarization. In polycrystalline samples
88 with randomly oriented fields the orientational averaging results in a so-called “one-third tail”
89 with $f_L = \frac{1}{3}$ ($f_T = \frac{2}{3}$). For single crystals, f_L varies between zero and unity as the orientation
90 between field and polarization changes from being perpendicular to parallel. λ_T and λ_L are the
91 transverse and longitudinal depolarisation rates of the μ -SR signals, respectively. $B_{\mu 1}$ and $B_{\mu 2}$
92 are the internal magnetic fields at the muon site, corresponding to high and low frequency com-
93 ponents, respectively. The third term describes the strongly damped magnetic component, which
94 occurs below 30 K only. We note that it is not possible to separate λ_{L3} from λ_{L1} and λ_{L2} and thus
95 it is fixed to 0 in the analysis.

96 **3 SQUID magnetometry**

97 Magnetization curves and zero-field-cooled/field-cooled susceptibility sweeps were carried out in
98 a SQUID magnetometer (Quantum Design MPMS-XL-7) on single crystals of CrI_3 where the rel-
99 ative orientation of the basal plane of the sample with the external magnetic field (both AC and
100 DC) is controlled. In order to measure the magnetization of CrI_3 in the presence of in-plane and
101 out-of-plane external magnetic fields, two different samples were prepared. For the out-of-plane
102 orientation, various single crystals were stacked one on top of the other and placed onto the surface
103 on the end of a polyethylene rod. This was carefully placed inside a silica tube with the crystals
104 facing upwards and another polyethylene rod placed on top trapping the crystal in between. For
105 the in-plane orientation, a polyethylene rod was cut longitudinally and a single crystal was glued
106 to the flat using Apiezon-N grease. These preparations were performed in an argon atmosphere
107 to avoid crystal degradation. The samples were then inserted inside an outer silica tube and flame
108 sealed at a pressure lower than 10^{-3} mbar.

109 **4 Synchrotron X-ray diffraction measurements**

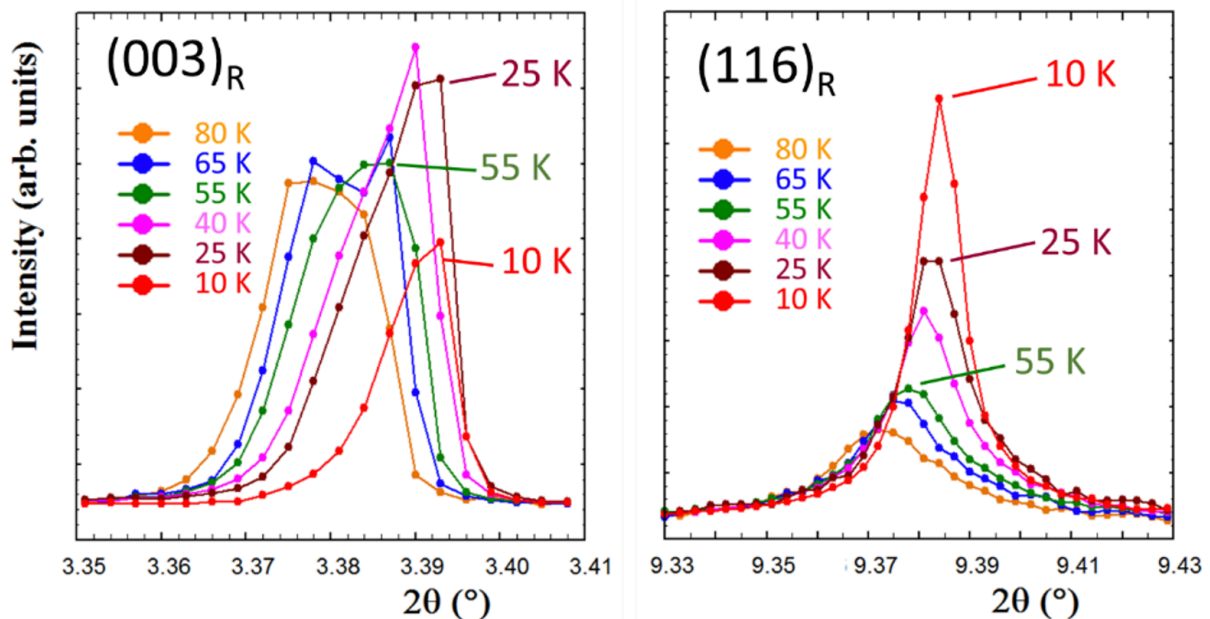
110 Synchrotron X-ray powder diffraction (SXRPD) measurements were performed on the BL04-
111 MSPD beam-line of the ALBA Synchrotron Light Facility (Barcelona, Spain) using the multi-
112 crystal analyser MAD detector system. The wavelength, $\lambda = 0.38670(3) \text{ \AA}$, was determined by
113 measuring a NIST standard silicon. This energy (32 KeV) was selected in order to avoid the flu-
114 orescence from iodine ions (33 KeV). The sample was loaded in a borosilicate glass capillary of
115 0.3 mm diameter to keep the absorption below 1, and was kept spinning during data acquisition.
116 Patterns between 10 and 300 K were collected using a Dynaflo liquid He cryostat. Measurements
117 were performed at fix temperatures in warming conditions, after a first cooling down to 10 K.



Supplementary Figure 2: **Temperature dependent X-ray diffraction.** a-c, Temperature evolution of selected monoclinic reflections below 80 K (MSPD Alba). Panels show the persistence of the monoclinic phase even at low temperatures. Notice the wide Lorentzian shape of the ($\pm 1L1$) family of planes.

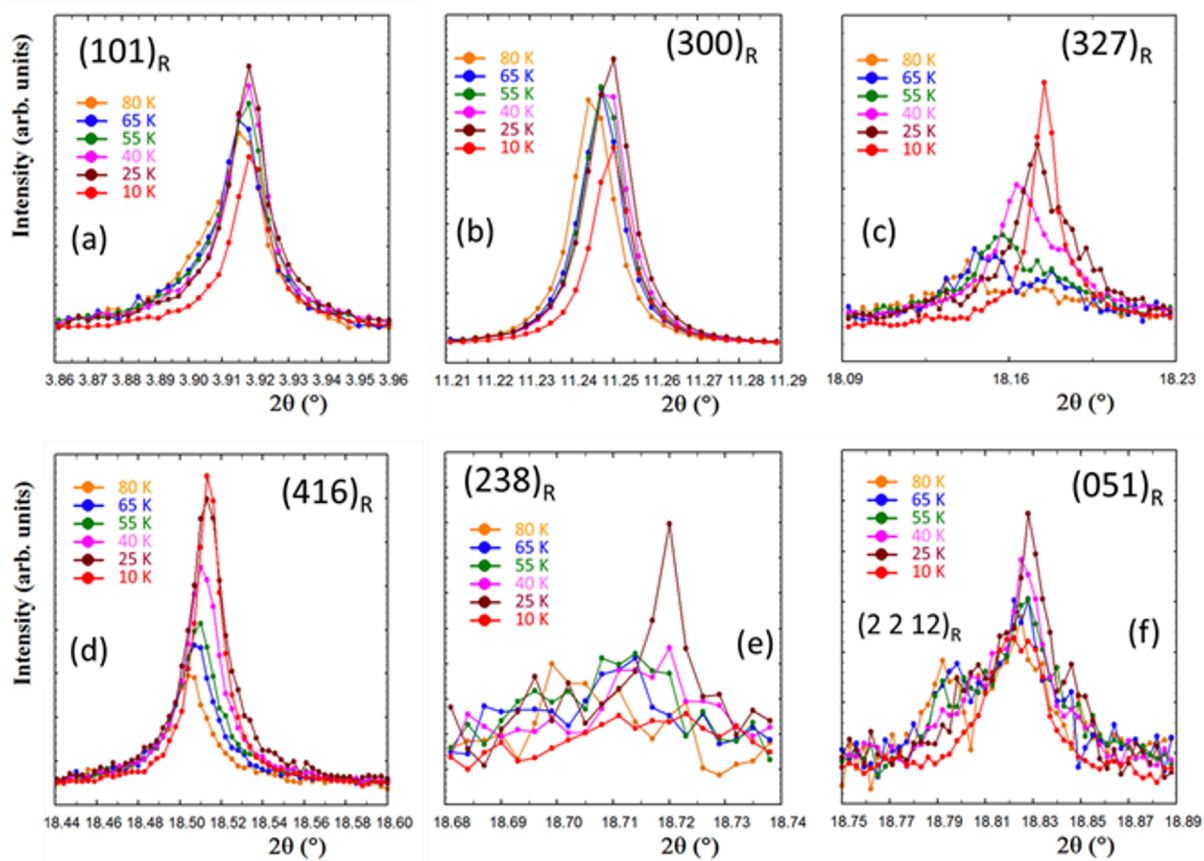
118 **5 Additional analysis on X-ray data**

119 To illustrate the coupling between structures and magnetic phases in a different way, Supplemen-
120 tary Figures S3–4 show the SXRPD intensity profiles of some R-peaks at the following selected
121 temperatures: 80, 65, 55, 40, 25 and 10 K. In Supplementary Figure S3 we compare the thermal
122 evolution of $(003)_R$ and $(116)_R$ profiles. Firstly, a sudden increase in the intensity is apparent in
123 both below 55 K. Secondly, and more interestingly, there is a remarkable intensity drop in $(003)_R$
124 below 25 K, which concurs with a large increase of the intensity of $(116)_R$. These small but reliable
125 changes strongly suggest a coupling between the structure and the magnetic transitions in CrI_3 .



Supplementary Figure 3: **Temperature dependent X-ray diffraction.** Anomalous temperature evolution close to the magnetic transition temperatures of the $(003)_R$ (left) and $(116)_R$ (right) rhombohedral peaks (MSPD Alba). Left: Sudden $(003)_R$ intensity increase below 55 K and sharp drop below 25 K. Right: Prominent $(116)_R$ intensity increase below 55 K and additional increase below 25 K, concurrent with the drop in $(003)_R$.

126 Similar intensity changes were observed in most of the R-peaks. As a further example,
127 Supplementary Figure S4 shows the changes at low temperatures in the intensity of six additional
128 diffraction peaks. A marked intensity raise below 55 K is apparent in $(327)_R$, $(416)_R$, $(238)_R$,
129 $(051)_R$. The highest intensity is reached at 25 K for $(101)_R$, $(300)_R$, $(238)_R$, $(051)_R$ whereas
130 a substantial intensity drop occurs at 10 K. The evolution of these peaks is reminiscent of that in
131 Supplementary Figure S3, and it is in contrast with reflections $(327)_R$ or $(416)_R$ in which (mirroring
132 the peak $(116)_R$ in Fig. S3) keeps growing also below 25 K. Therefore, this anomalous evolution
133 is consistent with small structural changes causing certain relative intensity variations within the
134 low-temperature range.



Supplementary Figure 4: **Temperature dependent X-ray diffraction.** Temperature evolution of some rhombohedral peaks of CrI₃ (MSPD Alba) showing different trends in the temperature dependence below 80 K.

135 **6 Supplementary References**

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