

1 Supplementary information for

2 **Microfluidic Manipulation by Spiral Hollow-Fiber Actuators**

3
4 Sitong Li¹, Rui Zhang², Guanghao Zhang¹, Luyizheng Shuai¹, Wang Chang¹, Xiaoyu
5 Hu¹, Min Zou¹, Xiang Zhou¹, Baigang An³, Dong Qian², Zunfeng Liu^{1*}

6
7 ¹State Key Laboratory of Medicinal Chemical Biology, College of Chemistry and
8 College of Pharmacy, Key Laboratory of Functional Polymer Materials, Frontiers
9 Science Center for New Organic Matter, Nankai University, Tianjin 300071, China

10 ²Department of Mechanical Engineering, University of Texas at Dallas, Richardson,
11 TX 75080, USA

12 ³School of Chemical Engineering, University of Science and Technology Liaoning,
13 Anshan 114051, China

14 E-mail: liuzunfeng@nankai.edu.cn

15
16
17
18 **This PDF file includes:**

19
20 Supplementary Note 1. Methods (Pages S2–S3)

21 Supplementary Note 2. Thermomechanical modeling (Pages S3–S7)

22 Supplementary Note 3. Optimum hollow fiber geometry analysis (Pages S7–S8)

23 Supplementary Figures 1 to 12 (Pages S9–S15)

24 Supplementary Tables 1 to 3 (Pages S16–17)

25 References (Page S18)

26

27 **Supplementary Note 1. Methods**

28 *1.1 Characterizations*

29 Linear coefficients of thermal expansion (CTE) of PEHF were measured from 0
30 to 100 °C at a heating rate of 3 °C min⁻¹ using a thermomechanical analyzer (TMA,
31 Q400 system, TA instruments). Because the thermal expansion of employed
32 polyethylene hollow fibre is anisotropic¹, the axial and the radial coefficients of
33 thermal expansion were tested individually. For each experiment, three heating scans
34 and two cooling scans were conducted. The linear CTE along each direction was
35 calculated based on the thermomechanical analysis results.

36 Two-dimensional wide-angle X-ray scattering (2D WAXS) experiments were
37 performed at a sample-to-detector distance of 100 mm using CuK α radiation ($\lambda = 1.54$
38 Å) by deploying a D8 Discover system (Bruker, Germany) equipped with a 2D
39 HI-Star detector and a 1.6 kW generator. Parallel PEHF were assembled, using a
40 video microscope to ensure that the X-ray beam irradiated the centers of the fibres.
41 The image acquisition time was 600 s. One-dimensional wide-angle X-ray scattering
42 (1D WAXS) experiments were performed using CuK α radiation and a 2 θ scanning
43 rate of 5° min⁻¹ on a Rigaku Smart Lab system (Japan) equipped with a 3.0 kW
44 generator.

45 *1.2 Calculation of the degree of orientation of crystals using WAXS analysis*

46 Despite inserting twists into a polymer fibre, the highly oriented semicrystalline
47 form is also the key to provide a fibre the actuation capability. The Herman's
48 orientation parameter (f) is employed to represent the polymer chain alignment of the
49 fibre, where $f = 1$ means that the polymer chains orient along the fibre axis
50 direction completely; $f = -0.5$ means that the polymer chains orient vertically to
51 the fibre axis direction completely. For a non-oriented sample, f equals to 0.

52 The Herman's orientation parameters (f_a , f_b , and f_c) for the a-axis, b-axis, and
53 c-axis, for a unit cell of polyethylene crystallite with orthorhombic symmetry, were
54 calculated as 0.12, -0.30, and 0.18, respectively, according to the literatures.²⁻³ These
55 values provide the evidences that the fibres are in highly oriented semicrystalline

56 form.

57 The calculation details are as follows.

58 The Herman's orientation parameter (f) is defined as:

$$59 \quad f = \frac{3\langle \cos^2 \Phi \rangle - 1}{2} \quad (1.1),$$

60 where $\langle \cos^2 \Phi \rangle$ is an orientation factor defined as:

$$61 \quad \langle \cos^2 \Phi \rangle = \frac{\int_0^\pi I_\Phi \cos^2 \Phi \sin \Phi d\Phi}{\int_0^\pi I_\Phi \sin \Phi d\Phi} \quad (1.2),$$

62 where I_Φ represents the diffraction intensity. The azimuthal curve obtained by
63 integration of diffraction intensity in the 2D WAXS pattern is used in Eq. (1.2) to
64 calculate $\cos^2 \Phi$ by the numerical integration process. Then the values of $\cos^2 \Phi$
65 are used in Eq. (1.1) to calculate the orientation function value f of the semicrystalline
66 polymers.

67 For polyethylene with orthorhombic symmetry of the unit cell, the a-axis (f_a) and
68 b-axis (f_b) orientation parameters were calculated from the intense reflections of (200)
69 and (020) planes based on the relation:

$$70 \quad \langle \cos^2 \Phi_{110} \rangle = 0.692\langle \cos^2 \Phi_{020} \rangle + 0.308\langle \cos^2 \Phi_{200} \rangle \quad (1.3),$$

71 where $\langle \cos^2 \Phi_{110} \rangle$ and $\langle \cos^2 \Phi_{200} \rangle$ are obtained from the azimuthal intensity
72 measurements on the (110) and (200) reflections, respectively. The parameter f_c is
73 estimated based on the orthogonal symmetry relationship: $f_a + f_b + f_c = 0$.⁴

74 The detailed calculations for f_a , f_b , and f_c are as follows. First, an azimuthal
75 curve was obtained by integration of diffraction intensity in the 2D WAXS pattern
76 over the azimuthal angle using the software of the GADDS (general area detector
77 diffraction system) for diffraction angles ($2\theta_d$) of 21.6° (110 plane) and 24° (200
78 plane), as shown in Supplementary Fig. 1c. Two identical peaks were observed in the
79 azimuthal curve (Supplementary Fig.1d) because of the symmetric diffraction pattern
80 in 2D WAXS. The f_a , f_b , and f_c of the PEHF are respectively 0.12, -0.30, and 0.18,
81 which indicate that the fibres are in highly oriented semicrystalline form.

82

83 **Supplementary Note 2. Thermomechanical modeling of hollow fibre actuator**

84 *1.1 Transient heat conduction analysis*

85 The temperature change of the hollow fibre actuator is contributed by the forced
86 convection of the internal flowing water as well as the natural convection of the
87 ambient (external) air. To study this transient heat conduction problem, we establish a
88 two-step model. First, we model the radial heat conduction at an arbitrary
89 cross-section of the hollow fibre actuator. Second, we build a simple axial heat
90 conduction model based on the assumption that heat conduction rate along the hollow
91 fibre axis is much slower compared to the water flow rate and the rate of heat
92 conduction in the radial direction. With this two-step model, we determine the hollow
93 fibre actuator temperature as a function of radial and axial coordinates and time.

94 The DSC results showed that there was no observable phase transition in the
95 PEHF for the investigated temperature range (Supplementary Fig. 1b), we assume that
96 there is no contribution of the phase transition to the volume change and the heat
97 transfer contributed to the volume expansion.

98 First, the governing equation of the transient heat conduction in radial direction
99 at an arbitrary cross-section of the hollow fibre is given by

$$100 \quad \frac{\partial^2 \theta}{\partial r^2} + \frac{1}{r} \frac{\partial \theta}{\partial r} = \frac{1}{\alpha} \frac{\partial \theta}{\partial t} \quad \text{in } R_i \leq r \leq R_o, \quad (2.1)$$

101 where $\theta(r, t)$ is the temperature, R_i and R_o are the inner and outer radius of the hollow
102 fibre, respectively. Thermal diffusivity α is defined by

$$103 \quad \alpha = \frac{k}{\rho c_p}, \quad (2.2)$$

104 in which k is the thermal conductivity, ρ is the mass density, c_p is the specific heat
105 capacity⁵⁻⁶. For PE, we have $k = 0.33 \text{ W (m}\cdot\text{K)}^{-1}$,⁷ $\rho = 920 \text{ kg m}^{-3}$,⁸ $c_p = 2300 \text{ J}$
106 $(\text{kg}\cdot\text{K})^{-1}$.⁸

107 The convective boundary conditions at the interior and exterior surfaces of the
108 hollow fibre are given by

109
$$k \frac{\partial \theta}{\partial r} \Big|_{r=R_i} = h_w (\theta|_{r=R_i} - \theta_w), \quad (2.3)$$

110
$$-k \frac{\partial \theta}{\partial r} \Big|_{r=R_o} = h_a (\theta|_{r=R_o} - \theta_a), \quad (2.4)$$

111 where h_w and h_a are the heat transfer coefficients of water and air, respectively.

112 Similarly, θ_w and θ_a are respectively the temperatures of water and air. The heat
 113 transfer coefficient of water depends on many factors such as hollow fibre internal
 114 geometry, surface roughness, water physical properties as well as the flow state
 115 (laminar or turbulent), *etc.* In this work, we employ $h_w = 2000 \text{ W (m}^2\cdot\text{K)}^{-1}$, which
 116 provides the best fit of theoretical predictions to the experimental measurements.

117 The initial condition is

118
$$\theta(t=0) = \theta_0, \quad (2.5)$$

119 where θ_0 is the initial temperature of the hollow fibre actuator.

120 The above 1D radial transient heat conduction equations can be solved either
 121 analytically using methods of separation of variables and superposition, which are
 122 well documented in ref. 10, or using numerical methods such as finite difference or
 123 finite element. In this work, we solved this 1D partial differential equation (PDE)
 124 using the PDE solver provided by MATLAB*.

125 Next, we assume that the axial heat conduction rate is much lower compared to
 126 the water flow rate. As such, the heat transfer in the axial direction is mainly
 127 contributed by the flowing water and the radial conduction. Therefore, by substituting
 128 water temperature as a function of axial coordinate and time, i.e. $\theta_w = \theta_w(x, t)$, in
 129 which x is the axial coordinate, into the above radial heat conduction equations, we
 130 can solve the hollow fibre temperature distribution in the axial direction.

131 Water temperature $\theta_w(x, t)$ depends on a few factors, such as initial water
 132 temperature, water flow rate, heat loss due to transfer to hollow fibre actuator and

*<https://www.mathworks.com/help/matlab/math/partial-differential-equations.html>

133 eventually to the ambient air, *etc.* For simplicity, we assume that the heat loss is
 134 relatively low when the flow rate is high. Therefore, we can write the water
 135 temperature as

$$136 \quad \theta_w(x, t) = \theta_{w0} H(\nu t - x), \quad (2.6)$$

137 where θ_{w0} is the initial water temperature, ν is the water flowrate and $H(\cdot)$ is the
 138 Heaviside step function.

139

140 2.2 Torsional actuation modeling

141 Torsional actuation mechanism of the twisted polymer fibres was studied in ref.
 142 10 and 11. The hollow fibre length l , the hollow fibre outer diameter d , and the
 143 helically oriented polymer chain length λ are related through

$$144 \quad \lambda^2 = [(\pi d T)^2 + 1] l^2, \quad (2.7)$$

145 where T is the twist density.

146 Upon heating, polymer chains will expand in the radial direction and contract in
 147 the axial direction, which can be accommodated by the changes in the hollow fibre
 148 length, the diameter and the twist density. In ref. 10, the change of hollow fibre length
 149 is considered small and the following equation for untwisting can be obtained by

$$150 \quad \Delta T = \left(\frac{\Delta \lambda}{\lambda} \frac{1}{\cos^2 \alpha_f} - \frac{\Delta d}{d} \right) T, \quad (2.8)$$

151 where bias angle $\alpha_f = \tan^{-1}(\pi d T)$ ¹².

152 Ref. 11 further assumed that λ is also constant. Therefore, the untwist ΔT is
 153 linearly proportional to the change in the hollow fibre diameter, i.e.

$$154 \quad \Delta T = -\frac{\Delta d}{d} T, \quad (2.9)$$

155 In this work, we employ Eq. (2.8) for torsional actuation modeling. It can be
 156 rewritten as

$$157 \quad \Delta T = \left(\alpha_\lambda \frac{1}{\cos^2 \alpha_f} - \alpha_d \right) \Delta \theta \cdot T, \quad (2.10)$$

158 where α_λ and α_d are the coefficients of thermal expansion of polymer chain in the
 159 axial direction and that of the hollow fibre in the radial direction, respectively. The $\Delta\theta$
 160 is the temperature change obtained from the previous transient heat conduction
 161 analysis. In this work, the average exterior hollow fibre temperature change is
 162 employed as the actuation temperature $\Delta\theta$. Axial and radial coefficients of thermal
 163 expansion of PEHFs were employed as $\alpha_\lambda = -(5.3 \pm 0.4) \times 10^{-4} \text{ K}^{-1}$ and
 164 $\alpha_d = (5.2 \pm 0.7) \times 10^{-4} \text{ K}^{-1}$, which were obtained from thermomechanical analysis
 165 (TMA).

166

167 *2.3 Tensile actuation modeling*

168 The tensile actuation mechanism of the twisted, mandrel-coiled polymer fibre is
 169 well documented in ref. 10, which is adopted in this work. The equation for tensile
 170 actuation is

$$171 \quad \frac{\Delta L}{L} = \frac{l^2}{NL} \Delta T \quad , \quad (2.11)$$

172 where L is the coil length, l is the twisted hollow fibre length, N is the number of coil
 173 turns and ΔT is the amount of untwisting that can be obtained from Eq. (2.10).

174

175 **Supplementary Note 3. The analysis for the optimum hollow fibre geometry**

176 To provide an estimation of the optimum configuration of the hollow fibre, we
 177 assume steady heat solution and constant temperature due to the small thickness of the
 178 hollow fibre. Based on the linear elastic solution for a thin-walled hollow fibre under
 179 pure torsion¹³, the generated torque M is related to the twist through

$$180 \quad M = 2\pi GJT, \quad (3.1)$$

181 in which G is shear modulus, J is the polar second moment of area. For the hollow
 182 fibre section $J = \frac{\pi}{2}(r_o^4 - r_i^4)$ where r_i and r_o are the inner and outer radius of the
 183 hollow fibre, respectively.

184 Applying Eq. (3.1) to the hollow fibre configurations before and after the

185 torsional actuation provides the torque release ΔM :

186
$$\Delta M = \pi G A_0 (r_i^2 + r_o^2) [(I + \alpha_d \Delta \theta)^4 (T + \Delta T) - T], \quad (3.2)$$

187 in which A_0 is the cross-section area of the hollow fibre.

188 We now introduce a simplified version of torsional actuation based on Ref. 10 :

189
$$\Delta T = \left(\frac{d_0}{d} - 1 \right) T. \quad (3.3)$$

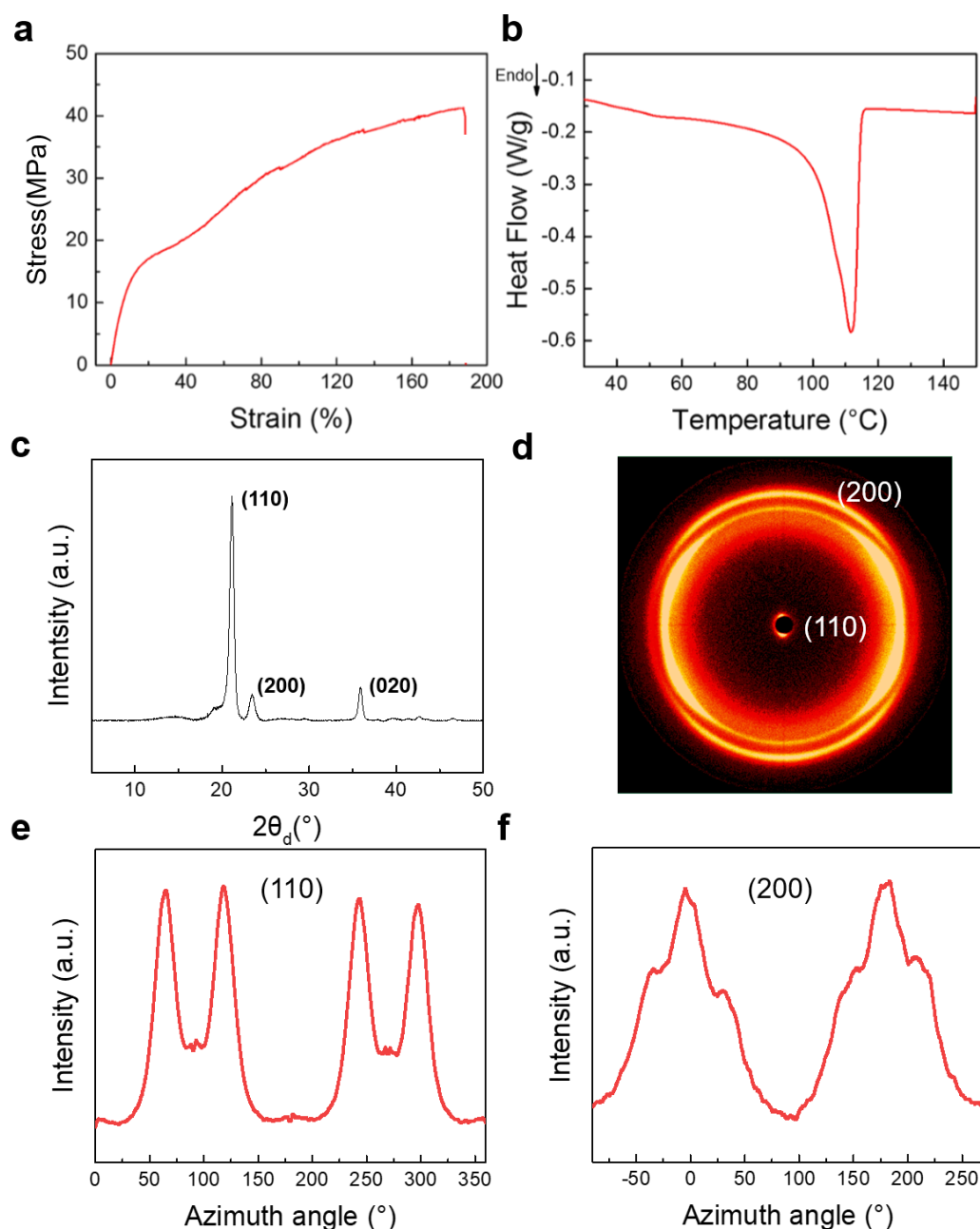
190 Substituting Eq. (3.3) into (3.2) gives

191
$$\Delta M = \pi G A_0 (r_i^2 + r_o^2) [(I + \alpha_d \Delta \theta)^3 - 1] T. \quad (3.4)$$

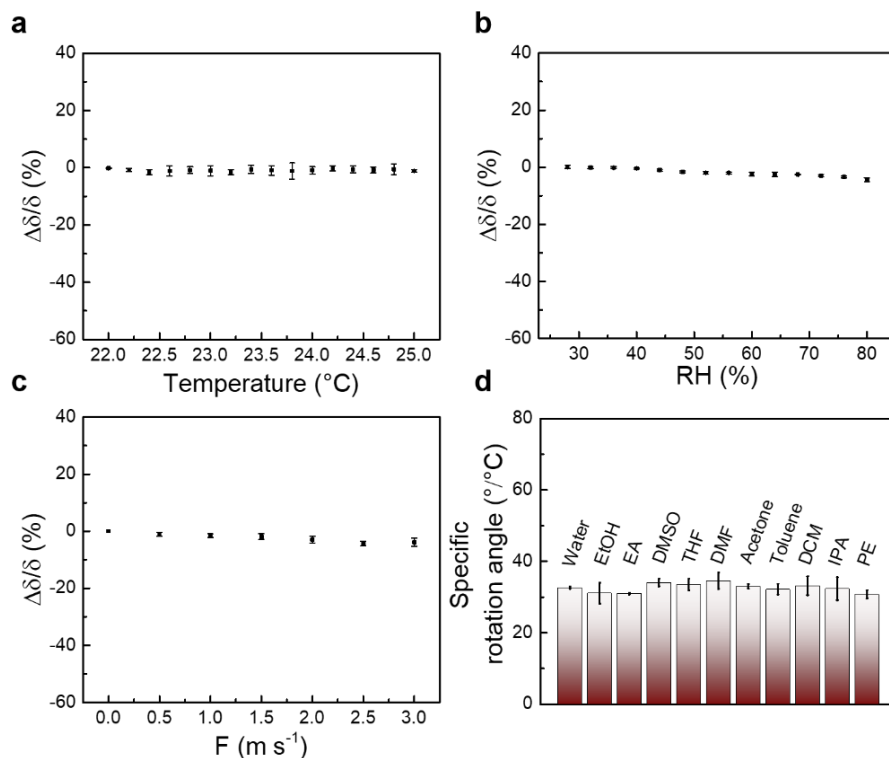
192 Based on Eq. (3.4), we conclude that the optimum geometry for maximizing the
193 torque output is to maximize $(r_i^2 + r_o^2)$, assuming that the cross-section area A_0 (or
194 equivalently the material consumption) is constant. This means that the material
195 should be distributed as far away as possible from the center of the cross-section,
196 which justifies the use of hollow fibre over solid cross-sections. It should be noted
197 that the above theoretically derived optimum geometry is subjected to the constraint
198 that the hollow fibre will not buckle under the applied torque. Thus, in practice a
199 hollow fibre with finite thickness should be used for the proposed actuation
200 application.

201

202 **Supplementary Figures**

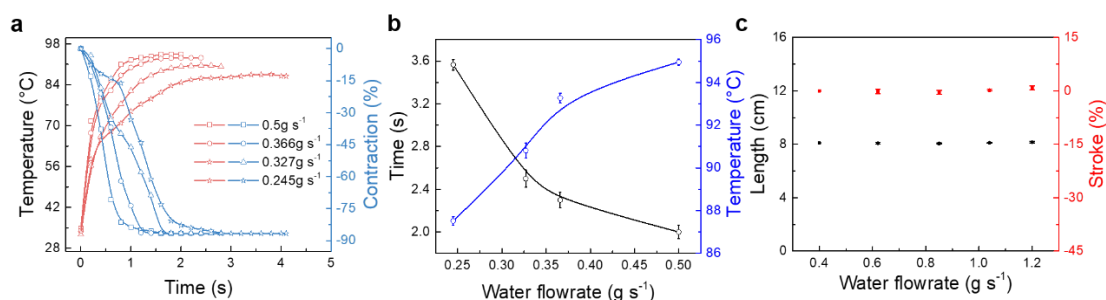


203
 204 **Supplementary Figure 1.** (a) The stress-strain curve of the PEHF₅₈₀₋₉₉₀. The gauge
 205 length was 34 mm, and the extension rate was 20 mm min⁻¹. (b) Differential scanning
 206 calorimetry (DSC) curve of the PEHF₅₈₀₋₉₉₀ with a heating rate of 5°C/min in N₂
 207 atmosphere. The melting enthalpy ($H_m=126.4 \text{ J g}^{-1}$) was obtained by integration of the
 208 endothermic melting peak. By considering that the melting enthalpy of the 100%
 209 crystalline polyethylene was $H_c=287.3 \text{ J g}^{-1}$ ¹⁴, we can obtain the crystallinity of
 210 low-density polyethylene (χ_c) from the equation $\chi_c=H_m/H_c=44\%$. (c) The intensity of
 211 diffraction peaks versus Bragg angle (2θ) in WAXS for PEHF₅₈₀₋₉₉₀. (d) 2D WAXS
 212 diffraction pattern for PEHF₅₈₀₋₉₉₀ showing anisotropic structure. (e, f) The azimuthal
 213 curves of (110) and (200) planes in PEHF₅₈₀₋₉₉₀ were obtained from (d).



214

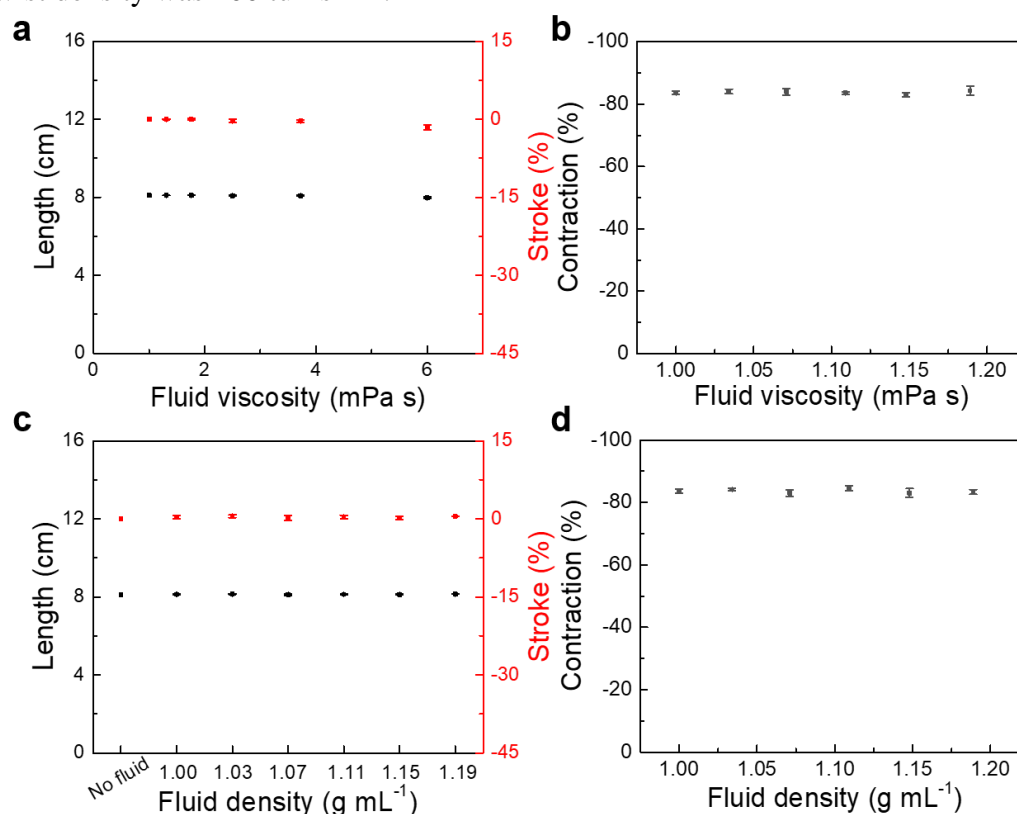
215 **Supplementary Figure 2.** The torsional actuation of the PEHF₅₈₀₋₉₉₀ actuator by
 216 flowing hot water at 1.72 g s⁻¹ at different actuation conditions: (a) at different
 217 environment temperatures, (b) at different environmental relative humidity, and (c) at
 218 different wind speed. (d) The torsional actuation of the PEHF₅₈₀₋₉₉₀ actuator by
 219 flowing different types of 40 °C liquid at a flow rate of 1.72 g s⁻¹. The δ and $\Delta\delta$ are the
 220 rotation angle during actuation and its change. If not specified, the room temperature
 221 is 25 °C, the environmental relative humidity is 40%, and the water flow rate is 1.72 g
 222 s⁻¹. If not specified, error bars in this figure and in the following figures indicate
 223 standard deviations. EtOH: ethanol; EA: ethyl acetate; DMSO: dimethyl sulfoxide;
 224 THF: tetrahydrofuran; DMF: N, N-dimethylformamide; DCM: dichloromethane;
 225 IPA: iso-propanol; PE: petroleum ether.
 226



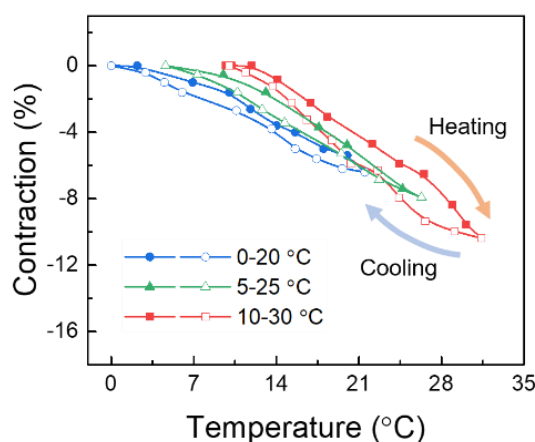
227

228 **Supplementary Figure 3.** (a) Contraction and hollow fibre surface temperature as a
 229 function of time for a homochiral PEHF₅₈₀₋₉₉₀ actuator by flowing 95 °C water with
 230 different flow rates. (b) Response time and hollow fibre surface temperature as a
 231 function of water flow rate for a homochiral PEHF₅₈₀₋₉₉₀ actuator by flowing 95 °C
 232 water. (c) Coil length and actuation stroke of the homochiral PEHF₅₈₀₋₉₉₀ actuator by

233 flowing 25 °C water at different flow rates. The spring index was 4.0, and the inserted
 234 twist density was 200 turns m⁻¹.

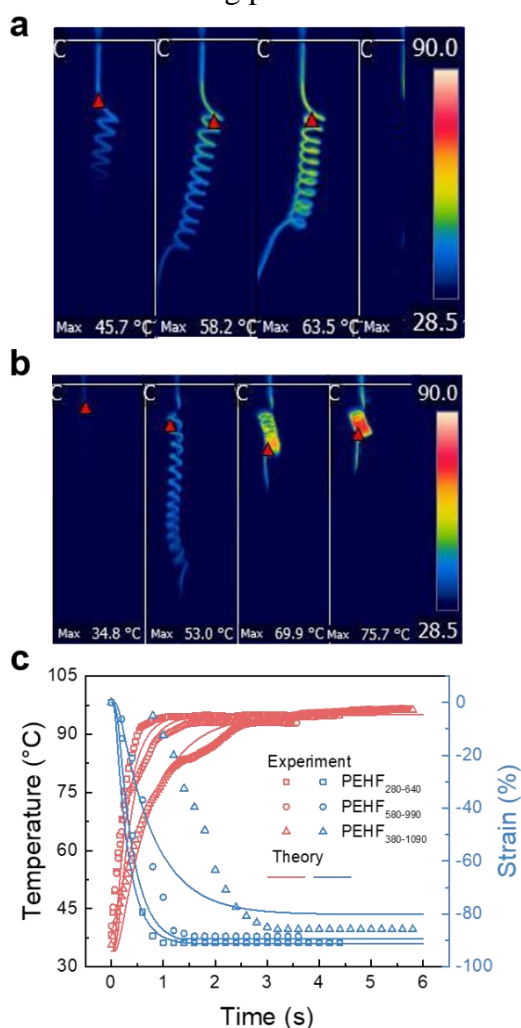


235
 236 **Supplementary Figure 4.** (a) The initial coil length and actuation stroke of the
 237 homochiral PEHF₅₈₀₋₉₉₀ actuator by flowing 25 °C glycerol/water solution with
 238 different viscosity. (b) Actuation stroke of the PEHF₅₈₀₋₉₉₀ actuator by flowing 80 °C
 239 glycerol/water solution with different viscosity. (c) The initial coil length and
 240 actuation stroke of the PEHF₅₈₀₋₉₉₀ actuator by flowing 25 °C aqueous NaCl solution
 241 with different densities. (d) Actuation stroke of the PEHF₅₈₀₋₉₉₀ actuator by flowing
 242 80 °C aqueous NaCl solution with different densities. The spring index was 4.0, and
 243 the twist density was 200 turns m⁻¹.

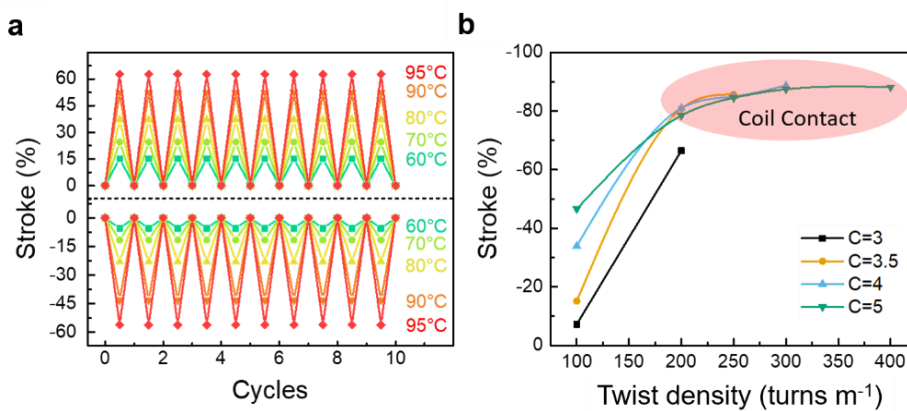


244
 245 **Supplementary Figure 5.** The contraction as a function of temperature for
 246 PEHF₅₈₀₋₉₉₀ actuator driven by consecutively flowing hot water and cold water with a
 247 temperature range. The spring index was 4.0, the inserted twist density was 200 turns

248 m^{-1} , and the flow rate was 1.72 g s^{-1} . The solid symbols indicate the heating process,
 249 and the open symbols indicate the cooling process.

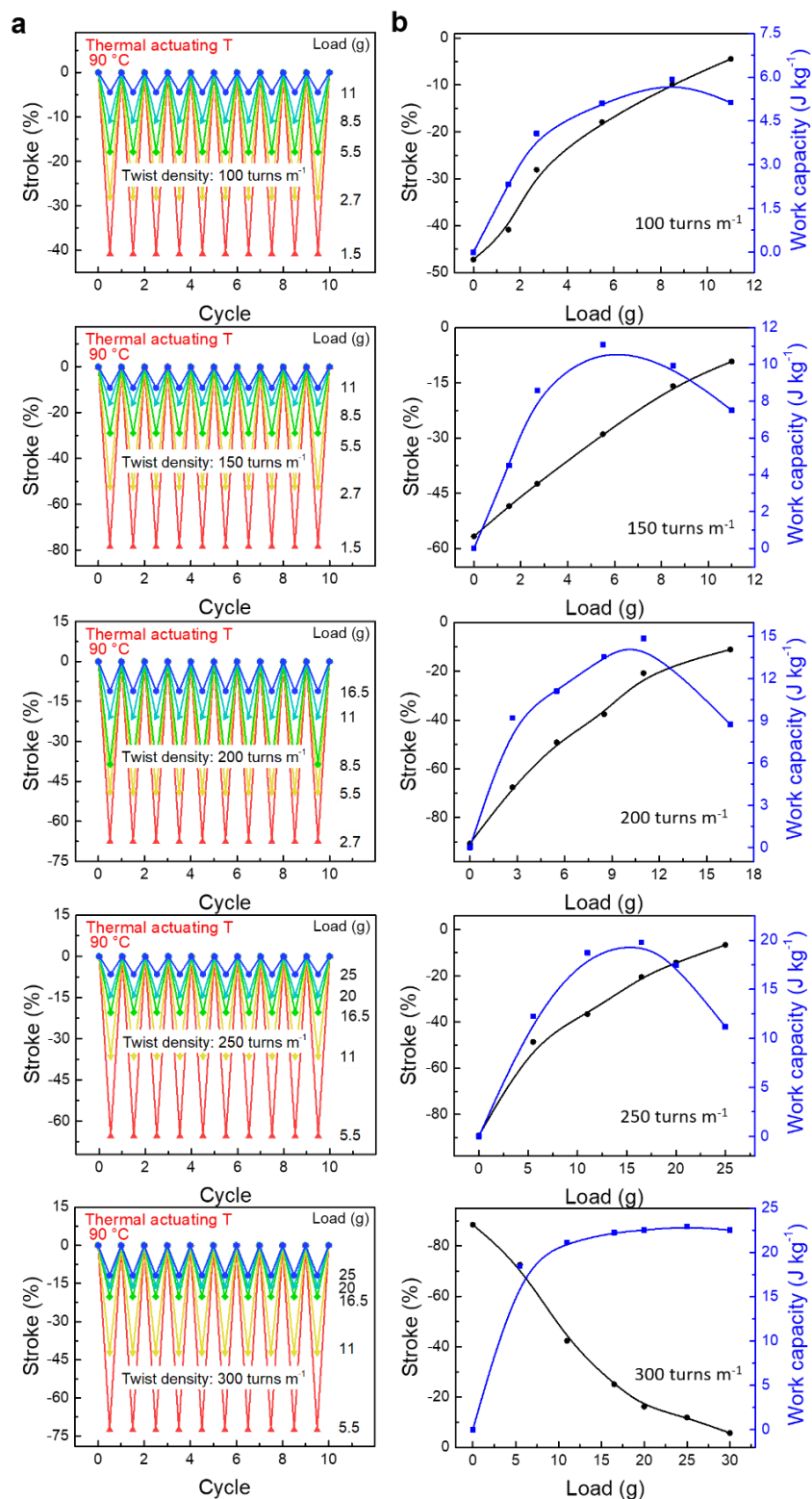


250
 251 **Supplementary Figure 6.** (a, b) Serial infrared images of the homochiral
 252 PEHF₃₈₀₋₁₀₉₀ (a) and PEHF₂₈₀₋₆₄₀ (b) actuators by flowing 95 °C water at different time
 253 (0 s, 0.2 s, 0.4 s, and 0.6 s). (c) Experimental results and theoretical analysis of the
 254 actuation strain and surface temperature of the PEHF actuators as a function of time.



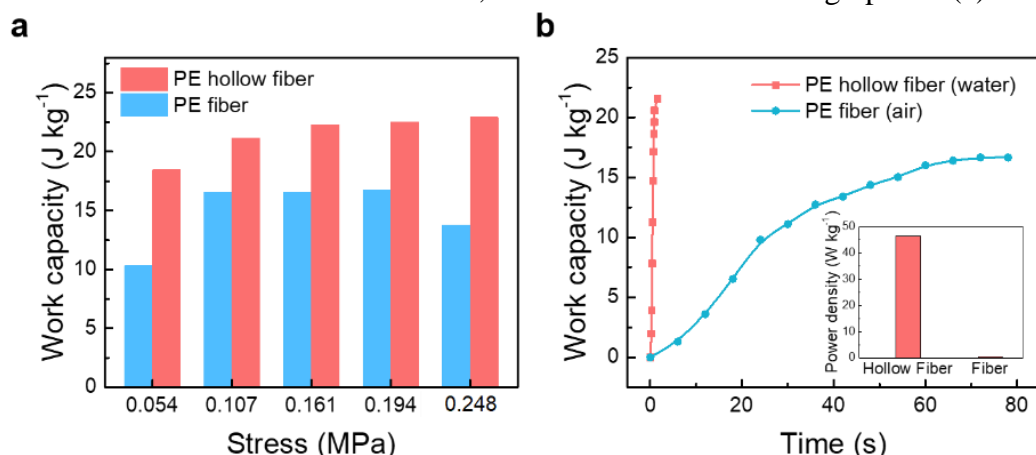
255
 256 **Supplementary Figure 7.** (a) Maximum actuation stroke for ten consecutive
 257 heating/cooling cycles for heterochiral and homochiral load-free PEHF₅₈₀₋₉₉₀ actuators
 258 driven by alternatively flowing hot water and 25 °C water. (b) Maximum actuation

259 stroke as a function of twist density for load-free PEHF₅₈₀₋₉₉₀ homochiral actuators
 260 with different spring indexes driven by 80 °C flowing water.

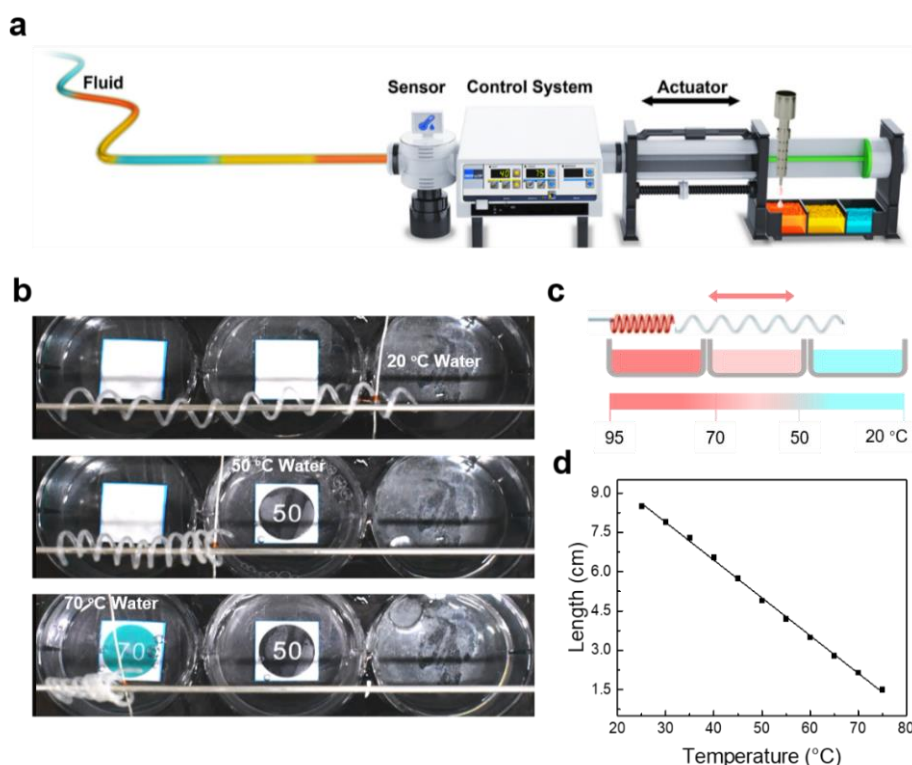


261
 262 **Supplementary Figure 8.** (a) Maximum contraction at different mass of the load for
 263 ten consecutive heating/cooling cycles of homochiral PEHF₅₈₀₋₉₉₀ actuators with a
 264 spring index of 4.0 and an inserted twist of 100, 150, 200, 250 and 300 turns m⁻¹ by
 265 alternatively flowing 90 °C and 25 °C water. (b) The actuation stroke and the work

266 capacity as a function of the mass of the applied load for the homochiral PEHF₅₈₀₋₉₉₀
 267 actuators with different twist densities, which were derived from graphs in (a).

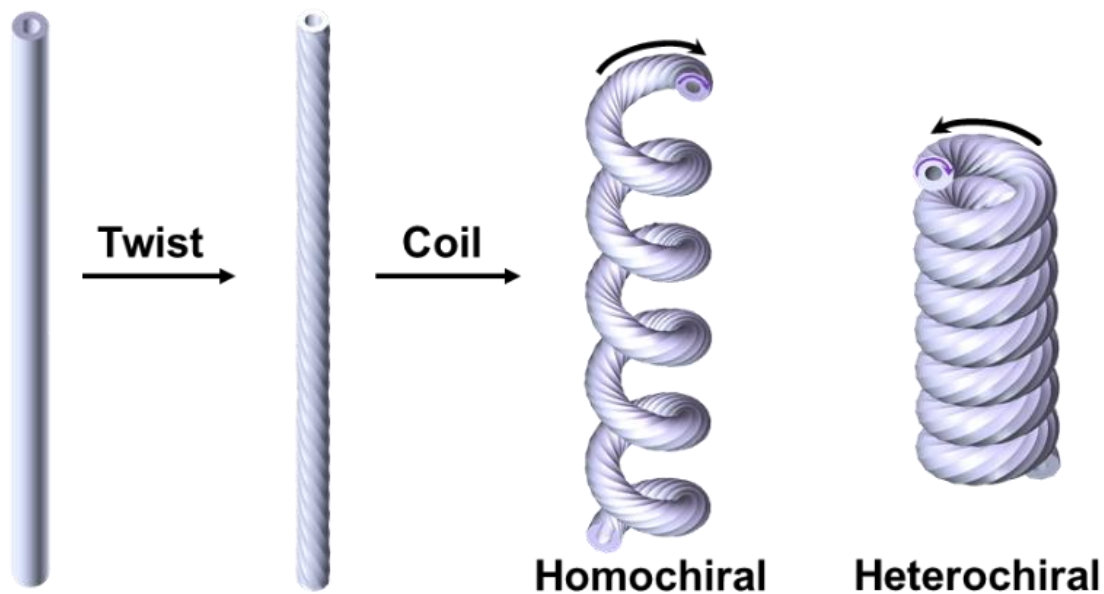


268
 269 **Supplementary Figure 9.** (a) Comparison of the work capacity at different isobaric
 270 stress for the homochiral PEHF₅₈₀₋₉₉₀ actuator driven by 90 °C water and for the PE
 271 solid fibre actuator driven by 90 °C water. (b) Comparison of the time-dependence of
 272 work capacity for the homochiral PEHF₅₈₀₋₉₉₀ actuator driven by 90 °C water and for
 273 the PE solid fibre driven by 90 °C air at the stress of 107 kPa, and the inset was the
 274 comparison of power density. The twist density was 300 turns m⁻¹, and the spring
 275 index was 4.0. The isobaric stress was calculated as the weight of the load divided by
 276 the cross-sectional area of sheath of the hollow fibre.

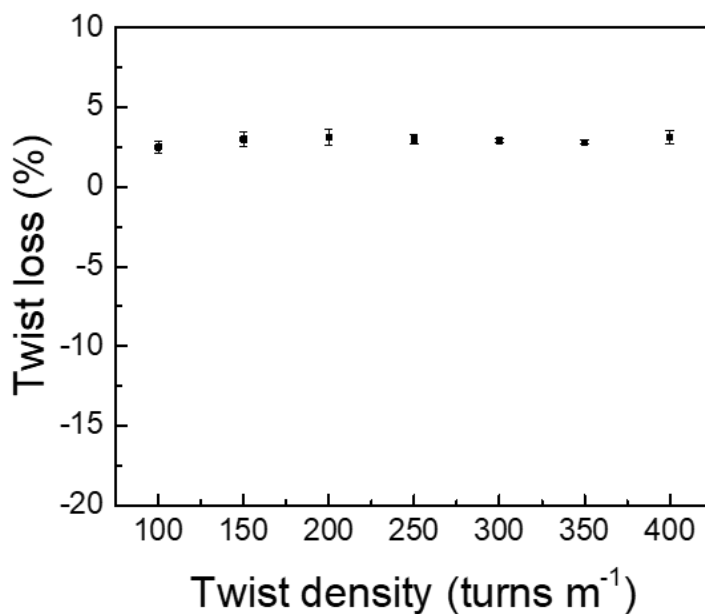


277
 278 **Supplementary Figure 10.** (a) Schematic diagram of a conventional microfluidic
 279 manipulation system containing liquid sensing, transporting and actuating devices.
 280 Optical images (b) and schematic illustration (c) of the contractile hollow fibre
 281 actuators used for sensing the liquid temperature and sorting the liquid into the

282 desired vessels. (d) The length of the hollow fibre actuators as a function of the
283 temperature of the flowing water for the homochiral PEHF₅₈₀₋₉₉₀ actuators.



284
285 **Supplementary Figure 11** Schematic of the twisted and coiled hollow-fibre
286 actuators
287



288
289 **Supplementary Figure 12.** The twist loss of the hollow fibre after twist insertion,
290 thermal annealing with both-end tethered at 108 °C for 1h, and cooling down to room
291 temperature and removing the tethering.
292

293 **Supplementary Tables:**

294 **Supplementary Table 1.** Response time and actuation parameters of the torsional

295 PEHF actuators in this work.

Materials	Twist density (turns m ⁻¹)	$\Delta\theta$ (°C)	Flowrate (g s ⁻¹)	Response time (s)
PEHF ₅₈₀₋₉₉₀	100	70	0.5	2.5
PEHF ₅₈₀₋₉₉₀	200	70	0.5	2.1
PEHF ₅₈₀₋₉₉₀	300	70	0.5	1.9
PEHF ₅₈₀₋₉₉₀	400	70	0.5	1.7
PEHF ₅₈₀₋₉₉₀	400	55	0.5	1.8
PEHF ₅₈₀₋₉₉₀	400	45	0.366	2.7
PEHF ₅₈₀₋₉₉₀	400	35	0.327	1.1
PEHF ₅₈₀₋₉₉₀	400	5	1.72	1.0

296

297 **Supplementary Table 2.** Response time and actuation parameters of the tensile PEHF

298 actuators in this work.

Materials	Twist density (turns m ⁻¹)	$\Delta\theta$ (°C)	Flowrate (g s ⁻¹)	Response time (s)
PEHF ₅₈₀₋₉₉₀	200	35	1.72	2.5
PEHF ₅₈₀₋₉₉₀	200	45	1.72	1.6
PEHF ₅₈₀₋₉₉₀	200	55	1.72	1.4
PEHF ₅₈₀₋₉₉₀	200	70	1.72	0.88
PEHF ₅₈₀₋₉₉₀	200	70	0.5	1.4
PEHF ₅₈₀₋₉₉₀	200	70	0.366	1.6
PEHF ₅₈₀₋₉₉₀	200	70	0.327	1.7
PEHF ₅₈₀₋₉₉₀	200	70	0.245	2.8
PEHF ₂₈₀₋₆₄₀	310	70	0.366	1
PEHF ₃₈₀₋₁₀₉₀	182	70	0.366	3.0

299

300

301 **Supplementary Table 3.** Resolution and the coil parameters of the homochiral PEHF
 302 hollow fibre actuators in this work.

Fibre Length (l , mm)	Coil Length (L , mm)	Number of Turns (N)	Twist density (T , turns mm^{-1})	Resolution ($l^2\Delta T/NL$, 10^{-3})
200	105	11	0.30	-3.0%
200	105	11	0.25	-2.0%
200	105	11	0.20	-1.3%
200	105	11	0.15	-0.78%
200	105	11	0.10	-0.43%
200	115	12	0.30	-2.5%
200	115	12	0.25	-1.7%
200	115	12	0.20	-1.1%
200	115	12	0.15	-0.66%
200	115	12	0.10	-0.36%
200	130	15	0.25	-1.2%
200	130	15	0.20	-0.78%
200	130	15	0.10	-0.26%
200	145	18	0.20	-0.58%
200	145	18	0.10	-0.19%

303

304 **References:**

- 305 1. Gururajan G. & Ogale A. A. Real-time crystalline orientation measurements during low-density
306 polyethylene blown film extrusion using wide-angle X-ray diffraction. *Polym. Eng. Sci.* **52**,
307 1532-1536 (2012).
- 308 2. Dees J. R. & Spruiell J. E. Structure development during melt spinning of linear polyethylene
309 fibers. *J Appl. Polym. Sci.* **18**, 1053-1078 (1974).
- 310 3. Choy C., Chen F. & Young K. Negative thermal expansion in oriented crystalline polymers. *J.*
311 *Polym. Sci. Polym. Phys. Ed.*, **19**, 335-352 (1981).
- 312 4. Stein R. S. The X-ray diffraction, birefringence, and infrared dichroism of stretched polyethylene.
313 II. Generalized uniaxial crystal orientation. *J Polym. Sci.* **31**, 327-334 (1958).
- 314 5. Jiang P., Li X., Xu R. & Zhang F. Heat extraction of novel underground well pattern systems for
315 geothermal energy exploitation. *Renew. Energ.* **90**, 83-94 (2016).
- 316 6. Olga 7 user manual. (2010).
- 317 7. Allan J., Pinder H. & Dehouche Z. Enhancing the thermal conductivity of ethylene-vinyl acetate
318 (EVA) in a photovoltaic thermal collector. *AIP Advances* **6**, 035011 (2016).
- 319 8. Zhang Z. & Lafleur P. G. Investigation of LDPE blown film instability induced by air cooling.
320 *Polym. Eng. Sci.* **48**, 1504-1510 (2008).
- 321 9. Hahn D. W. & Özisik M. N. Heat conduction. John Wiley & Sons (2012).
- 322 10. Haines C. S., et al. Artificial Muscles from Fishing Line and Sewing Thread. *Science* **343**,
323 868-872 (2014).
- 324 11. Aziz S, Naficy S, Foroughi J, Brown H. R. & Spinks G. M. Controlled and scalable torsional
325 actuation of twisted nylon 6 fiber. *J. Polym. Sci., Part B: Polym. Phys.* **54**, 1278-1286 (2016).
- 326 12. Leng X, et al. Tuning the reversibility of hair artificial muscles by disulfide cross-linking for
327 sensors, switches, and soft robotics. *Mater. Horiz.* **8**, 1538-1546 (2021).
- 328 13. Budynas R. G. & Nisbett J. K., *Shigley's Mechanical Engineering Design* (McGraw-Hill, New
329 York, 2008).
- 330 14. Kong J., Fan X., Xie Y. & Qiao W. Study on molecular chain heterogeneity of linear low-density
331 polyethylene by cross-fractionation of temperature rising elution fractionation and successive
332 self-nucleation/annealing thermal fractionation. *J. Appl. Polym. Sci.* **94**, 1710-1718 (2004).