

Supplementary Materials for  
**Ultrapерmeable 2D-channelled graphene-wrapped zeolite molecular sieving  
membranes for hydrogen separation**

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**The PDF file includes:**

Supplementary Text  
Figs. S1 to S12  
Tables S1 to S5  
Legend for movie S1  
References

**Other Supplementary Material for this manuscript includes the following:**

Movie S1

## Supplementary Text

### Characterization

#### #1 Field Emission Scanning Electron Microscopy (FE-SEM)

G-MFI nanoparticles (Fig. 1B) were observed using a scanning electron microscope (SEM, JEOL, JSM 7000F). Additionally, we observed MFI crystals and the cross-sectional area of the G-MFI membrane (Fig. S1I, J). The G-MFI particles and the membrane were measured by simply placing the membrane material onto a carbon tape. The cross-sectional area of the G-MFI membrane was observed by tilting the sample specimen during observation.

#### #2 Thermogravimetric analysis

Thermal analyses for the membranes were carried out using a thermogravimetric analysis (TGA) apparatus (STA7200, Hitachi) in an air atmosphere. We observed the weight losses for the corresponding components of our G-MFI membrane (Fig. S1D). Weight losses arising from the decomposition of  $\text{NH}_4\text{Cl}$  and graphene were observed.

#### #3 Raman spectroscopy

The defects on the G-MFI membrane after thermal treatment in air at different temperatures were examined with Raman spectroscopy (Renishaw inVia, IAB 8303). Raman spectra were observed after thermal treatment of the G-MFI membrane in the temperature range of 300 – 723 K (Fig. S1E).

#### #4 Transmission Electron Microscopy (TEM)

We grounded the G-MFI membrane into powder, dispersed with ethanol, and placed on a Cu grid for observation by transmission electron microscopy (TEM, JEOL JEM 2100). In a similar way, we prepared the graphene and MFI for the TEM observations. The TEM examination of the G-MFI membrane was carried out to observe the graphene-zeolite crystal face contacts (Fig. 1C,D). The nanowindows on the graphene were observed (Fig. 1E, Fig. S2,3). We observed contact between graphene and the (010) face of the MFI zeolite crystal, and the same orientation of the MFI crystal was used in the MD simulation study.

#### #5 X-ray diffraction and small-angle X-ray scattering

An X-ray diffractometer (XRD) (Rigaku,  $\text{CuK}\alpha$ ,  $\lambda = 0.15406$  nm) was used for diffraction measurement of the G-MFI membrane. XRD was measured before and after graphene wrapping (Fig. S5A) in transmission mode. We measured small-angle X-ray scattering (SAXS) for the G-MFI membrane and MFI membrane (Fig. S5B, C) using the synchrotron X-ray beam line BL8S3: wavelength (0.092 nm), camera length (0.45 m), and detector (R-AXIS, PILATUS, CCD) at the Aichi synchrotron radiation center. We calculated the radius of gyration using the Guinier approximation to discuss the porous structure of the G-MFI membrane.

#### #6 Gas adsorption

$\text{N}_2$  adsorption at 77 K for G-MFI in its powder and membrane forms was measured with an automatic volumetric gas adsorption apparatus (Quantacrome Autosorb iQ-MP) (Figs. 2A,B).  $\text{H}_2$ ,  $\text{CO}_2$ , and  $\text{CH}_4$  adsorption for the G-MFI membrane was measured using a

Microtrac MRB (BELSORP MAX) apparatus (Fig. 2C). The sample was degassed at 523 K for 3 h prior to the adsorption measurements.

#### #7 Gas permeance

Gas permeability measurements were carried out using a custom-made apparatus. Permeation tests for graphene-wrapped MFI membranes were carried out by using a pressure-driven method at 298 K. A membrane was fixed into a membrane module using O-rings. The pressure on the feed and permeate sides were set at 200 kPa and 100 kPa, respectively, and the permeate flow rate was measured using a bubble flow meter. In the case of separation tests for equimolar mixtures of H<sub>2</sub>/CH<sub>4</sub> and CO<sub>2</sub>/CH<sub>4</sub> the permeate flow of the mixture was measured by a bubble flow meter, and the composition of the permeate gas was determined using gas chromatography with a thermal conductivity detector (GC-TCD). The measurements were conducted after 1h of purging the equimolar H<sub>2</sub>/CH<sub>4</sub> and CO<sub>2</sub>/CH<sub>4</sub> mixtures at a feed pressure of 200 kPa, assuming that the H<sub>2</sub>, CO<sub>2</sub>, and CH<sub>4</sub> were pre-adsorbed on the G-MFI membrane. The steady permeabilities and selectivities were measured.

#### #8 Diffusion coefficient estimation by solution diffusion transport

The diffusion coefficient is related to the permeance by the well-known permeation equation (45):

$$J = -\frac{DS\Delta p}{\delta}, \quad (1)$$

where  $J$  is the molar flux of the permeate gas (mol m<sup>-2</sup> s<sup>-1</sup>),  $D$  is the diffusion coefficient (m<sup>2</sup> s<sup>-1</sup>),  $S$  is the solubility coefficient (mol m<sup>-3</sup> Pa<sup>-1</sup>),  $\Delta p$  is the transmembrane pressure (Pa), and  $\delta$  is the thickness of the membrane (m).

The  $J$  to  $\Delta p$  ratio is equal to the permeance  $P$  (mol m<sup>-2</sup> s<sup>-1</sup> Pa<sup>-1</sup>), and the diffusion coefficient is given by the following equation:

$$D = \frac{P\delta}{S} \quad (2)$$

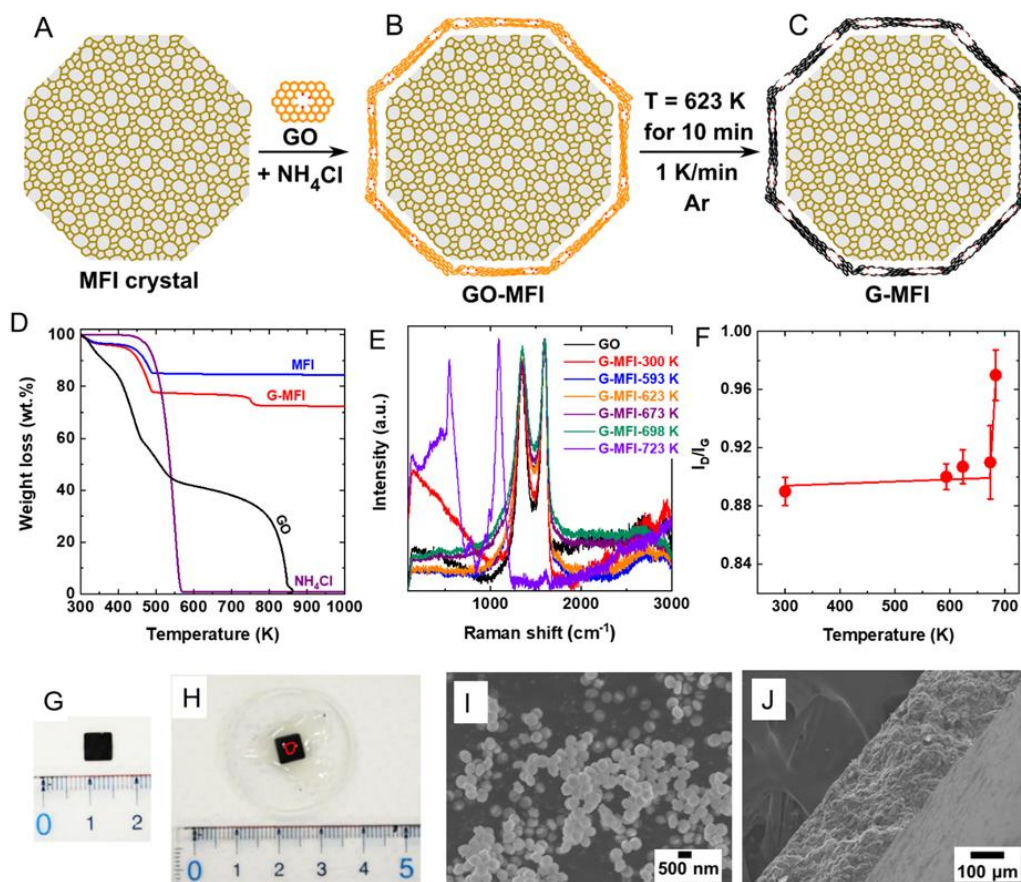
The permeance of H<sub>2</sub> through the G-MFI membrane was  $1.3 \times 10^{-5}$  mol m<sup>-2</sup> s<sup>-1</sup> Pa<sup>-1</sup> and the thickness was 150 μm. The solubility coefficient of H<sub>2</sub> in the G-MFI membrane was determined using a molecular dynamics simulation model based on Henry's law:

$$S = \frac{c}{p} \quad (3)$$

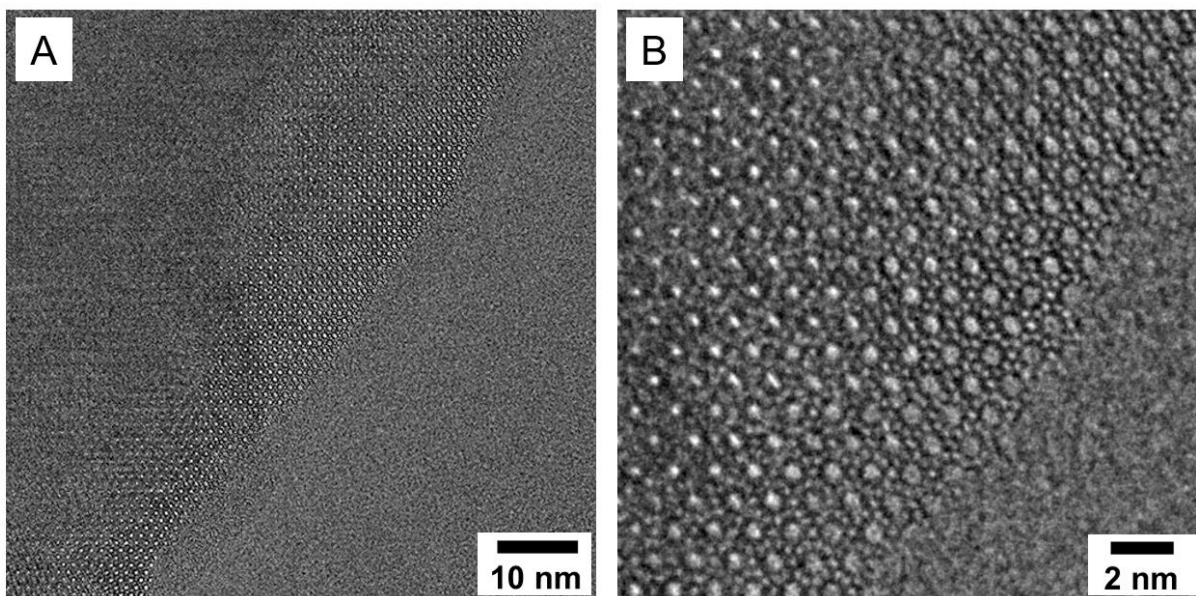
where  $c$  is the concentration of H<sub>2</sub> in the G-MFI membrane (mol m<sup>-3</sup>) and  $p$  is the pressure of H<sub>2</sub> over the membrane of 8.5 MPa. The concentration of H<sub>2</sub> was calculated from the number of H<sub>2</sub> molecules in the upper part of the G-MFI membrane model using MD simulations. The average number of H<sub>2</sub> molecules in the upper part of the membrane model was 4.0 after 3 ns of simulation time. The volume of the upper part of the G-MFI

membrane model was  $3.8 \times 10^{-27} \text{ m}^3$ , and a solubility coefficient of  $2.0 \times 10^{-4} \text{ mol m}^{-3} \text{ Pa}^{-1}$  was obtained using Equation (3). A diffusion coefficient of  $9.5 \times 10^{-2} \text{ cm}^2 \text{ s}^{-1}$  was calculated using Equation (2).

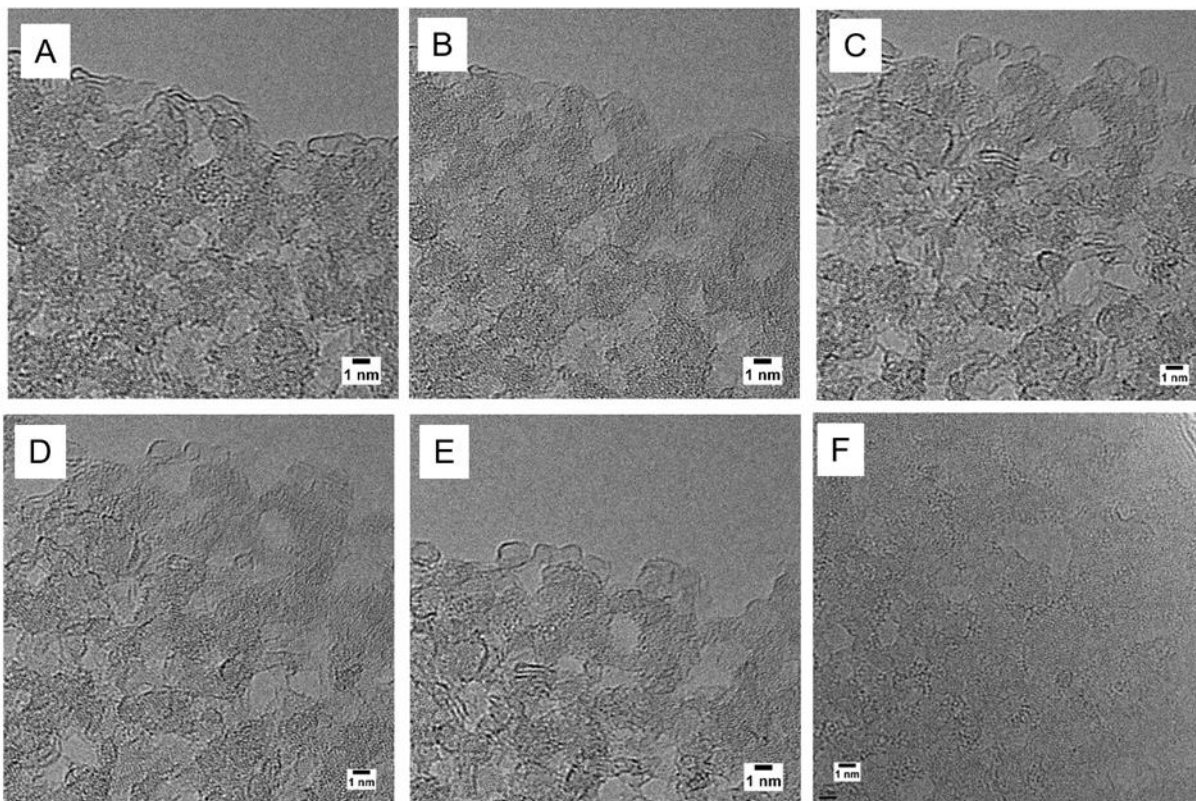
The diffusion coefficient for  $\text{CH}_4$  obtained using the aforementioned method was  $1.1 \times 10^{-4} \text{ cm}^2 \text{ s}^{-1}$ . The ratio of the diffusion coefficients of  $\text{H}_2$  and  $\text{CH}_4$  was calculated as the diffusivity selectivity. The ratio of the solubility coefficients of  $\text{H}_2$  and  $\text{CH}_4$  was calculated as the solubility selectivity, as described in the main text.



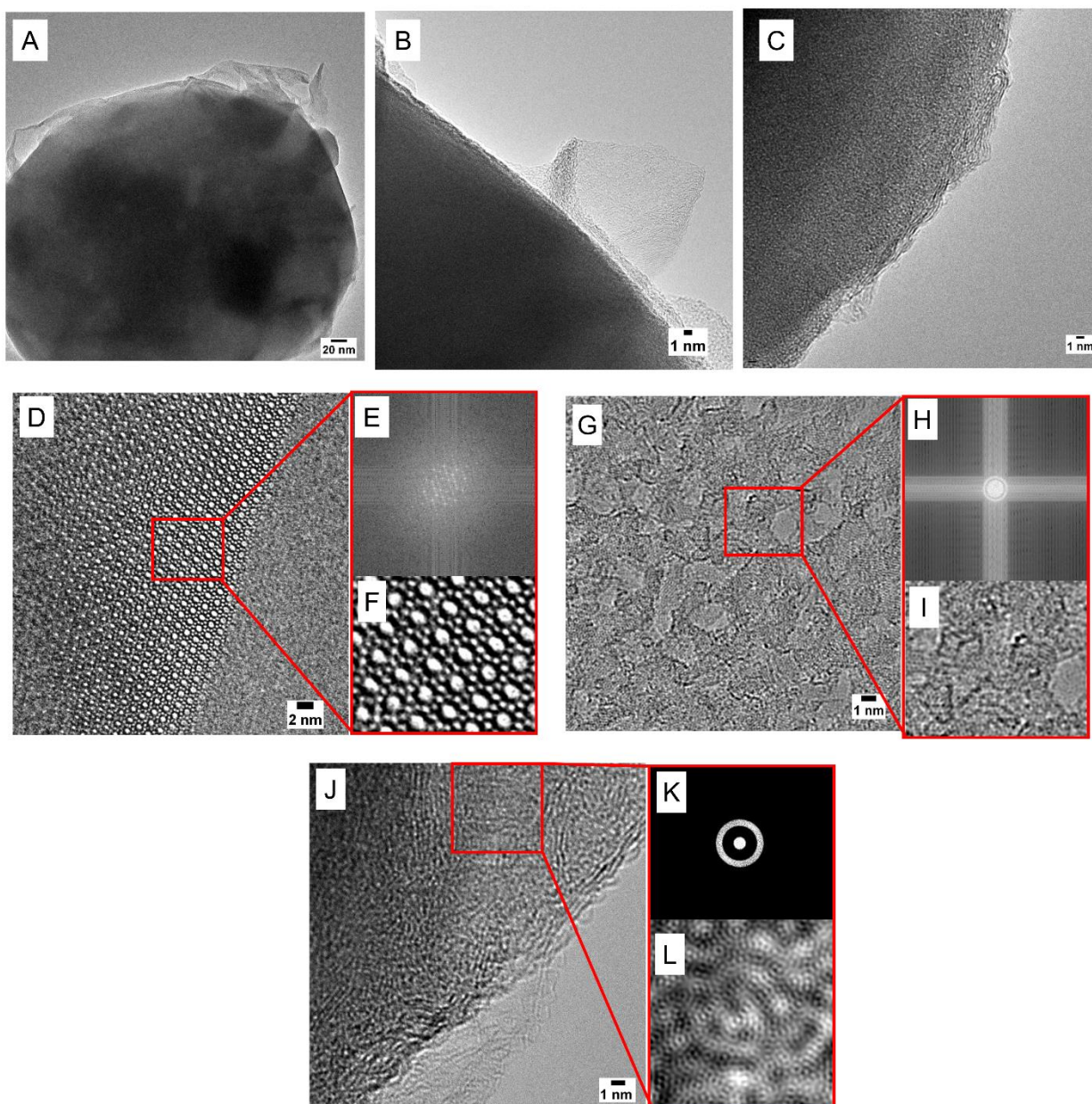
**Fig. S1. Flowchart of the fabrication route for the graphene-wrapped MFI (G-MFI) membrane, starting from the wrapping of a MFI crystal with GO to the final product of a G-MFI crystal and its thermal stability characterization.** (A) MFI crystal. (B) GO-wrapped MFI crystal. (C) G-MFI crystal. (D) TG curves for G-MFI crystals in air atmosphere at a heating rate of  $5 \text{ K min}^{-1}$ . (E) Raman spectra ( $\lambda = 532 \text{ nm}$ ) obtained for G-MFI crystals after thermal treatment in an air atmosphere at different temperatures. (F) The  $I_D/I_G$  ratio against temperature, showing that the  $I_D/I_G$  ratio almost does not increase with heating up to 673 K. (G) G-MFI membrane in pellet form. (H) G-MFI membrane mounted on a polyacrylate holder with araldite adhesive. The active area of  $3.2 \times 10^{-6} \text{ m}^2$  marked by the red circle was measured using ImageJ software. (I) SEM image of the MFI crystals, showing their spherical shape. (J) SEM image showing the cross-section of the membrane thickness.



**Fig. S2. TEM images of the (010) MFI crystal face.** (A-B) Elliptical nanochannels of the MFI crystal at different magnifications.

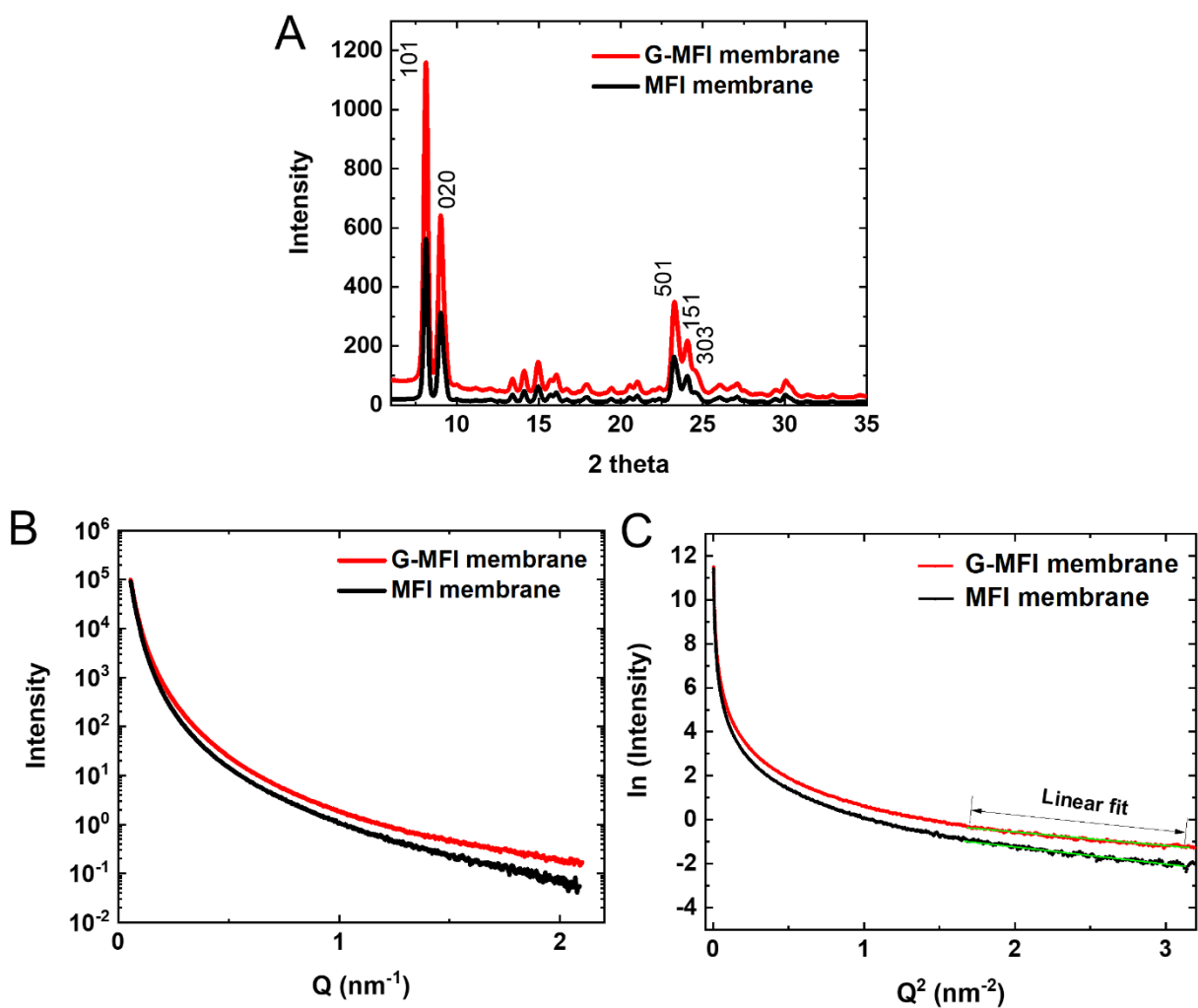


**Fig. S3. TEM images of nanowindows on graphene for graphene-wrapped MFI (G-MFI) powder. (A-F) Nanowindows for different G-MFI powder samples observed at different positions.**

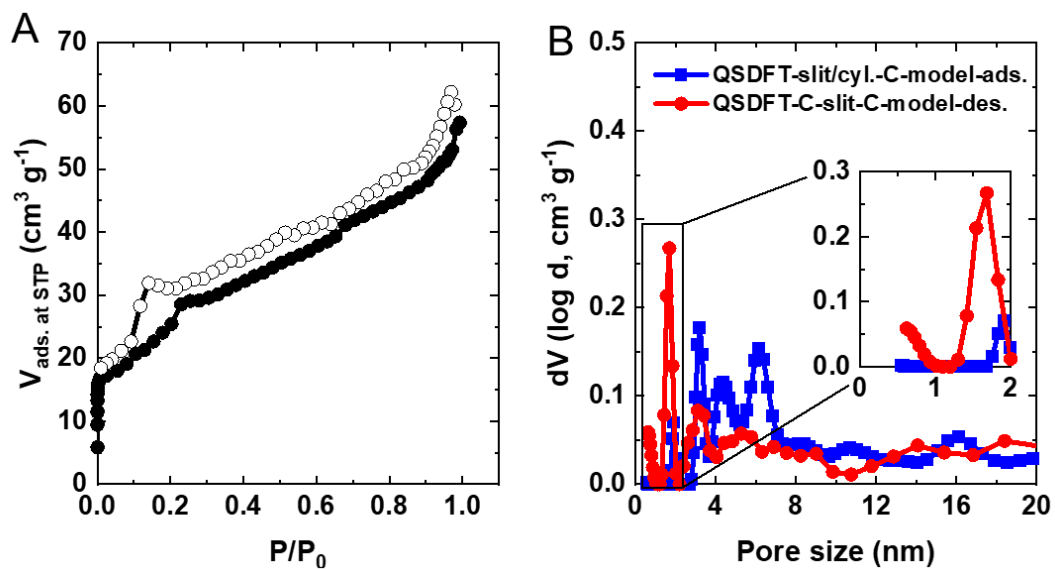


**Fig. S4. TEM images of the G-MFI zeolites.** (A) MFI zeolite wrapped with graphene. (B) Graphene layer attached to the surface of the MFI zeolite. (C) The graphene layers attached to the MFI crystal surface. (D) MFI crystal. (E) Fast Fourier Transform (FFT) and (F) inverse FFT for the SAED of the MFI crystal. (G) Graphene layer with nanowindows. (H) FFT and (I) IFFT for the SAED of the graphene. (J) Graphene layer on the MFI crystal surface. (K) FFT and (L) IFFT for the SEAD of the G-MFI.

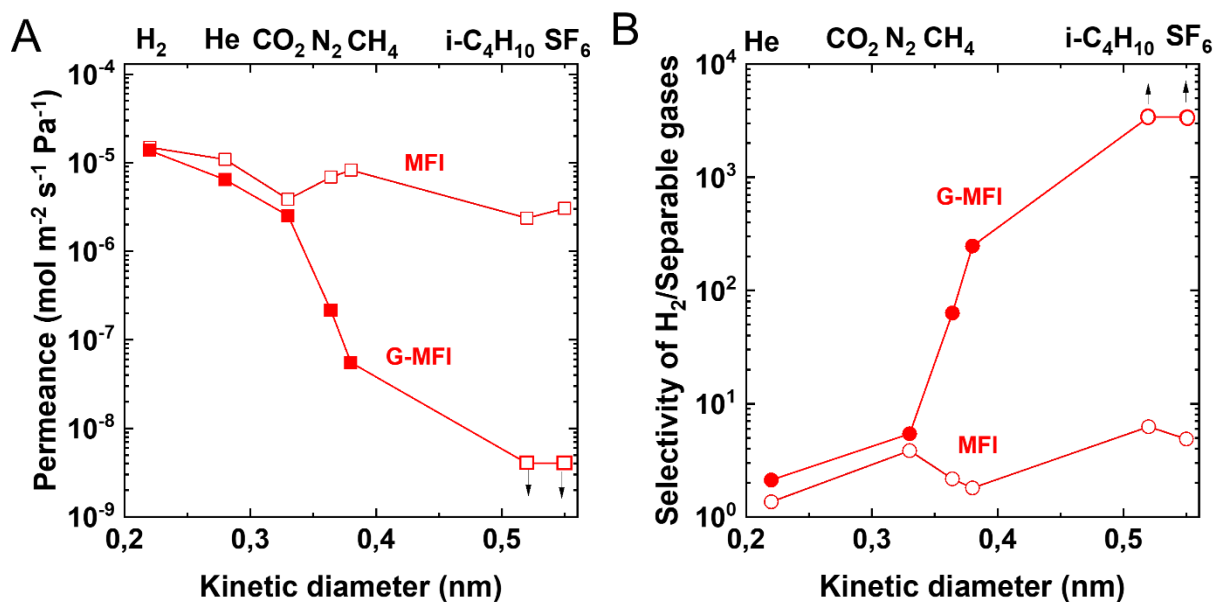




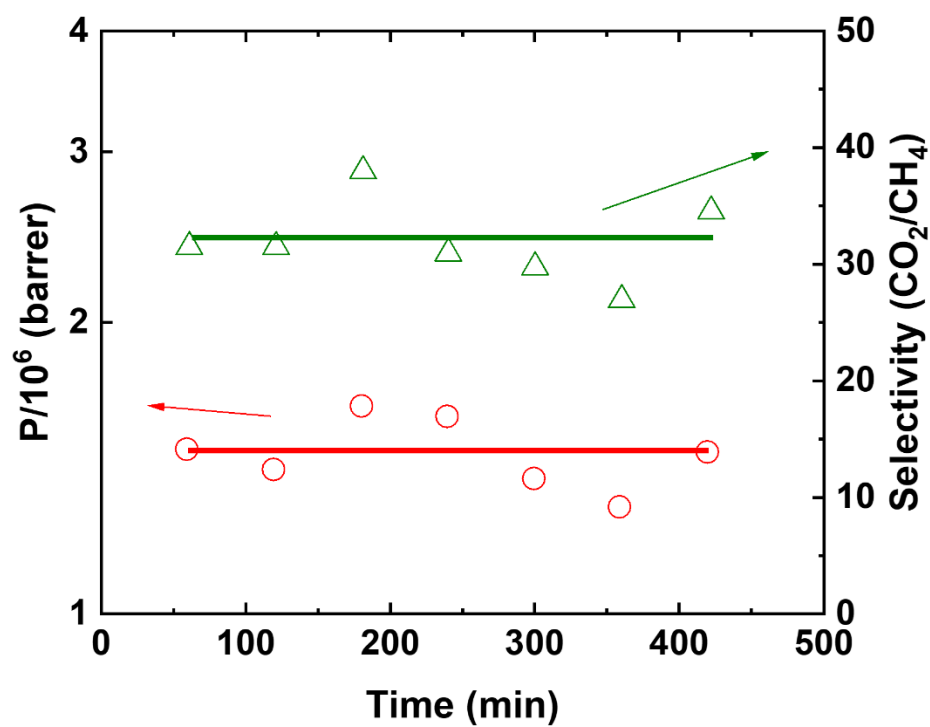
**Fig. S5. XRD and SAXS of the G-MFI and MFI membranes.** (A) XRD patterns for the graphene-wrapped MFI (G-MFI) and MFI membranes. (B) SAXS profiles for G-MFI and MFI membranes; intensity vs. scattering vector ( $Q$ ). (C) Guinier plot. The radii of gyration for G-MFI and MFI membranes of 0.48 nm and 0.60 nm were determined from the linear region using the Guinier approximation (Table S4).



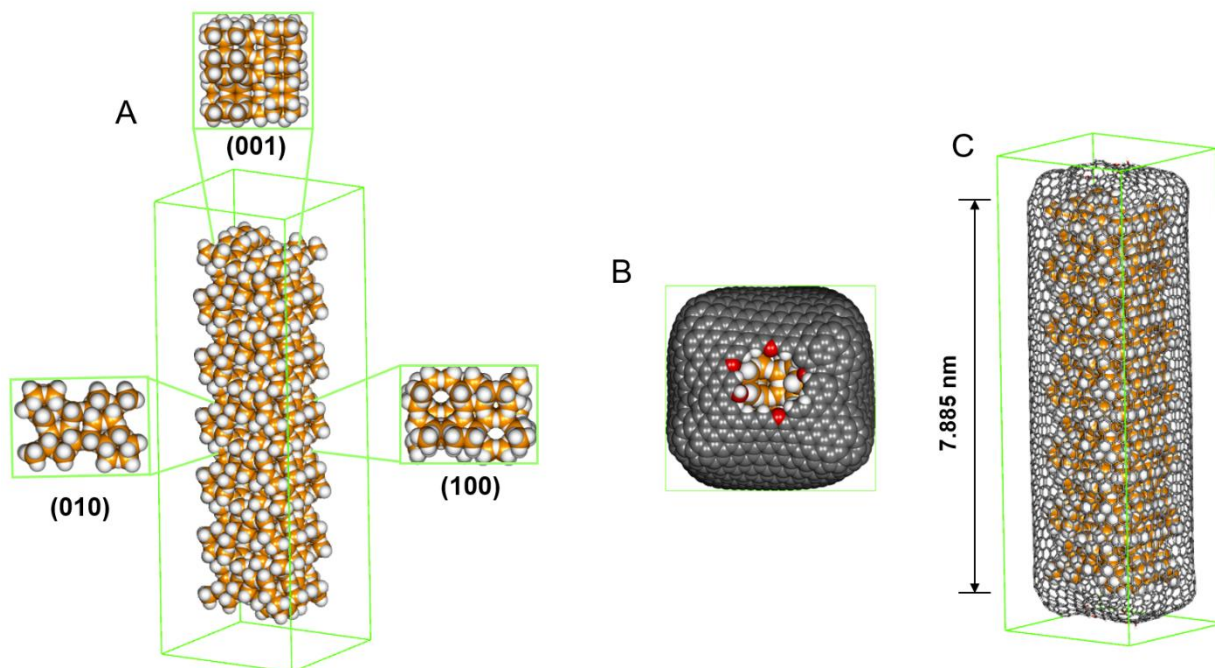
**Fig. S6. The N<sub>2</sub> adsorption isotherm at 77 K and pore size distribution.** (A) The N<sub>2</sub> adsorption isotherm obtained by subtraction of the adsorbed volume of N<sub>2</sub> of G-MFI with that of MFI membrane. (B) Pore size distribution of the G-MFI membrane obtained by QSDFT model. The QSDFT was applied to the adsorption branch for carbon slit/cylindrical pores (blue line) and to the desorption branch for carbon slit pores (red line).



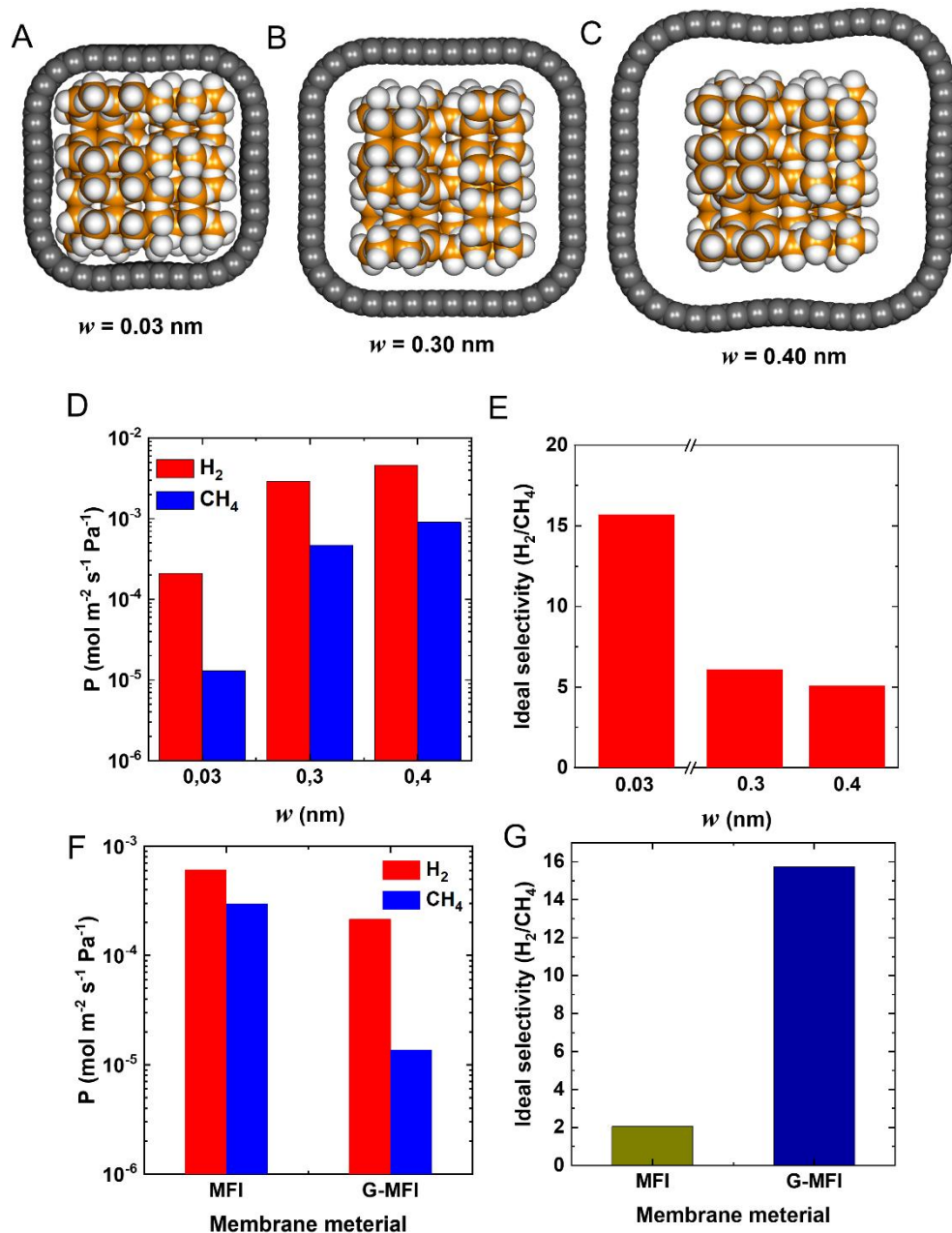
**Fig. S7. Comparison of separation characteristics of the G-MFI and MFI membranes.** (A) Permeability of a single gas against molecular size of a permeate gas. The lower side of the x-axis shows the kinetic diameter, and the upper side of the x-axis shows the corresponding gas molecule. (B) Selectivity of H<sub>2</sub> against the separable gases He, CO<sub>2</sub>, N<sub>2</sub>, CH<sub>4</sub>, i-C<sub>4</sub>H<sub>10</sub>, and SF<sub>6</sub>, as shown on the upper x-axis.



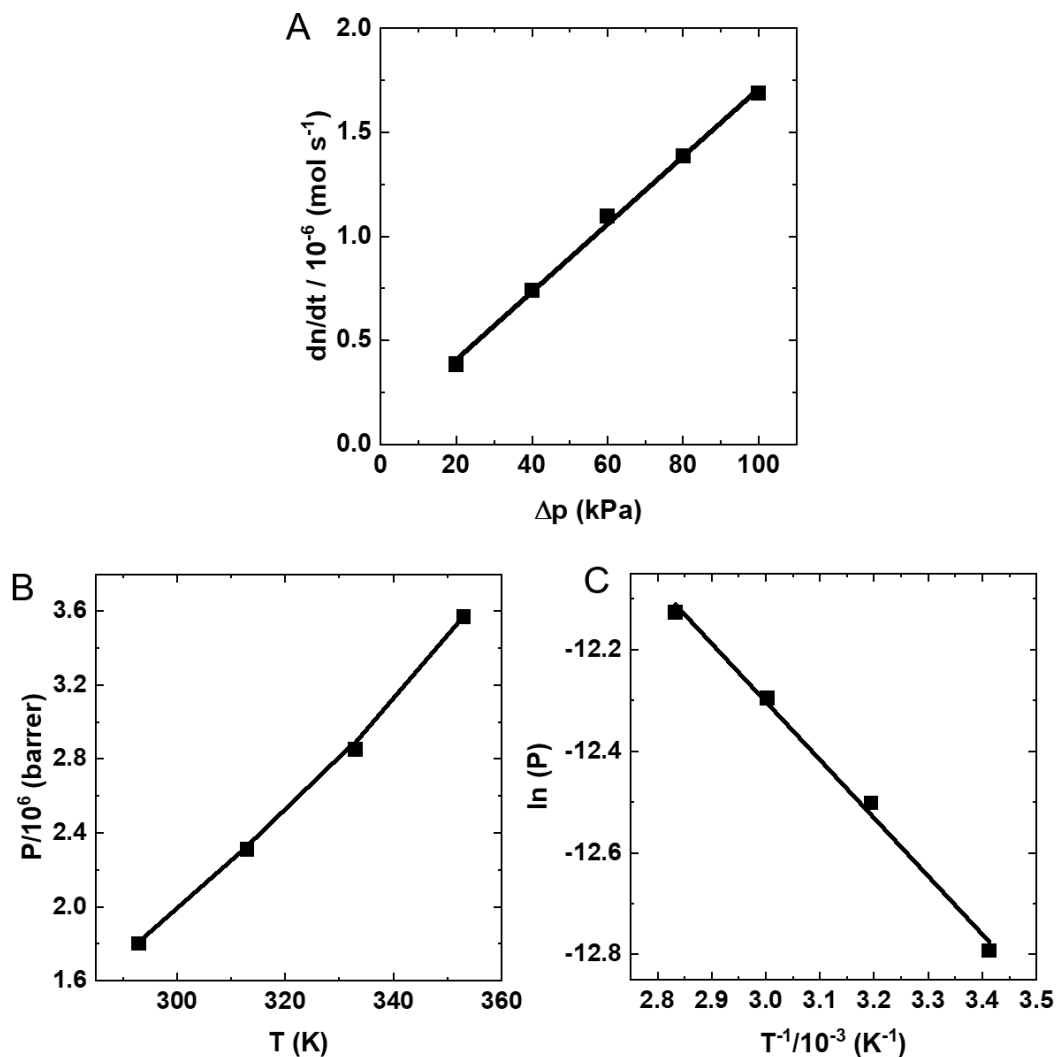
**Fig. S8.** Time-dependence of separation performance of G-MFI membrane. Permeability of CO<sub>2</sub> and selectivity of CO<sub>2</sub>/CH<sub>4</sub> mixed gas against time.



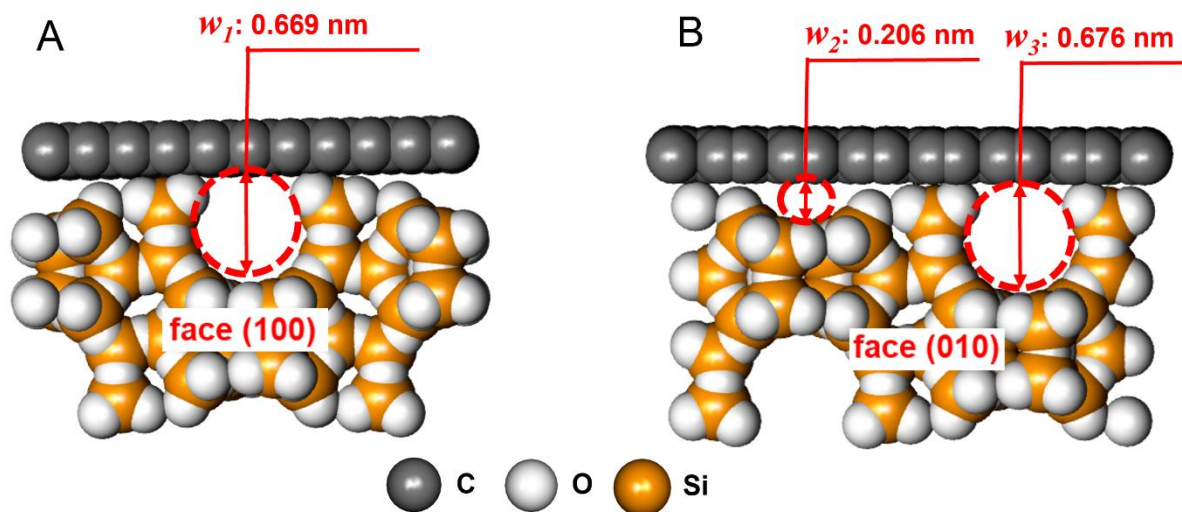
**Fig. S9. Crystal rod models of the graphene-wrapped MFI (G-MFI) membranes for molecular dynamics simulations.** (A) The crystal face structures of the MFI crystal; the (001) face is on the top, while the (010) and (100) faces are on the sides of the MFI crystal. The oxygen and silicon atoms in the MFI crystals are white and orange, respectively. (B) The G-MFI crystal viewed from the top; a ~1 nm diameter nanowindow on graphene located on the top of graphene. The oxygen atoms at the rim of the nanowindow are in red. Graphene is represented by the black color. (C) Side view of the MFI crystal wrapped with graphene whose length is 7.885 nm. The G-MFI model extended to 43.368 nm was also used as a model. For clarity, the atoms in the MFI crystal are drawn as silicon and oxygen atom spheres, graphene in figure (B) is drawn as carbon atom spheres, and the graphene in figure (C) is drawn with stick lines.



**Fig. S10. Effect of the graphene-to-MFI surface effective width ( $w$ ) on the permeance and ideal selectivity of G-MFI.** (A) Cross section of the graphene-wrapped MFI (G-MFI) model with  $w = 0.03$  nm. (B) Cross section of the G-MFI model of  $w = 0.30$  nm. (C) Cross section of the G-MFI model for  $w = 0.40$  nm. (D) Permeances of  $H_2$  and  $CH_4$  molecules against  $w$ . (E) Ideal selectivity against  $w$ . Comparison between permeance and ideal selectivity calculated by MD simulation using MFI and G-MFI membrane: (F) Permeance of  $H_2$  and  $CH_4$  through MFI and G-MFI, (G) Ideal selectivities of  $H_2/CH_4$  by MFI and G-MFI. The MFI crystal rod model of 7.885 nm was used for this simulation.



**Fig. S11. Pressure and temperature dependence.** (A) Molar flow of H<sub>2</sub> against transmembrane pressure (20 – 100 kPa) for the G-MFI membrane, suggesting the classical effusion mechanism in which the pore sizes of the G-MFI membrane are smaller than the mean free path of H<sub>2</sub>. (B) Permeability of H<sub>2</sub> against temperature at a transmembrane pressure difference of 20 kPa. (C) Arrhenius plot for the temperature dependence of the H<sub>2</sub> permeance.



**Fig. S12. Graphene-wrapped BEA zeolite interface.** (A) Effective width  $w_1$  between graphene and the (100) crystal face of the BEA crystal. (B) Effective widths  $w_2$  and  $w_3$  between graphene and the (010) crystal face of the BEA crystal. Experimental ideal selectivity of  $\text{H}_2/\text{CH}_4$  by G-BEA was 1.95.



**Table S1. Membrane characteristics.** Permeabilities and ideal selectivities are shown as cited in the Robeson plots of Figures 1A and 3D.

Material	$t$ ( $\mu\text{m}$ )	$T$ (K)	$\text{H}_2/\text{CO}_2$	$\text{H}_2/\text{CH}_4$	$\text{CO}_2/\text{CH}_4$	$P$ ( $\text{H}_2$ )		$P$ ( $\text{CO}_2$ )		$\Delta p$ (atm)	Year <sup>Ref.</sup>
						barrer	$\text{mol m}^{-2} \text{s}^{-1} \text{Pa}^{-1}$	barrer	$\text{mol m}^{-2} \text{s}^{-1} \text{Pa}^{-1}$		
Si-400 <sup>1</sup>	0.03	473	8	561	70	$1.43 \times 10^2$	$2.0 \times 10^{-6}$	$1.80 \times 10^1$	$2.0 \times 10^{-7}$	1.0	1998(46)
SiC	2	473	43	29	1.4	$7.00 \times 10^1$	$1.18 \times 10^{-8}$	$1.60 \times 10^1$	$2.8 \times 10^{-10}$	2.0	2007(47)
SAPO-34	5	295	-	-	95	-	-	$9.55 \times 10^2$	$6.4 \times 10^{-8}$	1.38	2008(48)
MFI	3	723	139	180	1.3	$3.7 \times 10^3$	$4.2 \times 10^{-7}$	$2.70 \times 10^1$	$3.0 \times 10^{-9}$	1.0	2009(12)
KUUST-1	60	298	4.5	7.8	-	$2.15 \times 10^3$	$1.2 \times 10^{-6}$	$5.0 \times 10^4$	$2.8 \times 10^{-7}$	1.0	2009(49)
ZIF-7	2	493	13	14.7	1.1	$2.72 \times 10^2$	$4.55 \times 10^{-8}$	$2.10 \times 10^1$	$3.5 \times 10^{-9}$	1.0	2010(50)
ZSM-5	9.5	723	25	-	-	$6.32 \times 10^3$	$2.2 \times 10^{-7}$	$1.22 \times 10^3$	$4.3 \times 10^{-8}$	1.0	2012(13)
rGO	0.01 8	298	279.4	117.6	0.42	$6.21 \times 10^0$	$1.0 \times 10^{-7}$	$1.8 \times 10^{-2}$	$3.6 \times 10^{-10}$	-	2013(51)
CAU-1	4	298	13.3	10.8	-	$2.27 \times 10^3$	$1.9 \times 10^{-7}$	$1.79 \times 10^2$	$1.5 \times 10^{-8}$	1.0	2013(52)
ZIF-8	6	303	5.8	11.1	1.9	$3.56 \times 10^3$	$19.9 \times 10^{-8}$	$6.10 \times 10^2$	$3.4 \times 10^{-8}$	0.2	2014(53)
ZIF-8/GO	20	523	22.4	198.3	8.8	$8.36 \times 10^3$	$1.4 \times 10^{-7}$	$3.88 \times 10^2$	$6.5 \times 10^{-9}$	1.0	2014(54)

MoS2	0.01 7	423	3.4	2.4	0.7	$4.68 \times 10^2$	$9.2 \times 10^{-6}$	$1.39 \times 10^2$	$2.7 \times 10^{-6}$	0.8	2015(55)
COF-MOF	60	298	12	6.2	-	$1.0 \times 10^5$	$5.5 \times 10^{-7}$	$8.30 \times 10^3$	$4.6 \times 10^{-8}$	1.0	2016(56)
2H MoS2	1	298	7.6	5.5	0.67	$1.18 \times 10^3$	$3.9 \times 10^{-7}$	$1.46 \times 10^3$	$4.9 \times 10^{-8}$	-	2016(57)
ZIF-8-PI <sup>2</sup> (30 %)	-	308	1.8	28	16	$2.58 \times 10^3$	-	$1.43 \times 10^3$	-	1.00	2016(58)
ZIF-90- 6FDA-TP (50 %)	76	308	2.8	101	36	$1.79 \times 10^2$	$7.8 \times 10^{-10}$	$6.30 \times 10^1$	$2.7 \times 10^{-10}$	9.8	2017(59)
MXene	2	298	238	780	3.4	$2.40 \times 10^3$	$3.7 \times 10^{-7}$	$1.0 \times 10^1$	$1.7 \times 10^{-9}$	1.0	2018(60)
BTESA-A 9:1 <sup>3</sup>	0.5	473	-	54	16.8	$1.90 \times 10^3$	$1.3 \times 10^{-6}$	$5.52 \times 10^2$	$3.7 \times 10^{-7}$	1.4	2020(61)
CHA	3	300	-	-	91	-	-	$3.1 \times 10^4$	$3.5 \times 10^{-6}$	1.0	2020(62)
Graphene	-	300	-	34	27	-	$4.0 \times 10^{-6}$	-	$3.2 \times 10^{-6}$	0.5-1.0	2021(63)
MOF-in- COF	1.00	298	33	38	-	$3.20 \times 10^3$	$1.1 \times 10^{-6}$	$1.00 \times 10^2$	$3.3 \times 10^{-8}$	1.00	2021(64)
RUB-15 <sup>4</sup>	0.3	423	3	27	7	$3.80 \times 10^1$	$4.1 \times 10^{-8}$	$1.00 \times 10^2$	$1.2 \times 10^{-8}$	-	2021(10)
CHFM <sup>5</sup>	3.0	403	84	5706	68	$4.75 \times 10^2$	$5.3 \times 10^{-8}$	$5.30 \times 10^1$	$5.9 \times 10^{-10}$	1.0	2021(65)

G-MFI (Single gas)	150	300	5.5	247	48	$5.80 \times 10^6$	$1.30 \times 10^{-5}$	$1.20 \times 10^6$	$2.5 \times 10^{-6}$	1.0	This work
G-MFI (Mixed gas)	150	300	3.0	50	32	$4.50 \times 10^6$	$1.00 \times 10^{-5}$	$1.40 \times 10^6$	$3.3 \times 10^{-6}$	1.0	This work

1-silica calcinated at 400 °C abbreviated as Si-400. 2-The MMM membrane containing polyimide (PI), containing 30 % ZIF-8; 2-4,40-hexa-fluoroisopropylidene bisphthalic dianhydride-triptycene polyimide (6FDA-TP), containing 50 % ZIF-90. 3-bis(triethoxysilyl)acetylene (BTESA) and (3-amino- propyl) triethoxysilane (APTES) precursors at the ratio of 9:1. 4-sodalite precursor has six-membered rings of SiO<sub>4</sub> tetrahedra. 5-cellulose-based asymmetric carbon hollow fiber membrane.

**Table S2. X-ray diffraction peak characteristics.** The positions and FWHMs of X-ray diffraction peaks of G-MFI and MFI.

Membrane	Plane	2 $\theta$ (°)	FWHM (°)
G-MFI	101	8.11	0.29
	020	9.04	0.37
	501	23.38	0.52
	151	24.04	0.43
	303	24.58	0.54
MFI	101	8.11	0.29
	020	9.04	0.37
	501	23.37	0.48
	151	24.03	0.41
	303	24.60	0.47

**Table S3. Porosity parameters of G-MFI and MFI membranes and their powders.** We obtained the specific surface area ( $S_{\text{BET}}$ ) from the BET plot (66) and the specific surface area ( $S_{\text{as}}$ ), micropore volume ( $V_{\text{Micro.}}$ ) and external surface area ( $S_{\text{ex.}}$ ) by subtracting pore effect method from high resolution  $\alpha_S$ -plot (67).

<b>Membrane material</b>	<b><math>S_{\text{BET}}</math> (<math>\text{m}^2 \text{g}^{-1}</math>)</b>	<b><math>S_{\text{as}}</math> (<math>\text{m}^2 \text{g}^{-1}</math>)</b>	<b><math>V_{\text{Micro.}}</math> (<math>\text{cm}^3 \text{g}^{-1}</math>)</b>	<b><math>S_{\text{ex}}</math> (<math>\text{m}^2 \text{g}^{-1}</math>)</b>
MFI (membrane)	340	557	0.10	66
G-MFI (membrane)	416	676	0.12	111
MFI (powder)	360	580	0.11	62
G-MFI (powder)	360	580	0.11	77

**Table S4. Summary of gyration radii determined from the linear regions of the Guinier plots for small-angle X-ray scattering of the G-MFI and MFI membranes.** The radius of gyration was determined for the thickness of the flat particle model using PRIMUS software.(68)

Range $Q^2$ (nm <sup>-2</sup> )	G-MFI		MFI	
	$R_g$ (nm)	$QR_g$ limit	$R_g$ (nm)	$QR_g$ limit
1.6 – 3.1	$0.48 \pm 0.00$	0.60 – 0.84	$0.60 \pm 0.04$	0.85 – 1.05

**Table S5. Force field parameters.** Atomic masses, charges, and Lennard-Jones(LJ)  $\sigma$  and  $\epsilon$  parameters used in MD simulations of H<sub>2</sub> and CH<sub>4</sub> permeance through the G-MFI membrane.

Molecule	Mass ( $1.66 \times 10^{-27}$ kg)	Charge (e)	LJ $\sigma$ (nm)	LJ $\epsilon$ (K)	Reference
H <sub>2</sub>	2.01588	0	0.2958	36.733	Ref.(69)
CH <sub>4</sub>	16.04276	0	0.373	148	Ref.(70)
C (-C=)	12.011	0	0.336	28	Ref.(18)
C (C-O-H)	12.011	0.15	0.355	35.2	
O (C-O-H)	15.9994	-0.585	0.307	85.5	
H (C-O-H)	1.00794	0.435	0	0	
C (C-O-C)	12.011	0.25	0.38	35.2	
O (C-O-C)	15.9994	-0.5	0.3	59.5	
C (C-H)	12.011	-0.115	0.355	35.2	
H (C-H)	1.00794	0.115	0.242	15.1	
Si	28.085	1.413	2.97	31.98	
O (Si-O-Si)	15.9994	-0.7065	3.011	52	Ref.(71)

**Movie S1. H<sub>2</sub>/CH<sub>4</sub> mixed gas separation by G-MFI membrane; H<sub>2</sub> molecules-red color, CH<sub>4</sub> molecules-blue color.** The H<sub>2</sub> molecules permeate faster than the CH<sub>4</sub> molecules through the G-MFI. The CH<sub>4</sub> molecules adsorb the the G-MFI and hinder the permeance of H<sub>2</sub> molecules. The selectivity of H<sub>2</sub>/CH<sub>4</sub> from the mixed gas is smaller than ideal selectivity of H<sub>2</sub>/CH<sub>4</sub> from the single gas, agreeing with the experimental results (Fig. 1A).



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