

Supplementary Information for

**Ion-Cluster-Mediated Ultrafast Self-Healable Ionoconductors for Reconfigurable Electronics**

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**Other Supplementary Materials for this manuscript include the following :**

Supplementary Movie 1 to 6

**Supplementary Table 1** Molecular characterizations of polymers prepared in this work.

Code	$M_w$ (g mol <sup>-1</sup> ) <sup>a</sup>	$M_n$ (g mol <sup>-1</sup> ) <sup>a</sup>	Polydispersity index ( $\mathcal{D}$ ) <sup>a</sup>	Mole fraction of hydroxyl parts (%) <sup>b</sup>
100-AAHA	946,000	430,000	2.2	0
92-AAHA	1,274,000	490,000	2.6	8
61-AAHA	1,250,600	481,000	2.6	39
90-DAHA	898,700	473,000	1.9	10

<sup>a</sup>Measured by SEC with poly(ethylene oxide) standards; <sup>b</sup>Determined by <sup>1</sup>H NMR spectroscopy.

**Supplementary Table 2** Comparison of healing mechanism, efficiency, temperature, and mechanical properties (Young's modulus and fracture strain) of previously reported self-healing platforms, including ionogels, hydrogels, and elastomers.

Materials	Type	Healing Mechanism	Healing Efficiency (%)	Healing Time (h)	Healing Temperature (°C)	Young's Modulus (MPa)	Fracture Strain (%)	Ref.
PAAm <sup>a</sup> /PNAGA <sup>b</sup>	Hydrogel	hydrogen bond	N.A.	0.5	90	0.048	665	1
UPyHCBA <sup>c</sup>	Hydrogel	hydrophobic interaction, hydrogen bond	N.A.	0.042	25	0.0017	1000	2
PAAc <sup>d</sup> /CTA <sup>e</sup>	Hydrogel	hydrophobic interaction	75	0.5	80	0.016	900	3
MPTC <sup>f</sup> /NaSS <sup>g</sup> /DMAEA <sup>h</sup> /BIS <sup>i</sup> copolymer	Hydrogel	electrostatic interaction	75	0.125	25	0.036	650 <sup>a</sup>	4
P(NaSS- <i>co</i> -DMAEA-Q)	Hydrogel	electrostatic interaction	82.3	24	25	0.08	1000	5
Zwitterionic nanocomposite hydrogel	Hydrogel	electrostatic interaction	74	24	25	0.0046 <sup>x</sup>	2000	6
Agel <sup>j</sup> -DCMC <sup>k</sup> hydrogel	Hydrogel	imine bond	90	1	37	0.03	140	7
P(AM- <i>co</i> -AEAM <sup>l</sup> ) hydrogel	Hydrogel	hydrogen bond, ionic interaction, Schiff base bond	53	48	25	0.02 <sup>x</sup>	1205	8
P(Urea-IL <sup>m</sup> -SPMA <sup>n</sup> )	Hydrogel	electrostatic interaction, hydrogen bond	91	24	25	0.18	720	9
hyperbranched polyurethane (HPDU)	Elastomer	hydrogen bond	96.5	10	90	0.8 <sup>x</sup>	3100	10
Dopamine acrylamide/butyl acrylate/Ca <sup>2+</sup>	Elastomer	hydrogen bond, metal coordination	37	24	25	3.9	310	11
poly(AAm/ChCl <sup>o</sup> - <i>co</i> -MA <sup>p</sup> /ChCl)	Elastomer	hydrogen bond	91	72	20	0.1	450	12
polyurethane acrylate	Elastomer	disulfide bond hydrogen bond	95	12	80	0.5 <sup>x</sup>	400	13
polyurethane (DA-PU)	Elastomer	hydrogen bond, donor-acceptor interaction	97	6.67	60	1.33 <sup>x</sup>	1900	14
PU-HEDS <sup>q</sup> -400	Elastomer	hydrogen bond, disulfide bond	95.4	24	50	5.04	553	15
Fe-Hpdca <sup>r</sup> -PDMS	Elastomer	metal-ligand interaction	90.3	48	25	0.54	1700	16
cellulose nanocrystal/[BMIM][HSO <sub>4</sub> ] composite	Ion gel	electrostatic interaction	90.3	24	25	0.016	620	17
PS <sup>s</sup> - <i>b</i> -P(DMAAm <sup>t</sup> - <i>r</i> -AAc)/[EMI][TFSI]	Ion gel	hydrogen bond	90	3	25	0.5 <sup>x</sup>	400	18
TetraPEG-Ant <sup>u</sup> /[EMI][TFSI]	Ion gel	photoisomerization	66	10	150	0.007 <sup>x</sup>	950	19
P(AzoMA- <i>r</i> -NIPAm)- <i>b</i> -PEO- <i>b</i> -P(AzoMA- <i>r</i> -NIPAm) <sup>v</sup> /[BMIM][PF <sub>6</sub> ]	Ion gel	photoisomerization	81	64	36	0.02 <sup>x</sup>	450	20

PVDF- <i>co</i> -HFP <sup>w</sup> -5545/[EMI][OTf]	Ion gel	ion-dipole interaction	100.3	6	50	0.10	1500	21
PVDF- <i>co</i> -HFP-5545/[EMI][TFSI]	Ion gel	ion-dipole interaction	99.1	24	50	0.21	2000	22
Cellulose/[BMI][Cl]	Ion gel	hydrogen bond	96	2	25	1 <sup>x</sup>	60	23
			<b>90.3</b>	<b>0.016 (1min)</b>				
<b>92-AAHA-IL</b>	<b>Ion gel</b>	<b>ion cluster</b>	<b>98.7</b>	<b>0.050 (3min)</b>	<b>25</b>	<b>0.84</b>	<b>1130</b>	<b>This work</b>
			<b>99.7</b>	<b>0.083 (5min)</b>				

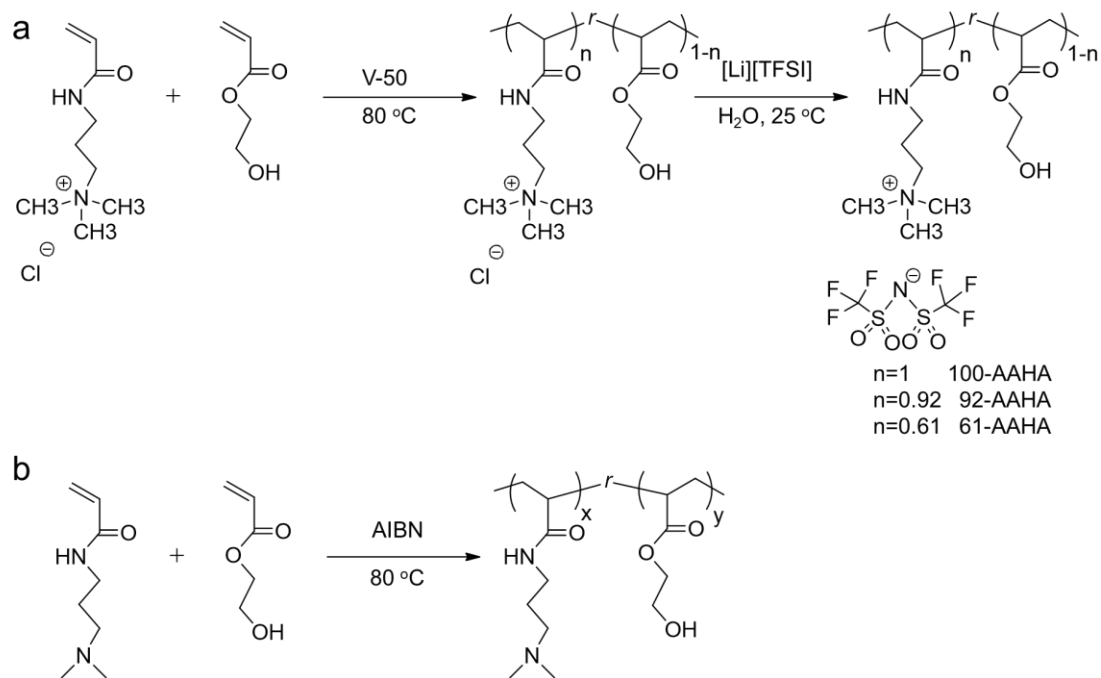
<sup>a</sup>PAAm: poly(acrylamide), <sup>b</sup>PNAGA: poly(*N*-acryloyl glycinamide), <sup>c</sup>UPyHCBA: 4-(6-(3-(6-methyl-4-oxo-1,4-dihydropyrimidin-2-yl)ureido)hexylcarbamoyloxy)butyl acrylate, <sup>d</sup>PAAc: poly(acrylic acid), <sup>e</sup>CTA: cetyltrimethylammonium, <sup>f</sup>MPTC: [3-(methacryloylamino)propyl]trimethylammonium chloride, <sup>g</sup>NaSS: sodium p-styrenesulfonate hydrate, <sup>h</sup>DMAEA: 2-(dimethylamino)-ethyl methacrylate, <sup>i</sup>BIS: N,N'-methylenebis(acrylamide), <sup>j</sup>Agel: amino-gelatin, <sup>k</sup>DCMC: dialdehyde carboxymethyl cellulose, <sup>l</sup>AEAM: 2-aminoethyl acrylamide hydrochloride, <sup>m</sup>Urea-IL: imidazolium-based ionic liquid (IL) monomers with urea groups, <sup>n</sup>SPMA: 3-sulfopropyl methacrylate potassium salt, <sup>o</sup>ChCl: choline chloride, <sup>p</sup>MA: maleic acid, <sup>q</sup>HEDS: 2,2'-hydroxy ethyldisulfide, <sup>r</sup>Hpdca: 2,6-pyridinedicarboxamide, <sup>s</sup>PS: poly(styrene), <sup>t</sup>DMAAm: N,N-dimethylacrylamide, <sup>u</sup>TetraPEG-Ant: tetra-arm poly(ethylene glycol)-anthracene, <sup>v</sup>P(AzoMA-*r*-NIPAm)-*b*-PEO-*b*-P(AzoMA-*r*-NIPAm): Poly(4-phenylazophenyl methacrylate-*random*-N-isopropylacrylamide)-*block*-poly(ethylene oxide)-*block*-P(4-phenylazophenyl methacrylate-*random*-N-isopropylacrylamide), <sup>w</sup>PVDF-*co*-HFP: Poly(vinylidene fluoride-*co*-hexafluoropropylene). <sup>x</sup>calculated from produced stress-strain curve.

**Supplementary Table 3** Vogel-Fulcher-Tammann (VFT) temperature fitting parameters of temperature dependence of the  $\alpha_2$  process.

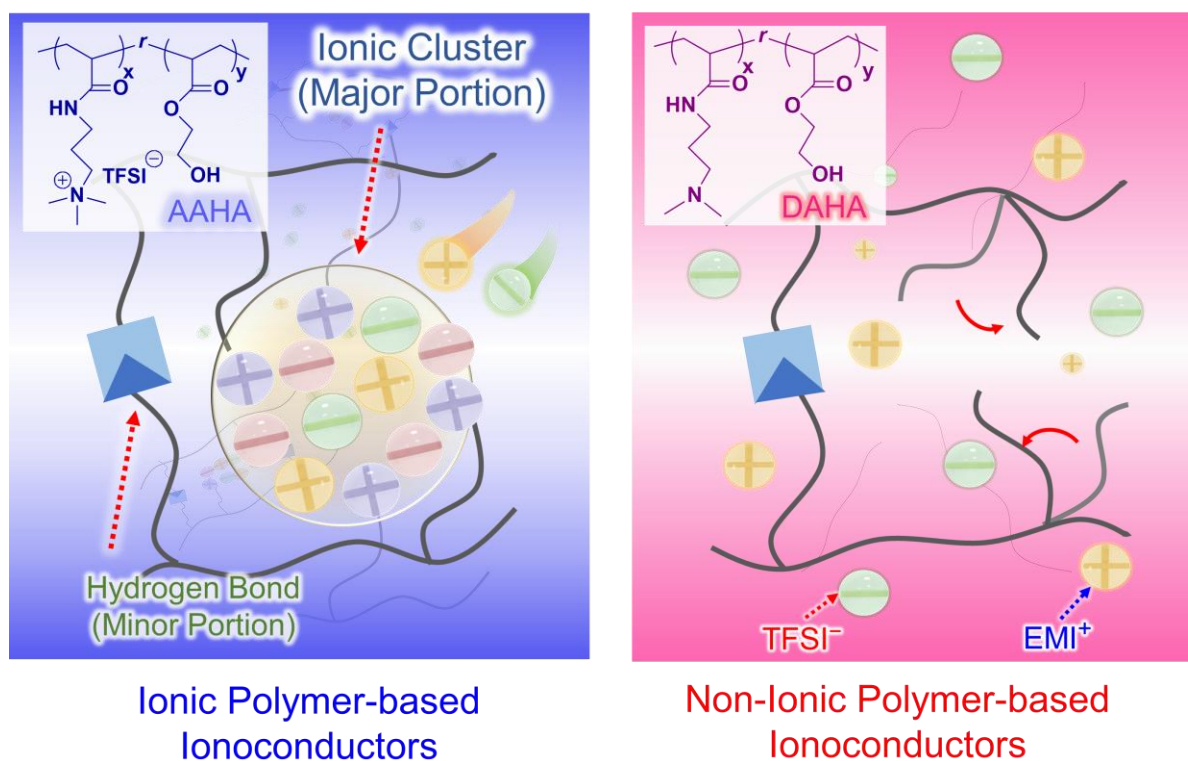
Sample	$\log(\omega_\infty)$ (rad s <sup>-1</sup> )	$D$	$T_0$ (K)	$T_g^a$ (K)	$E_a$ (kJ/mol)
92-AAHA-IL	10.6	6	171	256	9
90-DAHA-IL	12.6	11	165	250	15

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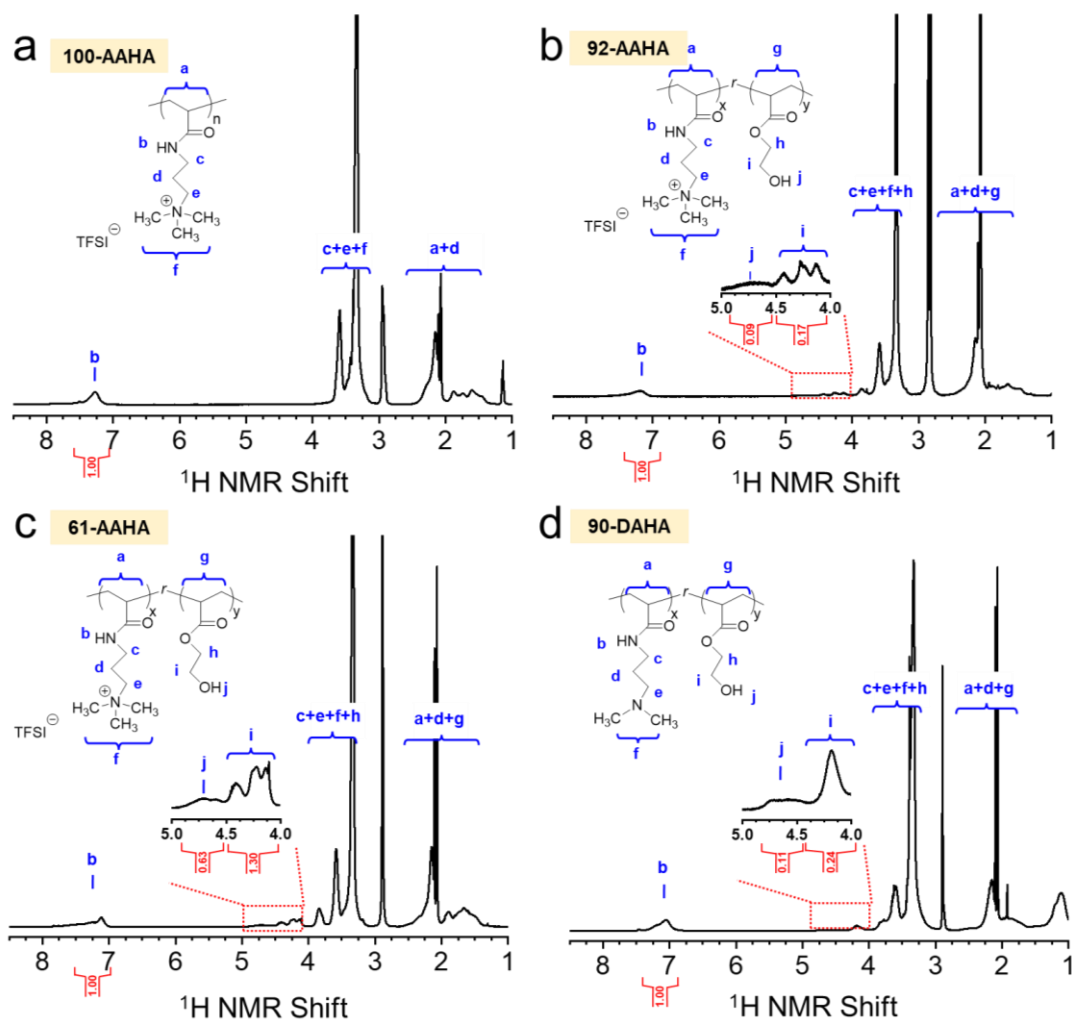
<sup>a</sup>Obtained from DSC thermograms (Supplementary Fig. 7).



**Supplementary Fig. 1 Synthetic strategy for copolymers employed in this work.** Synthetic routes for preparing **a** AAHAs and **b** DAHA.

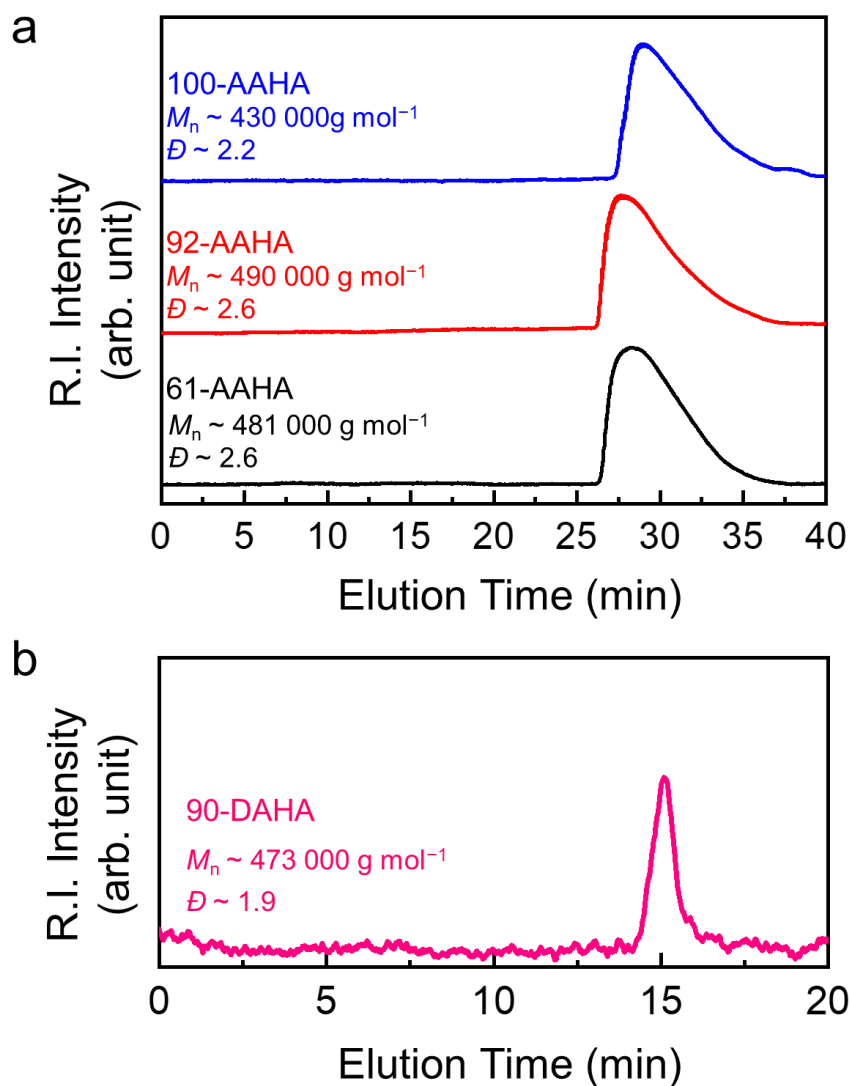


**Supplementary Fig. 2 Difference in molecular interaction depending on the presence of ionic moiety.** Schematic illustrations of ionic polymer-based ionoconductor (92-AAHA-IL), in which both hydrogen bonding and ion-cluster formation take place, and non-ionic polymer-based ionoconductors (90-DAHA-IL) only including hydrogen bonding interactions.

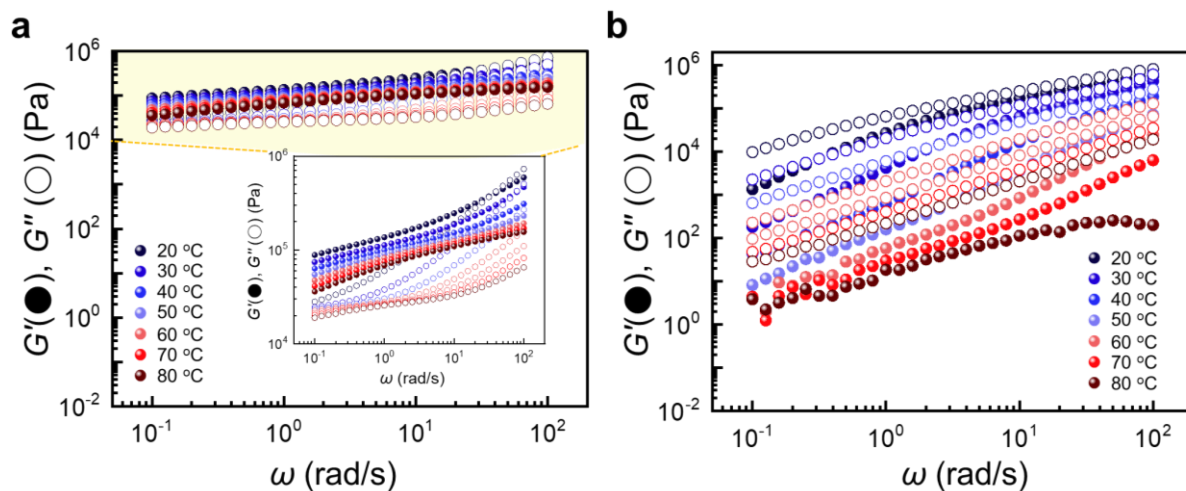


**Supplementary Fig. 3 Structural analysis of used polymers.**  $^1\text{H}$  NMR spectra of prepared polymers. **a** 100-AAHA, **b** 92-AAHA, **c** 61-AAHA, and **d** 90-DAHA.

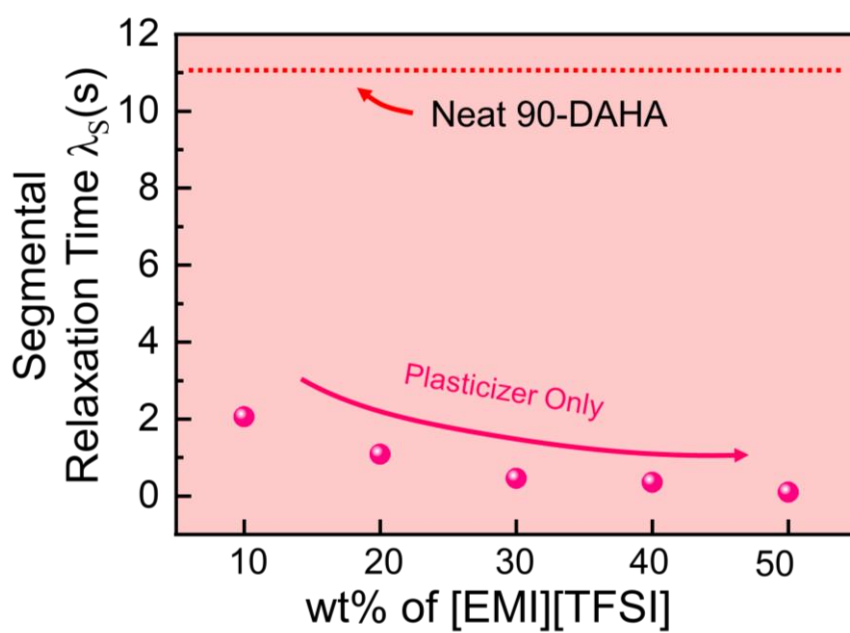




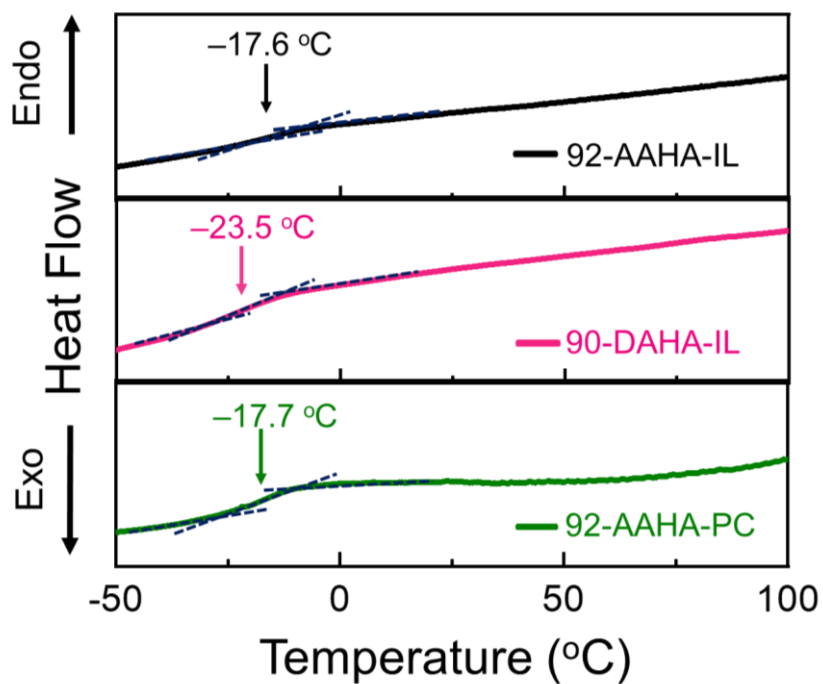
**Supplementary Fig. 4 Molecular characteristics of used polymers.** SEC traces of polymers employed in this study. **a** 100-AAHA, 92-AAHA, and 61-AAHA. To prepare the GPC samples, the synthesized AAHAs were dissolved in an aqueous buffer (pH = 3) containing 0.1 M sodium phosphate, and the operating flow rate was fixed at  $0.5\text{ mL min}^{-1}$ . **b** SEC trace of 90-DAHA dissolved in HPLC water with a flow rate of  $1.0\text{ mL min}^{-1}$ . All samples were calibrated with poly(ethylene oxide) standards.



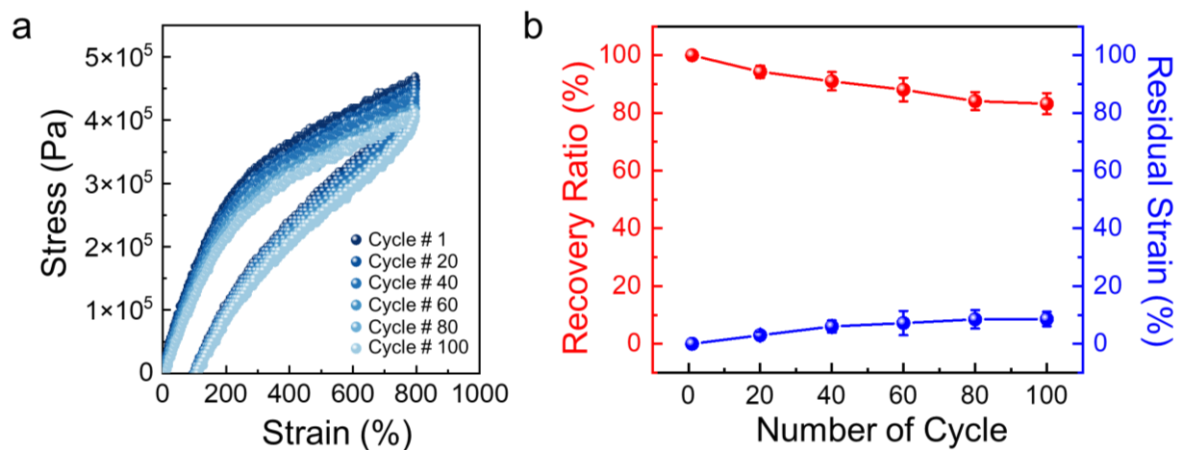
**Supplementary Fig. 5 Difference in rheological behaviors.** Frequency-dependent dynamic storage (filled circle) and loss (open circle) moduli at various temperatures for **a** 92-AAHA-IL and **b** 90-DAHA-IL.



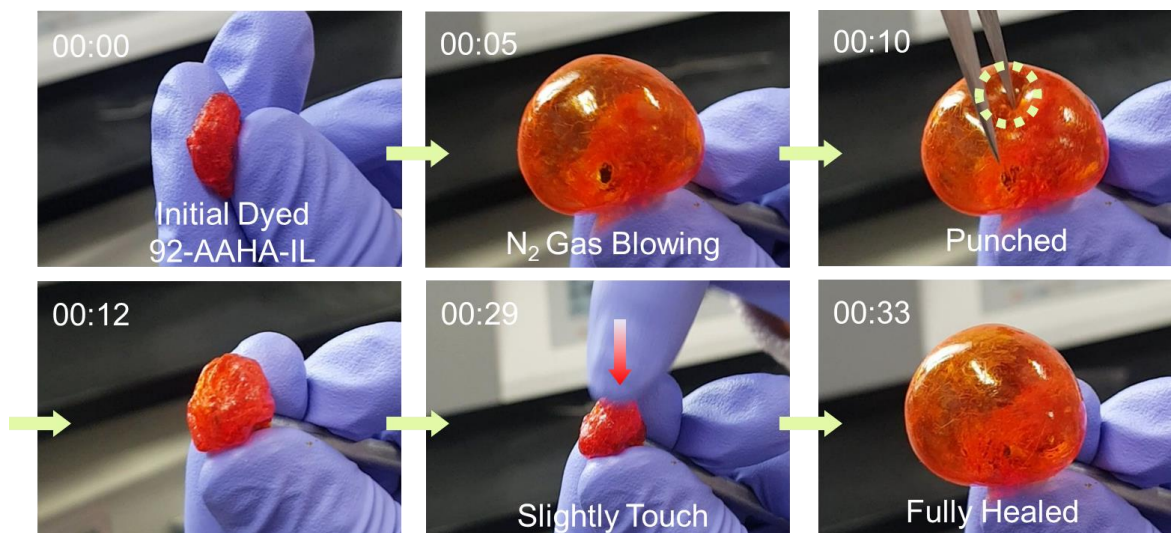
**Supplementary Fig. 6 Effect of [EMI][TFSI] on non-ionic polymer-based system.** Changes in segmental relaxation time of 90-DAHA-IL according to the weight fraction of the introduced [EMI][TFSI]. The red dotted-line indicates the segmental relaxation time of neat 90-DAHA.



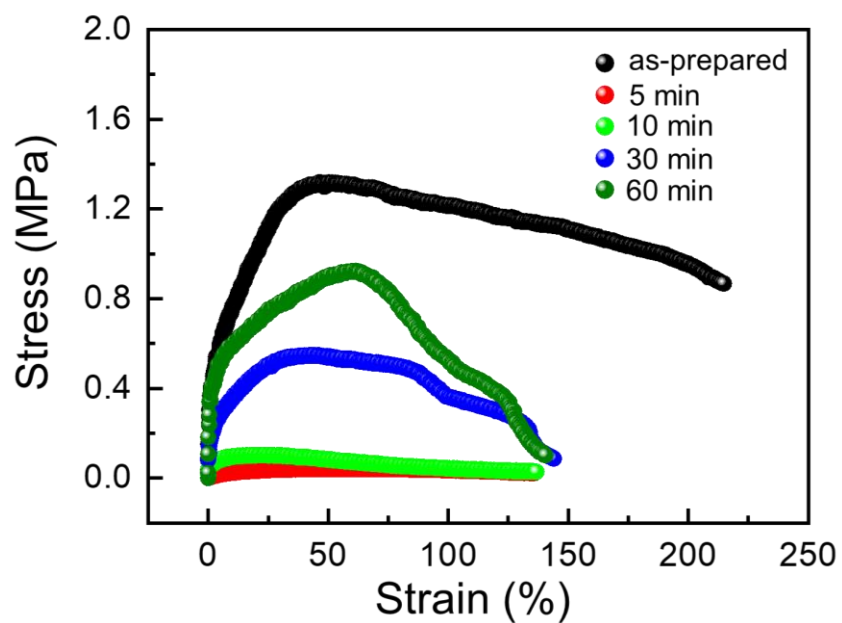
**Supplementary Fig. 7 Thermal properties of various ionoconductors with different copolymers.** DSC thermograms of 92-AAHA-IL, 90-DAHA-IL, and 92-AAHA-PC, obtained during the second run at a heating rate of 10 °C min<sup>-1</sup>.



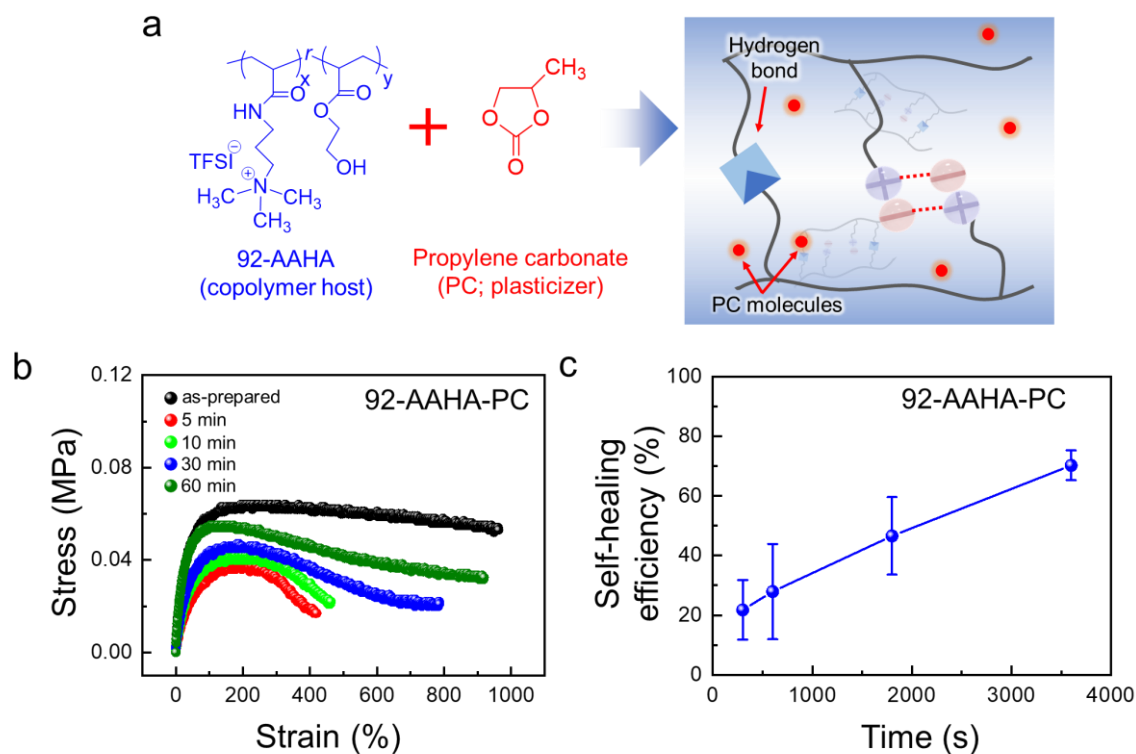
**Supplementary Fig. 8 Mechanical stability of 92-AAHA-IL.** **a** Stress-strain curves during cyclic stretching/releasing at 800% and **b** corresponding changes in recovery ratio and residual strain. The recovery ratio of ~83.2% and residual strain of ~8.6% after 100 consecutive cycles represent the high mechanical reliability of 92-AAHA-IL. Error bars indicate standard deviation.



**Supplementary Fig. 9 Demonstration of ultrafast self-healing performance of 92-AAHA-IL.** Snapshots from Supplementary Movie 1 showing the immediate self-healing capability of 92-AAHA-IL. The balloon-shaped 92-AAHA-IL was expanded by blowing N<sub>2</sub> gas and punched with a tweezer to induce mechanical damage. After self-healing with a slight touch to contact the injured region, 92-AAHA-IL fully prevented the leakage of N<sub>2</sub> molecules.

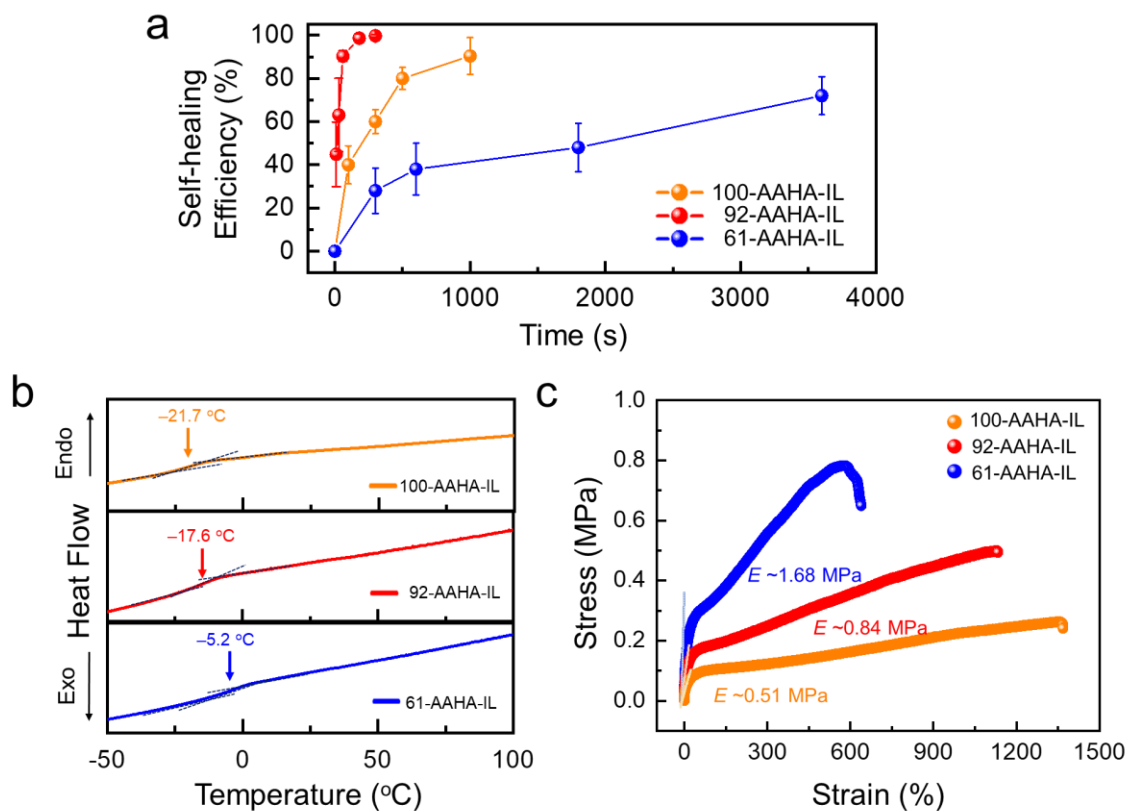


**Supplementary Fig. 10 Correlation between healing time and mechanical properties of 90-DAHA-IL.** Tensile stress-strain curves of as-prepared and healed 90-DAHA-IL (5–60 mins) at 25°C.

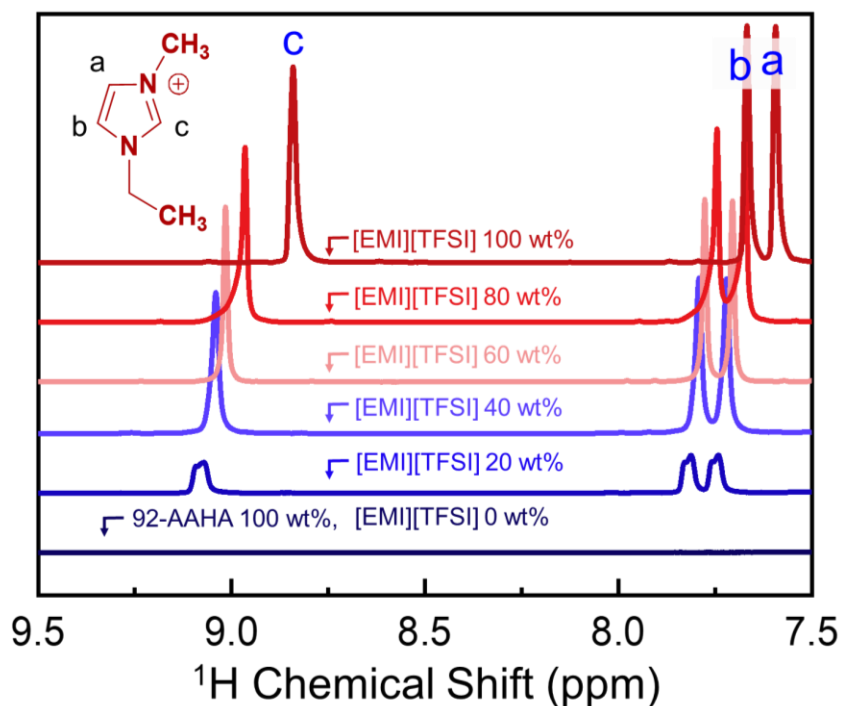


**Supplementary Fig. 11 Mechanical and self-healing characteristics of the system with no ionic liquids.** **a** Schematic illustration of 92-AAHA-PC. **b** Tensile stress–strain curves for self-healing tests of 92-AAHA-PC at 25 °C, and **c** self-healing efficiency as a function of healing time. The self-healing efficiency was calculated from the ratio of the toughness values before damage and after healing obtained from the stress–strain curves. Error bars indicate standard deviation.

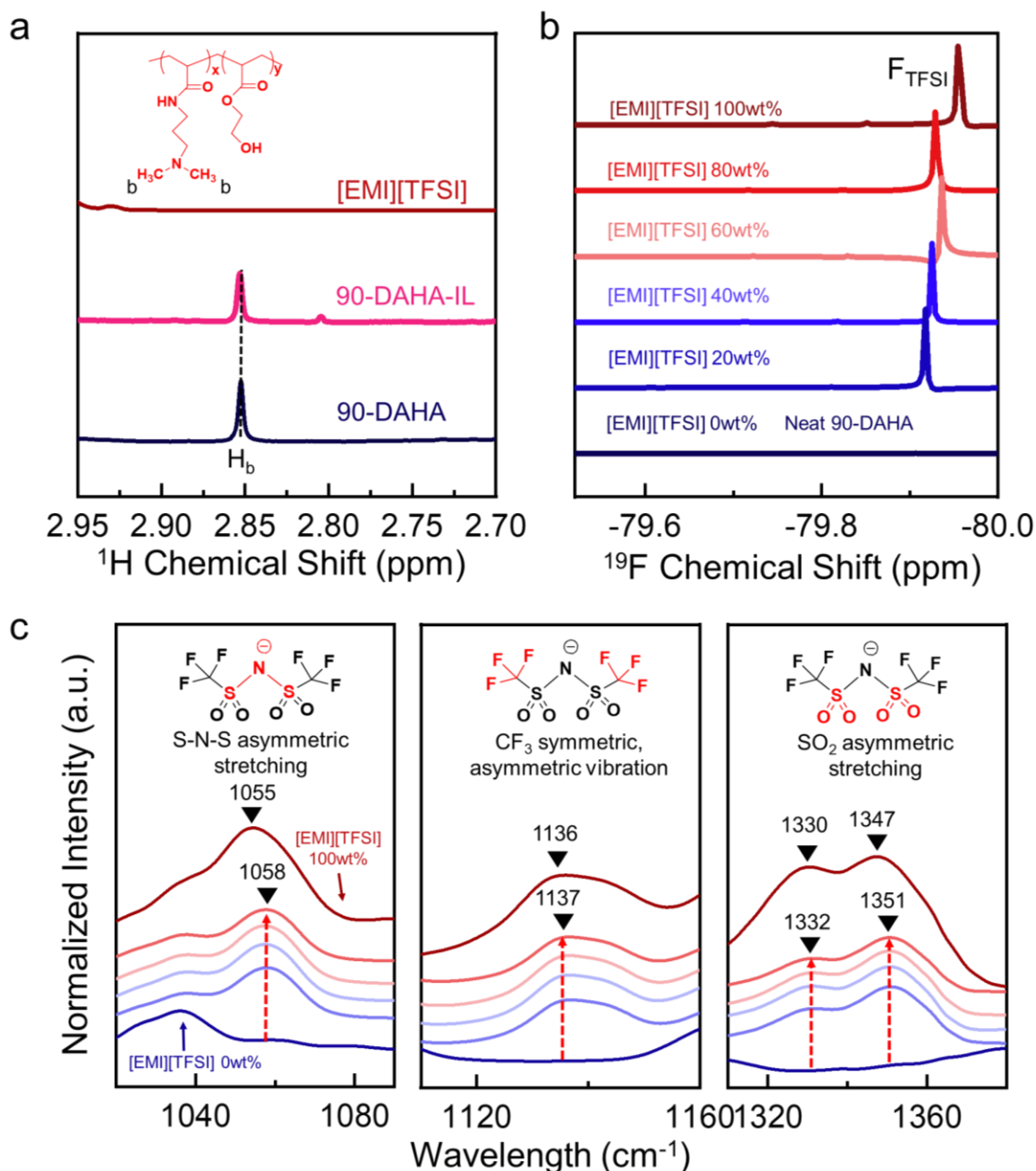




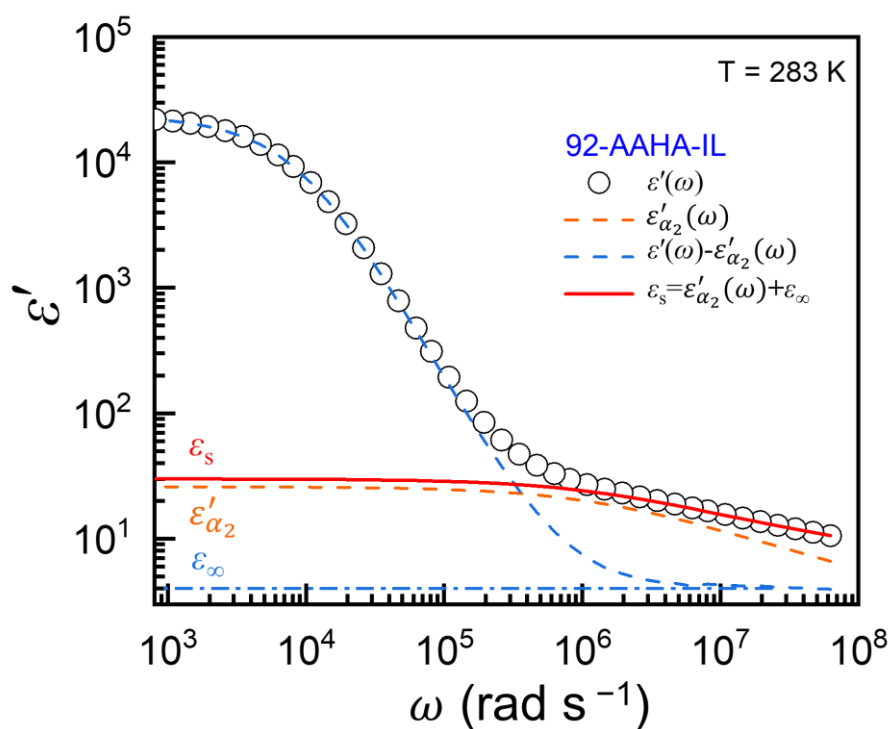
**Supplementary Fig. 12 Dependence of self-healing, thermal, and mechanical performance on the ionic content in AAHA-IL.** **a** Differences in self-healing efficiencies measured with different copolymer hosts. Error bars indicate standard deviation. **b** DSC thermograms for evaluating  $T_g$  values. **c** Tensile stress–strain curves for determining elastic moduli of 100-AAHA-IL, 92-AAHA-IL, and 61-AAHA-IL.



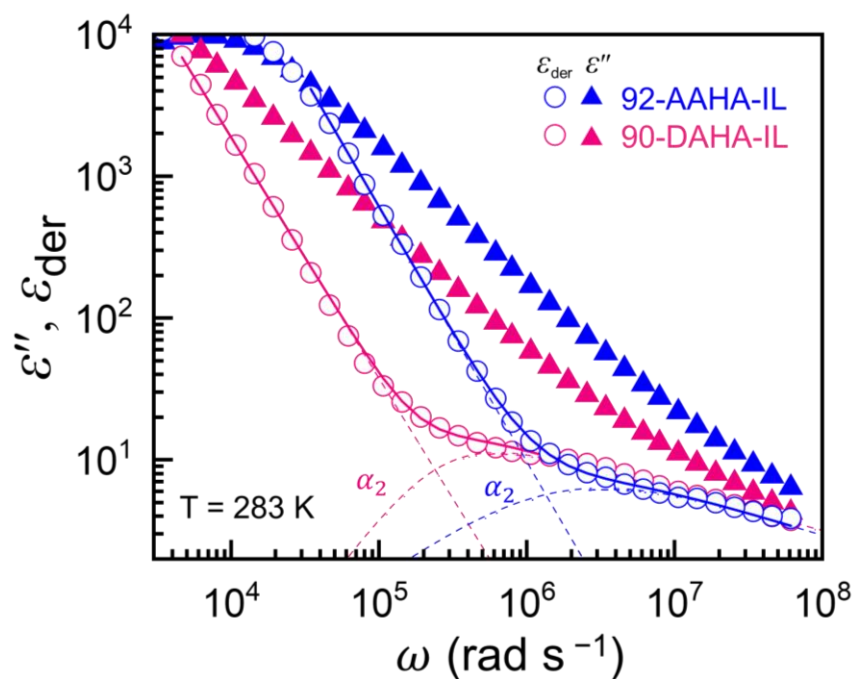
**Supplementary Fig. 13 Interaction between [EMI][TFSI] and 92-AAHA.**  $^1\text{H}$  NMR spectra of [EMI][TFSI] with various 92-AAHA-IL contents. The contents of [EMI][TFSI] were 0 (navy line, neat 92-AAHA), 20, 40, 60, 80, and 100 wt.% (red line, neat [EMI][TFSI]). The representative [EMI]<sup>+</sup> peaks were slightly shifted when the 92-AAHA was added, but no peaks below 7.5–9.5 ppm were observed for neat 92-AAHA.



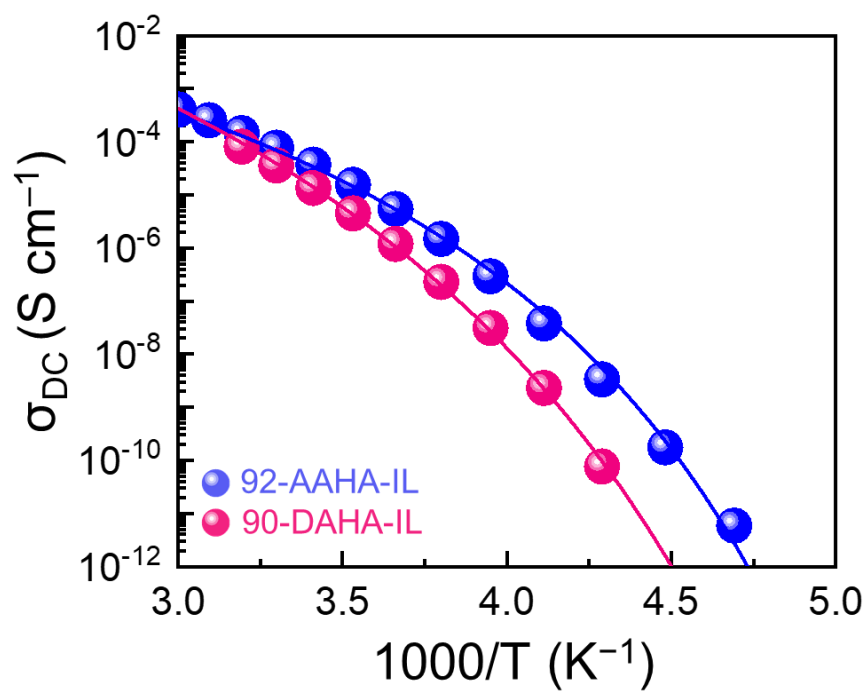
**Supplementary Fig. 14 Spectroscopic analysis of 90-DAHA-containing system.** **a** Comparison of  $^1\text{H}$  NMR spectra of 90-DAHA, 90-DAHA-IL, and [EMI][TFSI]. 90-DAHA-IL contained 30 wt.% [EMI][TFSI]. **b**  $^{19}\text{F}$  NMR spectra of 90-DAHA-IL with various [EMI][TFSI] contents. No  $[\text{TFSI}]^-$  peak was observed for neat 90-DAHA. **c** Changes in FTIR absorption peaks with respect to the fraction of  $[\text{TFSI}]^-$  in 90-DAHA-IL. The contents of [EMI][TFSI] were 0 (navy line, neat 90-DAHA), 20, 40, 60, 80, and 100 wt.% (red line, neat [EMI][TFSI]).



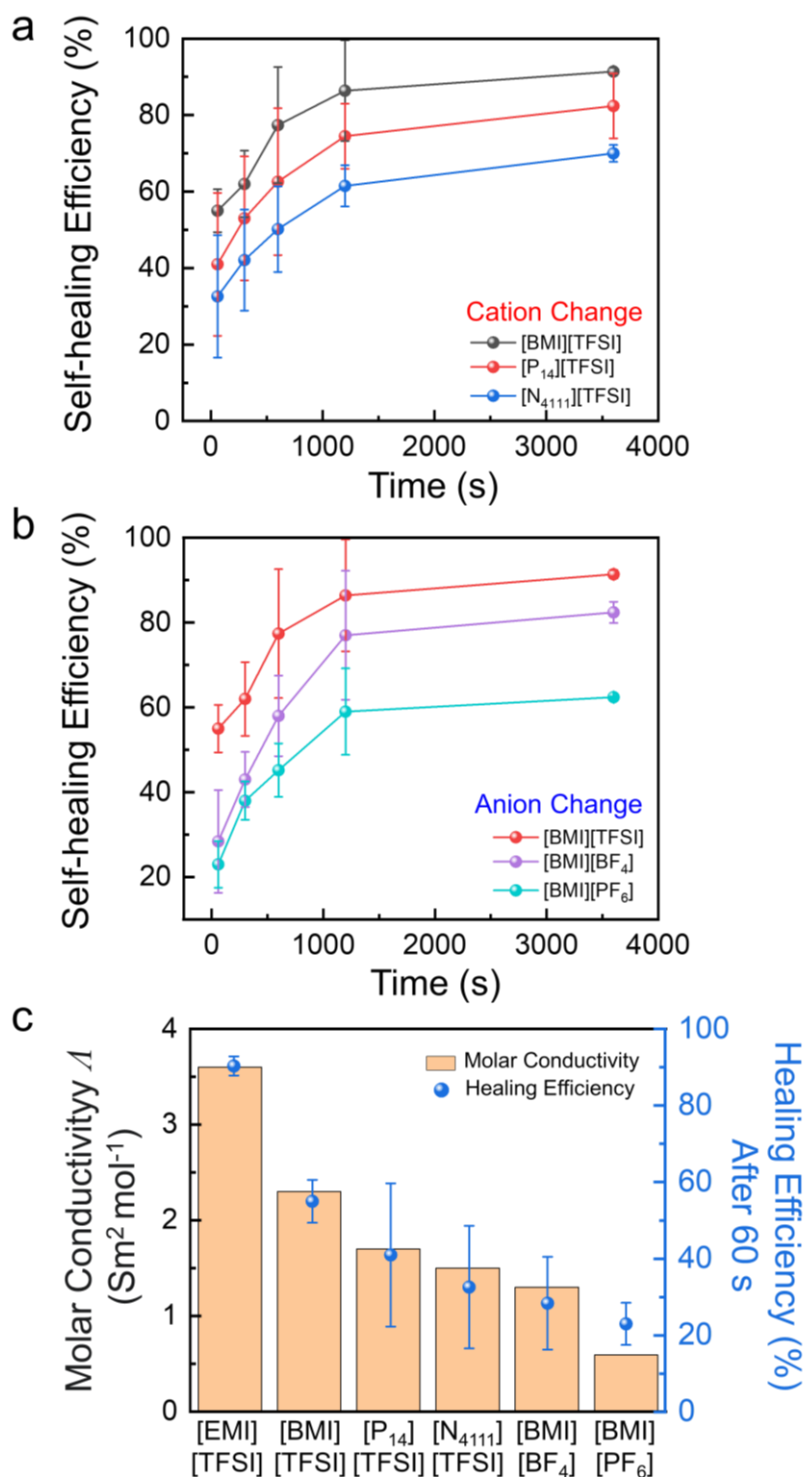
**Supplementary Fig. 15 Frequency dependent dielectric characteristics of 92-AAHA-IL.** Angular frequency dependence of dielectric permittivity  $\varepsilon'(\omega)$  of 92-AAHA-IL at 283 K and corresponding static dielectric constant  $\varepsilon_s$  (red solid line), determined as the sum of the frequency-dependent contributions from the  $\alpha_2$  process  $\varepsilon'_{\alpha_2}(\omega)$  (orange dashed line) and the high-frequency dielectric constant  $\varepsilon_\infty$  (blue dashed line).



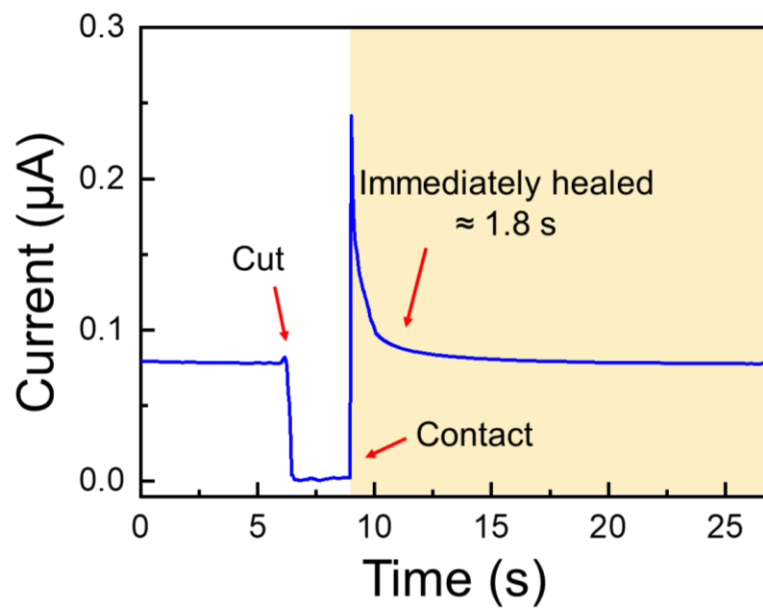
**Supplementary Fig. 16 Different  $\alpha_2$  process between ionic- and non-ionic polymer-based ionic conductors.** Derivative (open circles) and dielectric loss (filled triangles) spectra at 283 K and corresponding fits using a sum (solid line) of a power law for electrode polarization (dashed line) and a derivative form of the Havriliak-Negami function for the  $\alpha_2$  process (short dashed line).



**Supplementary Fig. 17 Temperature dependent ionic conductivity.** Temperature dependence of ionic conductivity for 92-AAHA-IL and 90-DAHA-IL.

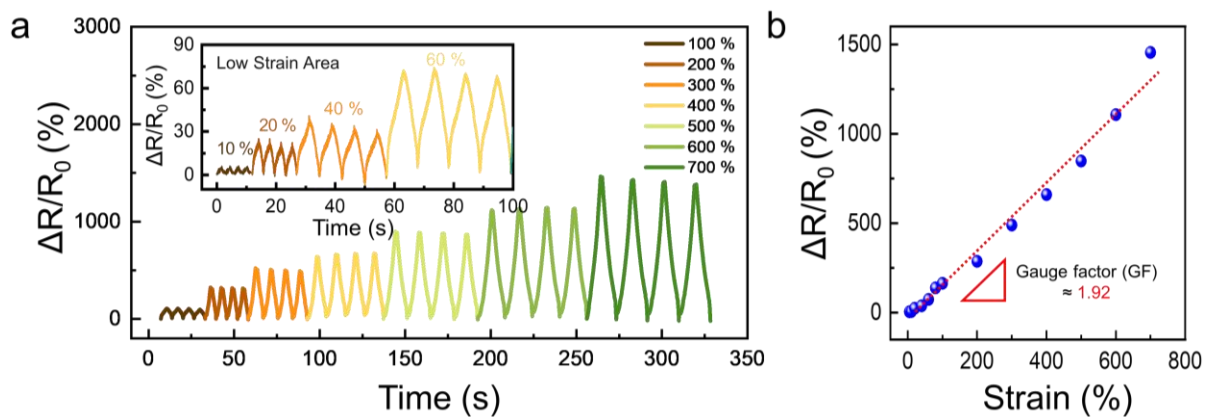


**Supplementary Fig. 18 Correlation between self-healing performance and type of ionic liquids.** Time-dependent self-healing efficiency of 92-AAHA-IL with selectively varied **a** cations and **b** anions of the ionic liquid. **c** Plots of molar conductivity ( $\Lambda$ ) and healing efficiency of ionic liquids at a repair time of 60 s. The  $\Lambda$  values were extracted from ref. 24. The self-healing processes of all ionic liquids were performed at 25 °C. Considering that a higher  $\Lambda$  of ionic liquids indicates an easier ion-ion dissociation and ion movement, the superior performance of the [EMI][TFSI]-containing 92-AAHA-IL can be rationalized in terms of the highest  $\Lambda$  of the ionic liquid additive. All error bars indicate standard deviation.

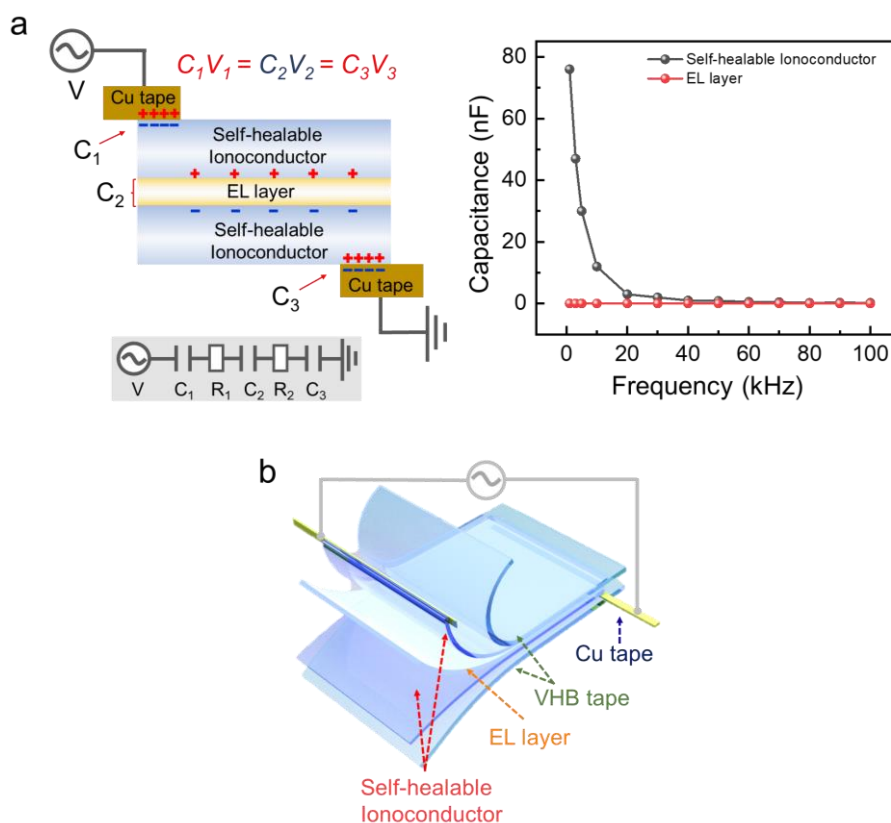


**Supplementary Fig. 19 Self-healing effect on electrical properties.** Self-healing of electrical properties of 92-AAHA-IL film. The flowing current was turned off when the film was cut in half, and was recovered immediately ( $\sim 1.8$  s) after reconnection.

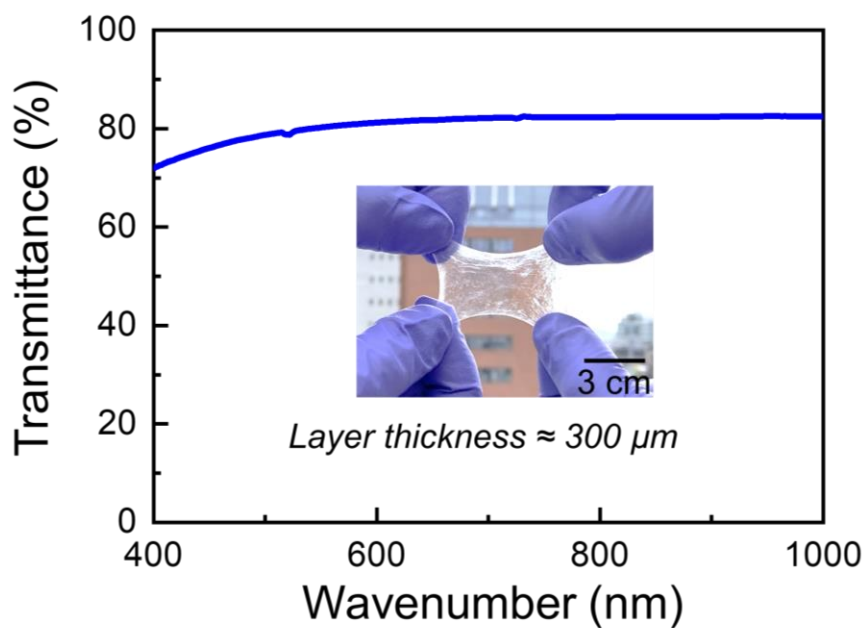




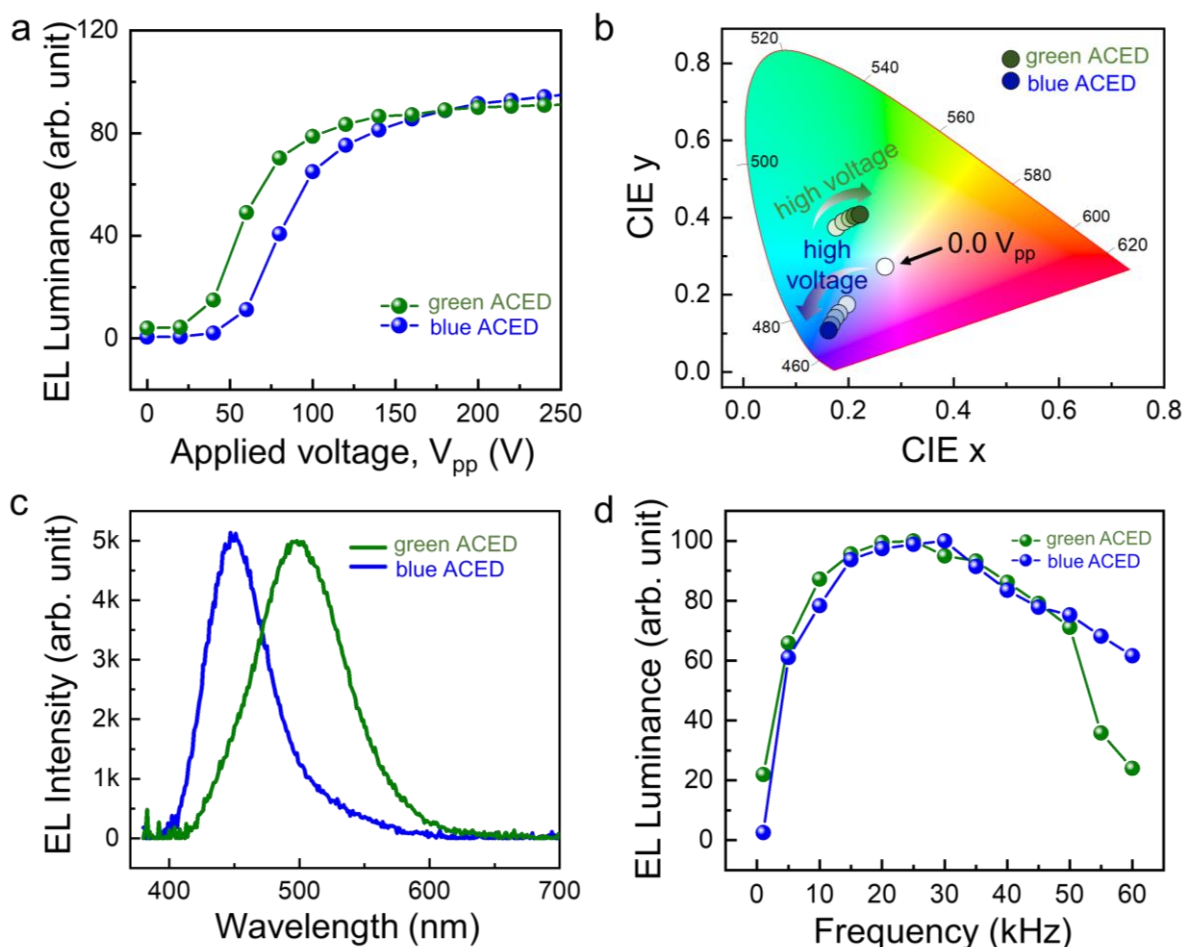
**Supplementary Fig. 20 Sensory performance of 92-AAHA-IL.** **a** Relative resistance variations ( $\Delta R/R_0$ ) under applied strains up to 700% (inset: relative resistance changes at 10, 20, 40, and 60% strains). **b** Variations in  $\Delta R/R_0$  over extended strain range, from which a GF value of  $\sim 1.92$  was extracted. It should be noted that the ionoconductors alone can serve as ionic sensing platforms with no additional components. Accordingly, strain-sensing platforms can be readily fabricated by simply connecting electronic wires.



**Supplementary Fig. 21 Application of 92-AAHA-IL in AC electroluminescent displays (ACEDs).** **a** Schematic diagram (left) of ACED and capacitance variations (right) of each component (EL layer and 92-AAHA-IL-based ionoconductor). To construct electrical double layers (EDLs), Cu tape was attached to both sides of the ionoconductor. To avoid unnecessary interfacial electrochemical reactions, the EDL area ( $A_{EDL}$ ) was a small portion of the total area of the ionoconductor ( $A_{Ion}$ ). The EDL and dielectric layer acted as a series capacitor, following the equation:  $Q = C_1V_1 = C_2V_2 = C_3V_3$ , where  $Q$  is the charge, while  $C$  and  $V$  represent the corresponding capacitance and voltage of each layer, respectively.<sup>25,26</sup> Accordingly, the EL layer had a relatively lower capacitance than the EDLs in ionoconductors in a lower frequency range (0.1 to 60 kHz), and the applied voltage was centered on the phosphor layer ( $C_1/C_2$  or  $C_3/C_2 \approx 1.8 \times 10^4$ ). **b** Schematic illustration of ACED. The EL device had a three-layer configuration where the EL phosphor layer was sandwiched between two symmetric ionoconductor layers.

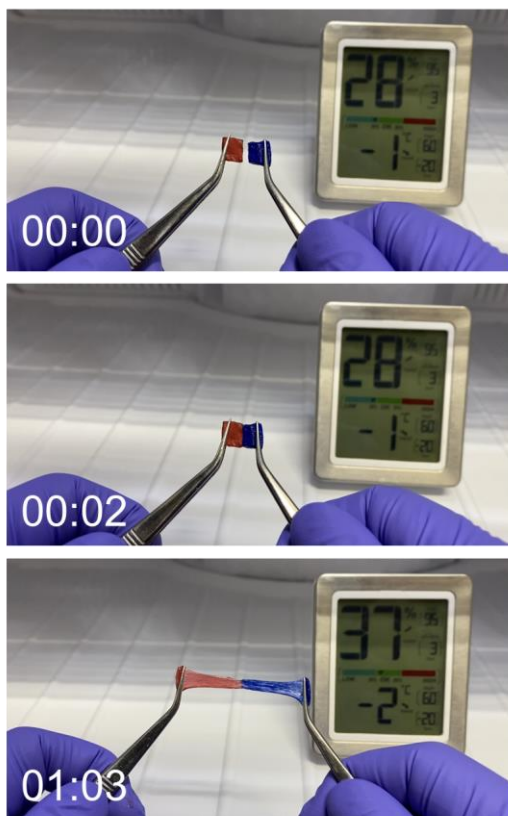


**Supplementary Fig. 22 Optical transparency of 92-AAHA-IL.** Transmittance spectrum of 300  $\mu\text{m}$ -thick 92-AAHA-IL film at various wavenumbers. The overall transmittance values were higher than  $\sim 70\%$  at 400–1100 nm.

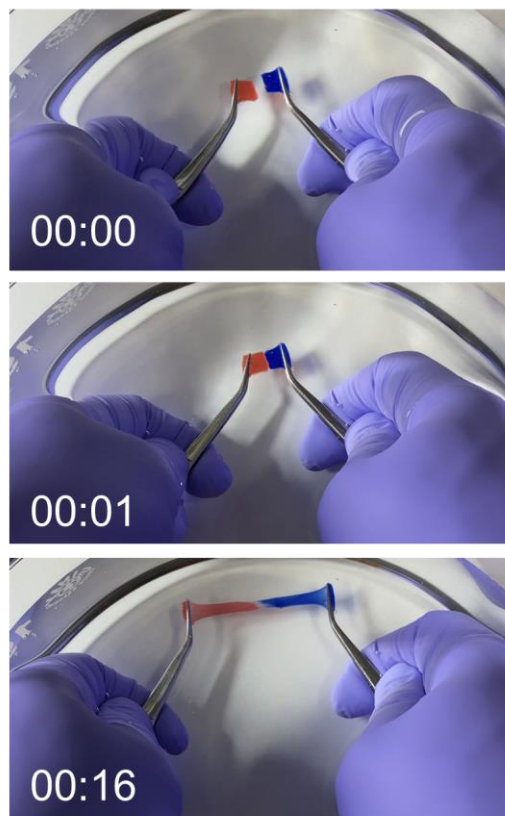


**Supplementary Fig. 23 Characteristics of AC electroluminescent displays (ACEDs).** **a** Luminance *versus* voltage characteristics and **b** CIE color coordinates of blue- and green-emitting ACED as a function of AC voltage at a frequency of 25 kHz. **c** EL emission spectra and **d** frequency dependence of EL luminance at an applied voltage of 200  $V_{pp}$ . The magnitude of the emission intensity was linearly related to the driving voltage power from the turn-on voltage ( $\sim 40 V_{pp}$ ) to the saturation of both EL brightness above 200  $V_{pp}$  (Supplementary Fig. 23a). Each green- and blue-emitting luminophore successfully served as a deep color-brightening ACED upon applying various voltages (Supplementary Fig. 23b). The normalized EL spectra of the ACEDs confirmed the blue and green emission with maximum EL wavelengths at  $\sim 450$  and  $\sim 500$  nm (Supplementary Fig. 23c). The analysis of the frequency dependence showed that the EL intensity rapidly increased up to 25 kHz at the applied voltage of 200  $V_{pp}$  (Supplementary Fig. 23d). However, further frequency increases (above 30 kHz) resulted in a steady reduction in luminance, due to the decrease in the electron acceleration.<sup>27</sup> Therefore, the operating conditions of ACEDs were optimized at an applied voltage of 200  $V_{pp}$  and a frequency of 25 kHz to synchronize the green- and blue-emitting systems.

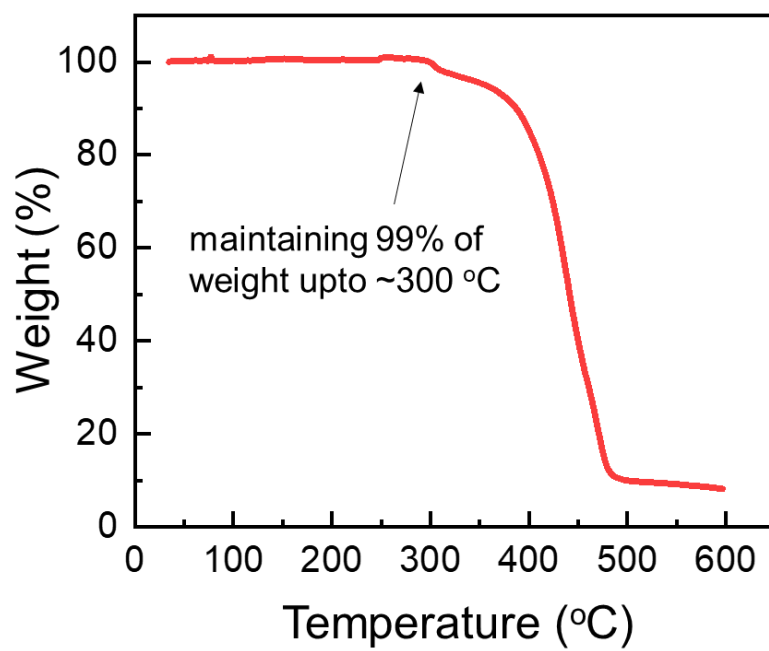
**a** Sub-zero  
Temperature Healing



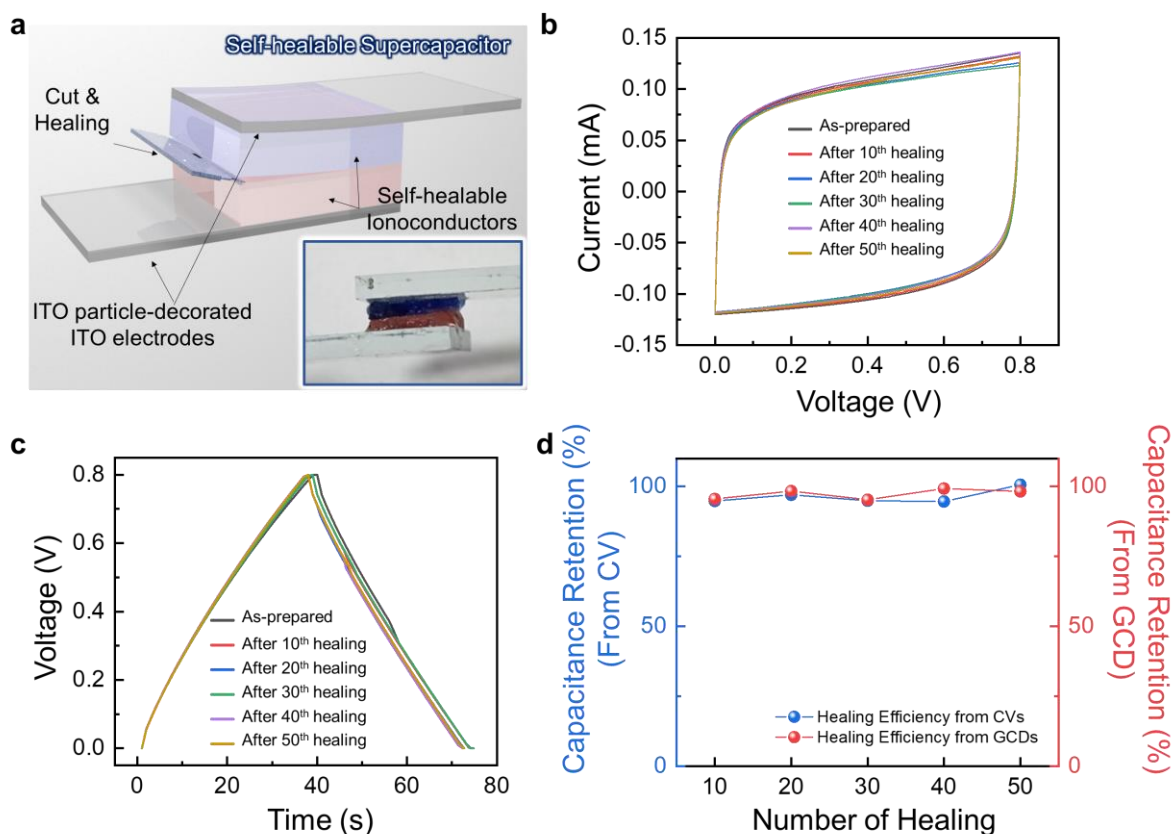
**b** Underwater Healing



**Supplementary Fig. 24 Successful self-healing in various conditions.** Successful self-healing of 92-AAHA-IL in extreme environments: **a** sub-zero temperature and **b** underwater environment.



**Supplementary Fig. 25 Thermal stability of 92-AAHA-IL.** TGA thermogram of 92-AAHA-IL, indicating its high thermal stability up to ~300 °C.



**Supplementary Fig. 26 Utilization of ultrafast self-healing performance of 92-AAHA-IL in healable supercapacitors.** **a** Schematic illustration of self-healable ionoconductor-based electrical double layer capacitor (EDLC). The inset displays a photograph of the actual device. **b** Cyclic voltammograms at a scan rate of  $25 \text{ mV s}^{-1}$  during self-healing cycles. The areal capacitance ( $C$ ) was calculated using the following equation:<sup>28</sup>  $C = \frac{1}{\Delta V(dV/dt)} \int I dV$ , where  $dV/dt$  and  $\int I dV$  are the voltage scan rate and the numerical integration of the current density in the half-cycle potential window ( $\Delta V$ ), respectively. **c** Galvanostatic charge-discharge (GCD) profiles at a current density of  $0.1 \text{ mA cm}^{-2}$ . The  $C$  values were extracted using the equation<sup>29</sup>  $C = I\Delta t/S\Delta V$ , where  $I$ ,  $\Delta t$ ,  $S$ , and  $\Delta V$  represent the discharging current, discharging time, electrode area, and potential window excluding the  $IR$  voltage drop, respectively. **d** Variation in capacitance retention ( $C/C_0$ ) as a function of number of healing cycles. A high  $C/C_0$  was maintained irrespective of the number of self-healing cycles, demonstrating the practical applicability of 92-AAHA-IL in self-healable energy storage devices.

## Supplementary References

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