Supporting Information

Side-chain length and dispersity in ROMP polymers with pore-generating side chains for gas separations

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Materials and methods

Materials: Dicyclopentadiene (Alfa Aesar), sodium hydride (Sigma–Aldrich), methyl iodide (Alfa Aesar), and Grubbs 2nd-generation catalyst (Sigma–Aldrich) were purchased from commercial sources and used as received. Anhydrous dimethylformamide (DMF) was purchased from Sigma–Aldrich in SureSeal bottles and dried over 4 Å molecular sieves prior to use. Anhydrous dichloromethane (CH_2Cl_2) was obtained from an INERT PureSolv MD5 solvent purification system and stored under Ar over 4 Å molecular sieves. All other solvents were purchased at ACS grade or higher and used as received. 1,4-Anthraquinone was purified with a silica plug (using dichloromethane as the eluent) prior to use.

Silica gel chromatography: Silica gel chromatography was performed on a Biotage Isolera flash chromatography system with Biotage SNAP Ultra columns containing HP-Sphere 25μm silica.

Nuclear magnetic resonance (NMR) spectroscopy: ¹H and ¹³C NMR spectra were obtained using Bruker Avance spectrometers at 400 or 600 MHz (100 or 150 MHz) for ¹H (¹³C), in deuterated solvents as specified, and referenced to the residual solvent signal. Spectra for quantitative integration were recorded using 16 scans and 5 s relaxation time.

Size exclusion chromatography (SEC): SEC was performed in HPLC-grade tetrahydrofuran using an Agilent 1260 Infinity system with a guard column (Agilent PLgel; 5 µm; 50 x 7.5 mm) and three analytical columns (Agilent PLgel; 5 μ m; 300 x 7.5 mm; 10⁵, 10⁴, and 10^3 Å pore sizes). The instrument was calibrated with polystyrene standards between 1.7 and 3150 kg mol−1. All runs were performed at 1.0 mL min−1 flow rate and 35 ºC. Molecular weight values were calculated using ChemStation GPC Data Analysis Software (Rev. B.01.01) based on the refractive index signal.

Matrix-assisted laser desorption/ionization (MALDI)–time of flight (TOF) mass spectrometry (MS): MALDI–TOF MS was performed on a Bruker Autoflex Speed machine using reflector mode and positive ionization. The compound *trans*-2-[3-(4-*tert*-butylphenyl)-2-methyl-2-propenylidene]malononitrile (DCTB) was used as the matrix.

Brunauer–Emmett–Teller (BET) surface area: BET surface areas of polymer powders were measured using N² sorption at 77 K using a Micromeritics ASAP 2020 analyzer. Powder samples were degassed under high vacuum at 120 °C for at least 4 hours prior to analysis.

Synthetic procedures and characterization

The synthetic procedures for the OMe monomer, oligomerization, and polymerization were previously reported^{1,2} and used without modification other than the separation step. Representative procedures are presented below.

OMe oligomer: OMe monomer was added to an oven-dried Schlenk flask, which was evacuated and backfilled with Ar three times. The monomer was heated at 220 °C for 18 h.

Separation of oligomers: The oligomer mixture was separated by silica gel chromatography using a Biotage Isolera flash chromatography system. Generally, a solvent gradient of 5% to 40% EtOAc/hexanes was successful in providing sufficient separation. Retention factor (R_f) decreases with increasing *n*. The isolated oligomers were dissolved in a small amount of CH_2Cl_2 and precipitated in MeOH prior to polymerization in order to remove impurities, presumably from the evaporated solvent used for chromatography. OMe *n*-mers were dried in the vacuum oven at 60 °C for at least 3 h.

MALDI–TOS MS was used to confirm the identity and purity of the separated OMe *n*mers. The observed *m*/*z* values match the expected values, as shown in **Table S1**.

Table S1. MALDI–TOF MS data and expected *m*/*z* values.

In addition to MALDI (see **Figure 1c**), ¹H NMR integration ratios were used to verify the separated OMe *n*-mers. **Figure S1** demonstrates the method used for NMR integration, and **Table S2** shows the expected and experimentally obtained ratios.

Figure S1. Example of the method used to obtain NMR integration ratios.

Table S2. NMR integration ratios of OMe *n*-mers.

Polymerization: OMe 4-mer (168 mg, 0.14 mmol, 1 equiv.) was added to an oven-dried Schlenk flask, which was evacuated and backfilled with Ar three times, and then dissolved in CH₂Cl₂ (1 mL). In a separate oven-dried vial, Grubbs $2nd$ -generation catalyst (1.18 mg, 0.0014 mmol, 0.01 equiv.) was dissolved in CH_2Cl_2 (0.4 mL). The catalyst solution was transferred by syringe into the oligomer solution, and the reaction mixture was stirred at room temperature for 18 h. The flask was unsealed and 1 drop of ethyl vinyl ether was added to quench the catalyst. The polymer solution was precipitated in methanol, and the solid was collected by vacuum filtration, washed with methanol, and dried under vacuum.

Table S3. Molecular weights of poly(OMe *n*-mer)s considered in this study.

	[M]/[I]	M_n (kDa)	
Poly(OMe 2-mer)	150		
Poly(OMe 3-mer)	-50		
Poly(OMe 4-mer)	つう	16	
Poly(OMe 5-mer)	0 ⁰		

Table S4. BET surface areas of poly(OMe 2-mer) through poly(OMe 5-mer) powders. The discrepancy between the reported BET surface area of OMe-ROMP here (484 m^2 g⁻¹) and the previously reported BET surface area² (146 m² g⁻¹) is attributed to variation in sample preparation and measurement techniques.³

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Figure S2. N₂ adsorption isotherms and pore size distributions (PSDs) calculated using the nonlocal density functional theory (NLDFT) using the standard slit carbon model of (a) poly(OMe 2-mer), (b) poly(OMe 3-mer), (c) poly(OMe 4-mer), (d) poly(OMe 5-mer), and (e) polydispersed OMe-ROMP obtained from Brunauer–Emmett–Teller (BET) analysis.

Membrane fabrication and treatment

Self-standing films of poly(OMe *n*-mer)s were made by dissolving polymers in chloroform to create ~3 wt% polymer solutions. The solutions were then cast into 50 mm diameter flat-bottom glass petri dishes that contained Norton® fluorinated ethylene propylene (FEP) liners (Welch Fluorocarbon). After 4–5 days of slow evaporation at room temperature in a fumehood, stable and defect-free films were formed.

Two different treatments were employed on the self-standing films. Thermally-treated films were dried at 120 °C for 24 h under vacuum to remove residual solvent, then dried at ambient conditions for 24 h and degassed under full vacuum at 35 °C for 8 h. Alcohol-treated films were soaked in either ethanol (poly(OMe 2-mer)) or methanol (poly(OMe 3-mer) through poly(OMe 5 mer)) for 48 h. After alcohol treatment, films were air-dried in a fumehood for 24 h before testing in the permeation system.

Polymer film density for thermally-treated samples was determined using Archimedes' principle using *n*-hexane as the buoyant liquid, since the density of water was expected to be close to the sample density. Measurements were conducted using a density measurement kit from Mettler Toledo (ME-DNY-4). The fractional free volume (FFV) was then calculated for each sample using the following equation:

$$
FFV = \frac{V - 1.3V_w}{V}
$$

where *V* is the molar volume of the polymer (cm³ mol⁻¹) and V_w is the van der Waals volume of the polymer $(cm³ mol⁻¹)$ determined using group contribution methods. "Method 1" refers to the method first developed by Bondi,⁴ and updated by Park and Paul⁵ in 1997 and van Krevelen⁶ in 2009 to account for larger functional groups. "Method 2" refers to the method updated by Wu et al. that accounts for new, unique structures that contain novel, contorted structural units and pendant groups that have been popularized since the first report of PIM-1 as a gas separation membrane.^{7,8}

Table S5. Density, van der Waals volumes (V_w) , and fractional free volume (FFV) of poly(OMe *n*-mer) samples from $n = 2-5$. "Method 1" refers to group contribution methods developed by Bondi, van Krevelen, and Park and Paul.^{4–6} "Method 2" refers to an updated group contribution method developed by Wu et al.⁷

n	Density $(g \text{ cm}^{-3})$	$\boldsymbol{V}_{\boldsymbol{W},\boldsymbol{Method}}$ 1 $\rm (cm^3\, mol^{-1})$	$\boldsymbol{V}_{w,Method~2}$ $(cm3 mol-1)$	FFV (Method 1)	FFV (Method 2)
	1.184 ± 0.005	362.64	325.51	0.077 ± 0.004	0.171 ± 0.004
	1.134 ± 0.009	529.11	482.15	0.140 ± 0.007	0.216 ± 0.007
	± 0.02	695.58	638.79	0.17 ± 0.02	± 0.02 0.24
	± 0.01	862.05	795.43	± 0.01	± 0.01

Figure S3. Fractional free volume (FFV) as a function of side-chain length (*n*). Red squares represent calculations using "Method 1".4–6 Black circles represent calculations using "Method 2".⁷

Pure-gas permeability measurements

Pure-gas permeability measurements of samples were performed on an automated constant-volume, variable-pressure permeation system from Maxwell Robotics. Polymer films were cut, placed on top of a hole in the center of a brass disk, and glued to the brass disk using epoxy glue (Devcon 5 min Epoxy). The glue was left to dry for at least 30 min. Afterwards, the polymer samples were sealed inside a stainless steel permeation cell (Millipore) and immersed in a water bath that was maintained at 35 °C using an immersion circulator (ThermoFisher SC150L). All gases used for testing (He, H_2 , CH₄, N₂, O₂, and CO₂) were ultra-high purity from Airgas.

The permeabilities of the six aforementioned gases were determined at \sim 1 bar. Before testing permeation, the testing chamber was dosed with \sim 2 bar of helium gas to ensure that no residual gas remained in the system. Then, the samples were held under vacuum at 35 °C for 8 h. Before switching to a new permeating gas for testing, samples were again dosed with \sim 2 bar of helium gas and held under vacuum for at least 1 h.

Pure-gas permeability (P) was calculated using the following equation:

$$
P = \frac{V_d l}{p_2 A R T} \left[\left(\frac{dp}{dt} \right)_{ss} - \left(\frac{dp}{dt} \right)_{leak} \right]
$$

in which V_d is the volume downstream of the film, *l* is the film thickness, p_2 is the upstream pressure, \vec{A} is the area of film exposed to the gas, \vec{R} is the ideal gas constant, \vec{T} is the absolute experimental temperature, $\left(\frac{dp}{dt}\right)_{ss}$ is the rate of pressure rise in the permeate at steady state, and $\left(\frac{dp}{dt}\right)_{leak}$ is the leak rate.⁹ The ideal gas selectivity $(\alpha_{i,j})$ was taken to be the ratio of the pure-gas permeabilities of the more permeable gas, *i*, to that of the less permeable gas, *j* (i.e., $\frac{P_i}{P_j}$). Diffusion coefficients for each gas were determined using the time-lag method, $D = \frac{l^2}{c}$ $\frac{1}{6\theta}$, in which θ is the time lag.¹⁰ Since the diffusion coefficients for smaller gases (i.e., He and H₂) were sometimes outside of the resolution of the acquisition time of the permeation system, which is approximately 1–2 s, diffusion coefficients for these two gases are not reported. Sorption coefficient were backcalculated using the sorption–diffusion model ($S = \frac{P}{R}$ $\frac{P}{D}$).¹¹ Error bars for permeability, diffusion coefficients, and sorption coefficients were determined using the error propagation method.¹² Thicknesses of each sample are shown in **Table S6**. Aged samples, which are shown in **Table S6** with their aging times, are separate samples from "fresh" (i.e., 1 day aged) samples.

Table S6. Gas separation performance of all poly(OMe *n*-mer)s in this study. Permeability coefficients (P) are given in barrer (10^{-10} cm³(STP) cm cm⁻² s⁻¹ cmHg⁻¹), diffusion coefficients (D) are given in 10^{-8} cm² s⁻¹, and sorption coefficients (S) are given in cm³(STP) cm⁻³ atm⁻¹. All data were obtained at 35 °C and ~1 bar upstream pressure.

Polymer	Treatment		He	$\overline{\textbf{H}}_2$	N ₂	$\overline{\mathbf{O}}_2$	$\overline{\text{CH}_4}$	CO ₂
Poly(OMe	120 °C 24 h	P	53 ± 2	$80 \pm$ 3	3.6 ± 0.1	0.4 12.7 \pm	\pm 0.2 4.7	$83.4 \pm$ 2.9
2 -mer $)$	vacuum. $\mathbf{1}$	D			3.6 \pm 0.1	8.7 0.2 \pm	$0.91 \pm$ 0.02	$4.3 \pm$ 0.1
	day aged	S			0.75 ± 0.03	1.1 0.1 \pm	3.8 \pm 0.2	$14.7 \pm$ 0.6
$l = 91 \text{ µm}$								
Poly(OMe	EtOH treat	P	107 ± 3	$154 \pm$ 5	13.4 0.4 \pm	33 \pm $\mathbf{1}$	13.9 0.4 \pm	190 6 \pm
2 -mer $)$	48 h, air-dry 24 h, 1 day	D \overline{s}			0.2 13.1 \pm	34.2 \pm 1.1 $0.73 \pm$	4.5 0.1 \pm	$13.7 \pm$ 0.2
$l = 79 \text{ µm}$	aged				0.76 ± 0.03	0.03	2.3 \pm 0.1	0.4 $10.4 \pm$
Poly(OMe	120 °C 24 h	\boldsymbol{P}	82 ± 5	$135 \pm$ 9	0.4 5.4 \pm	26.3 1.7 \pm	8.5 0.5 \pm	178 11 \pm
3 -mer $)$	vacuum, $\mathbf{1}$	D			6.1 0.1 \pm	20.6 0.2 \pm	2.0 \pm 0.1	$9.9 \pm$ 0.1
	day aged	S			± 0.1 0.7	\pm 0.1 1.0	3.2 \pm 0.2	0.9 $13.5 \pm$
$l = 130 \text{ µm}$ Poly(OMe	MeOH treat	\boldsymbol{P}	$\overline{169} \pm 6$	304 ± 11	0.7 19.0 \pm	2.3 65.7 \pm	28.5 \pm 1.0	17 471 \pm
3 -mer $)$	48 h, air-dry	D			0.3 11.6 \pm	35.4 1.2 \pm	0.1 3.5 \pm	$18.9 \pm$ 4.9
	24 h, 14 days	S			1.2 \pm 0.1	\pm 0.1 1.4	0.3 6.1 \pm	$18.7 \pm$ 0.8
$l = 91 \mu m$	aged							
Poly(OMe	120 °C 24 h	\pmb{P}	243 ± 10	462 ± 19	36.0 \pm 1.5	5 112 \pm	2.4 58.4 \pm	34 830 \pm
4 -mer $)$	vacuum, 1	D			22.3 0.9 \pm	55.6 \pm 3.7	0.3 6.7 \pm	$33.4 \pm$ 1.6
	day aged	\mathbf{s}	$\sqrt{2}$		1.2 \pm 0.1	1.5 ± 0.1	6.5 \pm 0.4	$18.6 \pm$ 1.2
$l = 85 \mu m$ Poly(OMe	120 °C 24 h	\boldsymbol{P}	$\frac{1}{243} \pm 11$	$\frac{1}{451}$ ± 21	34.4 \pm 1.6	108 5 \pm	2.6 55.6 \pm	788 36 \pm
4 -mer $)$	vacuum, 34	D			19.5 1.1 \pm	3.1 $\frac{49.6}{4}$ ±	6.1 0.4 \pm	29.8 \pm 1.8
	days aged	S			0.1 1.3 \pm	1.6 ± 0.1	0.5 6.8 \pm	$19.8 \pm$ 1.5
$l = 103 \mu m$								
Poly(OMe	120 °C 24 h	\boldsymbol{P}	$\frac{1}{422}$ ± 18	839 ± 35	79.8 3.3 \pm	224 9 \pm	137 6 \pm	1569 66 \pm
4 -mer $)$	vacuum,	D			1.9 45.0 \pm	8.4 96.8 \pm	14.9 \pm 0.3	55.8 \pm 2.8
	MeOH treat	\overline{s}			1.3 ± 0.1	1.7 ± 0.2	6.9 0.3 \pm	1.4 $21.1 \pm$
$l = 82 \mu m$	48 h, air-dry							
	24 h, 1 day							
Poly(OMe	aged 120 °C 24 h	\boldsymbol{P}	348 ± 23	717 ± 47	62.5 4.1 \pm	\pm 12 189	109 τ \pm	1355 89 \pm
4 -mer $)$	vacuum,	D			35.5 3.5 \pm	89.5 ± 10.1	12.1 \pm 1.1	54.0 \pm 5.5
	MeOH treat	\mathbf{s}			1.3 ± 0.2	1.6 ± 0.2	6.8 \pm 0.8	2.3 $18.8 \pm$
$l = 94 \mu m$	48 h, air-dry							
	24 h, 3 days							
	aged							
Poly(OMe	120 °C 24 h	\boldsymbol{P}	497 ± 39	1077 ± 85	107 \pm 8	± 24 309	185.1 ± 14.6	2247 \pm 177
5 -mer $)$	vacuum, 10	D			45.5 6.7 \pm	111 ± 19	16.2 \pm 2.3	67.8 \pm 10.4
$l = 82 \mu m$	days aged	S			1.8 \pm 0.3	2.1 ± 0.4	8.6 ± 1.4	$24.8 \pm$ 4.3
Poly(OMe	120 °C 24 h	\boldsymbol{P}	716 ± 27	1656 ± 62	$\overline{8}$ 214 \pm	552 ± 21	404 \pm 15	3762 ± 141
5 -mer $)$	vacuum,	D			69.2 \pm 1.6	146 7 \pm	$28.6~\pm~0.3$	$81.7 \pm$ 2.2
	MeOH treat	\mathbf{s}			\pm 0.1 2.3	\pm 0.2 2.8	10.6 ± 0.4	$34.5 \pm$ 1.6
$l = 138 \,\mathrm{\upmu m}$	48 h, air-dry							
	24 h, 1 day							
	aged							
Poly(OMe	120 °C 24 h	P	1076 ± 75	2476 ± 173	± 21 301	797 ± 56	567 ± 40	5324 ± 372
5 -mer $)$	vacuum, MeOH treat	D			102 \pm 4	203 ± 14	$39.9 \pm$ 1.0	129 \pm 6
$l = 134 \,\mathrm{\upmu m}$	48 h, air-dry	\mathbf{s}			\pm 0.2 2.3	\pm 0.3 3.0	10.8 ± 0.8	$31.5 \pm$ 2.6
	24 h, 2 days							
	aged							

Figure S4. Robeson plots of poly(OMe *n*-mer)s, OMe-ROMP, and CF₃-ROMP for (a) CO₂/CH₄, (b) H_2/CH_4 , and (c) H_2/N_2 gas pairs. Black and gray lines represent the 2008 and 1991 Robeson upper bounds, respectively.^{13,14} Filled shapes represent alcohol-treated samples, and open shapes represent thermally-treated samples.

Figure S5. Diffusion coefficient for (a) N_2 , (b) CH₄, and (c) CO₂ versus side-chain length (*n*). (d) Diffusivity selectivity for $n = 4$ and $n = 5$ uniform poly(OMe n -mer) and non-uniform OMe-ROMP with average $n = 4.5$ for CO_2/CH_4 , N_2/CH_4 , and CO_2/N_2 gas pairs.

Table S7. Diffusivity selectivity for fresh methanol-treated samples of poly(OMe 4-mer), OMe-ROMP (4.5 average side-chain length), and poly(OMe 5-mer).

	poly(OMe 4-mer)	OMe-ROMP (4.5 average) 2	poly(OMe 5-mer)
CO ₂ /CH ₄	4.35 ± 0.45	3.89 ± 0.25	3.04 ± 0.18
CO ₂ /N ₂	1.41 ± 0.12	1.34 ± 0.07	1.22 ± 0.04
O_2/N_2	2.43 ± 0.20	2.29 ± 0.15	2.06 ± 0.06
N_2 /CH ₄	3.08 ± 0.15	2.90 ± 0.05	2.48 ± 0.06

CO2-induced plasticization study

Figure S6. Hysteresis induced by conditioning of films at 51 bar of CO₂ for all samples in this study. Results for CF₃-ROMP, OMe-ROMP, and PIM-1 from our previous work² are included here for comparison.

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