Supplementary Information

Quantification of gas-accessible microporosity in metal-organic framework glasses

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Supplementary Notes 1 - Graphical representations of network topologies

Supplementary Figure 1. Graphical representations of the relevant network topologies under discussion here. The images were created based on the topological data from the *Reticular Chemistry Structure Resource Database* (http://rcsr.anu.edu.au). The cyan spheres represent the positions of the Zn²⁺ ions (i.e. the tetrahedral vertices of the network and the grey sticks represent the positions of the imidazolate-type linkers. Unit cells are shown with black lines. The three letter codes originate from the names of typical framework materials possessing these topologies: **cag**, CaGa2O4; **zni**, zinc imidazolate; **gis**, gismondine; **sod**, sodalite; **lta**, Linde type A zeolite.

Supplementary Methods 2 - X-ray powder diffraction data

Supplementary Figure 2. Profile fit (Pawley method) to the XRPD pattern of ZIF-4. The blue tick marks indicate the positions of allowed Bragg peaks.

Supplementary Figure 3. Profile fit (Pawley method) to the XRPD pattern of ZIF-zni. The blue tick marks indicate the positions of allowed Bragg peaks.

Supplementary Figure 4. Profile fit (Pawley method) to the XRPD pattern of zni_TZIF-4. The blue tick marks indicate the positions of allowed Bragg peaks.

Supplementary Figure 5. Profile fit (Pawley method) to the XRPD pattern of ZIF-62. The blue tick marks indicate the positions of allowed Bragg peaks.

Supplementary Figure 6. Profile fit (Pawley method) to the XRPD pattern of TIF-4. The blue tick marks indicate the positions of allowed Bragg peaks.

Intensity / arb. units	н						
	10 15 5	25 20	35 30	45 40 50			
2θ / \circ Supplementary Figure 6. Profile fit (Pawley method) to the XRPD pattern of TIF-4. The blue tick marks indicate the positions of allowed Bragg peaks. Supplementary Table 1. Unit cell parameters and corresponding R_{wp} , R_{exp} and χ values determined by the above displayed structureless profile fits (Pawley method).							
compound	$ZIF-4$	$zniTZIF-4$	ZIF-zni	ZIF-62	TIF-4		
crystal system	orthorhombic	tetragonal	tetragonal	orthorhombic	orthorhombic		
space group	Pbca	$I4_1cd$	$I4_1cd$	Pbca	Pbca		
a/\r{A}	15.488(3)	23.522(2)	23.499(2)	15.548(10)	15.483(7)		
b/\AA	15.528(3)	23.522(2)	23.499(2)	15.819(11)	15.789(8)		
c/\r{A}	18.113(4)	12.4512(11)	12.4691(12)	18.164(16)	18.113(10)		
α / \circ	90	90	90	90	90		
β / \circ	90	90	90	90	90		
$\gamma/$ °	90	90	90	90	90		
V/\AA ³	4356.1(16)	6889.1(14)	6885.6(16)	4468(6)	4431(4)		
$R_{\rm wp}$ / %	20.33	21.98	20.09	24.18	21.15		
$R_{\rm exp}$ / %	15.72	19.42	16.57	16.92	16.81		
χ	1.29	1.13	1.21	1.43	1.26		

Supplementary Table 1. Unit cell parameters and corresponding R_{wp} , R_{exp} and χ values determined by the above displayed structureless profile fits (Pawley method).

Supplementary Figure 8. XRPD pattern of a_gZIF-4.

Supplementary Figure 9. XRPD pattern of agZIF-zni.

Supplementary Figure 10. XRPD pattern of agZIF-62.

5 10 15 20 25 30 35 40 45 50 Inte n s
h units $2\theta l^{\circ}$ a_gZIF-4 *a*gZIF-zni'

Supplementary Figure 12. Overlay of the XRPD patterns of a_gZIF-zni' and a_gZIF-4' which were obtained by attempting to prepare the desired glasses without an isothermal segment in the TGA/DSC program (see Section S5.2). The black tick marks indicate the allowed Bragg peak positions for ZIF-zni (CCDC code IMIDZB).

Supplementary Methods 3 - Fourier-transform infrared (FTIR) spectroscopy data

The activation of all materials is demonstrated by the absence of the carbonyl stretching band of DMF at 1675 cm⁻¹. DMF is found in the as-synthesized materials as the template for the porous channels (except for ZIF-zni).^{1,2}

For ZIF-4 and the amorphous phases derived thereof, a broadening of the vibrational band at 835 cm⁻¹ ascribed to the out-of-plane ring deformation of the imidazolate linker³ is observed (see Supplementary Figure 13). For the **zni** phases (ZIF-zni and zni_TZIF-4) a much sharper band is observed which may be caused by the higher density of the material corresponding to fewer degrees of freedom for this vibration.

Supplementary Figure 13. Left**:** Stacked plot of the FTIR spectroscopy data of ZIF-4 and ZIF-zni and their thermal products. Right: Zoom in the region from 2000-500 cm–1. The discussed bands at 835 cm-1 are highlighted in light blue.

Supplementary Figure 14. Left: FTIR spectroscopy data of ZIF-62 and a_gZIF-62. Right: FTIR spectroscopy data of TIF-4 and a_aTIF-4.

Supplementary Methods 4 - 1H NMR spectroscopy data

The activation of all materials is demonstrated by the absence of signals ascribed to DMF. 1,2

For TIF-4, ZIF-62 and their corresponding glasses the ratio between the implemented linkers has been determined by the integral corresponding to the proton attached to the carbon atom between the two nitrogen atoms in the imidazolate type linkers. The corresponding signals are integrated in the spectra below.

A deeper inspection of the 1H NMR spectroscopy data unveiled additional small signals in the range of 9.0 ppm – 9.5 ppm and 7.5 ppm – 8.5 ppm for $a₉ZIF-4$ and $a₉ZIF-zni$ which are ascribed to aromatic or polyaromatic compounds formed as the result of partial framework decomposition during high temperature treatment under inert atmosphere (see Figure 2c for a zoom into the aromatic region). For agZIF-4 this has been already observed in the literature.⁴ However, as shown by the intensity of the ^{13}C satellite signals assigned to the protons of imidazole, the amount of decomposition is rather low.

Supplementary Figure 15. 1H NMR spectroscopy data of ZIF-4 recorded in a solvent mixture of DMSO-*d*⁶ and DCI/D₂O at room temperature.

Supplementary Figure 16. ¹H NMR spectroscopy data of a_TZIF-4 recorded in a solvent mixture of DMSO- d_6 and DCI/D₂O at room temperature.

Supplementary Figure 17. ¹H NMR spectroscopy data of zni_TZIF-4 recorded in a solvent mixture of DMSO-d₆ and DCl/D2O at room temperature.

Supplementary Figure 18. 1H NMR spectroscopy data of agZIF-4 recorded in a solvent mixture of DMSO-*d*⁶ and DCI/D₂O at room temperature.

Supplementary Figure 19. 1H NMR spectroscopy data of ZIF-zni recorded in a solvent mixture of DMSO-*d*⁶ and DCI/D₂O at room temperature.

Supplementary Figure 20. 1H NMR spectroscopy data of agZIF-zni recorded in a solvent mixture of DMSO-*d*⁶ and DCl/D2O at room temperature.

Supplementary Figure 21. 1H NMR spectroscopy data of ZIF-62 recorded in a solvent mixture of DMSO-*d*⁶ and DCI/D₂O at room temperature.

Supplementary Figure 22. 1H NMR spectroscopy data of agZIF-62 recorded in a solvent mixture of DMSO-*d*⁶ and DCl/D2O at room temperature.

Supplementary Figure 23. 1H NMR spectroscopy data of TIF-4 recorded in a solvent mixture of DMSO-*d*⁶ and DCI/D₂O at room temperature.

Supplementary Figure 24. 1H NMR spectroscopy data of agTIF-4 recorded in a solvent mixture of DMSO-*d*⁶ and DCI/D₂O at room temperature.

Supplementary Methods 5 - Thermal analysis

Supplementary Methods 5.1 - Differential scanning calorimetry (DSC)

For all DSC measurements a heating rate of $+10$ °C min⁻¹ was applied. Samples of ZIF-4 and ZIF-zni were heated to a maximum temperature of 600 °C. Samples of ZIF-62, TIF-4 and their corresponding glasses were heated to a maximum temperature of 485 °C.

Data analysis was performed with the TRIOS (v5.1.0.46403) software from TA Instruments. The melting temperatures (T_m) are determined as the peak offset, the glass transition temperatures (T_g) as the peak onset, whereas all other derived temperatures are defined as the peak temperature. The enthalpies are determined from the integral of the corresponding signal.

Supplementary Table 2. Summary of the phase transition temperature determined from the DSC data.

Supplementary Table 3. Summary of the phase transition enthalpies of ZIF-4 and ZIF-zni determined from the DSC data.

Supplementary Figure 25. DSC data for a_gZIF-62 showing a glass transition.

Supplementary Figure 26. DSC data for agTIF-4 showing a glass transition.

Supplementary Methods 5.2 - Simultaneous thermogravimetric analysis / differential scanning calorimetry (TGA/DSC)

Several different temperature programs were utilized for the preparation of the thermal products of the investigated ZIFs starting with material of the solvothermally synthesized corresponding crystalline precursor (see Supplementary Table 4).

Supplementary Table 4. Summary of the temperature programs applied for the materials' preparation via TGA/DSC experiments.

material	temperature program				
a⊤ZIF-4	RT (+10 °C min ⁻¹) \rightarrow 379 °C (-10 °C min ⁻¹) \rightarrow RT				
$zniTZIF-4$	RT (+10 °C min ⁻¹) \rightarrow 546 °C (-10 °C min ⁻¹) \rightarrow RT				
a_{α} ZIF-4	RT (+10 °C min ⁻¹) \rightarrow 578 °C -10 min \rightarrow 578 °C (-10 °C min ⁻¹) \rightarrow 50 °C \rightarrow RT				
a _a ZIF-zni	RT (+10 °C min ⁻¹) \rightarrow 578 °C -10 min \rightarrow 578 °C (-10 °C min ⁻¹) \rightarrow 50 °C \rightarrow RT				
a_q ZIF-62	RT (+10 °C min ⁻¹) \rightarrow 485 °C (-10 °C min ⁻¹) \rightarrow RT				
a_{α} TIF-4	RT (+10 °C min ⁻¹) \rightarrow 485 °C (-10 °C min ⁻¹) \rightarrow RT				

Supplementary Figure 27. TGA/DSC data for the preparation of a_TZIF-4. The cooling process (rate -10 K min⁻¹) was performed without data collection.

Supplementary Figure 28. TGA/DSC data for the preparation of zni_TZIF-4. The cooling process (rate –10 K min-1) was performed without data collection.

Supplementary Methods 6 - Photography

Supplementary Figure 33. Photographs of agZIF-4 obtained after the preparative TGS/DSC experiment.

Supplementary Figure 34. Photograph of agZIF-zni obtained after the preparative TGS/DSC experiment.

Supplementary Figure 35. Photograph of a₉ZIF-62 obtained after the preparative TGS/DSC experiment.

Supplementary Figure 36. Photograph of a_gTIF-4 obtained after the preparative TGA/DSC experiment.

Supplementary Methods 7 - Scanning electron microscopy

Supplementary Figure 37. SEM images of ZIF-zni (ground sample).

Supplementary Figure 38. SEM images of zni_TZIF-4 (ground sample).

Supplementary Figure 39. SEM images of agZIF-4 (ground sample). **Supplementary Figure 40.** SEM images of agZIF-zni (ground sample).

Supplementary Methods 8 - X-ray total scattering data

X-ray total scattering data have been collected for all investigated materials. From these data, pair distribution functions in the form *D*(*r*) have been calculated showing long range order correlations (up to at least 50 Å) for all crystalline materials (ZIF-4, ZIF -zni, zni $TZIF$ -4, ZIF -62 and TIF -4) whereas the last intense peak is found at approx. 5.9 Å for all amorphous materials (a_TZIF-4 , a_qZIF-4 , $a_qZIF-52$, a_qTIF-4 ; see Figure 2e and Supplementary Figure 42). This peak equals the Zn-Zn distance in the materials. The data are in accordance with previously reported total scattering data.⁴⁻ 8

Supplementary Figure 41. X-ray total scattering data in the form *S*(*Q*) for all investigated materials. Crystalline and amorphous phases are displayed in black or red, respectively. Data for ZIF-62 and agZIF-62 are taken from ref. 1**.**

Supplementary Figure 42. Pair distribution functions in the form $D(r)$ for all investigated materials. Crystalline and amorphous phases are displayed in black or red, respectively. Data for ZIF-62 and agZIF-62 are taken from ref. 1**.**

Supplementary Figure 43. Zoom into the background subtracted X-ray total scattering data of agZIF-4 and agZIF-zni compared to a simulated XRPD pattern of ZnO (crystallographic data taken from ref. 9). It becomes evident that the small but sharp peaks found in $\rm a_{g}Z$ IF-4 and $\rm a_{g}Z$ IF-zni can be ascribed to ZnO.

Supplementary Methods 8.1 - Fitting of the first sharp diffraction peak (FSDP)

The FSDP of the total scattering data in the form *S*(*Q*) (see Supplementary Figure 44) has been fitted to a pseudo-Voigt function for all investigated amorphous materials. From these fits we obtained the position of the FSDP (*Q*FSDP) and the peak width at half maximum (ΔQF_{SDP}) (see Supplementary Table 5). The calculations have been performed with the Fityk software package.10

Supplementary Figure 44. Overlay of X-ray total scattering data in the form *S*(*Q*) for all investigated amorphous materials. Zoom in the region from 0.5-5.5 Å-1. Data for agZIF-62 are taken from ref. 1**.**

Supplementary Table 5. Summary of the parameter obtained after fitting of the first sharp diffraction peak of the total scattering data for all investigated amorphous phases.

Supplementary Methods 9 - Isothermal gas physisorption data

N2 physisorption studies

Supplementary Figure 45. N₂ sorption isotherms recorded at 77 K for ZIF-4 (left) and the corresponding thermal products as well as ZIF-zni and its corresponding glass (right). Adsorption and desorption branches are shown as close and open symbols, respectively. ZIF-4 shows a previously reported breathing behaviour upon adsorption of N2.11–13

Supplementary Figure 46. N₂ sorption isotherms recorded at 77 K for ZIF-62 and a_gZIF-62 (left) and TIF-4 and a_gTIF-4 (right). Adsorption and desorption branches are shown as close and open symbols, respectively.

Supplementary Figure 47. N2 sorption isotherms recorded at 195 K for agZIF-4, agZIF-62 and agTIF-4. Adsorption and desorption branches are shown as close and open symbols, respectively. Notice that N2 is far above its critical temperature $(T_c = 126.192 \text{ K})$ at 195 K.

Ar physisorption studies

Supplementary Figure 48. Ar sorption isotherms recorded at 87 K for agZIF-4, agZIF-62 and agTIF-4. Adsorption and desorption branches are shown as close and open symbols, respectively. The negative uptake of agZIF-62 and a_g ZIF-4 is a measurement error due to the inaccessibility of the materials pores for Ar gas.

CO2 physisorption studies

Supplementary Figure 49. CO₂ sorption isotherms recorded at 273 K for ZIF-4 and ZIF-zni and their corresponding thermal products. Adsorption and desorption branches are shown as close and open symbols, respectively.

Supplementary Figure 50. CO2 sorption isotherms recorded at 273 K for ZIF-62 (data are taken from ref. 1) and a_q ZIF-62 (left) as well as TIF-4 and a_q TIF-4 (right). Adsorption and desorption branches are shown as close and open symbols, respectively.

Supplementary Figure 51. High-pressure CO2 adsorption isotherm of ZIF-62 recorded at 298 K with a maximum pressure of about 4130 kPa (corresponding to $p/p_0 = 0.64$). The absolute uptake (n_{absolute}) calculated from the measured excess uptake (n_{excess}) is shown: $n_{\text{absolute}} = n_{\text{excess}} + V_{\text{pore}} \cdot \rho_{\text{mol}}$; with $V_{\text{pore}} =$ pore volume of the adsorbent and ρ_{mol} = molar gas phase density of the adsorptive. The pressure dependent molar density of CO₂ were obtained from National Institute of Standards and Technology (NIST, https://webbook.nist.gov/chemistry/fluid/) and is based on an equation of state for CO₂ developed by Span and Wagner.¹⁴ The density data were fit to a fifth order polynomial (**Supplementary Figure** 52) in order to be utilized for the calculation of *n*absolute from *n*excess.

Supplementary Figure 52. Fifth order polynomial fit to the pressure-dependent molar gas phase density of CO₂ derived from NIST Chemistry Webbook (see caption of **Supplementary Figure** 51 for details).

The absolute $CO₂$ uptake recorded at 298 K and 4130 kPa amounts to 2.55 mmol $g⁻¹$. By considering the molar mass of $CO₂$ (44.009 g mol⁻¹) and the saturated liquid phase density of $CO₂$ at 298 K (0.7128 g cm⁻³), we determine a pore volume of ZIF-62 of 0.16 cm^3 g⁻¹. This is the same value as the one determined from the CO₂ sorption isotherms recorded at 195 K (applying the density of the supercooled liquid of CO₂ extrapolated to 195 K, see Supplementary Figure 62). Hence, the high-pressure $CO₂$ sorption data verify the robustness of our data analysis of the CO₂ sorption data recorded at 195 K.

*n***-Butane physisorption studies**

Supplementary Figure 53. *n*-Butane sorption isotherms recorded at 273 K for ZIF-4 and ZIF-zni and their corresponding thermal products. Adsorption and desorption branches are shown as close and open symbols, respectively.

Supplementary Figure 54. *n*-Butane sorption isotherms recorded at 273 K for ZIF-62 and TIF-4. Adsorption and desorption branches are shown as close and open symbols, respectively.

Supplementary Figure 55. *n*-Butane sorption isotherms recorded at 273 K for agZIF-62. The same material was repeatedly measured three times (colour code: 1st cycle black squares; 2nd cycle green squares; 3rd cycle red squares). Between the measurement, the materials were degassed for 30 min at 200 °C to desorb the residual *n*butane from the previous measurement. Adsorption and desorption branches are shown as close and open symbols, respectively. The repeatability of the experiment signifies that the observed hystereses are not associated with structural changes of the glass during gas sorption (i.e. a swelling or breathing behaviour).

Supplementary Figure 56. *n*-Butane sorption isotherms recorded at 293 K for a_gZIF-4, a_gZIF-62 and a_gTIF-4. Adsorption and desorption branches are shown as close and open symbols, respectively.

Propane and propylene physisorption studies

Supplementary Figure 57. Propane and propylene isotherms recorded at 293 K for ZIF-4 (left) and a_gZIF-4 (right). Adsorption and desorption branches are shown as close and open symbols, respectively. ZIF-4 shows similar total uptakes for both C3 hydrocarbons with a slightly higher affinity for propane at lower pressures which has been reported in the literature before.15

Supplementary Figure 58. Propane and propylene isotherms recorded at 293 K for ZIF-62 (left) and agZIF-62 (right). Adsorption and desorption branches are shown as close and open symbols, respectively.

Supplementary Figure 59. Propane and propylene isotherms recorded at 293 K for TIF-4 (left) and a_gTIF-4 (right). Adsorption and desorption branches are shown as close and open symbols, respectively.

Supplementary Figure 60. Propane isotherms recorded at 313 K for a_gZIF-4, a_gZIF-62 and a_gTIF-4. Adsorption and desorption branches are shown as close and open symbols, respectively.

Supplementary Figure 61. Propylene isotherms recorded at 313 K for agZIF-4, agZIF-62 and agTIF-4. Adsorption and desorption branches are shown as close and open symbols, respectively.

Supplementary Methods 9.1 - Summary of adsorption capacities of MOF glasses

Supplementary Table 6. Summary of maximum adsorption capacities of CO₂ at approx. 95 kPa at different temperatures of this study and the so far published data for other ZIF glasses. Capacities are given in mmol g–1.

Abbreviations: im- = imidazolate, bim- = benzimidazolate, 6-Cl-5-Fbim- = 6-chloro-5-fluorobenzimidazolate, 5-Cl-2-mbim- = 5-chloro-2-methylbenzimidazolate, 5-Fbim- = 5-fluorobenzimidazolate, 5-Clbim- = 5chlorobenzimidazolate, mbim– = 2-methylbenzimidazolate.

a a_oZIF-62(M)-bim_x refers to the glass derived from the former crystalline ZIF-62(M)-bim_x material with the generalized chemical composition M(im)_{2x}(bim)_x. The nomenclature is adopted from the corresponding publication.

^b The abbreviation a_gZIF-62(Zn)-bim_{0.35} equals a_gZIF-62 in this publication which was here shortened for clarity.
^c The molar capacities are approximated from figures in the literature. Therefore, the capacities (g cm⁻³, also read out from the corresponding figure) and then divided by the molar gas volume (STP).
^d The chemical composition for ZIF-8 and ZIF-62 is here reported as Zn(mim)₂ and Zn(im)_{1.75}(bim)_{0.25}, respect

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e Chemical composition for the crystalline precursor. Chemical composition of the glass is not reported.
^f The molar capacities are approximated from figures in the literatures. The capacities in cm³(STP) g⁻¹ were re

Supplementary Table 7. Summary of adsorption capacities of N₂ and investigated hydrocarbons at approx. 95 kPa at different temperatures. Capacities are given in mmol g⁻¹.

a The isotherms display a negative gas uptake for these materials. The pores are inaccessible for Ar at 87 K.

Supplementary Methods 9.2 - Surface area and pore volume analysis

BET²¹ surface areas were determined with the Quantachrome ASIQwin version 5.2 software. The applied relative pressure ranges and quality factors are given in Supplementary Table 8. We again note that the BET model is not applicable to microporous materials. Hence, the BET areas must be taken with great care and cannot be considered as absolute physical values. We provide the values of the BET areas for comparison purposes only.

The specific micropore volumes (*V*_{pore}) were calculated according to:

$$
V_{\text{pore}} = \frac{n_{\text{ads}}^{\text{max}} \cdot M_{\text{CO2}}}{\rho_{\text{sl}}} \text{ (see ref. 22)}
$$
 (1)

with $n_{\rm ads}^{\rm max}$ the specific molar amount of gas adsorbed (mmol of gas/g material) at 195 K and 95 kPa, M_{CO2} the molar mass of CO₂, and ρ_{s1} the density of the supercooled liquid at 195 K (that is 1.258 g cm⁻³). In analogy to previous reports²³, ρ_{liq} is obtained from the linear extrapolation of the tabulated liquid phase density of $CO₂$ from its triple point temperature (216.592 K) to 195 K (Supplementary Figure 62). Reference data for the liquid phase density of $CO₂$ as a function of temperature are taken from the NIST Chemistry Webbook (https://webbook.nist.gov/cgi/cbook.cgi?ID=C124389). The obtained values are summarized in Supplementary Table 8 and S9.

Supplementary Figure 62. Temperature dependent liquid phase density of CO₂ taken from NIST Chemistry Webbook (cyan squares) and linear fit to the data from 216.592 K to 225.592 K (purple line). Data points included in the fit are drawn as filled squares, data not included as open squares.

The experimental void fractions (eVF) were calculated according to:

 $eVF = V_{pore} \cdot \rho$ (2)

with the density ρ of the solid, either obtained from the crystallographic data (crystalline ZIFs) or obtained from the apparent density approximation (glassy ZIFs, see Section 9.4).

Supplementary Table 8. Pore analysis data for the CO₂ gas physisorption studies.

^a $p \approx 95$ kPa, applied value for the density of adsorbate in its supercooled liquid state $\rho_{\text{liq}}(CO_2 \otimes 195K)$

 $= 1.258$ g cm⁻³ (see Section S9.2 for further information)

b Applied value for cross sectional areas: σ (CO₂ @273 K) = 21 Å²/molecule and σ (CO₂ @195 K) = 17 Å²/molecule²⁴.

 $p_0(CO_2, 273K) = 3485 kPa^{25}$

 $d p_0(CO_2, 195K) = 191 kPa^{26}$

^e Raw data have been already included in ref. 1.

Supplementary Table 9. Comparison of the specific micropore volumes (V_{pore}) of a_gZIF-62 and a_gTIF-4 obtained from gas sorption isotherms of *n*-butane (@273 K) and CO₂ (@195 K).

	V_{pore} (<i>n</i> -butane) ^a	V_{pore} $({\rm CO}_2)^{\text{b}}$	ratio
a _g ZIF-62	0.043 cm ³ g ⁻¹	$0.12 \text{ cm}^3 \text{ g}^{-1}$	35%
ag TIF-4	$0.050 \text{ cm}^3 \text{ g}^{-1}$	$0.12 \text{ cm}^3 \text{ g}^{-1}$	42%

 α *p* \approx 95 kPa, applied value for the density of pure liquid adsorbate $\rho_{\text{liq}}(n\text{-}b$ utane @273K) = 0.601 g cm^{.3}.²⁷

 $p \approx 95$ kPa, applied value for the density of liquid adsorbate in its supercooled liquid state $\rho_{liq}(CO_2)$ ω 195K) = 1.258 g cm⁻³.

Supplementary Methods 9.3 **-** Pore size distribution analysis

In analogy to previous studies^{1,19}, experimental pore size distributions (PSDs) were derived from CO2 isotherms at 273 K with the nonlocal density functional theory (NLDFT²⁸, carbon equilibrium transition kernel at 273 K based on a slit pore model²⁹) using the Quantachrome ASIQwin version 5.2 software. This is the only NLDFT kernel implemented in the software packages of commercial gas physisorption analysers and thus often used in the MOF literature to derive PSDs.

Theoretical PSDs were calculated with the $Zeo++^{30}$ software package using the default CCDC radii (-ha 'high accuracy' flag³¹) and the implemented routine for pore size distributions. The probe size was 0.1 Å and 5000 Monte Carlo samples per unit cell were averaged. The structures for the calculations were taken from the CSD database (ZIF-4: CCDC code IMIDZB11; ZIF-zni: CCDC code IMIDZB). In both structures, missing hydrogen atoms have been added geometrically with Olex2³².

Supplementary Figure 63. Theoretical pore size distribution calculated with Zeo++ for crystalline ZIF-4 and ZIFzni in comparison to experimental pore size distributions calculated from the CO₂ isotherms recorded at 273 K using the NLDFT method for the crystalline and glassy phases of these materials.

Supplementary Figure 63 demonstrates that the PSDs derived from experimental CO₂ sorption isotherms of crystalline ZIF-4 and ZIF-zni recorded at 273 K by the NLDFT model (carbon, slit pore) are inconsistent with the theoretical PSDs derived from their crystal structures. It is evident that the NLDFT model for carbon materials with slit pore geometry is inappropriate for the calculation of PSDs of ZIF materials. This must be ascribed to the very different surface electrostatics (non-polar carbon vs. appreciably polar ZIFs) and the different pore geometries of the ZIFs. Since the PSDs of the crystalline ZIFs is not described accurately by the utilized NLDFT model, we conclude that also the PSDs previously derived for ZIF glasses via the same method^{1,19} do not represent a meaningful description of their pore structure.

Supplementary Methods 9.4 - Density approximation

The apparent densities of the glasses were determined from the correlation of the density to the specific pore volumes determined by CO₂ adsorption isotherms at 195 K.

Feasibility test

First, the feasibility of the correlation was proven with theoretical and experimental considerations for ZIF-4 and ZIF-zni. Therefore, the theoretical and experimental void fractions were calculated.

The theoretical void fractions (tVFs) – based on the crystal structures for both materials – were calculated with the implemented routine in Olex232 applying a probe radius of 1.6 Å and grid spacing of 0.2 Å. The structures were taken from the CSD database. In both structures, missing hydrogen atoms have been added geometrically with Olex2. The theoretical accessible pore space for ZIF-4 (CCDC code IMIDZB11) amounts to 28.7% and for ZIF-zni (CCDC code IMIDZB) to 7.5%.^a

The experimental void fractions (eVFs) were calculated from the specific micropore volumes (V_{core}) derived from the CO₂ adsorption isotherms at 195 K given in cm³ g⁻¹ (see Supplementary Table 8 and Supporting Information Section S9.2) multiplied by the crystallographic densities^b (ρ_{cryst}) of the materials (eVF = V_{pore} · ρ_{cryst} , see Supporting Information Section S9.2 for further details). The densities are 1.22 g cm⁻³ (ZIF-4) and 1.56 g cm⁻³ (ZIF-zni). The corresponding eVFs amount to 30.6% and 6.2% for ZIF-4 and ZIF-zni, respectively, which are in very good agreement with the tVFs, demonstrating the feasibility of the methodology.

Exponential fit

The V_{pore} vs. ρ_{cryst} data for the crystalline compounds ZIF-4, ZIF-zni were completed with the corresponding values for ZIF-62 and TIF-4 and fitted with an exponential fitting function (see Figure 3; R^2 -value = 0.998).

$$
V_{\text{pore}}(\rho) = a \cdot e^{-\frac{\rho}{b}} \tag{3}
$$

 V_{more} = specific pore volume and ρ = density

with $a = 377.6 \pm 120.1$ cm³ g⁻¹ and $b = 0.167 \pm 0.007$ g cm⁻³

Based on the fitting function the density of the ZIF glasses can be calculated from their experimental pore volumes.

^a The same calculation has also been performed for ZIF-62 (CCDC code SIWJAM) and TIF-4 (CCDC code QOSYAZ). Before the calculation, some disordered groups were resolved and solvent molecules were removed where present. tVF values are given in Figure 1. We note that the comparison of these values to the corresponding eVFs is not applicable, because of some unresolvable residual disorder leading to partially occupied secondary linkers (bim– or mbim–).

^b All densities for crystalline materials have been calculated from the mass of atoms in one unit cell and the unit cell volume determined via profile fits of XRPD data (see Supplementary Table 1).

Supplementary References

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