

# Supporting Information

## **Defect-Induced Dense Amorphous/Crystalline Heterophase Enables High-Rate and Ultra-Stable Sodium Storage**

Sahar Osman, Chao Peng\*, Fangkun Li, Haoliang Chen, Jiadong Shen, Zeming Zhong, Wenjie Huang, Dongfeng Xue, and Jun Liu\*

S. Osman, F. Li, H. Chen, J. Shen, Z. Zhong, W. Huang, Prof. J. Liu

School of Materials Science and Engineering and Guangdong Provincial

Key Laboratory of Advanced Energy Storage Materials

South China University of Technology

Guangzhou, Guangdong 510641, China

Email: msjliu@scut.edu.cn

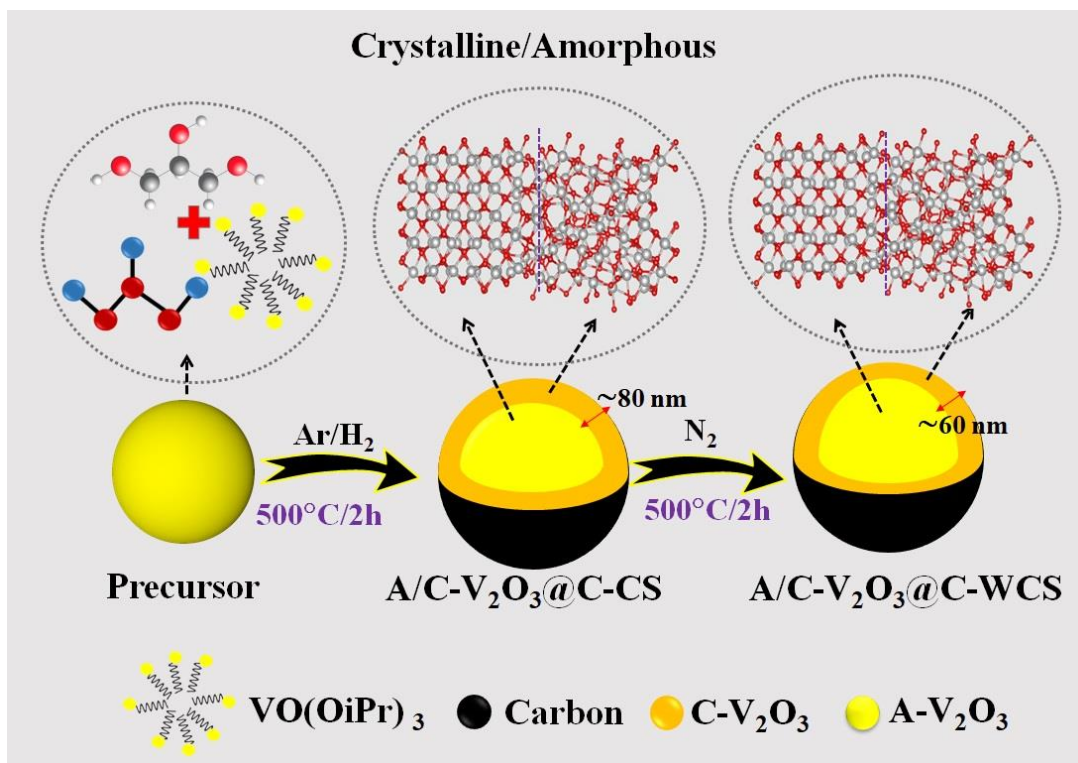
Dr. C. Peng, Prof. D. Xue

Multiscale Crystal Materials Research Center

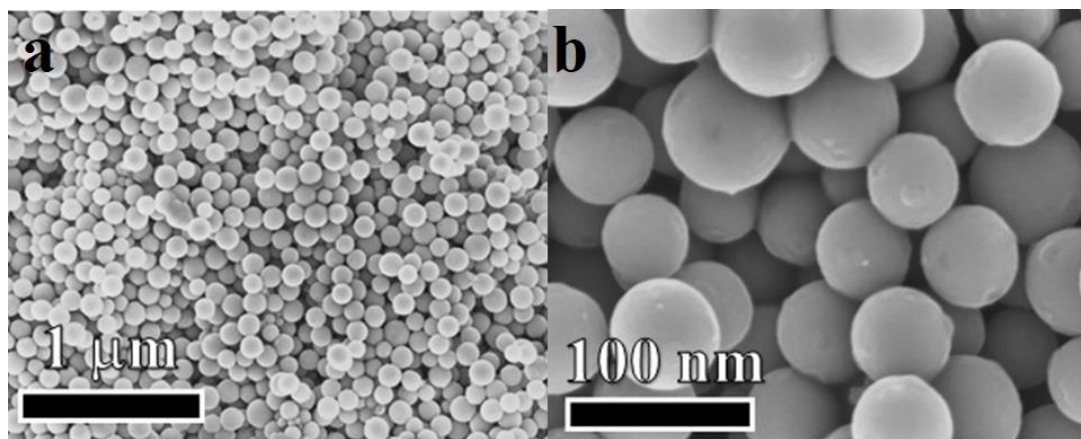
Shenzhen Institute of Advanced Technology

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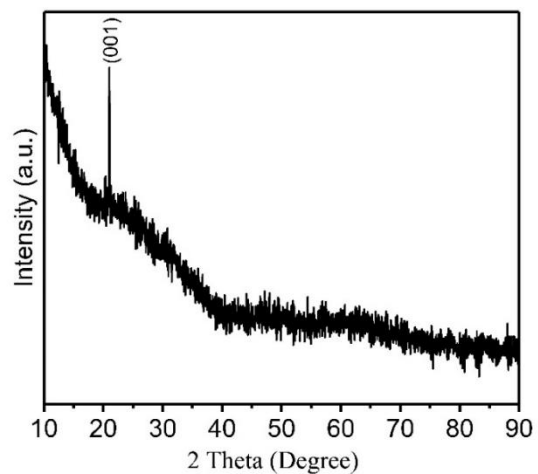
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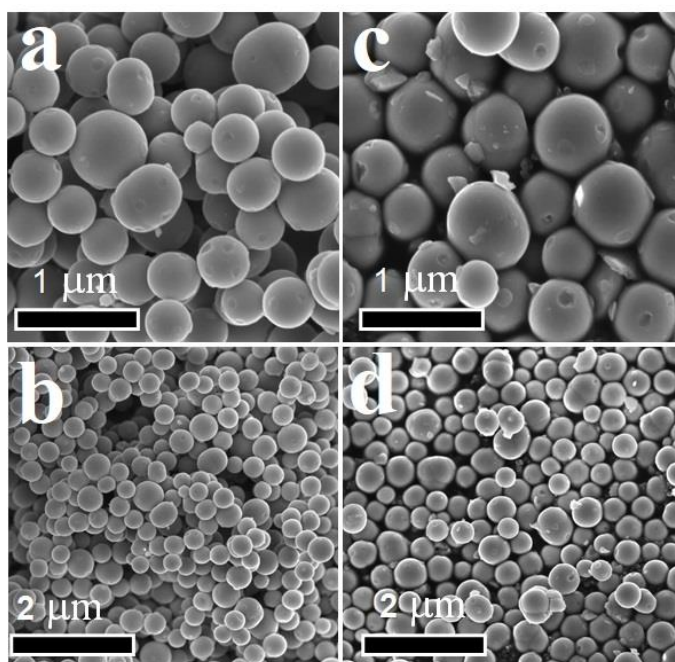
**Figure S1.** Schematic illustration of the synthesis of A/C-V<sub>2</sub>O<sub>3</sub>@C-CS and A/C-V<sub>2</sub>O<sub>3</sub>@C-WCS.



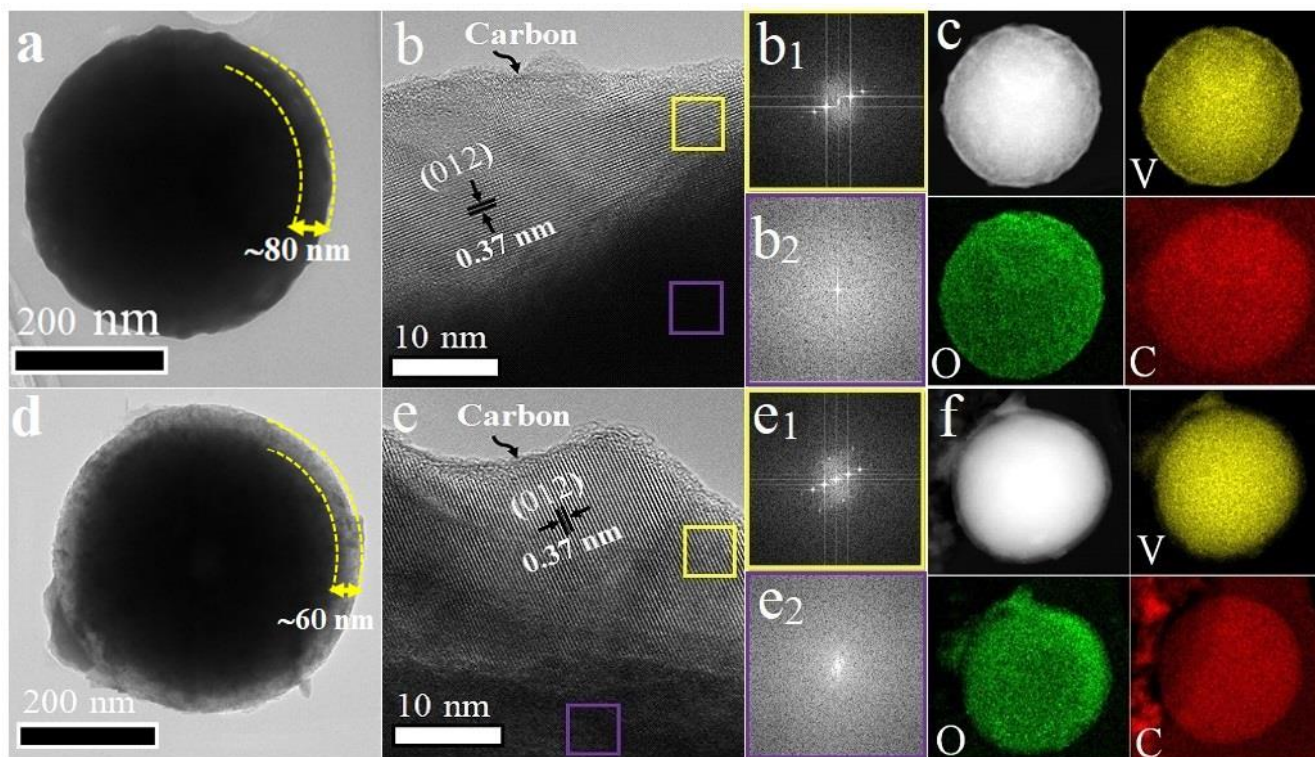
**Figure S2.** (a, b) Low/high magnification SEM images of vanadium-glycerate precursor.



**Figure S3.** XRD pattern of V-glycerate spheres.

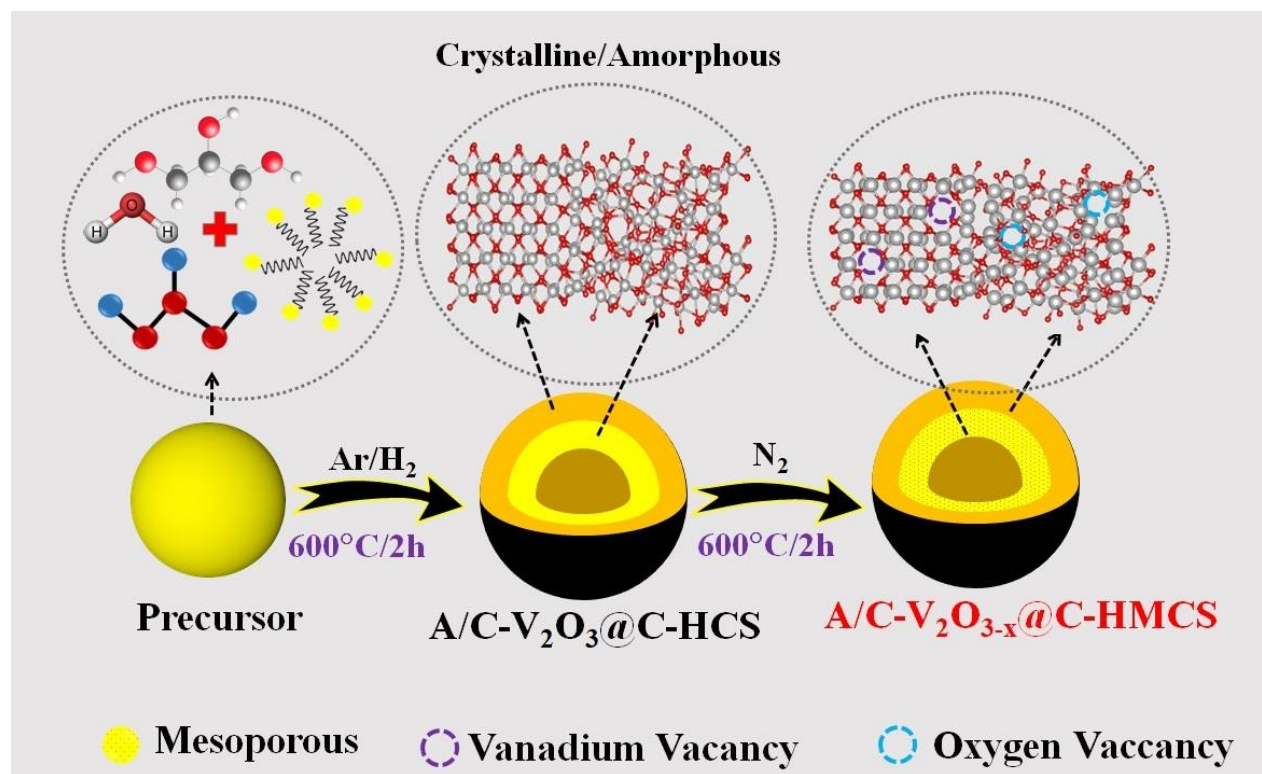


**Figure S4.** High/low-magnification SEM images of (a, b) A/C-V<sub>2</sub>O<sub>3</sub>@C-CS and (c, d) A/C-V<sub>2</sub>O<sub>3</sub>@C-WCS.

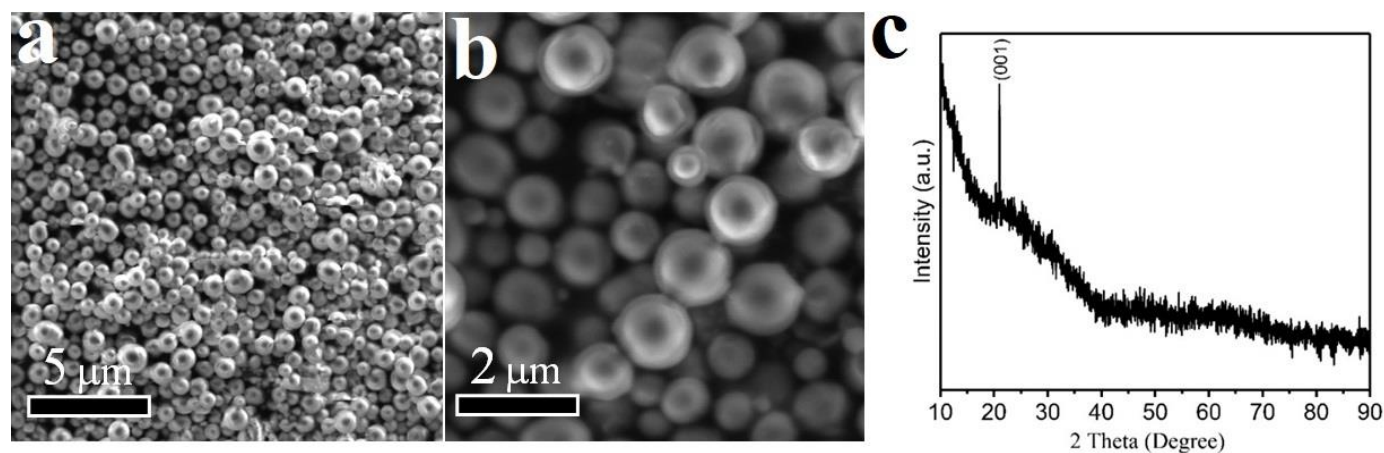


**Figure S5. Morphology and characterizations of A/C-V<sub>2</sub>O<sub>3</sub>@C-CS and /C-V<sub>2</sub>O<sub>3</sub>@C-WCS**  
 (a, d) TEM, (b, e) HRTEM images. (b<sub>1</sub>, e<sub>1</sub>) FFT patterns are taken from the corresponding yellow square and (b<sub>2</sub>, e<sub>2</sub>) violet square areas in (b, e), and (c, f) EDS mapping images of core-shell A/C-V<sub>2</sub>O<sub>3</sub>@C-CS and A/C-V<sub>2</sub>O<sub>3</sub>@C-WCS, respectively.

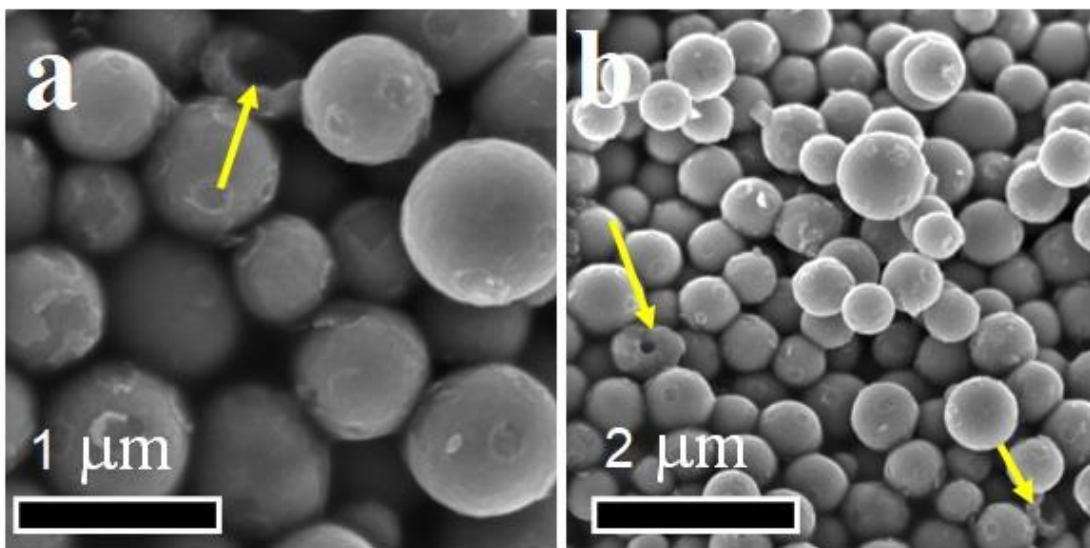




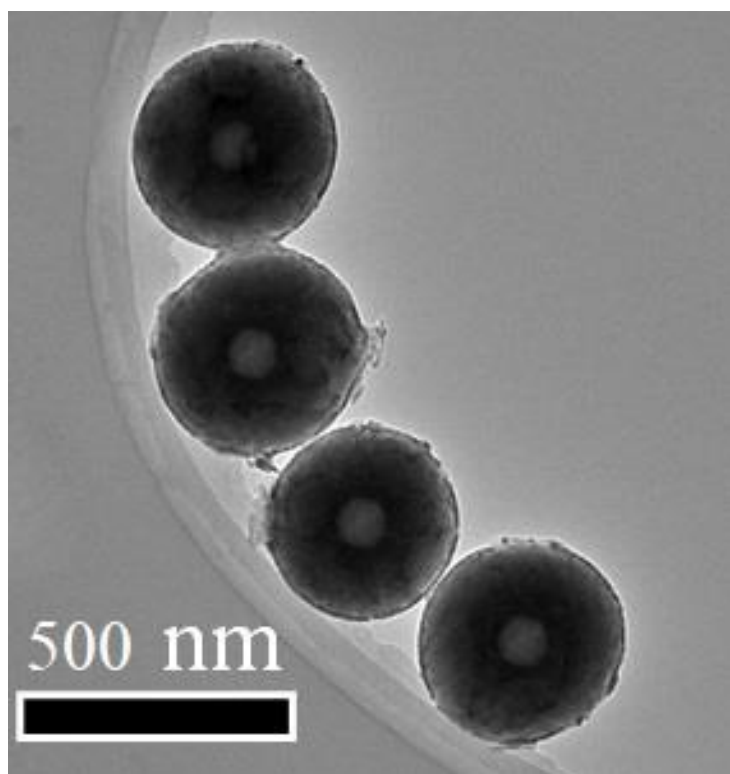
**Figure S6.** Schematic illustration of the synthesis of A/C-V<sub>2</sub>O<sub>3</sub>@C-HCS and A/C-V<sub>2</sub>O<sub>3-x</sub>@C-HMCS.



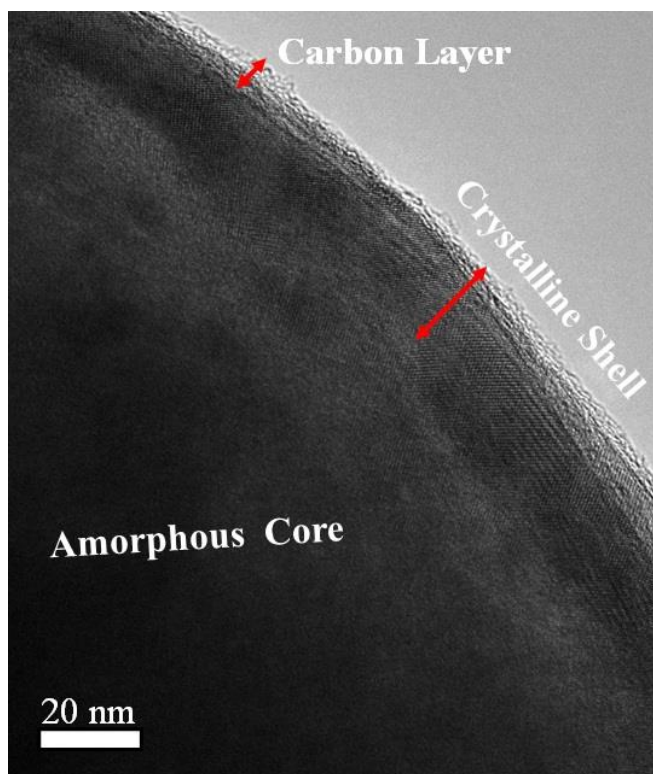
**Figure S7.** (a, b) Low/high magnification SEM images of vanadium hydrate-glycerate precursor. (c) its XRD pattern.



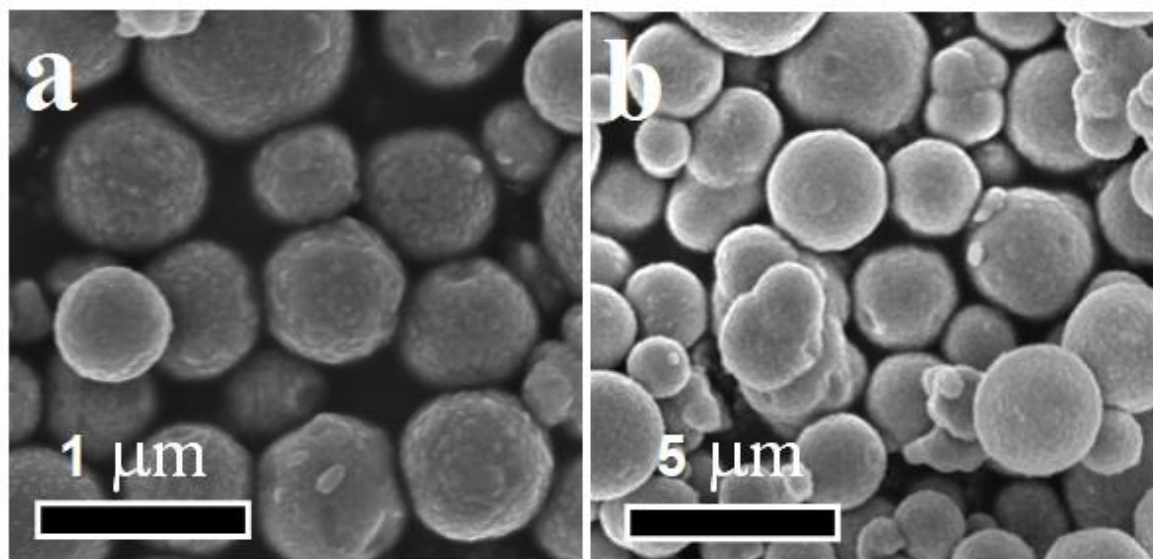
**Figure S8.** (a, b) High/low-magnification SEM images of A/C-V<sub>2</sub>O<sub>3</sub>@C-HCS.



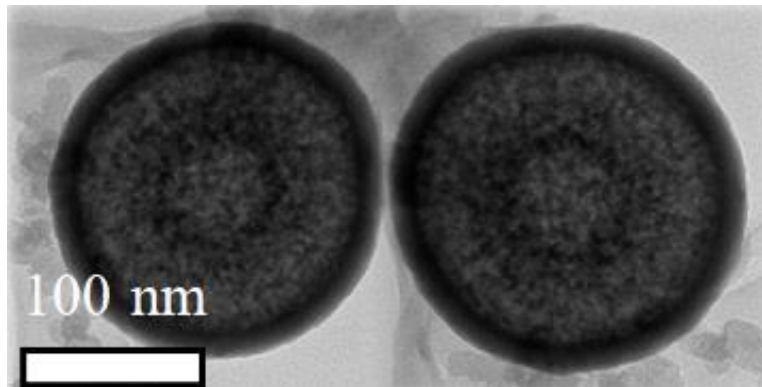
**Figure S9.** TEM image of A/C-V<sub>2</sub>O<sub>3</sub>@C-HCS.



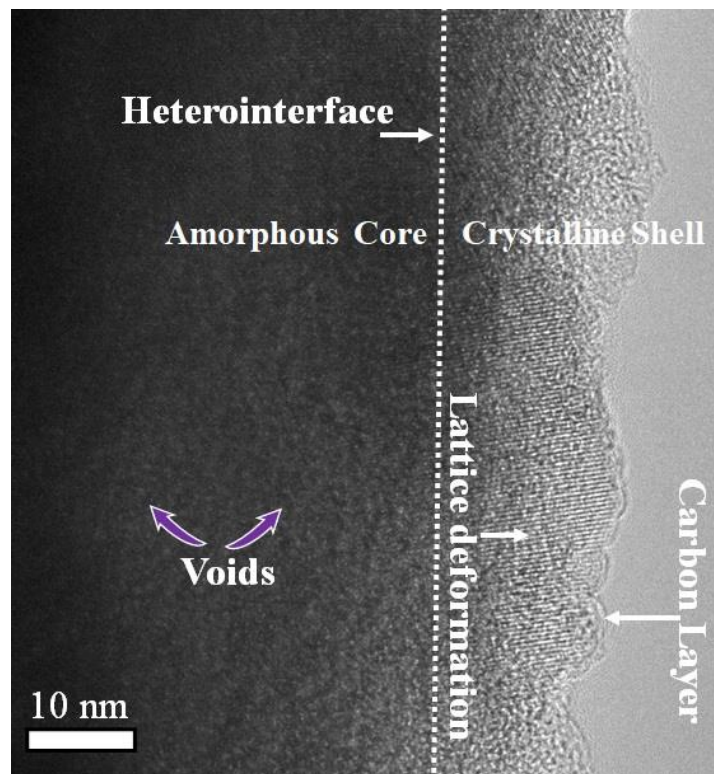
**Figure S10.** HTEM image of A/C-V<sub>2</sub>O<sub>3</sub>@C-HCS.



**Figure S11.** (a, b) High/low-magnification SEM images of A/C-V<sub>2</sub>O<sub>3-x</sub>@C-HMCS.

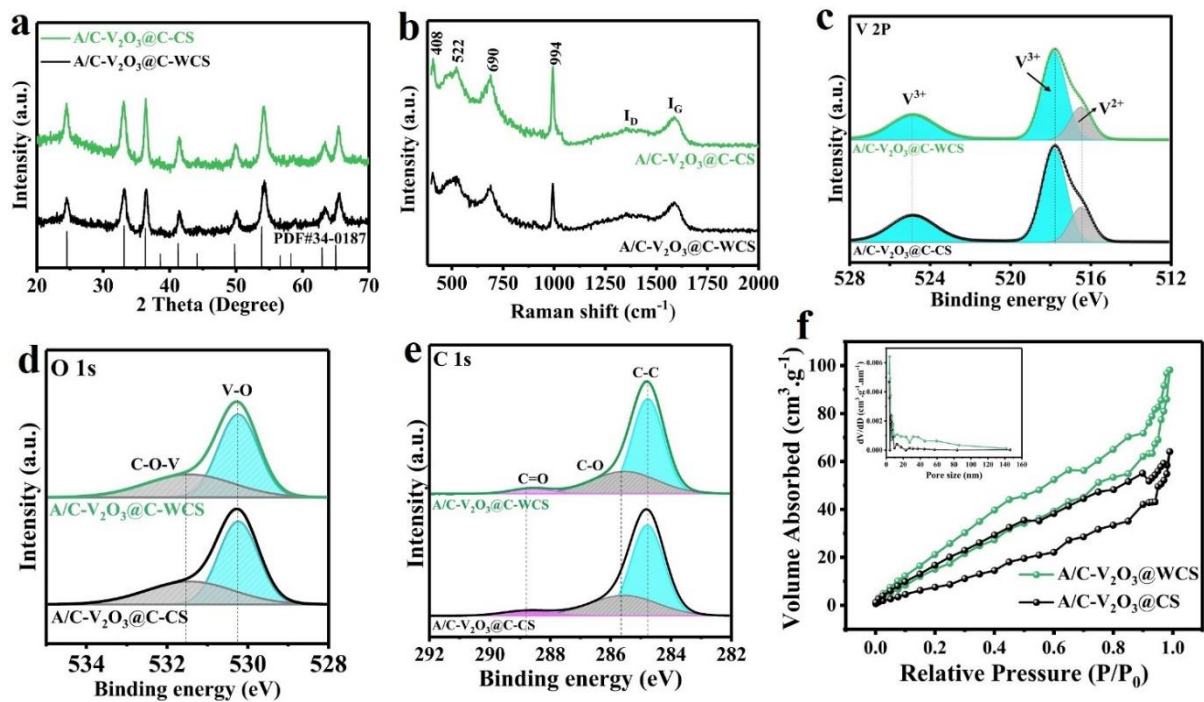


**Figure S12.** TEM image of A/C-V<sub>2</sub>O<sub>3-x</sub>@C-HMCS.

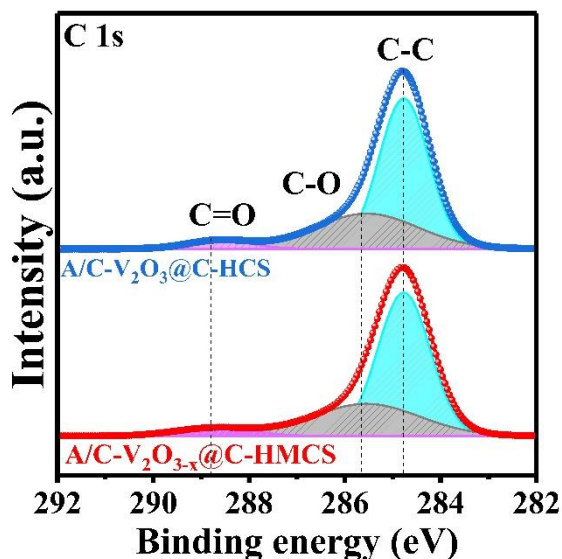


**Figure S13.** HRTEM image of A/C-V<sub>2</sub>O<sub>3-x</sub>@C-HMCS.

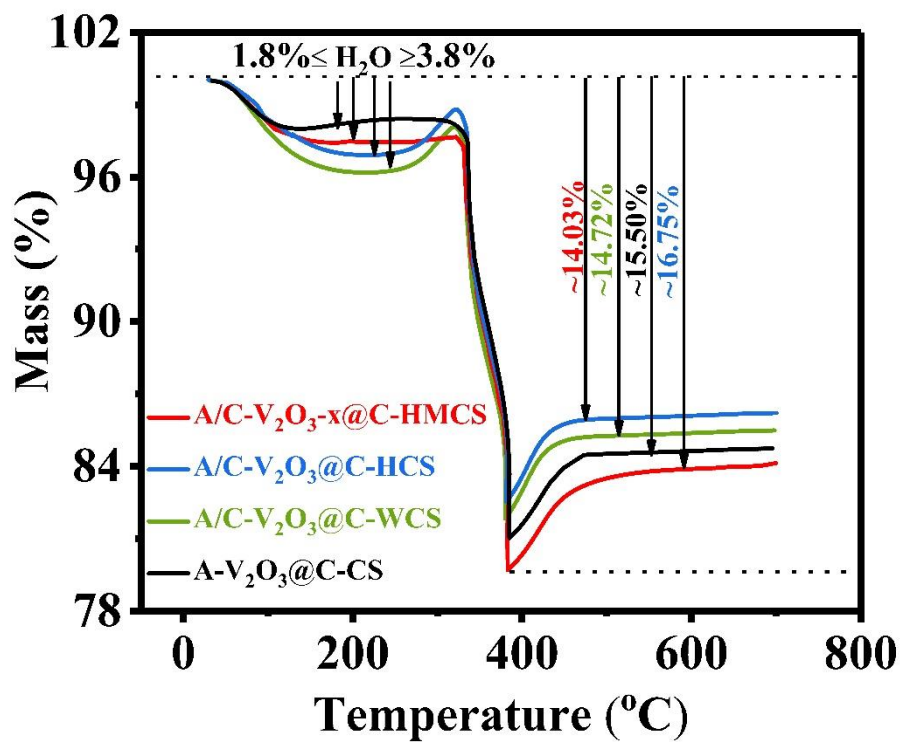




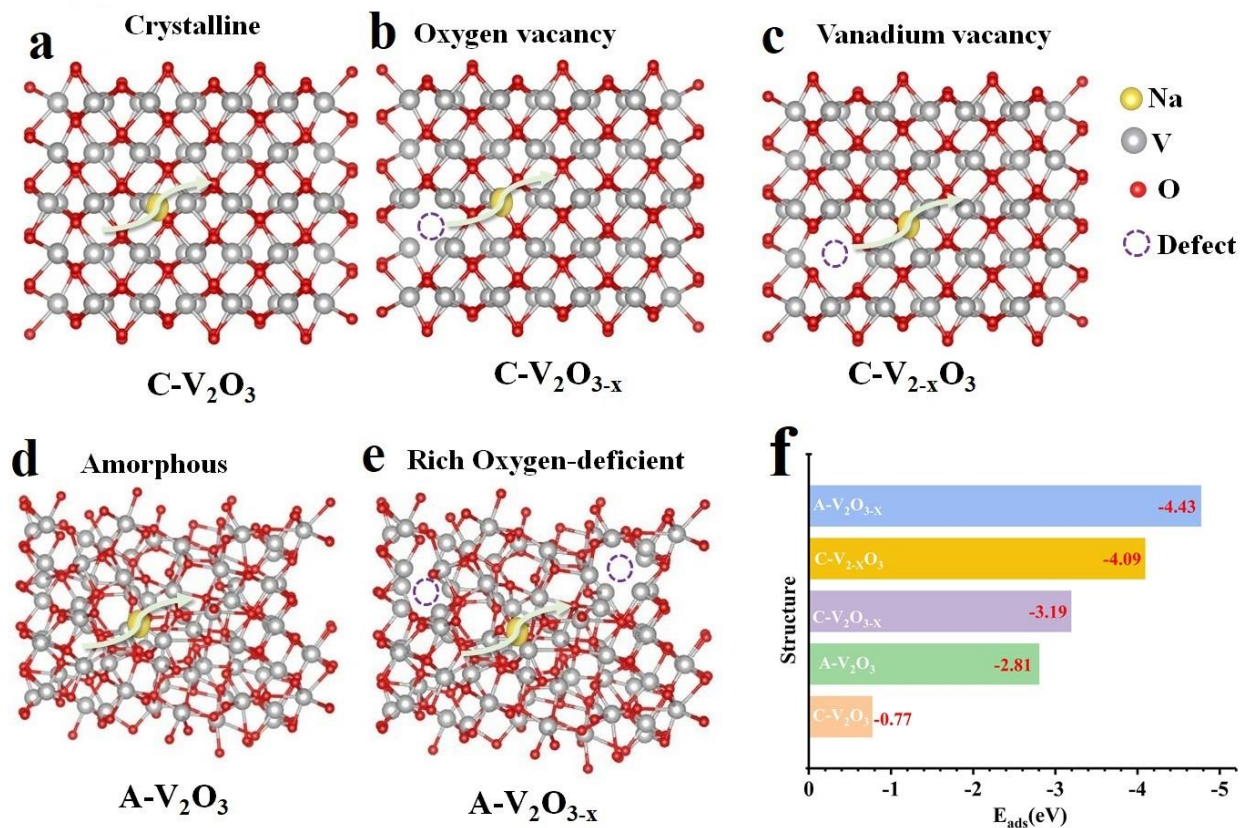
**Figure S14. Structural characterizations of A/C-V<sub>2</sub>O<sub>3</sub>@C-CS and A/C-V<sub>2</sub>O<sub>3</sub>@C-WCS:** (a) XRD patterns, (b) Raman spectra, high-resolution spectra of (c) V 2p, (d) O 1s, (e) C 1s. (f) N<sub>2</sub> adsorption-desorption isotherms and their pore size distributions (inset).



**Figure S15. C 1s high-resolution spectra of A/C-V<sub>2</sub>O<sub>3</sub>@C-HCS and A/C-V<sub>2</sub>O<sub>3-x</sub>@C-HMCS.**



**Figure S16.** Thermogravimetric analysis (TGA) curve for the A/C-V<sub>2</sub>O<sub>3</sub>@C-CS, A/C-V<sub>2</sub>O<sub>3</sub>@C-WCS, A/C-V<sub>2</sub>O<sub>3</sub>@C-HCS, and A/C-V<sub>2</sub>O<sub>3-x</sub>@C-HMCS.



**Figure S17.** Structure models of Na diffusion in (a) C-V<sub>2</sub>O<sub>3</sub>, (b) C-V<sub>2</sub>O<sub>3-x</sub>, (c) C-V<sub>2-x</sub>O<sub>3</sub>, (d) A-V<sub>2</sub>O<sub>3</sub>, (e) A-V<sub>2</sub>O<sub>3-x</sub>, and (f) their corresponding Na adsorption energies).

**Supplemental Table**  
in various V<sub>2</sub>O<sub>3</sub>

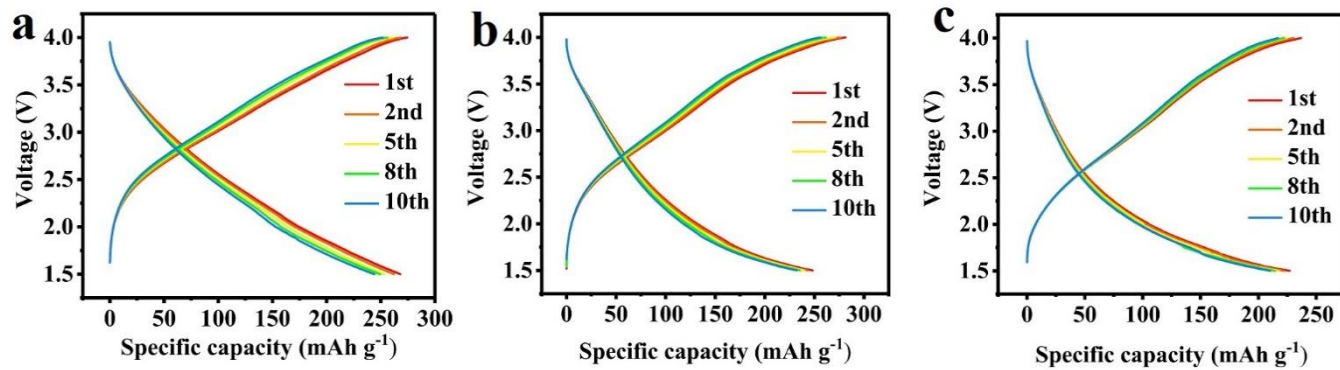
Structure	E <sub>ads</sub> (eV)
C-V <sub>2</sub> O <sub>3</sub>	-0.77
A-V <sub>2</sub> O <sub>3</sub>	-2.81
C-V <sub>2</sub> O <sub>3-x</sub>	-3.19
C-V <sub>2-x</sub> O <sub>3</sub>	-4.09
A-V <sub>2</sub> O <sub>3-x</sub>	-4.43
A/C-V <sub>2</sub> O <sub>3</sub> @C-HMCS	-1.77
A/C-V <sub>2</sub> O <sub>3-x</sub> @C-HMCS	-4.77

**S1: Adsorption energy of Na structures.**

## Theoretical discussion of Na in various $V_2O_3$ structures

The adsorption energy ( $E_{\text{ads}}$ ) of the Na atom at amorphous structure is calculated to be -2.81 eV which is much lower than that of crystalline structure (-0.77 eV), indicating that porous isotropic amorphous phase is favorable for Na intercalation (Figure S17a,d). In addition, the  $E_{\text{ads}}$  of C- $V_2O_{3-x}$  is even lower -3.19 eV (Figure S17b), which should be ascribed to oxygen vacancies, suggesting that the O-defective is highly beneficial for Na insertion. Distinctly, vanadium-deficient crystalline structure C- $V_{2-x}O_3$  displays lower  $E_{\text{ads}}$  -4.09 eV than crystalline, amorphous, and O-deficient crystalline structure, resulting in higher adsorbing ability and faster kinetics of Na storage (Figure S17c). This enhancement is attributed to a higher coordination number and radius of the vanadium atom compared to the oxygen atom, which leads to enlarged  $Na^+$  diffusion channels in the crystalline phase, thereby optimizing its kinetics. Further by combining the porous amorphous structure and oxygen-deficient, the A- $V_2O_{3-x}$  possesses a much lower  $Na^+$  adsorption energy of -4.43 eV compared to all single structures counterparts (Figure S17e), suggesting A- $V_2O_{3-x}$  is the most energetically favorable for Na insertion. This result suggests that tailoring unsaturated coordination sites is an effective strategy to enhance the adsorption of  $Na^+$  (Figure S17f, Table S1).

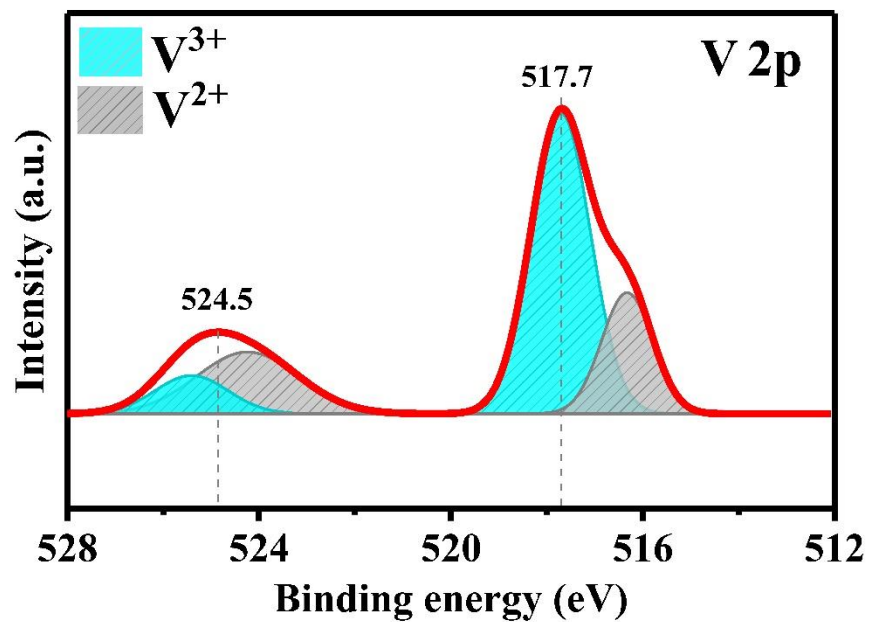




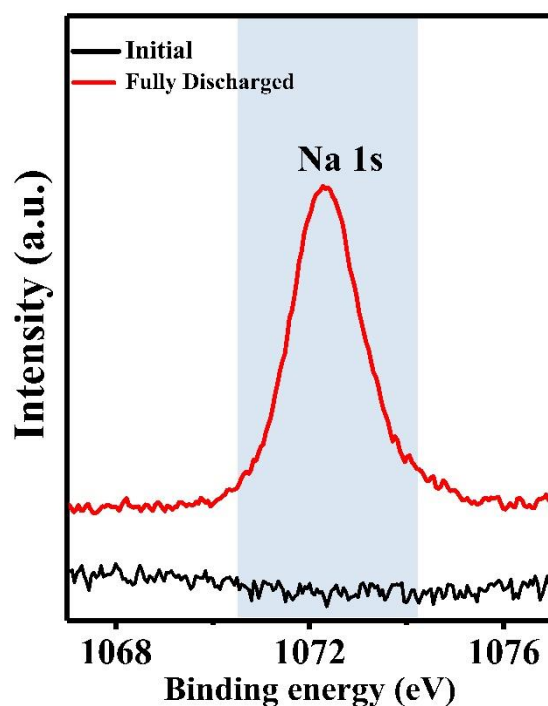
**Figure S18.** Discharge/charge profiles of (a) A/C-V<sub>2</sub>O<sub>3</sub>@C-HCS, (b) A/C-V<sub>2</sub>O<sub>3</sub>@C-WCS, and (c) A/C-V<sub>2</sub>O<sub>3</sub>@C-CS electrodes at 0.1 A g<sup>-1</sup>.

**Supplemental Table 2:** Comparison of the electrochemical performance with reported A/C-V<sub>2</sub>O<sub>3</sub>@C-M based electrodes.

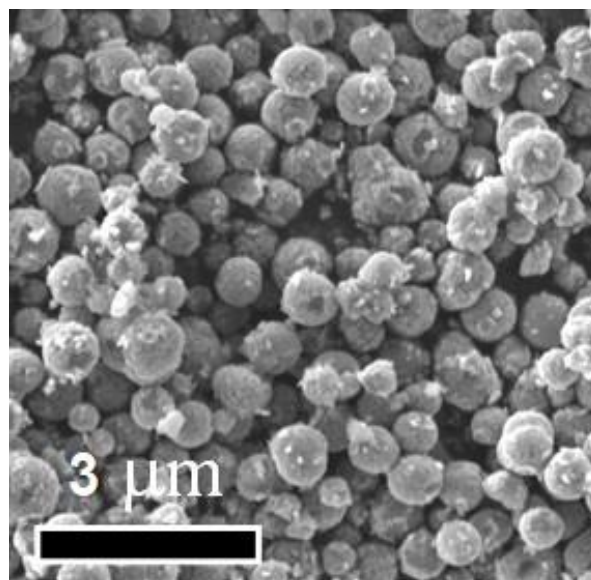
<b>Cathode description</b>	<b>Capacity Retention (% , cycles No)</b>	<b>Rate performance (mAh g<sup>-1</sup>)/ (A g<sup>-1</sup>)</b>	<b>Average voltage (V)</b>	<b>Ref</b>
A/C-V <sub>2</sub> O <sub>3</sub> @C-CS core-shell	93%,6000	92/10	2.6	This work
A/C-V <sub>2</sub> O <sub>3</sub> @C-WCS core-shell	100%,6000	123/10	2.7	
A/C-V <sub>2</sub> O <sub>3</sub> @C-HCS core-shell	100%,6000	155/10	2.8	
A/C-V <sub>2</sub> O <sub>3-x</sub> @C-HMCS core-shell	100%,6000	192/10	2.8	
NaV <sub>6</sub> O <sub>15</sub> /MWCNTs nanotube	96%,500	123.7/10	2.7	[1]
A-V <sub>2</sub> O <sub>5</sub> /C-WCS core-shell	95%,3000	148/5.0	3.0	[2]
Bilayered V <sub>2</sub> O <sub>5</sub>	85%,320	150/0.63	3.0	[3]
Additive-free V <sub>2</sub> O <sub>5</sub> thin film	100%,200	124/5.0	-0.25	[4]
Co <sub>0.16</sub> Zn <sub>0.09</sub> V <sub>2</sub> O <sub>5</sub> ·nH <sub>2</sub> O	97%,1000	90/3.0	0.7	[5]
cG/VO microspheres	73%,2000	214/4.0	1.0	[6]
VS <sub>4</sub> Nanoparticles	84%,1200	188.1/4.0	1.2	[7]
NaV <sub>6</sub> O <sub>15</sub> nanotube	94%,3000	105/2.5	2.5	[8]
NaVOPO <sub>4</sub> layered	67%, 1000	57/0.73	3.3	[9]
Na <sub>0.282</sub> V <sub>2</sub> O <sub>5</sub> nanorods	99%,1000	80/0.3	2.7	[10]
Na <sub>1.25</sub> V <sub>3</sub> O <sub>8</sub> nanowires	87%,1000	76/2.0	2.4	[11]
Na <sub>3</sub> V <sub>2</sub> (PO <sub>4</sub> ) <sub>3</sub> spheres	87.5%,1000	96/2.0	3.35	[12]
Na <sub>3</sub> V <sub>2</sub> (PO <sub>4</sub> ) <sub>2</sub> O <sub>2</sub> F nanotetraprisms	81%,2000	84/5.2	3.8	[13]
Na <sub>3</sub> V <sub>2</sub> (PO <sub>4</sub> ) <sub>2</sub> O <sub>2</sub> F carbon cloth	95%,1500	84/1.2	3.8	[14]
Na <sub>3</sub> V <sub>2</sub> (PO <sub>4</sub> ) <sub>2</sub> F <sub>3</sub> microcubes	98%,2000	53/2.2	3.6	[15]
Na <sub>6</sub> Fe <sub>5</sub> (SO <sub>4</sub> ) <sub>8</sub> /CNTs nano-architected	100.8%,1000	86.4/ 0.24	3.6	[16]



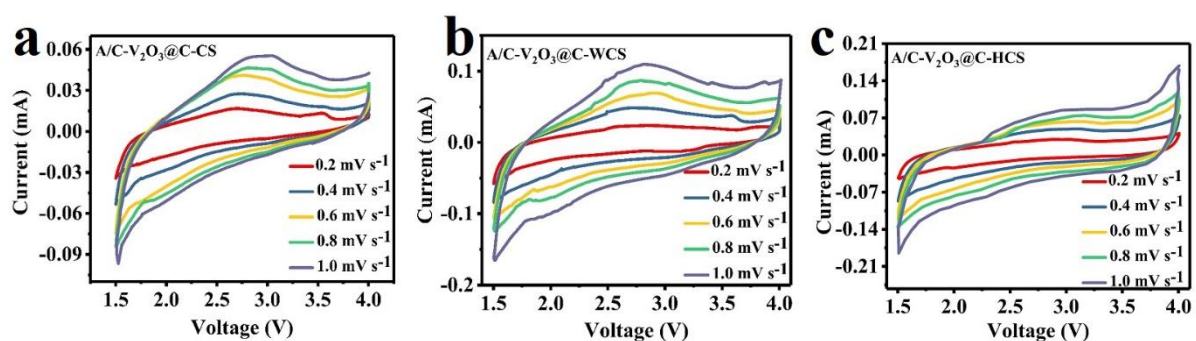
**Figure S19.** XPS spectrum of V 2p with 100 cycles for A/C-V<sub>2</sub>O<sub>3-x</sub>@C-HMCS electrode.



**Figure S20.** Ex-situ XPS Na 1st spectra of A/C-V<sub>2</sub>O<sub>3-x</sub>@C-HMCS electrodes at full discharge-stage (1.5 V).

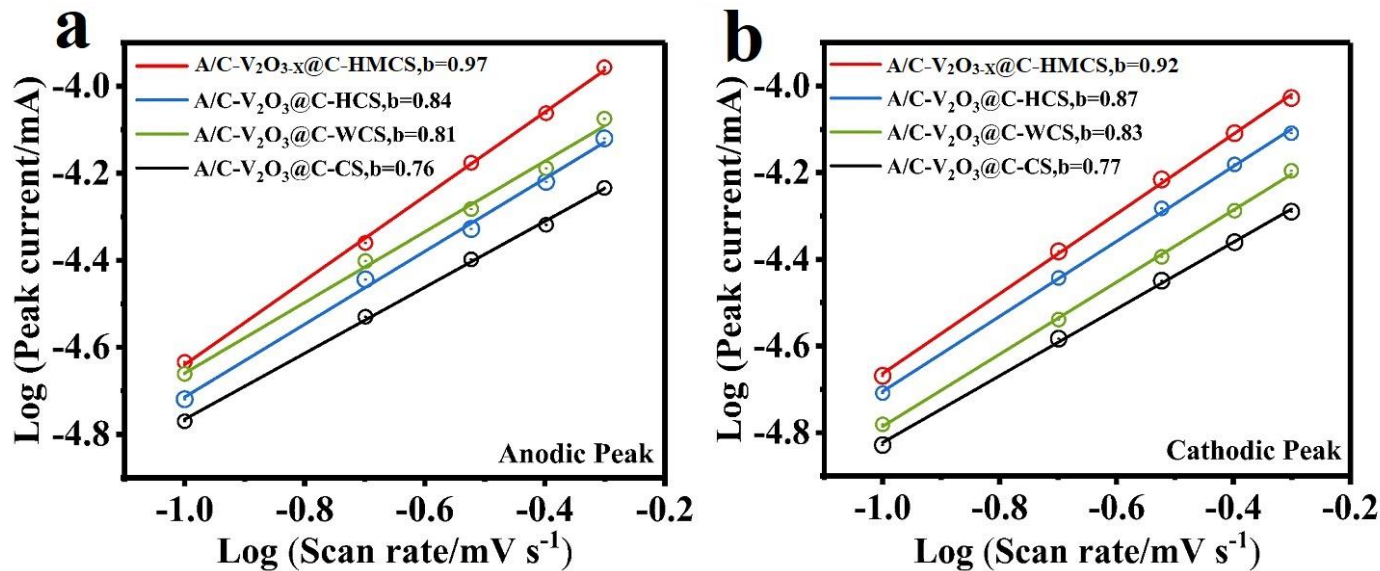


**Figure S21.** SEM images of A/C-V<sub>2</sub>O<sub>3-x</sub>@C-HMCS after 1000 cycles.

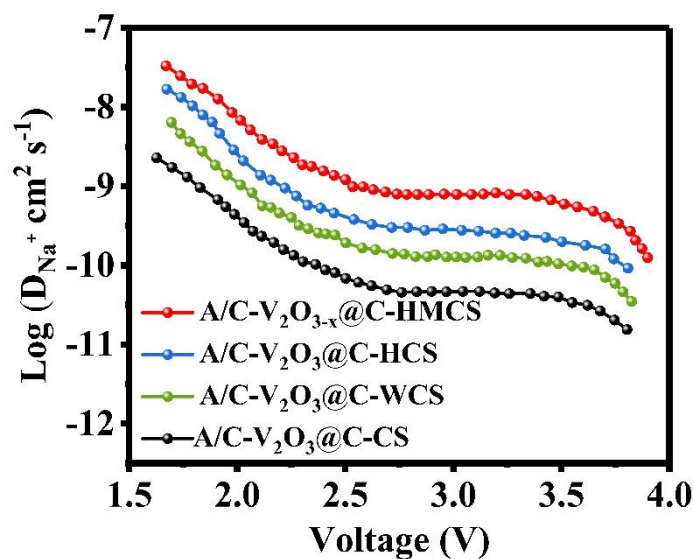


**Figure S22.** CV profiles at different scan rates (0.2-1.0 mV s<sup>-1</sup>) between 1.5-4.0 V (versus Na<sup>+</sup>/Na) of A/C-V<sub>2</sub>O<sub>3</sub>@C-CS, A/C-V<sub>2</sub>O<sub>3</sub>@C-WCS, and A/C-V<sub>2</sub>O<sub>3</sub>@C-HCS cathodes.





**Figure S23.** Calculated b-values from log (peak current) versus log (scan rate) plots for A/C- $V_2O_{3-x}$ @C-CS, A/C- $V_2O_3$ @C-WCS, A/C- $V_2O_3$ @C-HCS, and A/C- $V_2O_{3-x}$ @C-HMCS.



**Figure S24.** Diffusion coefficients calculated from the GITT during the desodiation process of A/C- $V_2O_3$ @C-CS, A/C- $V_2O_3$ @C-WCS, A/C- $V_2O_3$ @C-HCS, and A/C- $V_2O_{3-x}$ @C-HMCS.

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